



# **Fabrication and Applications of Ceramic-Based Nanofiber Materials Service in High-Temperature Harsh Conditions—A Review**

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**Abstract:** Ceramic-based nanofiber materials have attracted attention due to their high-temperature resistance, oxidation resistance, chemical stability, and excellent mechanical performance, such as flexibility, tensile, and compression, which endow them with promising application prospects for filtration, water treatment, sound insulation, thermal insulation, etc. According to the above advantages, we, therefore, reviewed the ceramic-based nanofiber materials from the perspectives of components, microstructure, and applications to provide a systematical introduction to ceramic-based nanofiber materials as so-called blankets or aerogels, as well as their applications for thermal insulation, catalysis, and water treatment. We hope that this review will provide some necessary suggestions for further research on ceramic-based nanomaterials.

Keywords: high-temperature; ceramic; nanomaterials; applications



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# 1. Introduction

In recent years, ceramic-based nanofiber materials have received more and more attention, and the publication of articles on ceramic-based nanofiber materials is also a growing trend. Regarding traditional ceramic materials, their brittleness greatly limits the development of their application areas, and flexible ceramic-based nanofibers are an efficient approach to improving this shortcoming. With the research and the derivation of many types of ceramic materials, these ceramic materials, because of their low density, low thermal conductivity, good chemical stability, and many other superior properties, are widely used in the field of high-temperature insulation, acoustics, catalysis, and other fields.

In this paper, we introduce ceramic-based nanofiber materials from two perspectives, the preparation of ceramic-based nanofiber materials and their application fields (as shown in Figure 1), in which their preparation is divided according to conventional ceramic fiber materials and ceramic aerogels. From these classifications, we describe the properties, chemical compositions, advantages, and disadvantages of the prepared materials in detail. In terms of applications, we highlight applications in high-temperature thermal insulation, air filtration, water treatment, sound absorption, electromagnetic wave insulation, battery separators, catalytic applications, etc. Finally, we offer an outlook on ceramic-based nanofiber materials.

nanofibers Oxide $Al_2O_3$ fiber $ZrO_2$ fiber Glass fiber	<ul> <li>Nitride Si<sub>3</sub>N<sub>4</sub> fiber</li> <li>Carbide SiC fiber</li> <li>Others</li> </ul>	aerogels • Oxide Al <sub>2</sub> O <sub>3</sub> aerogel ZrO <sub>2</sub> aerogel SiO <sub>2</sub> aerogel	<ul> <li>Nitride</li> <li>BN aerogel</li> <li>Si<sub>3</sub>N<sub>4</sub> aerogel</li> <li>Carbide</li> <li>SiC aerogel</li> </ul>	
	ŀ	Applications		
<ul> <li>Thermal insulation</li> <li>Air Filtration</li> <li>Water Treatment</li> <li>Sound Absorption</li> </ul>		<ul><li>Electromagnetic Wave Absorption</li><li>Battery Separators</li><li>Catalytic Application</li></ul>		



# 2. Ceramic Nanofibers

Conventional ceramic materials are usually sintered from natural raw materials such as clay and quartz, typical silicate materials. Their main elements are silicon, aluminum, and oxygen. These common ceramics are abundant sources and of low cost, and the mature process makes it easier to obtain the material. The composition can be divided into aluminosilicate, alumina, zirconia, quartz, glass fiber, etc.

#### 2.1. Oxide Fibers

#### 2.1.1. Al<sub>2</sub>O<sub>3</sub> Fiber

Aluminosilicate, alumina, and mullite mainly consist of aluminum–silicon refractories, but the content of Al<sub>2</sub>O<sub>3</sub> and the content of the main solid-phase composition are different.

Aluminosilicate belongs to a form of silicate whose composition is  $Al_2O_3$  and  $SiO_2$ , and the higher the  $Al_2O_3$  content, the better the heat resistance, as can also be seen in Table 1 [1]. These materials are excellent in terms of oxidation resistance, creep resistance, and high-temperature resistance, as well as thermal stability [2], and the high-temperature resistance reaches up to 1000–1500 °C for different chemical compositions; these mineral compositions can be divided into several categories of aluminosilicate fibers, as seen in Table 1 [1]. The first four categories, when placed at 1000 °C for a long time, will precipitate mullite, square quartz, and other crystals, but the corresponding fiber will become brittle, affecting the service life. The last two categories are crystalline fibers themselves, which are used at higher temperatures.

Aluminum silicate can be used as a heat insulation material for engine exhaust pipes and automobile silencers, the electric insulation of electric furnaces, metal melting, and high-temperature Karma high-pressure gas-filtering materials, etc. [3,4]. Aluminum silicate powder is often used as a fireproof material; for example, in the glass industry, it is used to build glass kilns, which indicates that its development prospects are very broad. In addition, it has been shown that the excellent electrochemical properties of this material can be used to prepare composite materials in the future, to further exploit its application value [5].

Alumina is another of the most important functional materials among inorganic materials, and it is one of the most researched materials nowadays. The main components of alumina fibers are  $Al_2O_3$  and a small amount of SiO<sub>2</sub>, which can stabilize the crystal and inhibit its growth. Alumina fibers are ultra-lightweight high-temperature insulation materials that have received much attention at home and abroad. Jia [6] summarized that there are many materials that can be processed into  $Al_2O_3$  fibers. Alumina fibers

have low thermal conductivity, low heating shrinkage, good chemical stability, good hightemperature resistance, and can be used at temperatures up to 1450–1600 °C, which is higher than ordinary aluminosilicate fibers, used as thermal insulation materials for nuclear reactors and space shuttles. In addition, they can also be used as catalyst carriers for the chemical industry. Their chemical resistance is also excellent for use in the fields of environmental protection and recycling technology.

	Class Number	1	2	3	4	5	6
	Grading temperature/°C	<1200	1260	1400	1400	1550	1600
	SiO <sub>2</sub>	53.9	53	45	55	15	5
Componente	$Al_2O_3$	43.4	47	55	41	85	95
Components	$Cr_2O_3$	-	-	-	4	-	-
	TiO <sub>2</sub>	1.7	-	-	-	-	-
	Fe <sub>2</sub> O <sub>3</sub>	0.8	-	-	-	-	-
	$K_2O + Na_2O$	0.2	-	-	-	-	-
	Content						
	Mullite	~65	65	75	57	54	18
Phase	Cristobalite	35	35	25	43	-	-
	Al <sub>2</sub> O <sub>3</sub>	-	-	-	-	-	-
	Structure	Amorphous	Amorphous	Amorphous	Amorphous	Polycrystalline	Polycrystalline

Table 1. Classification and chemical composition of aluminosilicate fibers [1].

The company 3M has developed many Nextel series alumina fibers, the earliest of which was the Nextel-312 fiber [7]. The  $B_2O_3$  component in this fiber was initially intended to increase the nucleation density of mullite and reduce grain growth [8], but it was found that  $B_2O_3$  volatilizes easily at high temperatures, resulting in insufficient thermal stability in Nextel-312. The Nextel-440 fiber, which has better high-temperature resistance, was prepared for this  $B_2O_3$  reduction component. Later, the Nextel-550 fiber completely removed B<sub>2</sub>O<sub>3</sub>, providing it with good high-temperature resistance and creep resistance compared to the previous two fibers. The development of the Nextel-610 and Nextel-720 fibers has proven that not all oxides have low tensile strength and resistance to high-temperature creep [9–11]. The Nextel-650 [12,13] fiber content includes ZrO<sub>2</sub>, a small amount of  $Y_2O_3$ , and a very small amount of  $Fe_2O_3$ , in addition to  $Al_2O_3$ . The Nextel-650 fiber has a monofilament tensile strength of 2.5–2.7 GPa; it hardly creeps at 1100 °C and exhibits good creep resistance [14]. In addition, 3M has found that the characteristics of fiber continuity can be used to prepare filter carriers with a porous network structure [15]. Jia [16] prepared flexible  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> nanostructured fibers using alumina and isopropanol as raw materials, glacial acetic acid and hydrochloric acid as catalysts, and polyvinyl pyrrolidone (PVP) as a spinning auxiliary. The characterization results showed that the fiber surface was dense, without holes and cracks. Its mechanical properties also showed good data. It was also found that the phase structure and morphology of alumina nanostructure fibers doped with SiO<sub>2</sub> changed significantly. The  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> fiber has a minimum thermal conductivity of 0.0476 W·m<sup>-1</sup>·K<sup>-1</sup> and is converted to  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> after calcination, and its thermal conductivity increases to 0.0773 W·m<sup>-1</sup>·K<sup>-1</sup>. Zhang [17] also used sol-gel combined with electrospinning to prepare flexible alumina nanostructured fibers. It was found that the addition of MgSO<sub>4</sub> could initially delay the phase transition of fibers from  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> to  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, and then effectively promote sintering. When subjected to heat and stress-strain, the fibers exhibited good thermal stability and flexibility. Doped CaO and  $SiO_2$  two-phase additives will cause the fiber to possess lower thermal conductivity, which has better applications in the field of thermal insulation.

Li [18] used aluminum powder as an aluminum source, tetraethyl orthosilicate as a silicon source, and polyethylene oxide as a presumptuous auxiliary, and prepared alumina fibers by sol–gel combined with electrospinning technology. Compared with pure alumina fiber, this flexible alumina nanostructured fiber has a dense surface without holes and cracks, showing good flexibility and impressive mechanical properties. Its tensile strength

is increased by 78%. In addition, its filtration efficiency is as high as 99.88%, indicating its good application prospects in high-temperature filtration and other fields. Wei [19] studied the heat treatment process of sol–gel to prepare alumina fibers. It was found that the heat treatment rate had a great influence on the performance of the fiber, and it is not suitable to use a high heating rate. In the heat treatment at 550 °C, the volume and mass shrinkage of alumina fibers accounted for a high proportion (up to 85%). In heat treatment with a temperature above 800 °C, the crystal phase will change from amorphous to  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> to mullite.

On the basis of alumina fibers, ultra-fine alumina fibers can be obtained by reducing the diameter of alumina fibers. Milanović [20] used electrospinning to prepare alumina fibers, adding magnesium oxide to the alumina structure for sintering. The sintered fiber is finer in diameter and smaller in volume, so as to show better flexibility and specific surface area. In addition, the design of a certain unique structure can also be used to significantly improve the performance of ultra-fine alumina fibers [21].

Almeida [22] studied a CeraFib75 alumina continuous fiber containing 175 µm mullite particles and trace  $\gamma$ -alumina. Compared to the previously developed Nextel-720 fiber (Nextel-720 fiber was once considered the commercial  $Al_2O_3$  fiber with the best resistance to high-temperature creep), it can exhibit higher strength retention at high temperatures. Song [23] used a sol–gel method to synthesize alumina fibers with two systems: AlCl<sub>3</sub>-Al powder– $H_2O$  and Al( $NO_3$ )<sub>3</sub>-Al powder– $H_2O$ . In the aluminum chloride system, an  $Al/AlCl_3$  molar ratio between 2 and 5 can lead to a better sol, while, for the aluminum nitrate system, the spinning range is narrow. It was found that Cl<sup>-</sup> is more difficult to decompose than NO<sub>3</sub><sup>-</sup>, but the phase transition temperature from  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> to  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> in the AlCl<sub>3</sub>-Al powder–water system is lower. When sintered above 800 °C, the fibers of the aluminum nitrate system are less broken and have fewer cracks than the fibers of the aluminum chloride system. Sun [24] synthesized an Al<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub> precursor sol with a certain viscosity with aluminum nitrate, isopropanol, and ethyl orthosilicate. After the hydrolysis polycondensation reaction, 10–20 nm sol particles would be generated in the sol system. In acidic conditions, Si-O-Si, Si-O-Al, Al-O-Al, and other structural units existed in the sol extraction product. Under aging, the system changes from sol to gelatinous, which is reversible and stable.

Chen [25] mixed alumina sol and silica sol, added spinning additives, and then obtained continuous alumina-based ceramic fibers by dry spinning and pyrolysis, and sintering. It was found that the ratio of aluminum powder to aluminum chloride, aluminum salt concentration, reaction time, oxygen content, and other factors had an effect on the sol. The pH value of the solution is a key factor in determining the performance of the aerosol. Tan [26] mixed acetic acid, ethyl orthosilicate, a sol stabilizer, an inorganic salt of aluminum, and aluminum powder, and reacted them at a certain temperature, with continuous stirring and condensation reflux, to obtain a precursor silicon-aluminum sol with good stability. Spinning additives were added to obtain a spinnable precursor sol. Alumina-based fibers that could be used in aerospace, automobiles, and composite reinforcements can be obtained through dry/wet spinning, drying, and sintering. At the same time, Tan also mixed aluminum nitrate and PVP with three different organic acidslactic acid, malic acid, and tartaric acid, respectively—to prepare  $Al_2O_3$ -based ceramic continuous fibers. Experiments showed that the solution spinning performance was the best when the mass ratio of aluminum nitrate, organic acid, and PVP was 10:3:1.5, without a dependency on the type of organic acid.

Mahapatra [27] prepared  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> microfibers with a diameter between 100 and 500 nm using aluminum acetate sol and PVP polymer solution as raw materials. The surface morphology and fiber diameter of the alumina fibers were measured. Characterized by crystal equality, it had a surface area of 40 m<sup>2</sup>/g and was highly crystalline in nature. Wang [28] successfully prepared a new independent  $\gamma$ -alumina fiber film with good flexibility, randomly arranged by nanofibers by electrospinning. This alumina fiber film possessed great

thermal stability (up to 900  $^{\circ}$ C) and high tensile strength. In addition, its filtration efficiency could be as high as 99.97%, indicating its potential in the field of high-temperature filtration.

By studying the preparation method of alumina, Li [29] found that by adjusting some process parameters, they could successfully design alumina fibers with different diameters in the sol–gel process, with different components and different structures, as shown in Figure 2. In addition, inorganic aluminum sources with a controlled number of organic components, an optimized calcination temperature and speed, etc., could be used to save costs and improve the overall performance of fibers.



**Figure 2.** Structure construction for sol–gel ultra-fine alumina fiber (adapted with permission from Ref. [29]. 2020, Elsevier).

Wang [30] successfully synthesized mesoporous alumina fibers with a surface area of 264.1  $m^2 \cdot g^{-1}$  using electrospinning and sol–gel methods. Calcined fibers exhibited an adsorption capacity of up to 781.25 mg  $g^{-1}$ . Li [31] combined the sol-gel method and heat treatment process to obtain an alumina fiber mat with good flexibility that can be applied to high-temperature processes, filtration, catalysis, and other fields. By adjusting the parameters, the diameter of the alumina fiber can be controlled to  $1 \sim 3 \mu m$ . Sedaghat [32] prepared an alumina mat by sol-gel centrifugal spinning using AlCl<sub>3</sub>·6H<sub>2</sub>O, Al powder, and SiO<sub>2</sub> as raw materials. The phase change temperature of this mat is 600 °C, and there is no amorphous phase at 800 °C, while  $\theta$ -alumina is the most notable component. The transition of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> is completed at 1200 °C, and the presence of SiO<sub>2</sub> inhibits the transition of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. The optimal proportion of Si in the alumina mat is 4%, and, if it exceeds this, mullite will form in the microstructure, as displayed in Figure 3. Akia [33] optimized the preparation of aluminum isopropyl and poly(vinyl alcohol) fine fibers, and calcined the fiber samples at different temperatures to obtain  $\gamma$  and  $\alpha$  phases. The analysis results showed that the surface area of the mesoporous  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> structure was 261 m<sup>2</sup>·g<sup>-1</sup>, and the average diameter of the crystalline alumina fine fibers was 272 nm.

The amount of alumina in mullite is between that of  $Al_2O_3$  fibers and aluminum silicate fibers, fluctuating between 72 and 78%. Mullite is a high-quality refractory material, mainly including high-purity electrofused mullite, ordinary electrofused mullite, all-natural bauxite concentrates, sintered mullite, and lightly burned mullite. Mullite is formed when aluminosilicates are artificially heated.

Chen [34] prepared mullite ceramic fibers with aqueous solutions of aluminum nitrate, isopropyl alumina, and tetraethylchlorosilicate. Spinnable sols exhibited high-viscosity shear thinning flow behavior during the gelation time. By temperature regulation, the degree of gelatinization could be stabilized at a certain value. Crack-free ceramic fibers with

a tensile strength of approximately 900 Mpa and a diameter between 45 and 50  $\mu$ m would be formed in the sintering conditions under 1100 °C. Zadeh [35] synthesized continuous mullite nanofibers using sol–gel and electrospinning techniques. Adjusting the content of polyvinyl butyral (PVB) can obtain materials of different qualities. When the PVB content is between 4 and 6%, after calcination at 1200 °C, very pure, smooth, uniform mullite nanofibers can be obtained. Moreover, the mullite nanofibers have a small diameter (85~130 nm). Song [36] prepared mullite using the same method. Due to densification, the average diameter of the fibers decreases from 318 to 261 nm when the temperature rises from 800 to 1200 °C. The surface of the mullite fibers becomes relatively smooth at 1000 °C due to the viscous flow of amorphous silica. At the same time, the fiber exhibits uniform characteristics at an Al/Si molar ratio of 2.98. The low modulus of elasticity and good flexural properties around 25.18 GPa reveal excellent flexibility. da Costa Farias [37] produced sub-micro-mullite fibers for the first time using solution blow spinning technology. This technology is not only effective but also has high yields. This new method produces mullite without cracks, with an average diameter of 800 nm and a high surface area.



**Figure 3.** Chemical composition of the alumina mat (4 wt% SiO<sub>2</sub>) at different temperatures with a holding time of 4 h by XRF analysis after the loss of ignition (L.O.I.) test (adapted with permission from Ref. [32]. 2006, Elsevier).

#### 2.1.2. ZrO<sub>2</sub> Fiber

ZrO<sub>2</sub> ceramics are also a potential material for high-temperature applications due to possessing excellent properties such as high melting and boiling points, high hardness, low thermal conductivity, high toughness, wear resistance, etc. Compared with other fibers, such as mullite, alumina, and alumina-silicate, ZrO<sub>2</sub> fibers could still maintain their integrity at a higher temperature of up to 2200 °C. At the same time, they have the advantages of corrosion resistance, oxidation, non-pollution, etc., and they are currently the most popular international refractory material.

There are three crystalline forms of pure  $ZrO_2$  at atmospheric pressure: monoclinic (m-ZrO<sub>2</sub>), tetragonal (t-ZrO<sub>2</sub>), and cubic zirconia (c-ZrO<sub>2</sub>) [38]. These three crystallines exist in different temperature ranges and can be transformed into each other. When the  $ZrO_2$  matrix is lowered from a high temperature to 900 °C, the tetragonal form will transform into the monoclinic form with volume expansion, cracks, and residual stresses, which appear inside the matrix and are prone to brittle fracture [39]. Rare earth oxides such as  $Y_2O_3$  and CeO<sub>2</sub> are therefore usually added to inhibit the fracture caused by the reverse transformation.

Han [40] examined thermal insulation tiles composed of short-cut zirconia fibers in an arc wind tunnel, which showed that zirconia can withstand very high temperatures. Similarly, zirconia fibers have some defects in their mechanical properties, and usually researchers add alumina to improve their properties. Li [41], using slurry impregnation and discharge plasma sintering techniques, prepared alumina-reinforced zirconia composites, and the toughening effect was nearly doubled, as also confirmed by Wang [42]. Lang [43] also prepared YSL composites with excellent properties based on this, which greatly enhanced the prospects of zirconia.

Wang [44] synthesized a variety of three-dimensional sponges based on oxide ceramics using an economic and effective blowing and spinning technology. Among them, the ZO<sub>2</sub> nanofiber sponge not only showed high-temperature resistance, but also showed excellent high-energy absorption and good compressive strain and recovery, and it can be used in high-temperature applications, catalysis, electricity, and other fields. Sun [45] prepared and characterized zirconium polyacetate (PZA) and found that the obtained cubic-phase  $ZrO_2$ fibers, after adding 6 mol%  $Y_2O_3$  and calcination at 1200 °C, possessed a grain production activation energy of approximately 24.18 kJ·mol<sup>-1</sup>. Moreover, the quality of the fiber can be further improved by spinning and heat treatment of the sol–gel of the aforementioned PZA. By using  $ZrOCl_2 \cdot 8H_2O$  (ZOC) and  $H_2O_2$  as raw materials, Liu [46] proposed a novel inorganic sol–gel method to prepare high-quality  $ZrO_2$ . It was found that the inorganic polyzirconium molecules in the spinning solution had a double-stranded structure, which resulted in the excellent strength and flexibility of the prepared  $ZrO_2$  fibers.

Wang [47] reported a nanofiber sponge based on yttrium-stabilized ZrO<sub>2</sub> (YSZ), exhibited in Figure 4. Porous three-dimensional sponges composed of YSZ nanofibers are not only lightweight but also elastic, both at room temperature and high temperatures. This sponge has great application potential in the field of filtration, with filtration efficiency of up to 99.4%.



**Figure 4.** Fabrication process of YSZ nanofiber sponge (adapted with permission from Ref. [47]. 2018, WILEY-VCH Verlag GmbH Co. KGaA).

#### 2.1.3. Glass Fiber

Glass fiber is an inorganic non-metallic material with excellent performance. It has the advantages of thermal insulation, heat resistance, and great mechanical properties, as well as some disadvantages, including brittleness, poor wear resistance, and a monofilament diameter of a few microns to twenty microns. Therefore, glass fiber is also often used as a reinforcing material for composite materials, electrical insulation materials, and thermal insulation materials.

Ma [48] showed, in their research experiments on carbon and glass fiber prepregs, that glass fibers do not decompose at 850 °C compared to carbon fibers. Both of them have an inhibitory effect on the pyrolysis of the composites, which is more pronounced for glass fibers. Sujatha [49] examined two different fiber glasses. One was an alkali-resistant

8 of 36

glass fiber, and the other was an electronic-grade glass fiber. Through experiments, it was found that the random addition of these two fibers can improve the unlimited compressive strength and energy absorption capacity of reinforced soil.

Wang [50] prepared glass fiber by using a high-temperature furnace and auxiliary raw materials. Through the test analysis, they found that the diameter of the glass fiber obtained by this method was 11.28  $\mu$ m. The average linear density value reached 4.42 g/km, and the average tensile strength of the glass fiber was 0.57 N. In addition, glass fiber also has good light absorption performance, and the acid and alkali resistance are significantly improved compared to those of unmodified glass fiber. Ma [51] found that glass fiber can improve the shortcomings of recycled concrete with many internal holes, poor compressive resistance, and poor acid and alkali resistance.

Quartz, mainly consisting of SiO<sub>2</sub>, is one of the most widely distributed minerals on the Earth's surface. It was used in early times to produce stone axes and other tools for survival and was later fused to create glass, which is also used in various areas, including metallurgy, construction, and the chemical industry. Quartz fiber is an inorganic fiber composed of high-purity quartz or natural crystal, generally a few microns to tens of microns in diameter. Quartz fiber is an excellent material for high-temperature resistance and is often used as a reinforcing phase for some composite materials. The SiO<sub>2</sub> content in quartz fiber is almost 100%. Its high-temperature resistance is higher than that of silica oxygen. In addition, quartz has a long-term usage temperature of up to 1200  $^{\circ}$ C. It also has superior electrical insulation, ablation resistance, and chemical stability, which is one of the reasons that quartz fiber is used in national defense, the military, aerospace, and other fields.

As mentioned above, oxide fibers have shown their potential for applications in high-temperature and harsh oxidative conditions, because of their excellent oxidation resistance and chemical stability. However, there exist shortcomings that have prevented their utilization in the grain-coarsening process over 1300 °C. Therefore, a new goal is to search for novel materials for higher-temperature utilization, such as carbide or nitride.

#### 2.2. Nitride Fibers

#### Si<sub>3</sub>N<sub>4</sub> Fiber

The most widely used among nitride ceramic fibers is silicon nitride  $(Si_3N_4)$ .  $Si_3N_4$  fiber is a type of ceramic fiber with high-temperature resistance and high strength. The high-temperature resistance of  $Si_3N_4$  fibers is particularly outstanding. The maximum use temperature can reach 1300 °C in oxidizing gases and up to 1800 °C in non-oxidizing gases. In addition, its mechanical and physical properties are also excellent. Based on the many excellent properties of  $Si_3N_4$  fiber, material scientists in various countries are very interested in it.  $Si_3N_4$  ceramic fibers have also undergone a very extensive research process. Thus far, the research on  $Si_3N_4$  ceramic fiber has achieved great results.

Zhang [52] synthesized a novel silicon nitride nanowire sponge. They used carbondoped silica sol foam as the skeleton structure and prepared it via a simple and effective carbon thermal reduction reaction. The silicon carbide nanowire microspheres prepared by this method have a uniformly curved morphology and high specific surface area due to the stable growth environment. Based on the various characterizations of silicon carbide nanowire sponges, it can be found that it provides new prospects for filtration, heat preservation, catalysts, and other fields. A schematic diagram of its microsphere formation is shown in Figure 5. Huo [53] sintered silicon nitride nanofiber knitted ceramic foam by an in situ reaction on the basis of a stable silicon foam. This new silicon carbide ceramic foam consists of a three-dimensional nanofiber assembly network with a diameter of 15–100 nm. The preparation method is simple and low-cost, which opens up a new opportunity for catalytic adsorption and high-temperature heat insulation and other fields.



**Figure 5.** Schematic diagram of the formation of nanowire-weaving microspheres (adapted with permission from Ref. [52]. 2019, The American Ceramic Society).

Compared to oxide fibers, nitride fibers displayed higher utilization potential in hightemperature conditions up to 1800 °C. However, due to their worse oxidation resistance, they are not suitable for utilization in oxidation conditions over 1800 °C for long periods. Therefore, it is necessary to choose a non-oxidized fiber covered with oxide to improve its high-temperature utilization, among which carbide is an excellent choice.

#### 2.3. Carbide Fibers

Eq. (1)

SiC Fiber

Silicon carbide ceramic fiber is a typical example of a carbide ceramic fiber. In addition to good mechanical properties, silicon carbide ceramics also have high bending strength, excellent oxidation resistance, good corrosion resistance, high wear resistance, and a low friction coefficient. In addition, the high-temperature mechanical properties of silicon carbide fibers are the best among known ceramic fibers. Silicon carbide fiber, which has many excellent properties, has become an important material in the field of high-end equipment and technology.

In the course of the research, it has been found that it is very challenging to produce high-performance materials that can be put to use. Most are hindered by chemical corrosion, low thermal stability, and complex preparation processes. Liu [54] reported a novel free-standing hollow SiC fiber pad, which is flexible and acid- and alkali-resistant. The preparation process is divided into three steps: preparation of a shell fiber, curing, and pyrolysis. The multilayer scattering mechanism of n-doped hollow silicon carbide fibers is shown in Figure 6. The obtained hollow structure has a cavity wall thickness of 1.5  $\mu$ m. After analyzing the morphology, composition, etc., of this hollow SiC fiber, it was found that it has the advantages of good flexibility, high thermal stability, corrosion resistance, a light weight, and low thermal conductivity. These excellent characteristics mean that the SiC fiber pad has very important applications in high-temperature thermal insulation, high-temperature catalysis, and other fields.

Li [55] successfully prepared microscale silicon carbide fiber pads via the blowing of an oil-in-water precursor emulsion, pre-oxidation, and high-temperature calcination. A schematic diagram of the process and formation mechanism is shown in Figure 7. This ultra-fine silicon carbide has excellent thermal stability and good semiconductor properties, which also indicate its considerable application potential in wave absorption, electronic semiconductors, energy storage, and so on.

Because of the coverage of  $SiO_2$ , the SiC fiber showed excellent performance compared to the oxide fiber and the nitride fiber, providing a new strategy to design the microstruc-



tures of fibers that can withstand high-temperature and oxidation conditions for long-term service.

**Figure 6.** Schematic illustration of integration of specific multilayer scattering mechanisms in the as-obtained N-doped hollow SiC fibers (adapted with permission from Ref. [54]. 2017, Royal Society of Chemistry).



**Figure 7.** Schematic of overall procedure and the formation mechanism of SiC fibers via emulsion–blow spinning (adapted with permission from Ref. [55]. 2017, Transactions of the Indian Ceramic Society).

#### 2.4. Other Fibers

There are many types of ceramic fibers, and the above are only a few of the more widely used ceramic-based fibers. Other materials, such as titanium dioxide (TiO<sub>2</sub>) [56–58] and zirconium carbide (ZrC) [59], are also excellent ceramic-based fiber materials and have good prospects in various application fields.

#### 3. Ceramic Aerogels

Aerogels are novel materials that have emerged in insulation research in recent years with nanoscale voids. A material becomes an aerogel when the gel is stripped of most of its solvent, or when the medium filling the gel space is a gas and the exterior is solid. Common methods for the synthesis of aerogels include the sol–gel method, sacrificial template method, solution spinning method, chemical vapor deposition method, etc. Figure 8 is a schematic diagram of another method of electrospinning to directly prepare ceramic aerogels [60]. However, there are two limitations in producing aerogels. One is the temperature, which is the reason that ceramic aerogels are very popular. The other is the inhibition of crystal transformation, which requires researchers to apply different treatments during the process. The main research directions of high-temperature aerogels are oxide aerogels, carbide aerogels, nitride aerogels, carbon aerogels, etc.



**Figure 8.** Illustration of 3D reaction electrospinning for direct fabrication of ceramic nanofibrous aerogels. (**a**) shows the 3D reaction electrospinning process. (**b**) shows the highly reactive colloidal particles forming highly crosslinked and robust skeletons via condensation and jet solidification. (**c**) shows the distance between the protonated oxygen atom of  $\equiv$ M-OH and the metal atom increase, creating a good leaving group (adapted with permission from Ref. [60]. 2022, The Authors).

#### 3.1. Oxide Aerogels

Common oxide aerogels are silicon oxide, aluminum oxide, and boron nitride. The classification depends on their chemical composition. Oxide aerogels can withstand very high temperatures and equally can exhibit defects such as inherent brittleness.

## 3.1.1. Al<sub>2</sub>O<sub>3</sub> Aerogel

The  $Al_2O_3$  aerogel has a wide range of applications in thermal insulation, sound absorption, and catalysis due to its unique nanoscale three-dimensional network skeleton structure, ultra-low density, high porosity, and low thermal conductivity.

Zu [61] used the acetone–aniline in situ water generation method in combination to prepare an ultra-high-temperature-resistant strong alumina aerogel, with its brittleness greatly improved. The morphology and appearance of the formed alumina aerogel are shown in Figure 9. This alumina aerogel could not only withstand a high temperature of 1300 °C but also possessed improved mechanical strength. Sun [62] prepared composite high-temperature thermal insulation tiles using a mullite fiber-toughened  $Al_2O_3$  aerogel, which enhanced the mechanical properties of the  $Al_2O_3$  aerogel while maintaining good high-temperature thermal insulation properties.

In addition, doping with some additives (rare earth oxides, SiO<sub>2</sub>, and other oxides) can increase the sintering temperature of Al<sub>2</sub>O<sub>3</sub> aerogels. Liu [63] prepared Al<sub>2</sub>O<sub>3</sub>-doped silica (SiO<sub>2</sub>) composite gels by the sol–gel method with a co-prepolymer, and, after freezedrying and high-temperature heat treatment, Al<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub> nanogels with a stable structure under high-temperature conditions were obtained. When the Al/Si molar ratio was 5:1, the aerogel maintained a large specific surface area (185 m<sup>2</sup>/g) and a uniformly distributed mesoporous structure (pore size of 6~8 nm) at 950 °C, whose infrared radiation (IR) reflectivity was as high as 40% in the wavelength range of 3~6 µm, effectively reducing the heat loss caused by IR thermal radiation. Zhou [64] prepared a Y<sub>2</sub>O<sub>3</sub>-Al<sub>2</sub>O<sub>3</sub> aerogel by adding Al<sub>2</sub>O<sub>3</sub>·6H<sub>2</sub>O to YCl<sub>3</sub>·6H<sub>2</sub>O, and the specific surface area still reached 380–400 m<sup>2</sup>·g<sup>-1</sup> after heat treatment at 1000 °C, indicating that the incorporation of Y<sub>2</sub>O<sub>3</sub> improved the high-temperature resistance and thermal stability of Al<sub>2</sub>O<sub>3</sub>.



**Figure 9.** Morphology and appearance of alumina aerogel (adapted with permission from Ref. [61]. 2013, American Chemical Society).

Electrostatic spinning technology can reduce the diameter of fibers to the nanometer level. The fibers obtained by this method possess excellent mechanical properties, electromagnetism, diverse chemical compositions, high specific areas, and other characteristics. Although the performance of alumina nanofibers is superior to that of alumina fibers, their mechanical properties still need to be improved. Li [18] doped SiO<sub>2</sub> in the preparation of alumina nanofibers, and the structural surface of the fibers was dense, with no pores and cracks. Moreover, it possessed excellent flexibility and mechanical strength. Corrias [65] utilized a sol–gel process to produce nanocrystalline  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and FeCo-Al<sub>2</sub>O<sub>3</sub> nanocomposite aerogels constituted of FeCo alloy nanoparticles dispersed in the  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> nanocrystalline matrix. This indicated that calcination at elevated temperatures will give rise to  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> that is stable up to 1000 °C.

Zhang [66] synthesized a novel type of ultra-high-temperature insulating and strong ceramic nanorod aerogel (CNRA) by using alumina nanorods as the basic manufacturing unit and a small amount of silica solvent as the binder. The CNRAs obtained have superior properties, such as ultra-high heat resistance, low thermal conductivity, and high mechanical strength, as shown in Figure 10. Therefore, they have great advantages in extremely harsh environments. Mei [67] selected the nonionic surfactant polyethylene glycol as the pore size regulator of an alumina aerogel and regulated the pore size distribution of alumina by changing the molecular weight and additional amount of polyethylene glycol in the sol–gel process.



**Figure 10.** Schematic illustration of the structural strength and thermal insulation performance of CNRAs (adapted with permission from Ref. [66]. 2021, American Chemical Society).

Ren [68] constructed a super-elastic silicon carbide aerogel using chemical vapor deposition and layer-by-layer self-assembly, as shown in Figure 11. This aerogel consists of a three-dimensional porous structure, and it possesses excellent particulate matter removal efficiency and high absorption capacity. Zou [69] synthesized alumina-based aerogels by acetone–aniline in situ aqueous formation, focusing on the effect of the modification temperature on the alumina-based aerogels. It was found that the morphology of the modified alumina-based aerogel will change from a network structure of needle-like particles to a larger sheet particle structure, and this structural transformation will cause the alumina-based aerogel to have better heat resistance. The thermal stability of the aerogel increases with the increase in the modification temperature. In addition, the modification can also inhibit the phase transition of alumina-based aerogels to  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>. Studies have also found that mullite felt and titanium dioxide can be added to alumina-based aerogels for further enhancement.



**Figure 11.** Suggested growth mechanism of SCS-NAs bulk-assembled by crosslinked SiC/SiOx core-shell nanofibers (adapted with permission from Ref. [68]. 2019, American Chemical Society).

#### 3.1.2. ZrO<sub>2</sub> Aerogel

Zirconia aerogel has drawn attention because of its high-temperature resistance up to 2200 °C. Meanwhile, researchers have found that doping with hybrid atoms or introducing nanofiber building blocks can further improve the performance.

Liu [70] prepared ZrO<sub>2</sub> fibers with nano-diameters and abundant hydroxyl groups by using polyacetylacetonate zirconium as a precursor and compounded them with a ZrO<sub>2</sub> aerogel to obtain high-performance fiber-composite ZrO<sub>2</sub> aerogel heat mats, whose thermal conductivity was 0.026, 0.037, and 0.058 W·m<sup>-1</sup>·K<sup>-1</sup> at 600, 800, and 1000 °C, respectively. Rosas [71] prepared ZrO<sub>2</sub> nanofibers by electrospinning. For calcination at different temperatures, the surface of the fiber appears granular and exhibits a lower BET area with the increase in temperature. In addition, the temperature increase also leads to a lower steady-state conversion value for the ZrO<sub>2</sub> fiber catalyst, where the semicrystalline tetragonal structure appears at 400 °C, and the positive monoclinic structure changes at 600 °C, but it leads to the growth of the monoclinic crystal size over 800 °C.

Chen [72] prepared a novel flexible zirconia nanofiber (ZNF) membrane by combining electrospinning and sol–gel methods, as shown in Figure 12. The flexibility and mechanical properties of ZNF membranes could be controlled by adjusting the grain phase and size of zirconia fibers. This flexible ZNF membrane has a high degree of filtration and excellent corrosion in acid and alkali solutions, which indicates its promising application prospects in the microfiltration of wastewater. Koo [73] reported a type of electrospun yttrium-stabilized zirconia (YSZ) nanofiber. The diameter of the electrospun YSZ nanofibers could be precisely controlled within the range of 200–900 nm by changing the viscosity of the precursor solution. The prepared YSZ had a wide, porous nanocrystalline structure with a high aspect ratio and high specific surface area.



**Figure 12.** FE-SEM images of (a) ZNF1@600, (b) ZNF1@800, (c) ZNF1@1000, and (d) ZNF0 membranes. The insets are the optical images of the corresponding membranes (adapted with permission from Ref. [72]. 2014, Royal Society of Chemistry).

#### 3.1.3. SiO<sub>2</sub> Aerogel

The SiO<sub>2</sub> aerogel [74,75] possesses very low thermal conductivity (0.01 W·m<sup>-1</sup>·K<sup>-1</sup>) and excellent thermal insulation [76] properties, as a promising material to replace traditional insulation materials. However, silica aerogels exhibit inherent brittleness and poor mechanical strength and flexibility. Therefore, research has focused on resolving the abovementioned weaknesses [77]. It was found that a fibrillated aerogel could improve these defects, but fiber reinforcement will lead to the deterioration of its thermal insulation properties and density, and the compatibility of the fiber–silica interface represents a new problem that needs to be solved. Lee [78] prepared submicron-level composite fibers of SiO<sub>2</sub>/TiO<sub>2</sub> with various components by calcination via sol–gel and electrospinning technology. It was noted that no gelling agents or binders were used in the preparation process, and the maximum amount of TiO<sub>2</sub> for suitable fiber formation was around 50 moles. Moreover, the calcination temperature and TiO<sub>2</sub> content greatly affected the surface morphology and crystallization behavior of the electrospinning fibers.

Tang [79] proposed the introduction of silica nanowires into the silica matrix as a secondary phase, which successfully solved this problem and synthesized a new silica nanowire–silica aerogel with excellent thermal insulation and mechanical properties. The data analysis clearly showed that the introduction of silica nanowires could improve both the performance and the mechanical properties of the composites with the increase in nanowires. Calisir [80] compared the nanofibrous silica mats produced via solution blowing and centrifugal spinning. It was found that samples fabricated by centrifugal spinning showed a more homogeneous and defect-free fibrous structure, with a higher average fiber diameter than in those obtained from solution-blowing. However, the solution blowing provided a more compact SiO<sub>2</sub> fibrous mat with a smaller pore size, which was suited to filtration and separation applications. Moreover, the fluffier centrifugal spinning SiO2 outputs could be advantageously adopted for thermal insulation applications.

Wang [81] integrated flexible electroamplified silicon nanofibers with rubber-like Si-O-Si bonding networks to prepare biomimetic nanofiber (SNF) aerogels with superelasticity. The prepared SNF aerogel possessed ultra-low density (>0.25 mg·cm<sup>-3</sup>), high porosity, temperature-invariant superelasticity (up to 1100 °C), and strong fatigue resistance. In addition, it also had fire resistance and ultra-low thermal conductivity for the ceramic itself of 0.024 W·m<sup>-1</sup>·K<sup>-1</sup>. It is a promising thermal insulation material, whose thermal



insulation performance is shown in Figure 13, and it opens up additional possibilities for thermal insulation materials' design and application.

**Figure 13.** (a) Optical image of large-sized SNF aerogels in different shapes for thermal insulation applications. (b) Thermal conductivity of SNF aerogels versus testing temperature. (c) Comparison of the thermal conductivity and maximum working temperatures of SNF aerogels and other reported cellular materials. (d) Ultralight and fire-resistant qualities of an SNF aerogel. (e) Thermal insulation capacity of an SNF aerogel. (f) Optical image of a tubular insulated SNF aerogel. (g) Infrared images of the thermal insulation process of a tubular SNF aerogel. (h) The time-dependent temperature profile of the inner and outer walls of the aerogel (adapted with permission from Ref. [81]. 2020, WILEY-VCH Verlag GmbH Co. KGaA).

Mi [82] reported a novel method for the preparation of three-dimensional silica fiber sponges. In the course of the study, it was found that three-dimensional fibers were easily collected on the aluminum foil using a properly aged tetraethyl orthosilicate (TEOS)/PVA solution. The prepared silica sponge has a continuous fiber structure, low bulk density (16 mg·cm<sup>-3</sup>), high surface area ( $6.5 \text{ m}^2 \cdot \text{g}^{-1}$ ), heat resistance, etc. Because of these excellent properties, the sponge can be used in various fields. Huang [83] proposed a silica nano-aerogel (SNFA) prepared by using electrospun silica nanofibers as a matrix and silica sol as a high-temperature nano-glue. This method addresses the disadvantage of the inherent brittleness of aerogels. The prepared silica aerogel exhibits low density, high-temperature resistance from flexibility to rigidity, adjustable mechanical properties, and so on.

Tepekiran [84] prepared silica-based nanofibers via centrifugal spinning and subsequent calcination. It was indicated that the 15 wt%TEOS/PVP sample possessed the highest flexibility and filtration efficacy among all the silica nanofibers. The average fiber diameter of the optimized web was found to be the lowest, around 521  $\pm$  308 nm, which resulted in enhanced filtration efficiency of around 75.89%.

Similar to oxide fibers, oxide aerogels also exhibit potential for applications in hightemperature oxidized conditions, especially due to their excellent thermal insulation performance. However, their shortcomings in the grain-coarsening process over 1300 °C still limit their utilization. Therefore, the goal has not changed as researchers are still searching for more efficient materials for higher-temperature utilization, such as aerogels of carbide or nitride.

# 3.2. Nitride Aerogels

Nitrides, with good thermal stability, chemical inertness, and a low coefficient of thermal expansion, have been used to synthesize aerogels with excellent high-temperature thermal and mechanical stability.

#### 3.2.1. BN Aerogels

BN has a structure similar to graphite crystals and has four different variants: hexagonal boron nitride (*h*-BN), rhombic boron nitride (*r*-BN), cubic boron nitride (*c*-BN), and fibrillated boron nitride (*w*-BN). BN fibers themselves have a low dielectric constant, high radiation absorption, high thermal conductivity, good mechanical properties, etc., and can be widely used in optical, thermal, and electrical applications, and so on. Moreover, porous boron nitride nanomaterials have the advantages of low density, a large specific surface area, superior mechanical properties, and good chemical stability. Boron nitride (BN) aerogel is a three-dimensional nanoporous material with a high specific surface area, large porosity, and low density. At the same time, it inherits good insulation, oxidation resistance, thermal stability, and chemical stability from conventional BN.

When combined with other different forms of aerogels, *h*-BN can improve the thermal and mechanical properties. Wang [85] prepared hexagonal boron nitride (*h*-BN)/TEMPOoxidized nanocellulose (TOCNF) aerogels with different pore structures, and then compounded TOCNF to prepare *h*-BN/TOCNF composite films. The results showed that when the solid content ratio of *h*-BN to TOCNF was 3:1, TOCNF could play a good role in dispersing *h*-BN, and the thermal conductivity of the composite film was the most efficient. The strength of the composite film was the best when the solid content ratio of *h*-BN and TOCNF was 1:1, the elongation at break was approximately 13%, and the tensile strength was 24.8 MPa, indicating its excellent mechanical properties.

It is well known that ceramic aerogels possess poor thermal stability and will degrade under thermal shock. On this basis, Xu [86] synthesized hexagonal boron nitride aerogels (hBNAGs) using a specially designed 3D graphene aerogel membrane plate. Its characterization is shown in Figure 14. The hBNAG has a double-paned metastructure hyperbolic structure with a negative Poisson's ratio (-0.25) and a negative thermal expansion coefficient ( $-1.8 \times 10^{-6}$  per °C). The hBNAGs synthesized by this method exhibit ultra-low density ( $0.1 \text{ mg}^{-1} \cdot \text{cm}^3$ ), superelasticity (up to 95%), thermal superinsulation ( $0.0024 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$  in vacuum, and  $0.0020 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$  in air), and thermal stability under sharp thermal shocks (~275 °C/s) and long-term high-temperature exposure (900 °C in air and 1400 °C in vacuum).

Lu [87] proposed an effective "growth-space assisted" method for the large-scale production of elastic silicon carbide nanowire aerogels (SiC NWAGs) with controllable density and shape, in view of the shortcomings of the complex and low-yield synthesis process of elastic ceramic aerogels. The SiC NWAGs obtained by this method show excellent deformability, processability, high-temperature stability, and fire resistance, etc., addressing the shortcomings of the high brittleness and poor high-temperature stability of traditional aerogels very well. This also expands the applications of aerogels in harsh environments and complex compositions.



**Figure 14.** Material characterization of *h*-BN aerogels. (**A**) An optical image showing an *h*-BN aerogel sample resting on the stamen of a flower. All tests were performed on ceramic aerogels with a density of 5 mg·cm<sup>-3</sup>, unless otherwise noted. (**B**) The lightest *h*-BN aerogel sample compared with other ultralight materials. The superscript numbers indicate the corresponding referenced work. (**C**) SEM image of *h*-BN aerogel. (**D**) SEM images of the double-pane wall structure of *h*-BN aerogel. Scale bars, 20 nm (adapted with permission from Ref. [86]. 2019, The American Association for the Advancement of Science).

Yang [88] prepared BN/SiOC aerogels by the sol–gel method, and the mechanical properties of SiOC aerogels were improved by adding BN particles to increase the density of the composite. The aerogel exhibited high compressive strength and high heat resistance at 1300 °C. Aravind [89] reported a porous silicon carbide oxide glass synthesized with a highly porous ambiguity material. The synthesized ambiguous gel had a very high surface area and pore volume, and the study found that the porosity remained unchanged during the pyrolysis of its ambiguous gel in an inert gas. Zeng [90] reported a simple cryocasting process (a method for the controlled freezing of an inorganic suspension in water) to prepare a boron nitride nanosheet (BNNS) aerogel, which had an ordered and anisotropic microstructure, as well as anisotropic superelasticity, high compressive strength, and great absorption capacity. It has wide application prospects in the fields of catalysts, environmental remediation, and energy absorption.

Li [91] developed a novel boron nitride nanoribbon aerogel by the high-temperature amination reaction of a melamine diborate precursor, as shown in Figure 15. Its extremely unique structure provides it with excellent temperature invariance, outstanding compressive elasticity, bending elasticity, torsional elasticity, cutting resistance, and recovery ability. In the range from liquid nitrogen temperatures (-196 °C) to over 1000 °C, these BN nanoribbon aerogels maintain superior mechanical superflexibility.



**Figure 15.** (a) The atomic structure of an M·2B cell in top (**left**) and side (**right**) view. The hydrogen bonds are presented with dashed lines. (b) Simulation setup of a single layer of M·2B embedded in solvent (left), snapshots of M·2B after equilibration in pure TBA (top) and pure water (bottom) (right). (c) Simulation of two crossed M·2B layers. (d) The side view of two crossed M·2B layers, linked to each other via the H-bond at the edge of the layer. (e) Simulated illustration of the H-bond converted to a BN bond at the edge of the layer (boric acid (B) and melamine (M)) (adapted with permission from Ref. [91]. 2019, WILEY-VCH Verlag GmbH Co. KGaA).

#### 3.2.2. Si<sub>3</sub>N<sub>4</sub> Aerogels

Silicon nitride ( $Si_3N_4$ )-based composites, with good wave transparency and acceptable high-temperature mechanical properties, are considered excellent candidates for aircraft radomes.

Kong [92] prepared Si<sub>3</sub>N<sub>4</sub> aerogels by the carbothermal reduction method with flowing nitrogen in order to break through the limitation of silica aerogels to 600 °C and to improve their stability. Ultra-light  $\alpha$ -Si<sub>3</sub>N<sub>4</sub> aerogels with controllable density can be prepared by a simple heat treatment process. They have excellent properties, such as fire resistance and elastic compressibility at 1200 °C, being a promising material for thermal insulation applications. Su [93] prepared ultra-light  $\alpha$ -Si<sub>3</sub>N<sub>4</sub> nano-felt aerogels (NBAs) with tunable density (1.8–9.6 mg·cm<sup>-3</sup>).  $\alpha$ -Si<sub>3</sub>N<sub>4</sub> NBA has ultra-high elastic compressibility, fire resistance, thermal insulation, and an ultra-low dielectric constant, as exhibited in Figure 16. With the use of silica sol and carbon black as raw materials, Zhang [52] synthesized Si<sub>3</sub>N<sub>4</sub> nanowire woven microspheres via a simple and efficient method. The Si<sub>3</sub>N<sub>4</sub> nanowire microspheres formed by this method have a uniformly curved morphology and high specific surface area and are very stable in a gas–solid growth environment.



**Figure 16.** Thermal conductivity versus the maximum working temperature of thermally insulating materials (adapted with permission from Ref. [93]. 2019, American Chemical Society).

Compared to oxide aerogel, nitride aerogel also displayed similarly higher utilization potential as nitride fibers in high-temperature conditions up to 1800 °C. However, there still exist the disadvantages of worse oxidation resistance and a lack of suitable utilization in high-temperature oxidation conditions for long periods. Therefore, it is necessary to choose a non-oxide aerogel with oxidation resistance for higher-temperature utilization, such as a carbide aerogel.

#### 3.3. Carbide Aerogels

Carbide itself has excellent high-temperature resistance, wear resistance, corrosion resistance, a high melting point, high hardness, good electrical conductivity, and good mechanical properties. Carbide aerogel is also the most promising material to resist temperatures above 1200 °C and is widely used in the aerospace field. Carbide aerogels have higher temperature resistance compared to traditional oxide aerogels, and their types are also very abundant. The preparation of carbide aerogels is complicated by the oxidation reaction of carbon, commonly known as SiC aerogels.

#### SiC Aerogel

Silicon carbide aerogel is the most popular carbide aerogel, which is usually obtained by mixing silica aerogel with carbon in a reduction reaction. SiC aerogels have excellent high-temperature resistance and stability and can be successfully applied above 1300 °C. The chemical and high-temperature stability and multifunctional properties of SiC aerogels indicate their potential application in electromagnetic absorption and thermal insulation in harsh environments.

Mahalingam [94] used a facile method consisting of simultaneous centrifugal spinning and solution blowing to mass-produce SiOC fibers from preceramic polymers. It was indicated that the fiber diameter and morphology were influenced by the rotating speed, working pressure, and polymer concentration, where non-porous fibers could be obtained by using a low-volatility binary solvent system and after pyrolysis at a high temperature. This fabrication method, based on pressurized gyration, showed great promise in the mass production of ceramic fiber bundles.

Liang [95] used an eggplant-derived method to fabricate a green and convenient SiC aerogel by using the naturally porous structure of eggplant decorated with in-situ-grown SiC nanowires. The SiC aerogel thus obtained has low density (Figure 17a), mechanical stability, thermal stability, and thermal insulation properties (Figure 17b). Su [96] successfully prepared silicon carbide aerogels with entangled nanowires and highly porous



structures by nucleating silicon carbide nanowires through a chemical deposition reaction on graphite caps.

**Figure 17.** (**a**) Thermal conductivity of carbon, SiC/C, and SiC aerogels and (**b**) heat resistance properties of the SiC aerogels (adapted with permission from Ref. [95]. 2019, Elsevier).

The porosity of single-porous silicon carbide is generally considered to be high at approximately 30%V/V, and its preparation process is usually more complicated. On this basis, Leventis [97] crosslinked the 3D sol–silicon nanostructure prepared by carbothermal reduction with polyacrylonitrile (PAN) synthesized by a simple mixed monomer to obtain silicon carbide with porosity of 70%V/V. The morphology of the silicon carbide network is constant at the processing temperature of 1300~1600 °C. Samples processed at 1200 °C are mesoporous and amorphous.

Due to their excellent heat resistance up to a maximum operating temperature of 1200 °C, oxidation resistance, and high strength, SiC fibers are also employed as high-temperature-resistant and reinforcing materials for a wide range of promising applications. Chen [98] experimentally showed that adjustable three-dimensional porous silicon carbide nanowire scaffolds can be prepared by adjusting the solid loading content in the mesh melamine foam template. These SiC nanowire scaffolds not only have high strength and good fire-stopping performance but also could efficiently absorb oil or act as catalysts, etc.

Song [99] prepared a multifunctional silicon carbide nanofiber aerogel (SiC NFAS) spring using a simple thermochemical process, as shown in Figure 18. This SiC NFAS has ultra-low density, ultra-low thermal conductivity, as well as great mechanical properties, showing strong superelasticity and cyclic fatigue resistance at temperatures ranging from low to high.



**Figure 18.** Schematic illustration of SiC NFAS fabrication and corresponding atomic structure model (adapted with permission from Ref. [99]. 2022, The Authors).

Owing to the surface oxidation of SiC, the covered  $SiO_2$  protected the inner SiC from further oxidation, which showed the excellent performance of the SiC aerogel compared to the oxide aerogel or that of nitride, and provided a new approach to design the microstructure of carbide aerogel to improve its oxidation resistance under high-temperature oxidation conditions, for future utilization in long-term service. On basis of the above-introduced ceramic-based materials, both the nanofibers and aerogels show their application potential in high-temperature harsh conditions. A comparison of the material and preparation methods of the different nanofibers and aerogels is shown in Table 2, which offers a reference for researchers regarding the main flexible nanomaterials with a series of materials, their fabrication processes, as well as their properties. Because of properties such as low density, porosity, a large surface area, low thermal conductivity, high-temperature resistance, and chemical stability, they show potential for the application fields of thermal insulation, air filtration, water treatment, sound absorption, electromagnetic wave insulation, battery separators, and catalytic applications.

Ceramics

Precursors

Aluminum acetate

Calcination Conditions	Methods	Products	References
20 °C·min <sup>-1</sup> , 1000 °C for 2 h, in air	Electrospinning	Nanofiber	[27]
1 °C·min <sup>-1</sup> , 600 °C for 2 h, 5 °C·min <sup>-1</sup> , 700−1000 °C for 2 h, in air	Electrospinning	Nanofiber	[28]
$4^{\circ}C{\cdot}min^{-1}$ , 600–1100 $^{\circ}C$ for 2 h, in air	Solution blow spinning	Nanofiber	[31]
$4^{\circ}C{\cdot}min^{-1}$ , 400–1200 $^{\circ}C$ for 4 h, in air	Centrifugal spinning	Nanofiber	[23]
$5 ^{\circ}\text{C}\cdot\text{min}^{-1}$ , 900 $^{\circ}\text{C}$ for 1 h, in air	Electrospinning	Nanofiber	[100]
$3\ ^\circ C \cdot min^{-1}$ , 650 $^\circ C/1200\ ^\circ C$ for 3 h, in air	Centrifugal spinning	Nanofiber	[33]
$2 \degree C \cdot min^{-1}$ , 800–1500 °C, in air	Electrospinning	Aerogel	[101]
800-1400 °C for 2 h in air	Floctrospipping	Nanofibor	[35]

Table 2. Materials and calcination condition

Solvents

Ethanol

A1-O-	Aluminum powder	H <sub>2</sub> O	$1  ^{\circ}\text{C-min}^{-1}$ , 600 $ ^{\circ}\text{C}$ for 2 h, 5 $ ^{\circ}\text{C-min}^{-1}$ , 700–1000 $ ^{\circ}\text{C}$ for 2 h, in air	Electrospinning	Nanofiber	[28]
711203	AlCl <sub>3</sub> ·6H <sub>2</sub> O, aluminum powder	H <sub>2</sub> O	$4\ ^{\circ}C\cdot min^{-1}$ , 600–1100 $^{\circ}C$ for 2 h, in air	Solution blow spinning	Nanofiber	[31]
	AlCl <sub>3</sub> ·6H <sub>2</sub> O, aluminum powder	H <sub>2</sub> O	$4\ ^\circ C \cdot min^{-1}$ , 400–1200 $^\circ C$ for 4 h, in air	Centrifugal spinning	Nanofiber	[23]
	Al(NO <sub>3</sub> ) <sub>3</sub> ·9H <sub>2</sub> O, aluminum isopropoxide	H <sub>2</sub> O	$5 {}^{\circ}\text{C} \cdot \text{min}^{-1}$ , 900 ${}^{\circ}\text{C}$ for 1 h, in air	Electrospinning	Nanofiber	[100]
	Aluminum isopropoxide	H <sub>2</sub> O	$3~{}^{\circ}\mathrm{C}{\cdot}\mathrm{min}^{-1}$ , 650 ${}^{\circ}\mathrm{C}/1200~{}^{\circ}\mathrm{C}$ for 3 h, in air	Centrifugal spinning	Nanofiber	[33]
	Aluminum trisec-butoxide, polyhydromethylsiloxane	lsopropanol, DMF, ethylacetoacetate	$2\ ^\circ C \cdot min^{-1}$ , 800–1500 $^\circ C$ , in air	Electrospinning	Aerogel	[101]
Mullite	Aluminium isopropoxide, Al(NO3) 3-9H2O, TEOS	H <sub>2</sub> O, ethanol	800–1400 °C for 2 h, in air	Electrospinning	Nanofiber	[35]
	Aluminum acetate, Colloidal silica	$H_2O$ , ethanol	$5\ ^\circ\mathrm{C\cdot min^{-1}}$ , 800 $^\circ\mathrm{C}$ for 1 h, 800–1200 $^\circ\mathrm{C}$ for 1 h in air	Electrospinning	Nanofiber	[36]
	Aluminum acetate, TEOS	H <sub>2</sub> O, ethanol	5 °C·min <sup>-1</sup> , 800 °C for 1 h, 1000 °C for 1 h, in air	Electrospinning	Nanofiber	[102]
	TEOS, Al(NO <sub>3</sub> ) <sub>3</sub> .9H <sub>2</sub> O	THF	600–1000 °C, in air	Solution blow spinning	Nanofiber	[37]
ZrO <sub>2</sub>	ZrOCl <sub>2</sub> ·8H <sub>2</sub> O	H <sub>2</sub> O	1 °C·min <sup>-1</sup> , 800 °C for 1 h, 5 °C·min <sup>-1</sup> , 1200 °C for 1 h. in air	Centrifugal spinning	Nanofiber	[45]
	Zirconium acetate	Acetic acid	200–1000 °C for 2 h, in air	Electrospinning	Nanofiber	[71]
	ZrOCl <sub>2</sub> ·8H <sub>2</sub> O	H <sub>2</sub> O	600-1300 °C, in air 5 °C, min <sup>-1</sup> 280 °C for 1 h 1 °C, min <sup>-1</sup> 800 °C for	Electrospinning	Nanofiber	[72]
	Zirconium acetate hydroxide	DMF	3 h, in air	Electrospinning	Nanofiber	[73]
	ZrOCl <sub>2</sub> ·8H <sub>2</sub> O	Ethanol, $H_2O$	$2 \degree \text{C} \cdot \text{min}^{-1}$ , 800 °C for 200 min, in air	Solution blow spinning	Nanofiber	[44]
	$ZrOCl_2 \cdot 8H_2O$	$H_2O_2, H_2O$	1.2–3 °C·min <sup>-1</sup> , 1300 °C form 3 h, in steam atmosphere	Centrifugal spinning	Nanofiber	[46]
	Zirconium n-propoxide	Ethanol	$2 \degree C \cdot min^{-1}$ , 800 °C for 200 min, in air	Solution blow spinning	Nanofiber	[47]
	TEOS	H <sub>2</sub> O	$5 ^{\circ}\text{C}\cdot\text{min}^{-1}$ , 600–1200 $^{\circ}\text{C}$ , in air	Electrospinning	Nanofiber (82,109–111), Aerogel (108)	[81,103–106]
<i></i>	TEOS	H <sub>2</sub> O	$5 \circ C \cdot min^{-1}$ , 800 °C for 2 h, in air	Electrospinning	Aerogel	[83]
$S_1O_2$	TEOS	Ethanol	$6 \degree \text{C} \cdot \text{min}^{-1}$ , 550 °C for 1 h, in air	Air-jet spinning	Nanofiber	[107]
	TEOS	Ethanol	$2 ^{\circ}\text{C}\cdot\text{min}^{-1}$ , 300, 600 and 900 $^{\circ}\text{C}$ , in air	Centrifugal spinning	Nanofiber	[84]
	TEOS	Ethanol	$10 ^{\circ}\text{C} \cdot \text{min}^{-1}$ , 250–1000 $^{\circ}\text{C}$ for 3 h, in air	Electrospinning	Nanofiber	[110]
BN	Boric acid, melamine	H <sub>2</sub> O, tertiary butyl alcohol	1200 °C for 3 h in NH <sub>3</sub>	Self-assembly	Aerogel	[91]
	Methyltrimethoxysilane, dimethyldimethoxysilane	Ethanol	$5\ ^{\circ}C\cdot min^{-1}$ , 1500 $^{\circ}C$ for 2 h in $N_{2}$	Chemical vapor deposition	Aerogel	[93]
	Silica sol, carbon black	/	$3 \degree C \cdot min^{-1}$ , 1600 °C for 3 h in N <sub>2</sub>	Chemical vapor deposition	Aerogel	[96]
$Si_3N_4$	PCS	DMF, THF	2 °C·min <sup>-1</sup> , 210 °C for 2 h in air, calcinated at 800 °C for 2 h, 1300 °C for 2 h in Ar	Polymer conversion	Aerogel	[111]
	PCS	Toluene, DMF	Stabilized at 170 °C for 3 h in air, calcinated at 1100–1500 °C in N <sub>2</sub> /calcinated at 1500 °C in Ar	Polymer conversion	Nanofiber	[54]
	Methyltrimethoxysilane, dimethyldimethoxysilane	Ethanol, H <sub>2</sub> O	$5\ ^{\circ}\text{C}\text{-min}^{-1}$ , 1550 $^{\circ}\text{C}$ for 2 h, in Ar	Chemical vapor deposition	Aerogel	[87]
	PCS	Xylene, H <sub>2</sub> O	Stabilized at 200 $^{\circ}\mathrm{C}$ for 10 h in air, calcinated at 1400 $^{\circ}\mathrm{C}$ for 2 h in Ar	Polymer conversion	Nanofiber	[55]
	Polysilocarbonsilane	/	Stabilized at 160–220 °C for 6–8 h in air, calcinated at 1800 °C in Ar	Melt spinning	Nanofiber	[112]

# 4. Applications

# 4.1. Thermal Insulation

Due to their low density, low thermal conductivity, and chemical stability, ceramic fibers have a wide range of applications in the field of thermal insulation [113], especially in aerospace and other fields. Generally, the lower the density, the lower the thermal conductivity and the better the thermal insulation performance of the material will be.

Si [103] combined silica nanofibers with an aluminum–borosilicate matrix to prepare ceramic nanofiber aerogels (CNFAs) with a layered structure and excellent elasticity. CNFAs exhibit good mechanical properties over a wide temperature range, as shown in Figure 19. The density and shape of such CNFAs is adjustable, showing the comprehensive characteristics of fly weight density >0.15 mg·cm<sup>-3</sup>, and they could be stabilized at 1100 °C to provide outstanding fire resistance and thermal insulation.



**Figure 19.** Mechanical properties of the CNFAs over a wide range of temperatures. (**a**–**c**) Storage modulus, loss modulus, and damping ratio of the CNFAs versus angular frequency (0.1 to 10 Hz) at temperatures from -100 °C to 500 °C, with an oscillatory value of 3%. (**d**) Compression and recovery study of the CNFAs after treatment at various temperatures for 30 min. (**e**) XRD patterns of CNFAs after treatment at 1000 °C, 1200 °C, and 1400 °C for 30 min. (**f**) SEM images of CNFAs after treatment at 1200 °C and 1400 °C for 30 min. Compression and recovery process of the CNFAs in the flame of (**g**) an alcohol lamp and (H) a butane blowtorch (adapted with permission from Ref. [103]. OpenAcess).

Jia [114] prepared a SiO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> composite ceramic sponge with temperature invariance and high compressibility in a simple manner, leading to an advanced anisotropic layered ceramic sponge with a density as low as 10 mg·cm<sup>-3</sup>, thermal conductivity as low as 0.034 W·m<sup>-1</sup>·K<sup>-1</sup>, and temperature-invariant compressive elasticity in the temperature range of  $-196\sim1000$  °C, as shown in Figure 20. The various excellent parameters make these ceramic sponges promising in terms of thermal insulation materials. Liu [54] prepared hollow silicon carbide fiber pads doped with nitrogen using sacrificial stencils and electrospinning. The cavity walls of the hollow fibers are thin, with excellent performance in terms of flexibility, acid and alkali resistance, non-combustibility, and high-temperature stability in air or inert gases. This material has thermal conductivity of  $0.026 \pm 0.013 \text{ W} \cdot \text{m}^{-1} \cdot \text{K}^{-1}$  and can be used as a high-temperature heat insulator. Su [96] reported a highly porous SiC nanowire aerogel (NWA) that improved the brittleness of traditional aerogels and the weakness of volume shrinkage at high temperatures. These SiC NWAs are composed of many interwoven nanowires, with extremely low density, large recoverable compressive strain, fire resistance, and high-temperature and oxidation resistance. These properties make it a promising fire-resistant and high-temperature insulation material.



**Figure 20.** Photographs showing the compressibility and fire resistance of the SAC sponges burned with a butane blowlamp or compressibility immersed in liquid N2 (adapted with permission from Ref. [114]. 2020, The Authors).

#### 4.2. Air Filtration

Air pollution is a serious problem worldwide, especially in developing countries, where the pollution is mainly derived from the particulate matter (PM) produced by cars, power plants, chemicals, etc. PM is also known as lung particle matter. It remains in the atmosphere for a long time, spreads across large distances, and contains a large number of harmful substances. In addition, it can easily remain in the bronchi and alveoli, and even enters the blood, which is extremely harmful to health. The effective removal of PM has become an urgent need for contemporary development. Therefore, an efficient filtration material is necessary to address this situation.

Jia [115] prepared an Al<sub>2</sub>O<sub>3</sub>-stabilized ZrO<sub>2</sub> (ASZ) submicron fiber air filter paper with excellent flexibility and stability for use in solution-blowing and calcination technology. The ASZ paper, with an area density of 56 mg·cm<sup>-2</sup>, has filtration efficiency of up to 99.56% at an airflow velocity of 5.4 cm·s<sup>-1</sup> for 15~615 nm NaCl particles. Furthermore, this foldable all-ceramic air filter material could be an important solution for particle removal in high-temperature exhaust gases.

Wang [47] reported a porous three-dimensional sponge composed of yttrium-stabilized  $ZrO_2$  (YSZ) nanofibers with a density of 20 mg·cm<sup>-3</sup> and a light weight. The elasticity of this sponge can be maintained at room temperature and high temperatures, and the filtration efficiency can reach 99.4% at room temperature and 99.97% at high temperatures, as shown



in Figure 21. The YSZ nanosponge, therefore, has promising application prospects, and a practical automotive exhaust filter with filtration efficiency of 98.3% has been assembled.

**Figure 21.** Filtration test at high temperature. (a) Schematic of high-temperature PM filtration measurement setup. (b) Concentration of PM particles before and after filtration at 750 °C. (c) PM0.3–2.5 and PM2.5–10 filtration efficiency at 750 °C. (d) Filtration efficiency of PM particles with different airflow velocities at 750 °C. (e) SEM images of YSZ nanofiber sponge after filtration at 750 °C. (f) PM measurement of automobile exhaust gas with YSZ nanofiber sponge filter. (g) Filtration efficiency for different particle sizes of automobile exhaust gas (adapted with permission from Ref. [47]. 2018, WILEY-VCH Verlag GmbH Co. KGaA).

Li [116] reported a typical structural model of a high-efficiency particulate air filter based on carbon nanotubes. High filtration efficiency can be obtained due to the aggressively high surface of the carbon nanotubes, and a low-pressure drop can be obtained by controlling the thickness and SVF value of the carbon nanofilm. Gradient nanostructures or layered nanostructures not only have higher filtration efficiency, but also have a longer service life and low pressure drop. Zhang [117] developed a novel hydroxyapatite (HAP) nanowire-based inorganic aerogel, shown in Figure 22, which can also be used as a highefficiency air filter for PM2.5, in addition to its advantages of high porosity, ultra-light density, and low thermal conductivity. Compared with organic aerogels, HAP nanowire aerogels have the advantages of environmental protection, low costs, and biocompatibility.



**Figure 22.** (**A**) Different types of breathing masks using the hydrophobic HAP nanowire aerogel filter. (**B**) Digital images of a hermetic chamber filled with highly polluted air with a very high concentration of PM, and a prototype filter model prepared using the hydrophobic HAP nanowire aerogel for air purification for 12 min. (**C**) Digital images of the hydrophobic HAP nanowire aerogel filter used in the prototype filter model after the air purification process for 2 h. (**D**) Digital image of the hydrochloric acid aqueous solution. (**E**) Digital image of the hydrochloric acid aqueous solution after dissolution of the hydrophobic HAP nanowire aerogel filter used for air purification for 2 h (adapted with permission from Ref. [117]. 2018, American Chemical Society).

# 4.3. Water Treatment

Water pollution is also a very serious problem, caused by the discharge of various types of sewage, oil spills, etc. For this reason, it is necessary to develop a system that can absorb and remove organic pollutants from polluted water. Many flexible ceramic fibers can be used as new materials to remove various organic pollutants.

Su [96] reported a highly porous three-dimensional silicon carbide nanowire aerogel (NWA) with recoverable compressibility and excellent high-temperature stability. The SiC NWA has high absorption capacity due to its high porosity and low density, as shown in Figure 23. The hydrophobic SiC NWA can selectively adsorb low-viscosity organic solvents. Moreover, this SiC NWA has elasticity and fire resistance, and could be used to absorb all organic solvents by squeezing NWA and burning it in air for heating. In addition, it was found that extrusion reabsorption does not lead to a decrease in absorption capacity, proving the SiC NWA as a promising material in sewage treatment and medicine

cabinets. Ren [68] prepared SiC nanofiber aerogels (SCS-NAs) with high hydrophobicity using octadecyl trichlorosilane (OTS) as a modifier, which can also absorb oily pollutants and organic solvents by extrusion and combustion, with potential application value in remedying chemical leakages and oil leakages.



**Figure 23.** Oil and organic solvent absorption properties of the SiC NWA. (**a**) Absorption process of kerosene (colored with Sudan III for clear presentation) within 5 s. (**b**) Recyclability of the absorption process. (**c**) Absorption capability of the SiC NWA for various organic liquids. (**d**) Recyclability of the SiC NWA for absorption of kerosene when using the squeeze method. (**e**) Recyclability of the SiC NWA for absorption of ethanol when using the direct combustion method (adapted with permission from Ref. [96]. 2018, American Chemical Society).

BN aerogels are also highly hydrophobic and can adsorb up to 160 times their weight in oil [118]. Xue [119] developed a multifunctional foam for cell networks composed of an interconnected nanotube hexagonal BN (*h*-BN) structure, which can well adsorb various hydrophobic oils in oil-contaminated water, with excellent antiseptic properties against strong acids and alkalis. This demonstrates that the three-dimensional tubular BN cellularnetwork foam (3D-BNF) is a very useful material for cleaning in very harsh environments. According to research statistics, most of the non-oxide ceramic materials are used for water treatment, such as SiC and BN, as mentioned above. For this reason, the application of oxide ceramic fiber as an absorbent material has yet to be studied [6].

#### 4.4. Sound Absorption

Noise pollution has become indispensable in modern life with the rapid development of construction and transportation. It causes great inconvenience and impacts our lives and work. It is therefore crucial to develop materials that can absorb and block noise pollution. Here, ceramic materials can be used as an efficient and suitable sound-absorbing material. Jia [114] prepared SiO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> composite ceramic sponges by a simple process. The layered structure, rough microfiber surface, and fiber vibrations of the SAC foam provide it with excellent sound absorption properties, as shown in Figure 24.



**Figure 24.** Acoustic absorption properties of the SAC sponges. (a) Optical images of the SAC sponges for acoustic absorption property determination. Scale bars, 2 cm. (b) Sound absorption coefficient of the SAC sponges with different thicknesses. (c) Noise reduction coefficient (NRC) of the SAC sponges with different thicknesses. (d) Comparison of the sound absorption properties of our SAC sponges with other sound absorption materials. (e) Schematic showing sound transmission though the SAC sponges (adapted with permission from Ref. [114]. 2020, The Authors).

#### 4.5. Electromagnetic Wave Absorption

With the rapid development of modern technology, electromagnetic pollution is becoming more and more serious, so it is of great significance to develop a material that can absorb electromagnetic waves. According to research, the most common electromagnetic-waveabsorbing material is SiC [120,121] fiber-based material, with an adjustable microstructure and dielectric properties.

Cai [122] prepared a series of highly porous, well-interconnected SiC@C nanowire foams (SCNFs) via a simple glucose solution permeation–carbonization method using silicon carbide nanowire aerogels as raw materials, which improved the shortcomings of the narrow absorption bandwidth and low absorption intensity of electromagnetic waves (EMWs). This SCNF has a density of 108 mg·cm<sup>-3</sup>, covering the entire absorption bandwidth of the X and Ku bands with an intensity of -52.5 dB. Moreover, it is advantageous for applications in extreme environments due to its great hydrophobicity and self-cleaning ability, as a promising new generation of EMW-absorbing materials.

Hou [123] prepared silicon carbide nanofiber pads with and without fiber arrangement by electrospinning and found that the arrangement of nanofibers can further improve the microwave absorption performance, as shown in Figure 25. In addition, the low weight fraction of SiC nanofiber pad silicone resin composites makes them a low-cost absorber. In addition, Hou [124] found that introducing hafnium carbide (HfC) into SiC nanofiber pads can not only improve their flexibility but also improves the dielectric properties and microwave absorption properties.



**Figure 25.** Electromagnetic attenuation mechanism and electromagnetic parameters of SCNWAG and SCNFs. (a) Schematic showing the EMW absorption mechanism of the SCNFs. (b) Real part of permittivity, (c) imaginary part of permittivity, and (d) tangent loss of the different samples (adapted with permission from Ref. [123]. 2017, American Chemical Society).

# 4.6. Battery Separators

Batteries are essential in our lives, among which Li-ion batteries provide efficient power for humans in many applications. However, they easily catch fire due to short circuits, which are mostly related to polymer separators, and they generally degrade immediately under sudden high temperatures. For this reason, it is necessary to develop ceramic fiber separators that withstand high temperatures.

Yan [106] prepared a series of oxide ceramic nanofiber films by sol–gel electrospinning technology, and the separator prepared by this film showed greater than 900% electrolyte absorption and high thermal insulation performance, which can dramatically improve the safety of lithium batteries. Zhao [125] prepared an elastic solid electrolyte to provide fast and continuous ion transport channels by filling the well-arranged  $L_{6.4}L_{a3}Zr_2Al_{0.2}O_{12}(LLZO)$  nanofiber film with a sol–gel electrospinning method, which showed high ionic conductivity of up to  $1.16 \times 10^{-4}$  at 30 °C. The elastic surface exhibited stable Li plating/stripping cycling over 700 h, the maximum in the present research, as shown in Figure 26. Therefore, most of the present research on ceramic fiber battery separators mainly focuses on LLZO materials. However, studies on the application of other materials in battery separators are also carried out to obtain safer and more successful chemical ceramic fiber separators.



**Figure 26.** Battery tests and characterizations. All the battery tests were carried out at RT. (**a**) Typical charge/discharge curves of the solid-state NCA/EACN/Li batteries at 0.2C. (**b**) The capacity and Coulombic efficiency of the NCA/EACN/Li batteries at 0.2C. The solid-state battery could efficiently power light-emitting diodes. (**c**) Impedance measurements for the NCA/EACN/Li battery before cycling. The inset figure is the equivalent circuit of the impedance data. (**d**) The capacity and Coulombic efficiency of the liquid NCA/Li batteries at 0.2C (adapted with permission from Ref. [125]. 2019, Elsevier).

#### 4.7. Catalytic Application

Ceramic fibers have high porosity and a beaten surface area and can be an attractive material for catalysts. Ceramic fibers have two forms in the field of catalysis: one is that the fiber itself has excellent catalytic activity for processing into ceramic flakes for use in the catalytic field, and the other is as a catalyst carrier.

Zhang [126] synthesized a composite nanofiber cathode composed of CeO<sub>2</sub> particles and La<sub>0.6</sub>Sr<sub>0.4</sub>Co<sub>0.2</sub>Fe<sub>0.8</sub>O3– $\delta$  (LSCF) by coaxial electrospinning, which had significant ORR activity and durability. At 700 °C, the polarization resistance of this cathode is approximately one fifth of that of LSCF powder. This LSCF/CeO<sub>2</sub> composite cathode possesses stability in anode separation cells.

When ceramic fibers are used as catalyst carriers, there are two methods by which to prepare the materials. One is to add a catalyst or catalyst precursor directly to the spinning solution, and the other is to load the catalyst onto the surface of the ceramic fiber after spinning calcination. Wang [127] synthesized Cu-Al<sub>2</sub>O<sub>3</sub> fiber membranes by electrospinning technology, which exhibited high Fenton catalytic activity under neutral conditions. Cheng [128] synthesized ZnO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> nanofibers by electrospinning and calcination. Silver nanofibers are modified on the surfaces of nanofibers by reduction, and Ag/ZnO/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> nanofibers exhibit the efficient catalytic degradation of methyl orange under ultraviolet irradiation.

#### 5. Summary and Prospects

Developments and applications of ceramic-based nanomaterials such as nanofiber materials and aerogels have achieved tremendous progress in the past few years, including several types of oxides, such as  $Al_2O_3$ ,  $ZrO_2$ , and glass; carbide for SiC; nitride for BN and  $Si_3N_4$ ; and other ceramic-based nanofibers. We, therefore, have presented a summary of these ceramic-based nanomaterials in terms of their types and their applications in the fields of thermal insulation, air filtration, water treatment, sound absorption, electro-

magnetic wave insulation, battery separators, catalysis, etc. Among them, we focused on summarizing the applications of ceramic-based nanomaterials for thermal insulation. In addition, in terms of improving the application of ceramic-based nanomaterials for thermal insulation, especially for high-temperature thermal insulation, we offer several pieces of advice on the development directions of ceramic-based nanofibers.

(1) To improve the efficiency of the thermal insulation of ceramic-based materials, ultra-fine nanofibers are essential. Therefore, it is necessary to employ methods such as centrifugal spinning, electrospinning, solution blow spinning, and self-assembly for the preparation of nanofibers.

(2) Polymers are usually used as thickeners in the spinning solution, which is removed after calcination and results in a fiber diameter reduction, porous surface formation, and decreased mechanical capacity. This phenomenon is eliminated by improving the solution by enhancing the solid content ratio as much as possible and avoiding polymer utilization.

(3) For some single-component ceramic-based nanofibers, their performance in terms of high-temperature resistance will decrease because of the nanometer effect. It is, therefore, necessary to add some other components to improve them, such as adding some yttrium oxide to improve the performance of  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> nanofibers.

(4) The required quantities of nanofiber materials in actual applications for thermal insulation and sound absorption are huge, but most of the spinning methods' efficiency meets the demands. Among all sol–gel routes, the solution blow spinning method possesses the highest efficiency for the preparation of nanofiber products as films and sponges, which should be given more research attention.

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#### References

- 1. Huang, Z. (Translator); Ceramic. In Ceramics (Part II); China Light Industry Press Ltd.: Beijing, China, 1989; pp. 207–214.
- 2. Xi, G. Thermophysical Properties of Inorganic Materials; Shanghai Science and Technology Press: Shanghai, China, 1981.
- 3. Okada, K.; Kato, H.; Nakajim, K. Preparation of silicon carbide fiber from activated carbon fiber and gaseous silicon monoxide. *J. Am. Ceram. Soc.* **1994**, 77, 1691–1693. [CrossRef]
- Zhou, X.; Li, Z. Reflections on the current situation and development strategy of China's specialty paper industry. *China Paper*. 2003, 1, 14–16.
- 5. Su, W. The present and future of ceramic fibers. Inorg. Salt Ind. 1982, 6, 47.
- Jia, C.; Xu, Z.; Luo, D.; Xiang, H.; Zhu, M. Flexible ceramic fibers: Recent development in preparation and application. *Adv. Fiber Mater.* 2022, *4*, 573–603. [CrossRef]
- 7. Johnson, D. Nextel-312 ceramic fiber from 3M. J. Coat. Fabr. 1981, 10, 282–296. [CrossRef]
- Hildmarm, B.O.; Schneider, H.; Schmücker, M. High temperature behavior of polycrystalline alumino-silicate fibres with mullite bulk composition: II, kinetics of mullite formation. J. Eur. Ceram. Soc. 1996, 16, 287–292. [CrossRef]
- Kaya, C.; Butler, E.G.; Lewis, M.H. Microstructurally controlled mullite ceramics produced from monophasic and diphasic sol-derived pastes using extrusion. J. Mater. Sci. 2003, 38, 767–777. [CrossRef]
- 10. Clauß, B. Fibers for Ceramic Matrix Composites; Wiley-VCH Verlag GmbH & Co. KGaA: Weinheim, Germany, 2008. [CrossRef]

- 11. Ohira, H.; Shiga, H.; Isamil, M.G.; Nakai, Z.; Akiba, T.; Yasuda, E. Compressive creep of mullite ceramics. *J. Mater. Sci. Lett.* **1991**, 10, 847–849. [CrossRef]
- 12. Wilson, D.M.; Visser, L.R. Nextel TM 650 ceramic oxide fiber: New alumilla-based fiber for high temperature composite reinforcement. *Ceram. Eng. Sci Proc.* 2000, 21, 363–373. [CrossRef]
- Poulon-Quintin, A.; Berger, M.H.; Bunsell, A.R. Mechanical and micro-structural characterization of Nextel-650 alumina-zirconia fibres. J. Eur. Ceram. Soc. 2004, 24, 2769–2783. [CrossRef]
- 14. Chen, D.; Han, W.; Li, S.; Lu, Z.; Qiu, H.; Shao, C.; Wang, C.; Wang, H.; Zhang, M.; Zhou, X.; et al. Preparation, Structure, Properties and Applications of Continuous Ceramic Fibers: Current Research Status and Development Direction. *Mod. Technol. Ceram.* **2018**, *39*, 151–222.
- 15. Song, S.; Liu, K. Specialty Ceramics and Refractories; Metallurgical Industry Press: Beijing, China, 2004.
- 16. Jia, Y. Preparation and Properties of Alumina Nanostructured Fibers. Ph.D. Thesis, Shandong University, Jinan, China, 2011.
- 17. Zhang, P. Electrostatic Spinning Preparation and Performance study of Nanostructured Alumina Fibers. Ph.D. Thesis, Shandong University, Jinan, China, 2012.
- Li, W.; Ma, X.; Zhang, X.; Luo, Y.; Luo, Q.; Yang, Y. Preparation of flexible alumina nanofibers and testing of their filtration properties. *Yunnan Chem.* 2021, 48, 37–39.
- 19. Wei, S.; Jin, L. Study of heat treatment process in alumina fiber preparation. Ind. Technol. Innov. 2022, 9, 32–37.
- 20. Milanović, P.; Vuksanović, M.M.; Mitrić, M.; Stojanović, D.B.; Kojović, A.; Rogan, J.R.; Jančić-Heinemann, R. Electrospun alumina fibers doped with ferric and magnesium oxides. *Sci. Sinter.* **2018**, *50*, 77–83. [CrossRef]
- Zhou, X.; Ju, J.; Li, Z.; Zhang, M.; Deng, N.; Cheng, B.; Kang, W. Design and fabrication of flexible mesoporous Si-doped Al2O3 ultrafine fibers by electro-blow spinning (EBS) technique. *Ceram. Int.* 2017, 43, 9729–9737. [CrossRef]
- 22. Almeida, R.S.M.; Bergmüller, E.L.; Lührs, H.; Wendschuh, M.; Clauß, B.; Tushtev, K.; Rezwan, K. Thermal exposure effects on the long-term behavior of a mullite fiber at high temperature. *J. Am. Ceram. Soc.* **2017**, *100*, 4101–4109. [CrossRef]
- Song, K.C.; Woo, K.J.; Kang, Y. Preparation of alumina fibers from aluminum salts by the sol-gel method. *Korean J. Chem. Eng.* 1999, 16, 75–81. [CrossRef]
- Sun, P.; Chen, X.; Gu, L.; Gu, L. Study on the microstructure of continuous Al<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub> fiber precursor sol. *Synth. Technol. Appl.* 2006, *3*, 18–21.
- 25. Chen, X.; Gu, L. Structural evolution of sol-gel derived mullite fibers with different solid contents during sintering. *J. Mater. Process Technol.* **2009**, 209, 3991–3998. [CrossRef]
- Tan, H.; Guo, C. Preparation of long alumina fibers by sol-gel method using malic acid. T Nonferrous Metals Soc 2011, 21, 1563–1567. [CrossRef]
- Mahapatra, A.; Mishra, B.G.; Hota, G. Synthesis of ultra-fine α-Al<sub>2</sub>O<sub>3</sub> fibers via electrospinning method. *Ceram. Int.* 2011, 37, 2329–2333. [CrossRef]
- 28. Wang, Y.; Li, W.; Xia, Y.; Jiao, X.; Chen, D. Electrospun flexible self-standing γ-alumina fibrous membranes and their potential as high-efficiency fine particulate filtration media. *J. Mater. Chem. A* **2014**, *2*, 15124–15131. [CrossRef]
- 29. Li, L.; Liu, X.; Wang, G.; Liu, Y.; Kang, W.; Deng, N.; Zhuang, X.; Zhou, X. Research progress of ultrafine alumina fiber prepared by sol-gel method: A review. *Chem. Eng. J.* **2021**, *421*, 127744. [CrossRef]
- Wang, Y.; Li, W.; Jiao, X.; Chen, D. Electrospinning preparation and adsorption properties of mesoporous alumina fibers. J. Mater. Chem. A 2013, 1, 10720–10726. [CrossRef]
- 31. Li, L.; Kang, W.; Zhao, Y.; Li, Y.; Shi, J.; Cheng, B. Preparation of flexible ultra-fine Al2O3 fiber mats via the solution blowing method. *Ceram. Int.* **2015**, *41*, 409–415. [CrossRef]
- Sedaghat, A.; Taheri-Nassaj, E.; Naghizadeh, R. An alumina mat with a nano microstructure prepared by centrifugal spinning method. J. Non-Cryst. Solids 2006, 352, 2818–2828. [CrossRef]
- Akia, M.; Capitanachi, D.; Martinez, M.; Hernandez, C.; de Santiago, H.; Mao, Y.; Lozano, K. Development and optimization of alumina fine fibers utilizing a centrifugal spinning process. *Microporous Mesoporous Mater.* 2018, 262, 175–181. [CrossRef]
- 34. Chen, X.; Gu, L. Spinnablity and structure characterization of mullite fibers via sol-gel-ceramic route. *J. Non-Cryst. Solids* 2009, 355, 2415–2421. [CrossRef]
- Zadeh, M.M.A.; Keyanpour-Rad, M.; Ebadzadeh, T. Synthesis of mullite nanofibres by electrospinning of solutions containing different proportions of polyvinyl butyral. *Ceram. Int.* 2013, 39, 9079–9084. [CrossRef]
- 36. Song, X.; Ma, Y.; Wang, J.; Liu, B.; Yao, S.; Cai, Q.; Liu, W. Homogeneous and flexible mullite nanofibers fabricated by electrospinning through diphasic mullite sol-gel route. *J. Mater. Sci.* 2018, *53*, 14871–14883. [CrossRef]
- Da Costa Farias, R.M.; Menezes, R.R.; Oliveira, J.E.; de Medeiros, E.S. Production of submicrometric fibers of mullite by solution blow spinning (SBS). *Mater. Lett.* 2015, 149, 47–49. [CrossRef]
- Fu, C.; Li, S.; Bai, J.; Liu, J. Research on preparation technology of alumina/zirconia precursor fiber spinning liquid. *Mater. Rep.* 2015, 29, 68–72.
- Zhong, L.; Cao, L.; Huang, J.; Liu, Y.; OuYang, H.; Wang, Q. Research progress of oxide fiber reinforced oxide ceramic matrix composites. J. Shaanxi Univ. Sci. Technol. 2022, 40, 121–133.
- 40. Han, D.; Zhen, S.; Sun, X.; Zhang, S.; Fang, K.; Ai, B.; Tao, L.; Wang, C.; Sun, H. Study on heat transfer performance of zirconia fiber high-temperature thermal insulation materials. *Ceramic* 2022, *7*, 9–12.

- 41. Li, X. Preparation and Performance Study of Continuous Alumina Fiber Reinforced Zirconia Ceramic Matrix Composites. Master's Thesis, Shanghai Jiao Tong University, Shanghai, China, 2020.
- 42. Wang, J. Research on Preparation Process of Alumina Fiber Reinforced Zirconia/Alumina Composites. Master's Thesis, Shandong University, Jinan, China, 2007.
- 43. Lang, Y. Preparation and Performance Study of Fiber-Reinforced Porous YSZ Ceramic Materials. Ph.D. Thesis, Tsinghua University, Beijing, China, 2014.
- 44. Wang, H.; Zhang, X.; Wang, N.; Li, Y.; Feng, X.; Huang, Y.; Zhao, C.; Liu, Z.; Fang, M.; Ou, G.; et al. Ultralight, scalable, and high-temperature-resilient ceramic nanofiber sponges. *Sci. Adv.* **2017**, *3*, e1603170. [CrossRef]
- 45. Sun, G.; Du, X.; Zhang, M.; Zhou, C.; Chen, J.; Liu, F. Fabrication of zirconia fibers by a sol-gel combined rotational centrifugal spinning technique. *Trans. Indian Ceram. Soc.* **2014**, *73*, 228–232. [CrossRef]
- 46. Liu, H.Y.; Chen, Y.; Liu, G.S.; Pei, S.G.; Liu, J.Q.; Ji, H.; Wang, R.D. Preparation of high-quality zirconia fibers by super-high rotational centrifugal spinning of inorganic sol. *Mater. Manuf. Process.* **2013**, *28*, 133–138. [CrossRef]
- 47. Wang, H.; Lin, S.; Yang, S.; Yang, X.; Song, J.; Wang, D.; Wang, H.; Liu, Z.; Li, B.; Fang, M.; et al. High-temperature particulate matter filtration with resilient yttria-stabilized ZrO<sub>2</sub> nanofiber sponge. *Small* **2018**, *14*, 1800258. [CrossRef]
- Ma, J.; Jia, X.; Tang, J.; Zhang, X.; Dai, S.; Yang, X. Comparison of pyrolysis and combustion characteristics of carbon fiber, glass fiber/epoxy resin. J. Compos. Mater. 2022, 1–10. [CrossRef]
- 49. Sujatha, E.R.; Atchaya, P.; Darshan, S.; Subhashini, S. Mechanical properties of glass fibre reinforced soil and its application as subgrade reinforcement. *Road Mater. Pavement Des.* **2021**, *22*, 2384–2395. [CrossRef]
- 50. Wang, B.; Hou, Y.; Chen, W.; Wu, Q.; Hu, J.; Shan, Z.; Zhao, H.; Cheng, S.; Wang, Z.; Cao, G. Preparation of glass fiber from blast furnace slag and its properties. *Sci. Technol. Innov.* **2022**, *4*, 35–38.
- 51. Ma, Q. Research on chemical attack resistance of glass fiber recycled concrete. Guangdong Build. Mater. 2022, 38, 11–14.
- 52. Zhang, X.; Huo, W.; Chen, Y.; Hu, Z.; Wang, Y.; Gan, K.; Yang, J. Novel micro-spherical Si<sub>3</sub>N<sub>4</sub> nanowire sponges from carbon-doped silica sol foams via reverse templating method. *J. Am. Ceram. Soc.* **2019**, *102*, 962–969. [CrossRef]
- 53. Huo, W.; Zhang, X.; Xu, J.; Hu, Z.; Yan, S.; Gan, K.; Yang, J. In situ synthesis of three-dimensional nanofiber-knitted ceramic foams via reactive sintering silicon foams. *J. Am. Ceram. Soc.* **2019**, *102*, 2245–2250. [CrossRef]
- 54. Liu, Y.; Liu, Y.; Choi, W.C.; Chae, S.; Lee, J.; Kim, B.S.; Park, M.; Kim, H.Y. Highly flexible, erosion resistant and nitrogen doped hollow SiC fibrous mats for high temperature thermal insulators. *J. Mater. Chem. A* **2017**, *5*, 2664–2672. [CrossRef]
- Li, Y.; Yan, G.; Zhao, Y.; Kang, W.; Li, L.; Zhuang, X.; Chen, B.; Li, X. Emulsion-Blow Spun Self-Sustained Crystalline β-Silicon Carbide (SiC) Fiber Mat and Its Conductivity Property. *Trans. Indian Ceram. Soc.* 2017, 76, 159–164. [CrossRef]
- Zhang, M.; Wang, Y.; Zhang, Y.; Song, J.; Si, Y.; Yan, J.; Ma, C.; Liu, Y.; Yu, J.; Ding, B. Conductive and elastic TiO<sub>2</sub> nanofibrous aerogels: A new concept toward self-supported electrocatalysts with superior activity and durability. *Angew. Chem. Int. Ed.* 2020, 59, 23252–23260. [CrossRef]
- 57. Santos, A.M.C.; Mota, M.F.; Leite, R.S.; Neves, G.A.; Medeiros, E.S.; Menezes, R.R. Solution blow spun titania nanofibers from solutions of high inorganic/organic precursor ratio. *Ceram. Int.* **2018**, *44*, 1681–1689. [CrossRef]
- 58. Costa, D.L.; Leite, R.S.; Neves, G.A.; Lima Santana, L.N.; Medeiros, E.S.; Menezes, R.R. Synthesis of TiO<sub>2</sub> and ZnO nano and submicrometric fibers by solution blow spinning. *Mater. Lett.* **2016**, *183*, 109–113. [CrossRef]
- 59. Li, F.; Kang, Z.; Huang, X.; Zhang, G. Fabrication of zirconium carbide nanofibers by electrospinning. *Ceram. Int.* **2014**, 40, 10137–10141. [CrossRef]
- 60. Cheng, X.; Liu, Y.T.; Si, Y.; Yu, J.; Ding, B. Direct synthesis of highly stretchable ceramic nanofibrous aerogels via 3D reaction electrospinning. *Nat. Commun.* 2022, 13, 2637. [CrossRef]
- Zu, G.; Shen, J.; Zou, L.; Wang, W.; Lian, Y.; Zhang, Z.; Du, A. Nanoengineering Super Heat-Resistant, Strong Alumina Aerogels. *Chem. Mater.* 2013, 25, 4757–4764. [CrossRef]
- 62. Sun, J.; Hu, Z.; Wu, W.; Zhou, J.; Li, J. Preparation and performance of alumina aerogel composite high temperature heat insulation tile. *Aerosp. Mater. Technol.* **2017**, *47*, 33–36+41.
- 63. Liu, X.; Su, L.; Miao, L.; Zhou, J.; Yang, B.; Wu, B.; Xia, R.; Cao, M.; Qian, J. Preparation and performance study of high temperature resistant Al2O3-SiO2 nano aerogel. *Acta Ceram. Sin.* **2021**, *42*, 620–625.
- 64. Zhou, J.; Chen, X.; Song, H.; Hu, Z.; Sun, C.; Yao, X. Effect of yttrium oxide doping on the structure and properties of Al<sub>2</sub>O<sub>3</sub> bulk aerogels. *Ceram. Bull.* **2010**, *29*, 1002–1006.
- 65. Corrias, A.; Casula, M.F.; Falqui, A.; Paschina, G. Preparation and Characterization of FeCo-Al<sub>2</sub>O<sub>3</sub> and Al<sub>2</sub>O<sub>3</sub> Aerogels. *J. Sol-Gel Sci. Technol.* **2004**, *31*, 83–86. [CrossRef]
- 66. Zhang, E.; Zhang, W.; Lv, T.; Li, J.; Dai, J.; Zhang, F.; Zhao, Y.; Yang, J.; Li, W.; Zhang, H. Insulating and robust ceramic nanorod aerogels with high-temperature resistance over 1400 C. ACS Appl. Mater. Interfaces **2021**, *13*, 20548–20558. [CrossRef]
- 67. Mei, J. Functionalization of Alumina Aerogels and Their Applications. Master's Thesis, Southeast University, Nanjing, China, 2019.
- Ren, B.; Liu, J.; Rong, Y.; Wang, L.; Lu, Y.; Xi, X.; Yang, J. Nanofibrous Aerogel Bulk Assembled by Cross-Linked SiC/SiOx Core-Shell Nanofibers with Multifunctionality and TemperatureInvariant Hyperelasticity. ACS Nano 2019, 13, 11603–11612. [CrossRef]
- 69. Zou, W.; Wang, X.; Wu, Y.; Zu, G.; Zou, L.; Zhang, R.; Yao, X.; Shen, J. Highly thermally stable alumina-based aerogels modified by partially hydrolyzed aluminum tri-sec-butoxide. *J. Sol-Gel Sci. Technol.* **2017**, *84*, 507–514. [CrossRef]

- Liu, B. High-Strength Design and Mechanism Study of Zirconia-Based Nanofiber Composite Nanoporous Aerogel Materials. Ph.D. Thesis, Institute of Advanced Materials, Shandong Academy of Sciences, Jinan, China, 2020.
- 71. Ruiz-Rosas, R.; Bedia, J.; Rosas, J.M.; Lallave, M.; Loscertales, I.G.; Rodríguez-Mirasol, J.; Cordero, T. Methanol decomposition on electrospun zirconia nanofibers. *Catal. Today* **2012**, *187*, 77–87. [CrossRef]
- 72. Chen, Y.; Mao, X.; Shan, H.; Yang, J.; Wang, H.; Chen, S.; Tian, F.; Yu, J.; Ding, B. Free-standing zirconia nanofibrous membranes with robust flexibility for corrosive liquid filtration. *RSC Adv.* **2014**, *4*, 2756–2763. [CrossRef]
- 73. Koo, J.Y.; Hwang, S.; Ahn, M.; Choi, M.; Byun, D.; Lee, W. Controlling the diameter of electrospun yttria-stabilized zirconia nanofibers. J. Am. Ceram. Soc. 2016, 99, 3146–3150. [CrossRef]
- 74. Zhang, X.; Chen, Z.; Zhang, J.; Yu, Q.; Cui, S. Rapid synthesis of silica aerogels by microwave irradiation. *J. Porous Mater.* **2021**, *28*, 1469–1479. [CrossRef]
- Zhang, X.; Chen, Z.; Zhang, J.; Ye, X.; Cui, S. Hydrophobic silica aerogels prepared by microwave irradiation. *Chem. Phys. Lett.* 2021, 762, 138127. [CrossRef]
- 76. Ye, X.; Chen, Z.; Li, M.; Wang, T.; Wu, C.; Zhang, J.; Zhou, Q.; Liu, H.; Cui, S. Effect of heat treatment temperature on melamine sponge reinforced silica aerogel. *Mater. Res. Express* **2019**, *6*, 125517. [CrossRef]
- 77. Ye, X.; Chen, Z.; Zhang, J.; Wu, C.; Xiang, J. SiC network reinforced SiO2 aerogel with improved compressive strength and preeminent microwave absorption at elevated temperatures. *Ceram. Int.* **2021**, *47*, 31497–31505. [CrossRef]
- Lee, S.W.; Kim, Y.U.; Choi, S.S.; Park, T.Y.; Joo, Y.L.; Lee, S.G. Preparation of SiO<sub>2</sub>/TiO<sub>2</sub> composite fibers by sol-gel reaction and electrospinning. *Mater. Lett.* 2007, *61*, 889–893. [CrossRef]
- Tang, X.; Sun, A.; Chu, C.; Yu, M.; Ma, S.; Cheng, Y.; Guo, J.; Xu, G. A novel silica nanowire-silica composite aerogels dried at ambient pressure. *Mater. Des.* 2017, 115, 415–421. [CrossRef]
- Calisir, M.D.; Kilic, A. A comparative study on SiO<sub>2</sub> nanofiber production via two novel non-electrospinning methods: Centrifugal spinning vs solution blowing. *Mater. Lett.* 2020, 258, 126751. [CrossRef]
- Wang, F.; Dou, L.; Dai, J.; Li, Y.; Huang, L.; Si, Y.; Yu, J.; Ding, B. In situ synthesis of biomimetic silica nanofibrous aerogels with temperature-invariant superelasticity over one million compressions. *Angew. Chem. Int. Ed. Engl.* 2020, 59, 8285–8292. [CrossRef]
- Mi, H.Y.; Jing, X.; Huang, H.X.; Turng, L.S. Instantaneous self-assembly of three-dimensional silica fibers in electrospinning: Insights into fiber deposition behavior. *Mater. Lett.* 2017, 204, 45–48. [CrossRef]
- 83. Huang, T.; Zhu, Y.; Zhu, J.; Yu, H.; Zhang, Q.; Zhu, M. Self-reinforcement of light, temperature-resistant silica nanofibrous aerogels with tunable mechanical properties. *Adv. Fiber Mat.* 2020, *2*, 338–347. [CrossRef]
- Tepekiran, B.N.; Calisir, M.D.; Polat, Y.; Akgul, Y.; Kilic, A. Centrifugally spun silica (SiO<sub>2</sub>) nanofibers for high-temperature air filtration. *Aerosol. Sci. Technol.* 2019, 53, 921–932. [CrossRef]
- 85. Wang, X.; Sun, M.; Wang, X.; Wang, S.; Bian, H.; Wu, W.; Dai, H. Study on pore structure regulation of hexagonal boron nitride/nanocellulose aerogel and thermal conductivity of composite film. *Made China Pap.* **2021**, *40*, 36–43.
- Xu, X.; Zhang, Q.; Hao, M.; Hu, Y.; Lin, Z.; Peng, L.; Wang, T.; Ren, X.; Wang, C.; Zhao, Z.; et al. Double-negative-index ceramic aerogels for thermal superinsulation. *Science* 2019, 363, 723–727. [CrossRef] [PubMed]
- 87. Lu, D.; Su, L.; Wang, H.; Niu, M.; Xu, L.; Ma, M.; Gao, H.; Cai, Z.; Fan, X. Scalable fabrication of resilient SiC nanowires aerogels with exceptional high-temperature stability. *ACS Appl. Mater. Interfaces* **2019**, *11*, 45338–45344. [CrossRef]
- 88. Yang, H.; Li, C.; Yue, X.; Huo, J.; Ye, F.; Liu, J.; Shi, F.; Ma, J. New bn/sioc aerogel composites fabricated by the sol-gel method with excellent thermal insulation performance at high temperature. *Mater. Des.* **2020**, *185*, 108217. [CrossRef]
- 89. Aravind, P.R.; Soraru, G.D. Porous silicon oxycarbide glasses from hybrid ambigels. *Microporous Mesoporous Mater.* **2011**, 142, 511–517. [CrossRef]
- 90. Zeng, X.; Ye, L.; Yu, S.; Sun, R.; Xu, J.; Wong, C.P. Facile preparation of superelastic and ultralow dielectric boron nitride nanosheet aerogels via freeze-casting process. *Chem. Mater.* **2015**, *27*, 5849–5855. [CrossRef]
- Li, G.; Zhu, M.; Gong, W.; Du, R.; Eychmüller, A.; Li, T.; Lv, W.; Zhang, X. Boron nitride aerogels with super-flexibility ranging from liquid nitrogen temperature to 1000 °C. *Adv. Funct. Mater.* 2019, 29, 1900188. [CrossRef]
- Kong, Y.; Zhang, J.; Zhao, Z.; Jiang, X.; Shen, X. Monolithic silicon nitride-based aerogels with large specific surface area and low thermal conductivity. *Ceram. Int.* 2019, 45, 16331–16337. [CrossRef]
- Su, L.; Li, M.; Wang, H.; Niu, M.; Lu, D.; Cai, Z. Resilient Si<sub>3</sub>N<sub>4</sub> nanobelt aerogel as fire-resistant and electromagnetic wavetransparent thermal insulator. ACS Appl. Mater. Interfaces 2019, 11, 15795–15803. [CrossRef]
- Mahalingam, S.; Pierin, G.; Colombo, P.; Edirisinghe, M. Facile onepot formation of ceramic fibres from preceramic polymers by pressurised gyration. *Ceram. Int.* 2015, 41, 6067–6073. [CrossRef]
- 95. Liang, C.; Wang, Z. Eggplant-derived SiC aerogels with high-performance electromagnetic wave absorption and thermal insulation properties. *Chem. Eng. J.* **2019**, *373*, 598–605. [CrossRef]
- 96. Su, L.; Wang, H.; Niu, M.; Fan, X.; Ma, M.; Shi, Z.; Guo, S.W. Ultralight, recoverable, and high-temperature-resistant SiC nanowire aerogel. *ACS Nano* **2018**, *12*, 3103–3111. [CrossRef]
- 97. Leventis, N.; Sadekar, A.; Chandrasekaran, N.; Sotiriou-Leventis, C. Click synthesis of monolithic silicon carbide aerogels from polyacrylonitrile-coated 3D silica networks. *Chem. Mater.* **2010**, *22*, 2790–2803. [CrossRef]
- 98. Chen, Y.; Ola, O.; Liu, G.; Han, L.; Hussain, M.Z.; Thummavichai, K.; Wen, J.; Zhang, L.; Wang, N.; Xia, Y.; et al. Multifunctional porous SiC nanowire scaffolds. *J. Eur. Ceram. Soc.* **2021**, *41*, 3970–3979. [CrossRef]

- 99. Song, L.; Fan, B.; Chen, Y.; Gao, Q.; Li, Z.; Wang, H.; Zhang, X.; Guan, L.; Zhang, R. Ultralight and hyperelastic SiC nanofiber aerogel spring for personal thermal energy regulation. *J. Am. Ceram. Soc.* **2022**, *11*, 1235–1248. [CrossRef]
- Song, X.; Zhang, K.; Song, Y.; Duan, Z.; Liu, Q.; Liu, Y. Morphology, microstructure and mechanical properties of electrospun alumina nanofibers prepared using different polymer templates: A comparative study. J. Alloy. Compd. 2020, 829, 154502. [CrossRef]
- 101. Liu, R.; Dong, X.; Xie, S.; Jia, T.; Xue, Y.; Liu, J.; Jing, W.; Guo, A. Ultralight, thermal insulating, and high-temperature resistant mullite-based nanofibrous aerogels. *Chem. Eng. J.* **2019**, *360*, 464–472. [CrossRef]
- Song, X.; Liu, W.; Wang, J.; Xu, S.; Liu, B.; Cai, Q.; Ma, Y. Highly aligned continuous mullite nanofibers: Conjugate electrospinning fabrication, microstructure and mechanical properties. *Mater. Lett.* 2018, 212, 20–24. [CrossRef]
- Si, Y.; Wang, X.; Dou, L.; Yu, J.; Ding, B. Ultralight and fire-resistant ceramic nanofibrous aerogels with temperature-invariant superelasticity. *Sci. Adv.* 2018, 4, eaas8925. [CrossRef] [PubMed]
- 104. Wang, X.; Dou, L.; Li, Z.; Yang, L.; Yu, J.; Ding, B. Flexible hierarchical ZrO<sub>2</sub> nanoparticle-embedded SiO<sub>2</sub> nanofibrous membrane as a versatile tool for efficient removal of phosphate. *ACS Appl. Mater. Interfaces* **2016**, *8*, 34668–34676. [CrossRef] [PubMed]
- 105. Mao, X.; Si, Y.; Chen, Y.; Yang, L.; Zhao, F.; Ding, B.; Yu, J. Silica nanofibrous membranes with robust flexibility and thermal stability for high-efficiency fine particulate filtration. *RSC Adv.* **2012**, *2*, 12216–12223. [CrossRef]
- 106. Yan, J.; Zhao, Y.; Wang, X.; Xia, S.; Zhang, Y.; Han, Y.; Yu, J.; Ding, B. Polymer template synthesis of soft, light, and robust oxide ceramic films. *Iscience* **2019**, *15*, 185–195. [CrossRef]
- 107. Lu, B.; He, Y.; Duan, H.; Zhang, Y.; Li, X.; Zhu, C.; Xie, E. A new ultrahigh-speed method for the preparation of nanofibers containing living cells: A bridge towards industrial bioengineering applications. *Nanoscale* **2012**, *4*, 1003–1009. [CrossRef]
- Ren, L.; Simmons, T.J.; Lu, F.; Rahmi, O.; Kotha, S.P. Template free and large-scale fabrication of silica nanotubes with centrifugal jet spinning. *Chem. Eng. J.* 2014, 254, 39–45. [CrossRef]
- Ren, L.; Ozisik, R.; Kotha, S.P. Rapid and efficient fabrication of multilevel structured silica micro-/nanofibers by centrifugal jet spinning. J. Colloid Interface Sci. 2014, 425, 136–142. [CrossRef]
- 110. Loccufier, E.; Geltmeyer, J.; Daelemans, L.; D'hooge, D.R.; De Buysser, K.; De Clerck, K. Silica nanofibrous membranes for the separation of heterogeneous azeotropes. *Adv. Funct. Mater.* **2018**, *28*, 1804138. [CrossRef]
- 111. An, Z.; Ye, C.; Zhang, R.; Zhou, P. Flexible and recoverable SiC nanofiber aerogels for electromagnetic wave absorption. *Ceram. Int.* **2019**, *45*, 22793–22801. [CrossRef]
- 112. Zheng, C.; Li, X.; Yu, Y.; Wang, H.; Cao, F.; Zhao, D. Study of high temperature resistant SiC(Al) fibers precursor— Polyaluminocarbosilane fibers. *Acta Polym. Sin.* 2006, *6*, 768–773. [CrossRef]
- Zhang, J.; Zhang, J.; Sun, Q.; Ye, X.; Ma, X.; Wang, J. Sol-Gel Routes toward Ceramic Nanofibers for High-Performance Thermal Management. *Chemistry* 2022, 4, 1475–1497. [CrossRef]
- 114. Jia, C.; Li, L.; Liu, Y.; Fang, B.; Ding, H.; Song, J.; Liu, Y.; Xiang, K.; Lin, S.; Li, Z.; et al. Highly compressible and anisotropic lamellar ceramic sponges with superior thermal insulation and acoustic absorption performances. *Nat. Commun.* 2020, 11, 3732. [CrossRef] [PubMed]
- 115. Jia, C.; Liu, Y.; Li, L.; Song, J.; Wang, H.; Liu, Z.; Li, Z.; Li, B.; Fang, M.; Wu, H. A foldable all-ceramic air filter paper with high efficiency and high-temperature resistance. *Nano Lett.* **2020**, *20*, 4993–5000. [CrossRef]
- 116. Li, P.; Wang, C.; Zhang, Y.; Wei, F. Air filtration in the free molecular flow regime: A review of high-efficiency particulate air filters based on carbon nanotubes. *Small* **2014**, *10*, 4543–4561. [CrossRef]
- Zhang, Y.; Zhu, Y.; Xiong, C.; Wu, J.; Chen, F. Bioinspired ultralight inorganic aerogel for highly efficient air filtration and oil-water separation. ACS Appl. Mater. Interfaces 2018, 10, 13019–13027. [CrossRef]
- Song, Y.; Li, B.; Yang, S.; Ding, G.; Zhang, C.; Xie, X. Ultralight boron nitride aerogels via template-assisted chemical vapor deposition. *Sci. Rep.* 2015, *5*, 10337. [CrossRef] [PubMed]
- 119. Xue, Y.; Dai, P.; Zhou, M.; Wang, X.; Pakdel, A.; Zhang, C.; Weng, Q.; Takei, T.; Fu, X.; Popov, Z.I.; et al. Multifunctional superelastic foam-like boron nitride nanotubular cellular-network architectures. *ACS Nano* **2017**, *11*, 558–568. [CrossRef]
- 120. Ye, X.; Zhang, J.; Chen, Z.; Xiang, J.; Jiang, Y.; Xie, F.; Ma, X. Microwave absorption properties of Ni/C@ SiC composites prepared by precursor impregnation and pyrolysis processes. *Def. Technol.* 2021; *in press.* [CrossRef]
- 121. Ye, X.; Chen, Z.; Li, M.; Wang, T.; Wu, C.; Zhang, J.; Zhou, Q.; Liu, H.; Cui, S. Hollow SiC foam with a double interconnected network for superior microwave absorption ability. *J. Alloys Compd.* **2020**, *817*, 153276. [CrossRef]
- 122. Cai, Z.; Su, L.; Wang, H.; Niu, M.; Gao, H.; Lu, D.; Li, M. Hydrophobic SiC@ C nanowire foam with broad-band and mechanically controlled electromagnetic wave absorption. *ACS Appl. Mater. Interfaces* **2020**, *12*, 8555–8562. [CrossRef] [PubMed]
- 123. Hou, Y.; Cheng, L.; Zhang, Y.; Yang, Y.; Deng, C.; Yang, Z.; Chen, Q.; Du, X.; Zheng, L. SiC nanofiber mat: A broad-band microwave absorber, and the alignment effect. *ACS Appl. Mater. Interfaces* **2017**, *9*, 43072–43080. [CrossRef] [PubMed]
- 124. Hou, Y.; Cheng, L.; Zhang, Y.; Yang, Y.; Deng, C.; Yang, Z.; Chen, Q.; Du, X.; Zhao, C.; Zheng, L. Enhanced flexibility and microwave absorption properties of HfC/SiC nanofiber mats. *ACS Appl. Mater. Interfaces* **2018**, *10*, 29876–29883. [CrossRef]
- Zhao, Y.; Yan, J.; Cai, W.; Lai, Y.; Song, J.; Yu, J.; Ding, B. Elastic and well-aligned ceramic LLZO nanofiber based electrolytes for solid-state lithium batteries. *Energy Storage Mater.* 2019, 23, 306–313. [CrossRef]
- 126. Zhang, W.; Wang, H.; Guan, K.; Wei, Z.; Zhang, X.; Meng, J.; Liu, X.; Meng, J. La0. 6Sr0. 4Co0. 2Fe0. 8O3– δ/CeO2 heterostructured composite nanofibers as a highly active and robust cathode catalyst for solid oxide fuel cells. ACS Appl. Mater. Interfaces 2019, 11, 26830–26841. [CrossRef]

- 127. Wang, Y.; Li, J.; Sun, J.; Wang, Y.; Zhao, X. Electrospun flexible self-standing Cu–Al2O3 fibrous membranes as Fenton catalysts for bisphenol A degradation. *J. Mater. Chem. A* 2017, *5*, 19151–19158. [CrossRef]
- 128. Cheng, Z.; Zhao, S.; Han, L. A novel preparation method for ZnO/γ-Al<sub>2</sub>O<sub>3</sub> nanofibers with enhanced absorbability and improved photocatalytic water-treatment performance by Ag nanoparticles. *Nanoscale* **2018**, *10*, 6892–6899. [CrossRef]

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