



## Article

# Use of Ethylamine, Diethylamine and Triethylamine in the Synthesis of Zn,Al Layered Double Hydroxides

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**Abstract:** Amines with two carbon atoms in the organic chain [ethylamine (EA), diethylamine (DEA), triethylamine (TEA)] have been used as precipitant agents to obtain a hydrotalcite-like compound with Zn (II) and Al (III) as layered cations and with nitrate anions in the interlayered region to balance the charge. This Layered Double Hydroxide was prepared following the coprecipitation method, and the effect on the crystal and particle sizes was studied. Also, the effect of submitting the obtained solids to hydrothermal post-synthesis treatment by conventional heating and microwave assisted heating were studied. The obtained solids were exhaustively characterized using several instrumental techniques, such as X-ray diffraction, Thermal Analysis (DTA and TG), Chemical Analysis, Infrared Spectroscopy (FT-IR), determination of Particle Size Distribution and BET-Surface area. Well crystallized solids were obtained showing two possible LDH phases, depending on the orientation of the interlayer anion with respect to the brucite-like layers. The results indicated that there is a certain influence of the amine, when used as a precipitating agent, and as a consequence of the degree of substitution, on the crystallinity and particle size of the final solid obtained. The LDHs obtained using TEA exhibited higher crystallinity, which was improved after a long hydrothermal treatment by conventional heating. Regarding the shape of the particles, the formation of aggregates in the former solid was detected, which could be easily disintegrated using ultrasound treatments, producing solid powder with high crystallinity and small particle size, with homogeneous size distribution.

**Keywords:** LDHs; hydrotalcite; amines; coprecipitation method; hydrothermal treatment; crystallinity



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## 1. Introduction

The term Layered Double Hydroxides (LDHs) is used to name synthetic or natural hydroxides with a layered structure and with at least two types of metal cations in the main layers, which are positively charged, and contain anionic species in the interlayer space. This large family of compounds is also called anionic clays, in comparison with cationic clays, which, in their interlayer region, contain cations to balance the negative charge of the layers [1]. They are also known as hydrotalcite-like compounds, because hydrotalcite is the most abundant mineral with this layered structure in Nature. These materials are not as abundant in Nature as the analogous cationic clays, but they are very easy to synthesize with a tuned composition and are generally not very expensive.

The layered structure of the LDH consists of brucite-like layers with divalent cations occupying octahedral spaces formed by  $\text{OH}^-$  ions  $[\text{M}(\text{OH})_6]$ , in which an isomorphic, and partial, substitution of divalent cations by trivalent cations has taken place, leaving the layers positively charged. The electroneutrality of the compound is achieved by the incorporation of anions in the interlayer space [2]. The chemical composition of LDHs is described by the chemical formula:  $[\text{M}_{1-x}^{\text{II}}\text{M}_x^{\text{III}}(\text{OH})_2]^{x+} [\text{A}^{m-}]_{\frac{x}{m}} \cdot n\text{H}_2\text{O}$ , where  $\text{M}^{\text{II}}$  and  $\text{M}^{\text{III}}$  are the divalent and trivalent metal cations, respectively, and  $\text{A}^-$  is the interlayer anion, with  $x$  defined as the  $\text{M}^{\text{III}} / (\text{M}^{\text{II}} + \text{M}^{\text{III}})$  ratio [1,3]. Both organic and inorganic anions

can be incorporated into the interlayer space, in a wide range of sizes and charges [1,4]. In addition, in the interlayer space there are randomly arranged water molecules, and this interlayer region has quasi-liquid behavior. The wide range of anions and divalent and trivalent cations that can be used to prepare LDH, provide them with a diversity of compositions.

In recent decades, LDHs have established themselves as promising materials, due to their properties and applications in a number of fields, such as water remediation [5–7], catalysis [8–10], drug delivery [11,12], electroactivity [13], biomedicine [14,15], and others. Their applications as anion exchanging solids depend on the interlayer anion and its affinity for the LDH layers. Thus, Miyata et al. [16] have reported an order of anionic selectivity for MgAl hydrotalcite-like materials, which could be applicable to other combinations of elements [17]. According to this, for applications involving anion exchanging, LDHs containing nitrate have a higher ion exchange facility than those containing carbonate in the interlayer space.

As above mentioned, the synthesis of LDHs in the laboratory is relatively simple and cheap, and they could be synthesized by many different methods. The method selected has an impact on the properties of the final solid and, therefore, on its subsequent application. Among the different methods described, the most widely used is the coprecipitation method, due to its great ease of use and reproducibility [3]. This method is based on precipitation by the slow dropwise addition of a solution containing the mixture of  $M^{II}$  and  $M^{III}$  salts solution in a fixed ratio and the anion to an alkaline solution, working at constant pH. The addition of a second alkaline solution allows the pH of the precipitation medium to be maintained during the precipitation of the cations [3]. Many parameters are relevant to monitoring the process, such as the type and concentration of cations and anions, the precipitation medium, the pH and the temperature [18,19]. The optimal pH depends on the nature of the cations to be incorporated into the structure; thus, Klopogge et al. [20] reported that the best crystallinity for samples of Zn/Al LDH were exhibit in the pH range 11–12. Moreover, when the coprecipitation method is carried out, the incorporation of the carbonate anion into the interlayer space is very difficult to avoid, due to the fact that it is the anion with the highest affinity for the LDH layers. However, its incorporation can be prevented by using decarbonated water and bubbling  $N_2$ .

LDH synthesized by the coprecipitation method generally exhibits low crystallinity with a high degree of aggregation and a wide particle size distribution. The most common way to obtain more uniform particle properties, with an improved crystallinity is by an aging process. For this reason, the coprecipitation method is generally followed by a long aging period, from 10 to 80 h, and often longer [19]. Also, post-synthesis treatment heating at moderate temperatures is used; for instance, by gentle-to-gentle reflux or by hydrothermal treatment. The aging presumably occurs through the Ostwald ripening process, in which larger and more perfect crystallites grow at the expense of smaller particles in solution by dissolution/precipitation processes [21]. Therefore, microwave-assisted hydrothermal treatment, or hydrothermal treatment by conventional heating, have been widely used with the aim of improving structural and textural properties [22–26].

In our previous work [27], Zn/Al-LDH in molar ratio 2:1 was synthesized using amines with one carbon atom in the organic chain [methylamine (MMA), dimethylamine (DMA) and trimethylamine (TMA)], such as precipitant agents. Furthermore, the evolution on the properties caused by hydrothermal treatment of samples synthesized with amines using the following two ways of heating was studied: conventional heating and microwave assisted heating. Highly crystalline LDH with nitrate anion in the interlayer was obtained using DMA as the precipitant agent and, after submitting the solid to conventional hydrothermal treatment. In the present work, we reported the use of amines with two carbon atoms in the organic chain [ethylamine (EA), diethylamine (DEA), triethylamine (TEA)] as precipitant agents in the synthesis of Zn,Al-LDH in molar ratio 2:1. Furthermore, the solids were treated hydrothermally by conventional heating and microwave-assisted heating to improve the

crystallinity and properties of the solids. The effect of the treatment methods was analyzed for each amine-assisted synthesis condition.

## 2. Materials and Methods

### 2.1. Materials

$\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  (98–102%),  $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  (98–102%) and NaOH (98%) were purchased from Panreac and used as received. An aqueous solution of ethylamine (70% in  $\text{H}_2\text{O}$ ) was purchased from Alfa Aesar (Ward Hill, MA, USA). Diethylamine (99.5%) was purchased from Panreac (Barcelona, Spain). Triethylamine ( $\geq 99\%$ ) was purchased from Sigma Aldrich (Burlington, MA, USA).

### 2.2. Synthesis

The coprecipitation method was used to prepare the desired solids [3]. In order to avoid the intercalation of carbonate anions, decarbonated water solutions and nitrogen atmosphere were used during the synthesis. In order to prepare the solution of the metal cations, 0.3 L in 2.5 M concentration of their nitrate salts in a  $\text{M}^{\text{II}}/\text{M}^{\text{III}}$  molar ratio 2:1 was prepared. For the precipitation medium, an aqueous solution of 4.5 M concentration of the desired amine was prepared. The metal cation solution was added dropwise to the amine solution, which was maintained under vigorous magnetic stirring. The pH of the precipitation media was kept at a preselected pH value of 10 by adding the required amount of a 2 M NaOH solution using a 240 CRISON pH-burette. After complete addition, the aqueous suspension of the precipitated solid was stirred for 1 h at room temperature and, then, the slurry obtained was subjected to different ageing treatments: (i) a portion of the sample without any hydrothermal treatment was kept as a reference; (ii) a portion was subjected to a hydrothermal treatment by microwave-assisted heating (MW) for 60 or 300 min at 90 °C; (iii) a portion was subjected to a hydrothermal treatment by conventional heating (HT) for 1 or 7 days at 90 °C using a home-made stainless-steel bomb lined with Teflon. Hydrothermal treatment by microwave-assisted heating was carried out in a MILESTONE ETHOS PLUS microwave oven, where the aqueous suspension was placed in Teflon digestion vessels, sealed and mounted on a turntable inside the oven. The programmed temperature was controlled by a thermocouple immersed in a reference vessel and software provided by the manufacturer. The different portions of the solid suspensions were separated and washed by centrifugation with distilled water until reaching a pH close to 7, in order to eliminate the equilibrium ions of the starting salts. Finally, the solids were dried at 40 °C in an oven under air atmosphere.

The samples were labelled according to the preparation procedure: reaction medium, hydrothermal treatment method and time period to which they had been subjected. So, a sample labeled as ZA2XYt, ZA2 represents the cations (Zn–Al, molar ratio 2/1); X represents the precipitation media (depending on the amine used EA, DEA or TEA); Y represents the aging treatment (STH for the reference sample, HT or MW); and t the treatment duration (in minutes for MW, in days for HT).

### 2.3. Characterization

A Yobin Ivon Ultima II apparatus at NUCLEUS (University of Salamanca, Salamanca, Spain) was used for elemental chemical analysis of Zn and Al by ICP-OES.

A Siemens D-5000 instrument was used to record Powder X-ray diffraction (PXRD) patterns using  $\text{Cu-K}\alpha$  radiation ( $\lambda = 1.54050 \text{ \AA}$ ) with a scanning rate of  $2^\circ/\text{min}$  from  $5^\circ$  to  $70^\circ$  ( $2\theta$ ). The Scherrer equation was used to calculate the crystallite sizes from the FWHM (Full Width at Half Maximum) of the diffraction maximum (001). The Warren correction for instrumental line broadening was taken into account, but the possible contribution of disorder effects and/or lattice strains to the peak broadening was ignored.

A Perkin-Elmer Spectrum One instrument was used to record the FT-IR spectra by transmission with a nominal resolution of  $2 \text{ cm}^{-1}$  from  $4000 \text{ cm}^{-1}$  to  $450 \text{ cm}^{-1}$ , using KBr pressed pellets.

SDT Q600 equipment from TA Instruments was used to carry out the thermogravimetric (TG) and differential thermal analyses (DTA). The thermal analyses were carried out by heating from room temperature to 900 °C at a rate of 10 °C/min under continuous oxygen (L'Air Liquide, 99.995%) flow (50 mL/min).

A Micromeritics Gemini VII 2390t apparatus was used to record the nitrogen (L'Air Liquide, 99.999%) adsorption–desorption isotherms at −196 °C, and to calculate the specific surface area and porosity data. The apparatus was calibrated with He (L'Air Liquide, 99.999%). Before measurements, the samples were pretreated at 110 °C for 2 h under a stream of N<sub>2</sub> in a Micromeritics FlowPrep 060 Sample Degass System.

A Diffraction Mastersizer 2000 equipment from Malvern Instruments was used to determine the particle size distribution (PSD) by Laser Diffraction. Using the dispersion unit Hydro 2000 from Malvern Instruments, the solid was dispersed in water at 25 °C (approx. 0.05 vol.%), and, after measuring the PSD for the dispersed samples, ultrasounds were applied in situ to disaggregate the particles.

### 3. Results and Discussion

#### 3.1. Element Chemical Analysis

Table 1 gives the Zn/Al molar ratio values and chemical formulae of the samples synthesized in the presence of the different amines used as precipitant agents. The chemical formulae were deduced from the results of elemental chemical analysis and thermogravimetric analysis (see below).

**Table 1.** Element chemical analysis results and the chemical formulae of each sample.

Sample	Al <sup>a</sup>	Zn <sup>a</sup>	Zn/Al <sup>b</sup>	x <sup>c</sup>	Formulae
ZA2EASTH	8.32	40.27	2.00	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.44 H <sub>2</sub> O
ZA2EAMW60	7.98	38.63	2.00	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.48 H <sub>2</sub> O
ZA2EAMW300	7.96	38.30	1.99	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.50 H <sub>2</sub> O
ZA2EAHT1	9.84	46.51	1.95	0.34	LDH * + ZnO + Al <sub>2</sub> O <sub>3</sub>
ZA2EAHT7	11.05	52.01	1.94	0.34	LDH * + ZnO + Al <sub>2</sub> O <sub>3</sub>
ZA2DEASTH	7.51	38.44	2.11	0.32	[Zn <sub>0.68</sub> Al <sub>0.32</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.32</sub> · 0.57 H <sub>2</sub> O
ZA2DEAMW60	7.67	38.87	2.09	0.32	[Zn <sub>0.68</sub> Al <sub>0.32</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.32</sub> · 0.55 H <sub>2</sub> O
ZA2DEAMW300	7.64	38.31	2.07	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.55 H <sub>2</sub> O
ZA2DEAHT1	7.73	38.47	2.05	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.51 H <sub>2</sub> O
ZA2DEAHT7	8.37	41.88	2.07	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.42 H <sub>2</sub> O
ZA2TEASTH	7.64	37.56	2.03	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.55 H <sub>2</sub> O
ZA2TEAMW60	7.62	37.62	2.04	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.51 H <sub>2</sub> O
ZA2TEAMW300	7.60	37.39	2.03	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.52 H <sub>2</sub> O
ZA2TEAHT1	7.67	37.82	2.04	0.33	[Zn <sub>0.67</sub> Al <sub>0.33</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.33</sub> · 0.50 H <sub>2</sub> O
ZA2TEAHT7	8.13	41.16	2.09	0.32	[Zn <sub>0.68</sub> Al <sub>0.32</sub> (OH) <sub>2</sub> ](NO <sub>3</sub> ) <sub>0.32</sub> · 0.43 H <sub>2</sub> O

<sup>a</sup> Mass percentage. <sup>b</sup> Molar ratio. <sup>c</sup> Al/(Al + Zn) molar ratio. \* It was not possible to determine the chemical formula of the LDH phase.

In all cases, the Zn/Al molar ratio approached the value of 2, suggesting a complete precipitation of the existing cations in the synthesis medium. In some cases, a small deviation could be observed, but never more than 5%.

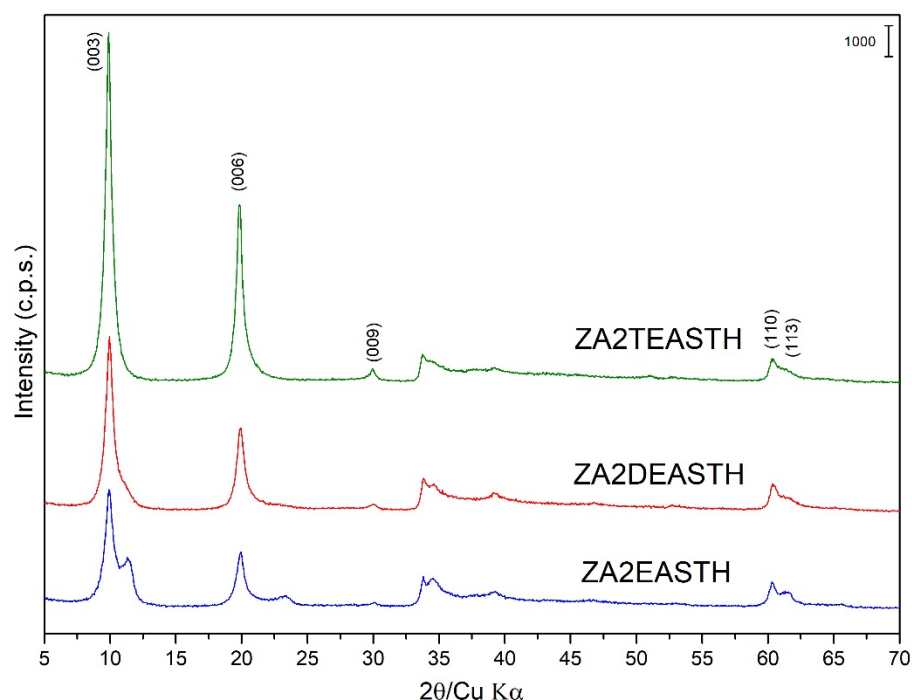
For the determination of the chemical formula of each of the samples, the amount of nitrate anion was calculated from the Al/(Al + Zn) molar ratio, assuming that it was the only interlayered anion, as observed by FT-IR spectroscopy, neutralizing the positive charge excess of the layers.

Unlike samples synthesized without amines or in the presence of methylamine, for dimethylamine or trimethylamine samples synthesized in the presence of EA, DEA or TEA have water content per chemical formula which is generally lower, as reported in our previous work [27]. The more regular stacking of the octahedral layers, as observed by PXRD, leads to a decrease in the number of water molecules per unit formula, and this fact is more evident in the samples having HT hydrothermal treatment.

On the other hand, samples ZA2EAHT1 and ZA2EAHT7 were mostly or completely formed by ZnO, as deduced from the PXRD analysis. However, after chemical analysis, the presence of Al in the solid sample could be determined, indicating the formation of an amorphous phase containing aluminum [28], not observed by X-ray diffraction. Therefore, as there was a mixture of phases in the solid sample, it was not possible to determine the amount of Zn and Al forming the LDH structure (if its collapse was not complete); thus, making it difficult to determine its chemical formula.

### 3.2. Powder X-ray Diffraction (PXRD)

The samples synthesized using EA, DEA and TEA as modifiers of the precipitation medium were also obtained in the form of microcrystalline powder. Figure 1 shows the PXRD diagrams of these samples without hydrothermal treatment. The positions and relative intensities of the recorded diffraction peaks revealed a layered structure of the solids, characteristic of an ordered  $3R_1$  polytype of solids with the LDH structure (JCPDS: 22-0700) [29–31]. In all cases, the most intense diffraction peak, attributed to the diffraction plane (003) of the crystal structure, was recorded at a position  $10.0^\circ$  ( $2\theta$ ), with a spacing of 8.93 Å. This spacing is in agreement with the values reported by Miyata et al. [16] for LDH with nitrate as the interlayer anion arranged in a perpendicular orientation to the brucite-like layers and with a  $M^{2+}/M^{3+}$  molar ratio close to 2. Confirming the layered structure, diffraction peaks corresponding to crystallographic planes (006) and (009) were recorded at values close to  $19.9^\circ$  ( $2\theta$ ) and  $30.0^\circ$  ( $2\theta$ ), respectively, and with spacings of 4.46 Å and 2.98 Å. Reflections corresponding to diffraction planes (110) and (113) were recorded at  $60.3^\circ$  ( $2\theta$ ) and  $61.3^\circ$  ( $2\theta$ ), with spacings of 1.53 Å and 1.51 Å, respectively.



**Figure 1.** Powder X-ray diffraction diagrams of the samples prepared in the presence of EA, DEA and TEA with no hydrothermal treatment.

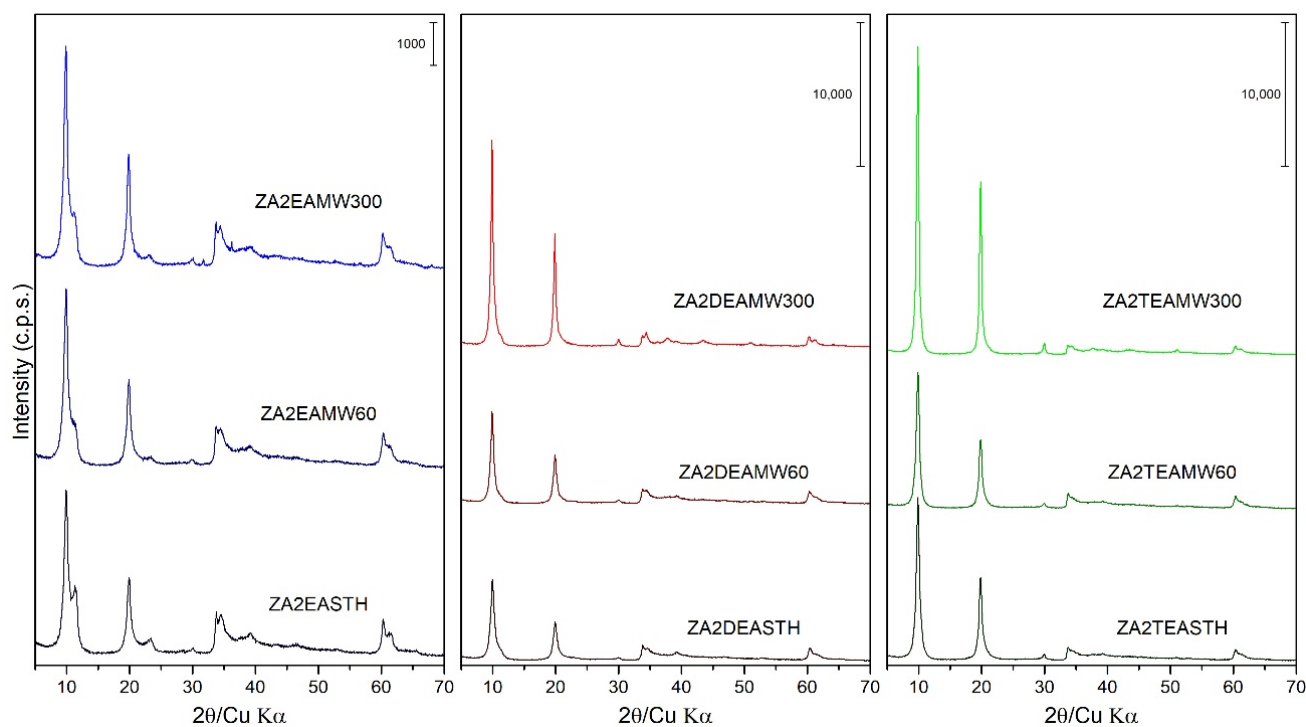
As can be seen in Figure 1, the solids prepared using DEA or TEA as precipitant agents showed a single crystallographic phase. However, in the case of the sample prepared in the presence of EA, diffraction peaks corresponding to a second LDH phase could be distinguished in the PXRD diagram. Thus, it is possible to distinguish the most characteristic peak of this secondary phase, which was recorded close to  $11.33^\circ$  ( $2\theta$ ) and ascribed to the diffraction peak (003), with a spacing of 7.81 Å. In this case, the diffraction angle for this diffraction peak was slightly higher than that found for the sample synthesized in the



absence of amines [27], where, unlike the shoulder observed for the sample prepared in the absence of amines, a well-defined maximum could be clearly distinguished. In this second phase with a spacing of 7.81 Å for the peak, due to the (003) planes, the interlaminar nitrate anions were arranged with their molecular plane parallel to the plane of the brucite-like layers [32–34].

Observing the width and profile of the peaks ascribed to the diffraction plane (003) close to 10° (2θ), the crystallinity of these samples decreased in the order: ZA2TEASTH > ZA2DEASTH > ZA2EASTH. Again, it could be observed how the use of amines (and the nature of these) in the precipitation medium modifies the crystallinity of the solids, to the point of two phases coexisting, with nitrate anion in the interlayer space depending on the amine used.

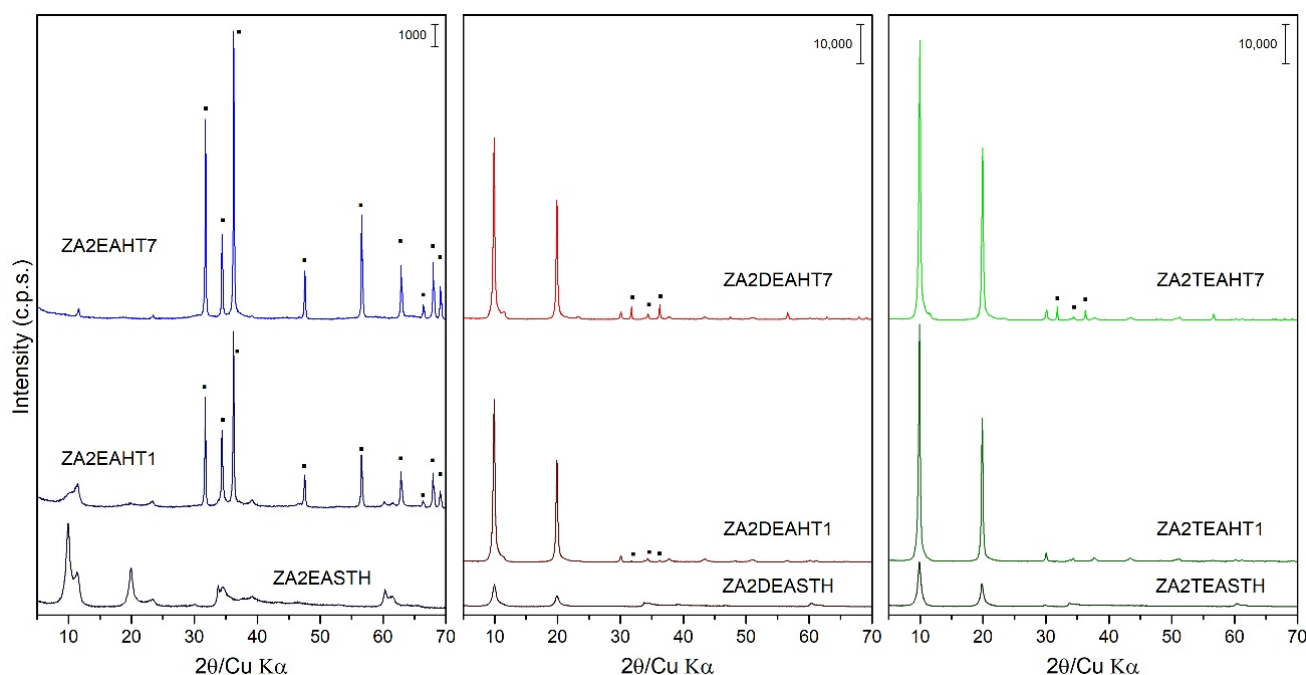
The samples synthesized in the presence of EA, DEA and TEA were also subjected to hydrothermal treatments, using two heating routes: microwave-assisted heating (MW) and conventional oven heating (HT). Figure 2 includes the PXRD plots of these samples with different MW hydrothermal treatment periods. Observing the profile of the diffraction peaks and their relative intensities as the treatment time increased, it can be seen how, unlike the samples prepared in our previous work [27], the application of MW hydrothermal treatment increased the crystallinity of the solids as the MW treatment time increased. Comparing the width of the diffraction peaks (003) of the samples subjected to a longer treatment time, that is, the samples with 300 min of treatment, the following decreasing order of crystallinity could be established as a function of the amine used: ZA2TEAMW300 > ZA2DEAMW300 > ZA2EAMW300. It is noteworthy that, as the MW hydrothermal treatment time was prolonged on the samples synthesized with EA, the relative intensity of the diffraction peak (003) of the phase with the nitrate anion parallel to the brucite-like layers did not increase as the treatment time increased, going from being a well-defined peak in sample ZA2EASTH to being a shoulder of the peak (003) in sample ZA2EAMW300.



**Figure 2.** PXRD of the samples prepared in the presence of EA, DEA and TEA with 60 and 300 min of MW treatment.

The application of a conventional hydrothermal treatment (HT) resulted in a greater increase in the crystallinity of the solids synthesized in the presence of DEA and TEA.

However, the same effect is not observed for the samples synthesized in the presence of EA, where the application of HT treatment resulted in the collapse of the layered structure, as can be observed in Figure 3. Thus, the PXRD recorded for samples synthesized using EA showed the formation of a zinc oxide (ZnO) like zincite phase (JCPDS: 00-036-1451 [29]). This ZnO phase is identified mainly by the reflections recorded in the range of  $28^{\circ}$ – $38^{\circ}$  ( $2\theta$ ). While with 1 day of HT treatment the collapse of the LDH phase was not complete, as can be seen from its corresponding PXRD diagram, after 7 days of treatment the collapse was complete. So, the main diffraction peaks of sample ZA2EAHT7 corresponded to the ZnO phase and, with very little intensity, the diffraction peaks at  $11.6^{\circ}$  and  $23.4^{\circ}$  ( $2\theta$ ) corresponded to the crystallographic planes (003) and (006), respectively, of the LDH phase with the interlayer nitrate arranged parallel to the plane of the brucite-like layers.



**Figure 3.** PXRD of the samples prepared in the presence of EA, DEA and TEA with 1 and 7 days of HT treatment. (■) ZnO phase.

In the case of the samples synthesized using DEA or TEA, segregation of a small amount of the zincite phase (ZnO) could also be observed as the HT treatment time increased. Thus, from the profile of the diffraction peaks (003) and their relative intensities for these two series of samples, it can be observed how with 1 day of treatment, solids of high crystallinity were obtained without segregation of the ZnO phase when TEA was used as the precipitant agent, and with a very small amount of this phase when DEA was used. Therefore, the segregation of ZnO decreased as the degree of substitution of the amino group in the compound used as precipitating agent increased. Comparing the width of the diffraction peaks (003) of the samples subjected to a longer treatment time (samples with 7 days of treatment) the following decreasing order of crystallinity can be established as a function of the amine used: ZA2TEAHT7 > ZA2DEAHT7. Table 2 shows the amount of zincite phase present in the sample calculated from the calibration line with the ratio of the areas of the characteristic peaks of the zincite phase and the LDH phase [35]. Thus, it is shown how the amount of zincite phase increased as the HT treatment time increased, finding an amount of approximately 2% of ZnO in the sample ZA2DEAHT7. By extrapolation of the calibration line to samples synthesized using EA, for the ZA2EAHT1 sample, 67% was ZnO and for longer treatments it was close to 100% (Table 2).

**Table 2.** ZnO content in the Zn and Al samples prepared in the presence of EA, DEA and TEA with HT treatment.

Sample	(101) Peak Area (ZnO) <sup>a</sup>	(003) Peak Area (LDH) <sup>a</sup>	Area Ratio (101)/(003)	ZnO Content <sup>b</sup>
ZA2EAHT1	1807.0	1424.0	1.268961	67
ZA2EAHT7	2776.0	-	-	≈100
ZA2DEAHT1	122.1	18,339.8	0.006659	0.32
ZA2DEAHT7	790.7	19,232.1	0.041114	2.14
ZA2TEAHT1	-	-	-	-
ZA2TEAHT7	502.6	26,627.0	0.018876	0.97

<sup>a</sup> a.u. <sup>b</sup> Mass percentage.

The lattice parameters of the prepared solids were calculated from the positions of the diffraction peaks due to the (003) and (110) planes [30]; being,  $c = 3 \cdot d(003) \approx 26.6\text{--}26.8 \text{ \AA}$ , and  $a = 2 \cdot d(110) \approx 3.069\text{--}3.077 \text{ \AA}$ . In addition, from the value of the Full Width at Half Maximum (FWHM) of reflection 003, the crystallite size (D) in the  $c$  direction was calculated using the Scherrer equation,  $D = k\lambda/\beta\cos\theta$  [36,37], where  $k$  is a constant, taken in this case as 0.9;  $\lambda$  is the wavelength of the radiation used;  $\beta$  the FWHM and  $\theta$  the diffraction angle; correction due to instrumental broadening was not applied. The values calculated for the samples obtained are included in Table 3, together with the calculated values for the number of stacked layers.

**Table 3.** Lattice parameters  $c$  and  $a$ , average crystal size D and number of stacked layers for the samples obtained.

Sample	$c$ (Å)	$a$ (Å)	D (Å)	Number of Stacked Layers
ZA2EASTH	26.80	3.0697	111	12
ZA2EAMW60	26.80	3.0697	118	13
ZA2EAMW300	26.80	3.0720	135	15
ZA2EAHT1	-	-	-	-
ZA2EAHT7	-	-	-	-
ZA2DEASTH	26.67	3.0673	126	14
ZA2DEAMW60	26.80	3.0697	148	17
ZA2DEAMW300	26.80	3.0720	210	23
ZA2DEAHT1	26.67	3.0743	284	32
ZA2DEAHT7	26.80	3.0766	334	37
ZA2TEASTH	26.80	3.0697	150	17
ZA2TEAMW60	26.80	3.0697	142	16
ZA2TEAMW300	26.80	3.0697	218	24
ZA2TEAHT1	26.80	3.0766	319	36
ZA2TEAHT7	26.67	3.0743	322	36

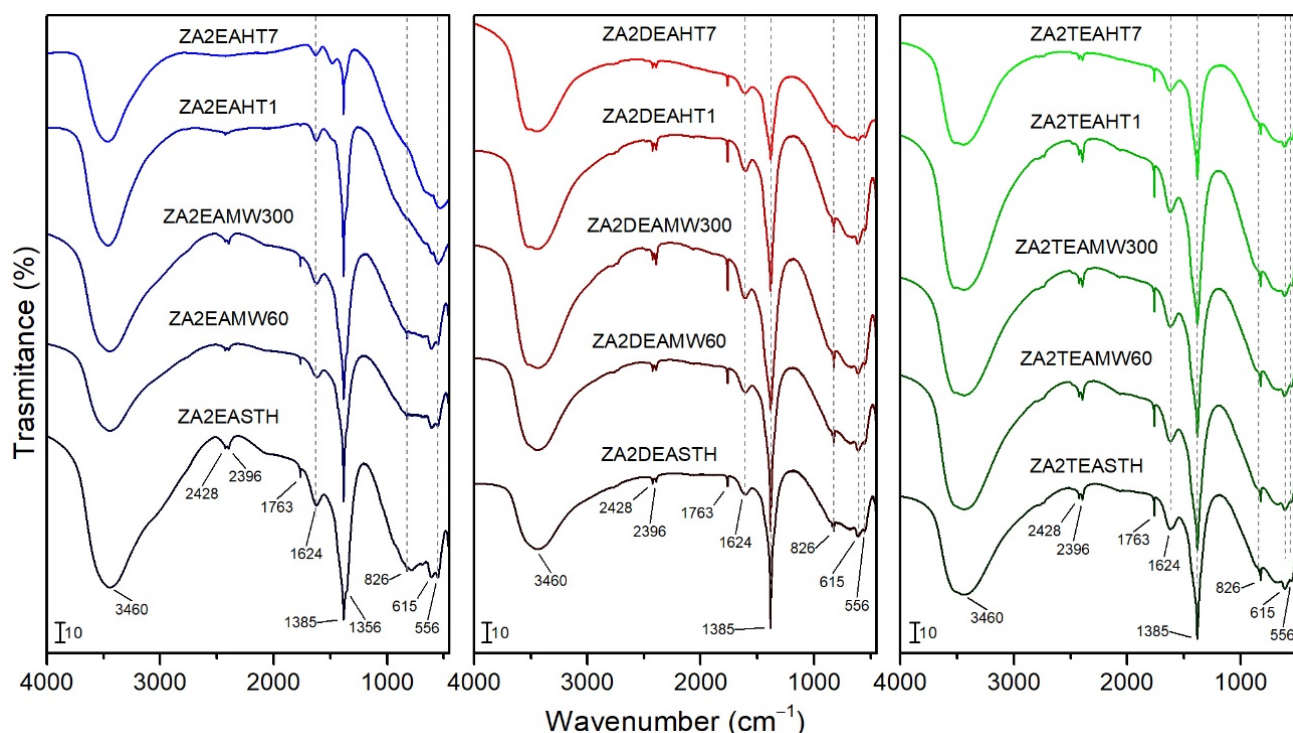
The similarity of the lattice parameter  $a$  value, with differences of less than 1%, are coherent with the homogeneity of metals composition in the samples.

The crystallite size (D) values highlight how the conventional hydrothermal treatment led to a greater increase in the crystallinity of the solids. When the MW treatment was applied, it was observed that, as the treatment time increased the crystal size increased, resulting in the obtaining of, in both the DEA and TEA series, crystal sizes around 200 Å. However, when HT treatment was applied, it could be observed that the samples synthesized in the presence of TEA with only one day of treatment reached a crystal size close to 320 Å, which was practically maintained, even if the HT treatment time increased. In the case of samples prepared in the presence of DEA, with one day of HT treatment a crystal size higher than 280 Å was obtained; reaching a crystal size of 334 Å when the treatment was extended to 7 days.



### 3.3. FT-IR Spectroscopy

FT-IR spectra of the synthesized solids are plotted in Figure 4. The broad band at  $3460\text{ cm}^{-1}$  is ascribed to the stretching vibration modes of the hydroxyl groups and the water molecules of the interlayer space. At lower wavenumbers it was possible to observe a bands at  $615$  and  $556\text{ cm}^{-1}$  caused by M-OH vibration modes. The vibration ascribed to the bending mode of the interlayer water molecules was recorded at  $1624\text{ cm}^{-1}$  [18,38,39]. The bands at  $2396$  and  $2428\text{ cm}^{-1}$  could be attributed to atmospheric  $\text{CO}_2$  weakly bonded on the LDH surface.



**Figure 4.** FT-IR spectra of the samples prepared in the presence of EA, DEA and TEA with no hydrothermal treatment and with MW and HT treatments.

In all cases, the characteristic bands of the vibrational modes of the nitrate anion molecules were recorded, confirming its presence as an interlayer anion. Thus, bands at  $1385\text{ cm}^{-1}$  and  $826\text{ cm}^{-1}$  can be observed, assigned to the  $\nu_3(\text{E}')$  and  $\nu_2(\text{A}_2'')$  vibrational modes, respectively, of  $\text{NO}_3^-$  with a  $\text{D}_{3h}$  symmetry [19,38]. At  $1763\text{ cm}^{-1}$  a narrow band can be observed, corresponding to the combination of the vibrational modes  $\nu_1(\text{A}_1')$  at  $1068\text{ cm}^{-1}$  and  $\nu_4(\text{E}')$  at  $692\text{ cm}^{-1}$  of nitrate, the latter not clearly observed in the infrared spectra. For sample ZA2EASTH, a shoulder can be seen at  $1356\text{ cm}^{-1}$  (Figure 4), which could be due to nitrate anions in parallel orientation in the second LDH phase observed by PXRD. As observed by PXRD for the samples synthesized in the presence of EA after MW hydrothermal treatment, a crystallinity increase of the phase with the anions in perpendicular orientation took place in detriment of the phase with the anions in parallel orientation to the layers. This effect was reflected in the FT-IR spectra, where the shoulder at  $1356\text{ cm}^{-1}$  became less evident (Figure 4). On the other hand, when DEA or TEA were used as precipitation agents, such a shoulder was also not observed at  $1356\text{ cm}^{-1}$ .

When the sample synthesized in the presence of EA was subjected to a HT hydrothermal treatment process, as observed in the PXRD studies, the structure collapsed segregating the zinc oxide in its zincite phase, being almost complete with long treatment times; this caused the FT-IR spectra of these samples to change slightly. In Figure 4 the bands attributed to the vibrational modes of the hydroxyl groups and water molecules (band positions already mentioned above) can be identified. In addition, the vibration band at

1385  $\text{cm}^{-1}$ , attributed to the presence of the nitrate anion of the LDH phase recorded with low intensity in the PXRD diagrams of the sample, can be clearly observed. As a result of the formation of ZnO in the HT-treated samples, a band at the limit of the spectrum, around 470  $\text{cm}^{-1}$ , attributed to the stretching vibrational mode of ZnO, was observed in the FT-IR spectra [40].

In the spectra of the samples synthesized in the presence of DEA and TEA, no different bands were observed for the hydrothermally treated samples, both MW and HT, with respect to that of the samples that received no treatment. It should be noted that the higher crystallinity of the samples resulted in a better ordering of both the interlaminar anions and the layered structure, giving rise to a regularity that was reflected in a subtle increase in intensity and narrowing of the vibration bands.

Neither in the FT-IR spectra recorded for the samples without hydrothermal treatment, nor for the samples subjected to MW or HT hydrothermal treatments, were bands corresponding to the vibrational modes of the amines used during the synthesis observed.

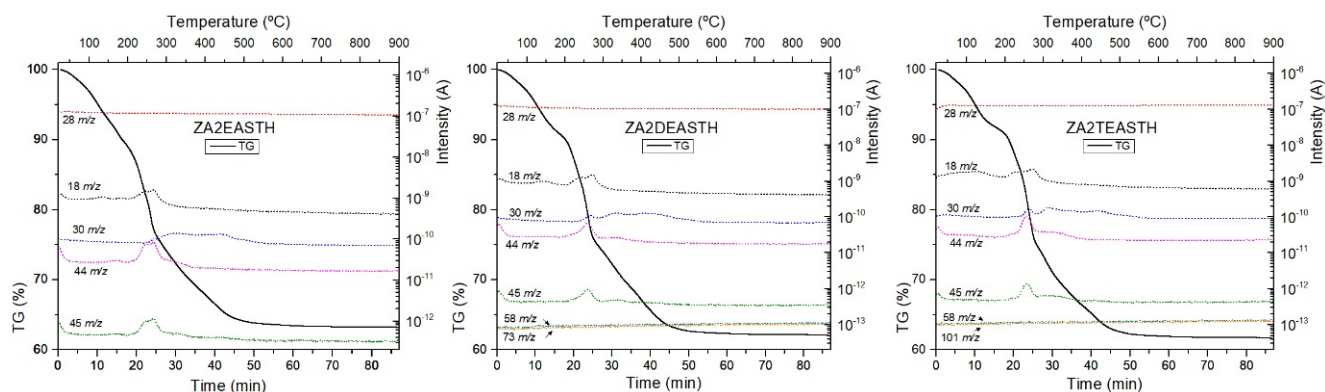
### 3.4. Thermal Analysis

The thermal analysis of the samples prepared using EA, DEA or TEA as precipitant agents were carried out to determine their stability and evolution to mixed oxides. During the thermal analysis, mass spectrometry (MS) of the gases and vapors formed during the process (EGA, evolved gas analysis) was carried out. To identify the masses of the generated species, a complete mass spectrum was initially recorded and, in a second step, the MS analysis was performed by fixing these masses and following the change in their intensities throughout the analysis. The reference MS of the expected evolved gases was also taken into account. The signals monitorized corresponded to  $\text{H}_2\text{O}$  ( $m/z = 18$ ),  $\text{N}_2$  and  $\text{CO}$  ( $m/z = 28$ ),  $\text{NO}$  ( $m/z = 30$ ),  $\text{N}_2\text{O}$  ( $m/z = 44$ ),  $\text{NO}_2$  ( $m/z = 45$ ), EA (fragment at  $m/z = 30$ , 44 and 45), DEA (fragments at  $m/z = 58$  and 73) and TEA (fragments at  $m/z = 58$  and 101).

TG curves of the samples without hydrothermal treatment (STH) are included in Figure 5, together with the tracked masses of the gases generated during the process. In all curves, the typical decomposition stages of LDH compounds can be identified. Three decomposition stages could be identified in all TG curves. First, below 180  $^\circ\text{C}$ , water removal was observed, as shown by the MS peak at  $m/z = 18$ . The second stage of decomposition was observed up to around 300  $^\circ\text{C}$ , and corresponded to the release of water from the condensation of hydroxyl groups of the brucite-like layers. Finally, a steady mass loss was observed between 300 and 700  $^\circ\text{C}$ , which corresponded to the removal of interlayer nitrate species. The MS signals recorded in this temperature range corresponded to formation of species such as  $\text{NO}$ ,  $\text{NO}_2$ , and  $\text{N}_2\text{O}$ , from nitrate decomposition. In all the curves, the process of elimination of the hydroxyl groups practically overlapped with the last stage of decomposition/removal of the interlaminar anion. However, there was better differentiation of the first stage from the second decomposition stage as the degree of substitution of the amino group in the compound used as precipitating agent increased. Thus, for the ZA2TEASTH sample a small plateau could be observed around 200  $^\circ\text{C}$ .

From the tracking of the  $m/z$  signals of each amine, the absence of amine residues in the final solids could be concluded, as could also be observed by FT-IR spectroscopy. Only in the case of the sample prepared in the presence of EA, could the mass at 45  $m/z$  be attributed to that amine. However, some of the masses associated with EA overlap with the signals of the decomposition products of the nitrate anion ( $\text{N}_2\text{O}$ ,  $\text{NO}$  and  $\text{NO}_2$ ). The absence of EA was confirmed because the mass tracking curves had the same profile as those of  $\text{NO}_x$  gas formation in the solids synthesized in the presence of DEA and TEA.

For the hydrothermally treated samples, both MW and HT, had similar TG curves recorded, in which the aforementioned plateau was more evident as a result of the increase in the crystallinity of the samples.



**Figure 5.** TG curves, in  $O_2$  atmosphere, of the samples prepared in the presence of EA, DEA and TEA with no hydrothermal treatment and tracking of the characteristic  $m/z$  signals.

On the other hand, it can also be observed that, as the crystalline regularity of the solids increases, the total mass loss decreased (Table 4), due to the lower amount of water retained in the solids. Table 4 shows that as the hydrothermal treatment time increased, the total mass loss was lower, being in all cases between 30 and 40%. Only in the case of the samples synthesized in the presence of EA and with HT hydrothermal treatment was a total mass loss less than 30% observed. This was due to the formation of ZnO in these samples, where after 7 days of treatment practically the whole structure had collapsed and the mass loss observed in the TG curve was due to the small amount of the LDH phase, the retained water and the decomposition of the nitrate anion.

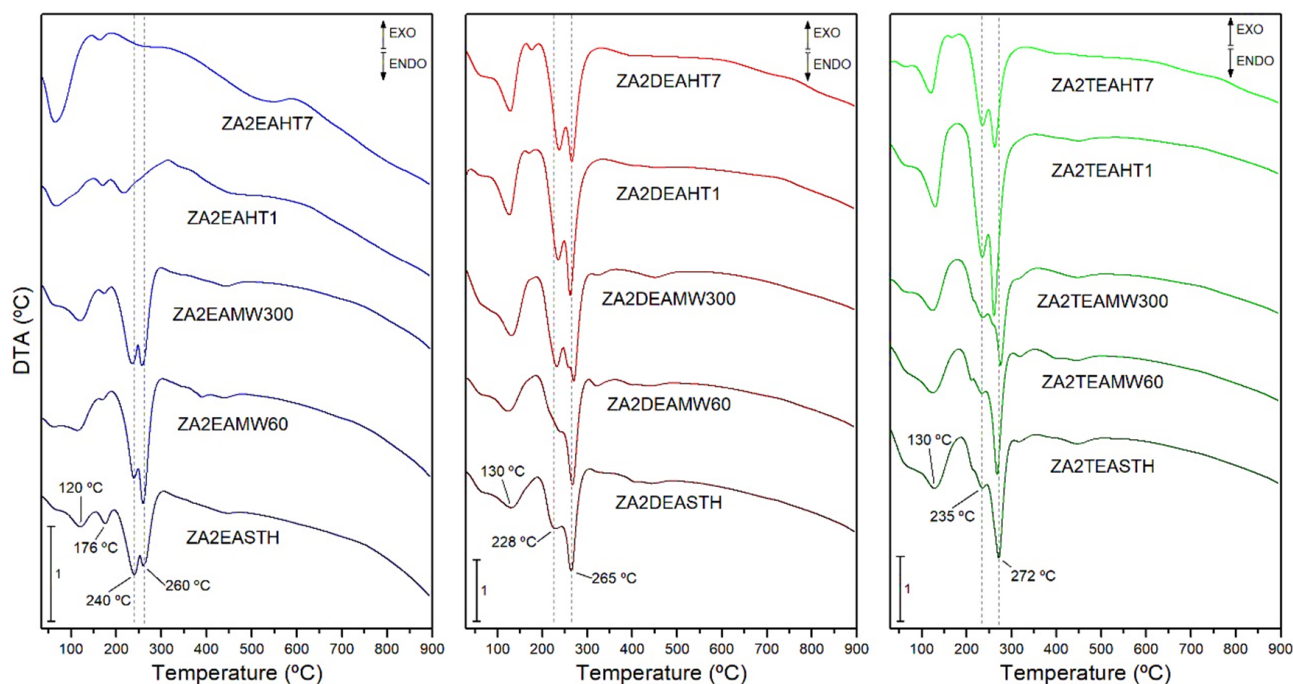
**Table 4.** Total weight loss and  $H_2O$  molecules per chemical formula calculated for each sample.

Sample	Weight Loss (%)	$H_2O$ Molecules Per Chemical Formula ( $n$ )
ZA2EASTH	36.6	0.44
ZA2EAMW60	36.7	0.48
ZA2EAMW300	38.0	0.50
ZA2EAHT1	22.6	-
ZA2EAHT7	14.6	-
ZA2DEASTH	38.1	0.57
ZA2DEAMW60	37.8	0.55
ZA2DEAMW300	38.2	0.55
ZA2DEAHT1	37.3	0.51
ZA2DEAHT7	31.0	0.42
ZA2TEASTH	38.7	0.55
ZA2TEAMW60	38.3	0.51
ZA2TEAMW300	38.5	0.52
ZA2TEAHT1	37.9	0.50
ZA2TEAHT7	32.8	0.43

The DTA curves of the samples both without hydrothermal treatment and with hydrothermal treatments, MW or HT, are included in Figure 6. In all cases, endothermic minima associated with the different decomposition processes of the samples can be observed.

The minimum recorded at 120–130 °C in the DTA curves was associated with the process of release of the water retained in the interlayer space. However, in the case of the samples synthesized in the presence of EA, another minimum could be observed at 176 °C, which could correspond to the elimination of water molecules retained more strongly in the structure of the layered solid. On the other hand, at temperatures above 200 °C a minimum associated with the process of elimination of hydroxyl groups in the form of water vapor and the decomposition of the nitrate anion was found. In many cases this minimum presented a shoulder at lower temperature, or was even dissociated into two

clearly distinguishable minima. This dissociation and, therefore, differentiation in the decomposition processes, became more evident as the crystalline regularity in the solids increased after the application of a hydrothermal treatment. On the other hand, it is worth mentioning that in all cases the main minimum was found at higher temperatures than in the case of the solids synthesized in our previous work using methylamine, dimethylamine or trimethylamine as precipitant agents, and even in the case of the solids synthesized in the absence of amines [27]. Thus, for the sample synthesized in the presence of DEA, this minimum was found at 260 °C, while for the sample synthesized in the presence of TEA, the minimum shifted to 272 °C.



**Figure 6.** DTA curves, in O<sub>2</sub> atmosphere, of the samples prepared in the presence of EA, DEA and TEA with no hydrothermal treatment and with MW and HT treatments.

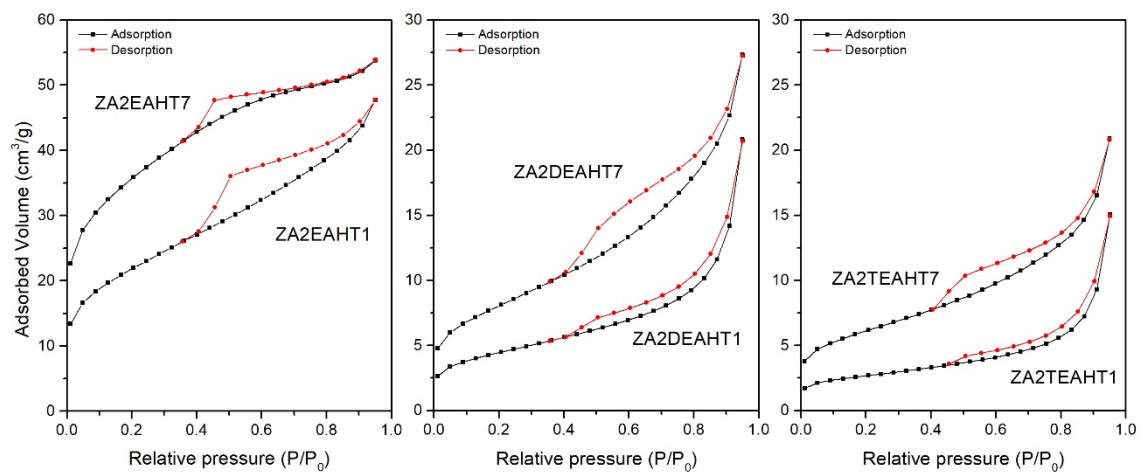
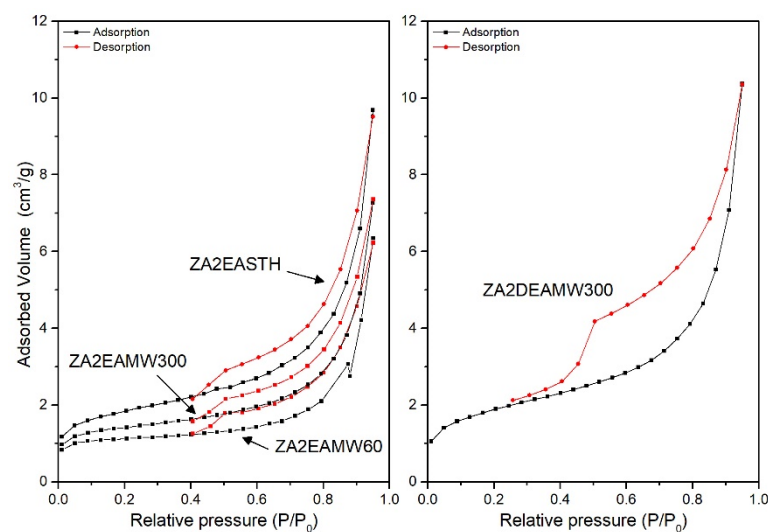
### 3.5. Specific Surface Area and Porosity

The textural properties of the synthesized solids were studied from the N<sub>2</sub> adsorption–desorption isotherms at −196 °C. Table 5 includes the values of the specific surface areas calculated by the BET ( $S_{\text{BET}}$ ) method [41,42], the pore volume ( $V_{\text{pore}}$ ) and the average pore diameter calculated by the BJH method [42,43] for the synthesized samples. For the samples prepared in the presence of DEA and TEA, both without hydrothermal treatment and with MW treatment, the adsorption capacity was below the confidence limit of the equipment used. Only for the samples with HT hydrothermal treatment (and sample ZA2DEAMW300) did the adsorption measurements present confidence values. This behavior was not observed for samples prepared in the presence of EA, although the samples presented low  $S_{\text{BET}}$  values, close to the detection limit. Figures 7 and 8 include the corresponding adsorption–desorption isotherms, which correspond to type II according to the IUPAC classification [44,45], corresponding to adsorption on non-porous or mesoporous adsorbents, where adsorption can occur without monolayer-multilayer restrictions. Moreover, it can be observed how all of them presented a hysteresis cycle, corresponding to the H3 type according to the IUPAC classification [42], indicating that adsorption took place in slit-shaped pores formed by layer-like particles.



**Table 5.** BET specific surface area, pore volume and average pore diameter of the samples prepared.

Sample	$S_{\text{BET}}$ ( $\text{m}^2/\text{g}$ )	$V_{\text{pore}}$ ( $\text{mm}^3/\text{g}$ )	BJH Desorption Average Pore Diameter (nm)
ZA2EASTH	6.1	14.9	9.1
ZA2EAMW60	3.4	9.8	9.6
ZA2EAMW300	4.5	11.2	9.6
ZA2EAHT1	74.3	73.7	4.6
ZA2EAHT7	118.5	83.2	3.0
ZA2DEASTH	-	-	-
ZA2DEAMW60	-	-	-
ZA2DEAMW300	6.4	16.0	6.9
ZA2DEAHT1	15.3	32.2	7.6
ZA2DEAHT7	28.2	42.2	5.3
ZA2TEASTH	-	-	-
ZA2TEAMW60	-	-	-
ZA2TEAMW300	-	-	-
ZA2TEAHT1	9.0	23.3	9.9
ZA2TEAHT7	21.0	32.4	5.7

**Figure 7.** Nitrogen adsorption-desorption isotherms for HT treated samples prepared in the presence of EA, DEA and TEA.**Figure 8.** Nitrogen adsorption-desorption isotherms for MW treated samples prepared in the presence of EA and DEA.



In all cases  $S_{\text{BET}}$  values were higher than those obtained for the samples obtained with the amines used in our previous work and, also, the values were higher than those found for the samples synthesized in the absence of amines [27]. On the contrary, smaller pore diameter sizes were recorded.

In view of the results included in Table 5, the segregation of ZnO in the samples synthesized in the presence of EA with HT treatment led to a substantial increase in the specific surface area. Solids with higher porosity were obtained, where the pore volume increased, with smaller pore diameter sizes with respect to the sample without hydrothermal treatment.

In the case of the solids synthesized in the presence of DEA and TEA, after the application of a HT hydrothermal treatment, according to the PXRD results, an increase in the crystallinity took place, as a consequence of better ordering of the brucite-like layers. Together with the increase in the crystallinity of the solids, an increase in the  $S_{\text{BET}}$  could be observed and adsorption–desorption curves could be recorded in both cases. Moreover, the prolongation of the HT treatment resulted in higher crystallinity linked to an increase of the  $S_{\text{BET}}$  value, where, in both series of samples, the  $S_{\text{BET}}$  value for sample with 7 days of treatment was twice that for the sample with one day of treatment. However, it is important to remember the presence of approximately 2% of ZnO in sample ZA2DEAHT7, which could justify the difference in specific surface area with respect to that of sample ZA2TEAHT7.

Figure 8 shows the adsorption–desorption curves for the samples without hydrothermal treatment and with MW treatment. While the sample without hydrothermal treatment had an  $S_{\text{BET}}$  value close to  $6 \text{ m}^2/\text{g}$ , for samples synthesized using EA as precipitant agent, when a MW hydrothermal treatment was applied the surface area decreased. In the three curves a similar behavior against the desorption process can be observed, with slightly larger pore sizes when MW treatment was applied. On the other hand, when DEA or TEA were used as precipitant agents, only the adsorption–desorption curve for sample ZA2DEAMW300 was recorded. The  $S_{\text{BET}}$  value for this sample was  $6.4 \text{ m}^2/\text{g}$ , higher than that found for the analogous sample synthesized in the presence of EA.

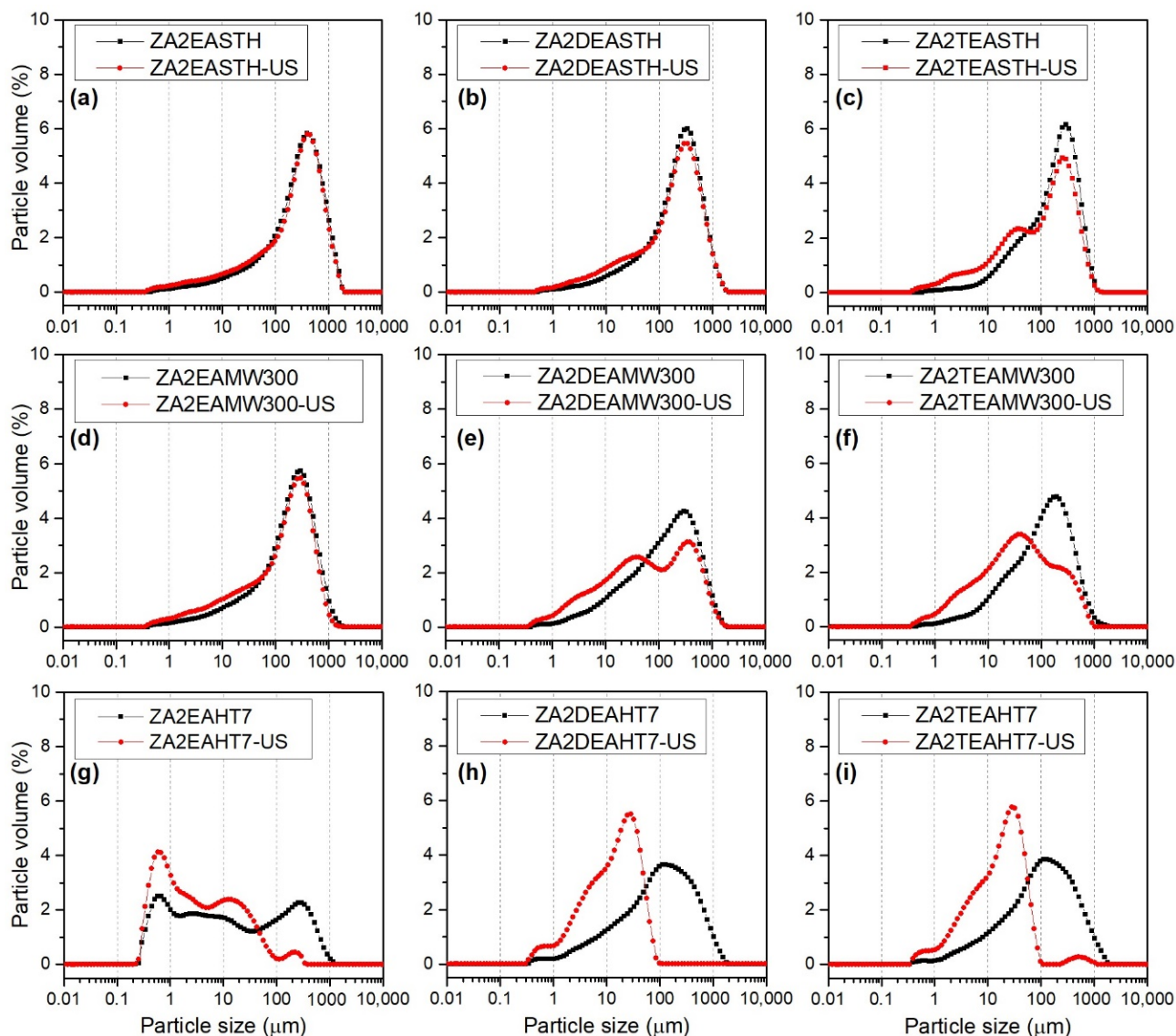
### 3.6. Particle Size Distribution

Figure 9 shows the particle size distribution curves of the samples synthesized in the presence of EA, DEA and TEA without hydrothermal treatment and after the application of the longest periods of both hydrothermal treatments: MW and HT. For each of the samples two distribution curves are represented: (i) the distribution curve of the sample in aqueous suspension, black curve, and (ii) the distribution curve of the sample in aqueous suspension subjected to sonication treatment for 15 min, directly in the particle size analyzer, red curve. Sonication treatment is often used to disaggregate primary particles, changing the distribution curves.

The samples without hydrothermal treatment presented a monomodal size distribution centered between 300 and 400  $\mu\text{m}$ . Although, as the degree of amino group substitution increased, a small shoulder could be observed at lower size values, approximately at 20  $\mu\text{m}$ , as observed in Figure 9a–c. The application of ultrasound for 15 min did not have a great impact on the distribution curves; only in the case of sample ZA2TEASTH was the aforementioned shoulder accentuated at 20  $\mu\text{m}$ . This fact indicates that samples prepared in the presence of EA and DEA give rise to robust particles that are more difficult to disaggregate than particles obtained using TEA as the precipitant agent.

The particle size distribution curves for the samples subjected to MW hydrothermal treatment for 300 min are included in Figure 9d–f. While the sample synthesized in the presence of EA presented the same profile as its version without hydrothermal treatment, broader distribution curves were observed for samples ZA2DEAMW300 and ZA2TEAMW300. After the application of ultrasound, practically no changes in the distribution were observed for sample ZA2EAMW300, indicating the difficulty in disaggregating its particles, showing the same robustness as the STH sample. In the case of samples pre-

pared in the presence of DEA and TEA, after the application of ultrasound, disaggregation to smaller particle size took place, resulting in bimodal distribution curves with maxima at 40  $\mu\text{m}$ . Moreover, after the application of ultrasound, it can be observed how these distributions became wider.



**Figure 9.** Particle size distribution before (black) and after (red) sonication in water suspension of synthesized samples (a) ZA2EASTH, (b) ZA2DEASTH, (c) ZA2TEASTH, (d) ZA2EAMW300, (e) ZA2DEAMW300, (f) ZA2TEAMW300, (g) ZA2EAHT7, (h) ZA2DEAHT7, (i) ZA2TEAHT7.

The distribution curves for the samples subjected to HT hydrothermal treatment for 7 days are included in Figure 9g–i, where it is observed that the samples synthesized in the presence of DEA and TEA presented the same behavior as the samples with MW hydrothermal treatment over 300 min, obtaining monomodal particle size distributions that covered a wide range of sizes. However, when an ultrasound treatment was applied, the particles disaggregated, obtaining monomodal distributions at smaller particle sizes with a maxima between 20 and 30  $\mu\text{m}$ , which presented, in both cases, a shoulder at even smaller particle sizes, around 5  $\mu\text{m}$ . Furthermore, in both samples it was observed how the size distributions became narrower. For example, the particle size distribution of sample ZA2DEAHT7 decreased from a maximum at 130  $\mu\text{m}$  to a maximum at 30  $\mu\text{m}$ , with a shoulder at 6  $\mu\text{m}$ . In contrast, the curve of sample ZA2EAHT7 presented slightly different

behavior, where a multimodal distribution was obtained over a wide range of particle sizes, with maxima at 0.6, 2.7, 10 and 300  $\mu\text{m}$ . After the application of ultrasound, the disintegration of the larger particles occurred, also resulting in a multimodal distribution with maxima at 0.6, 2.5, 12 and 225  $\mu\text{m}$ , with a higher percentage of smaller particles. The different behavior of sample ZA2EAHT7 was due to fact that in this sample the hydrotalcite-type structure had completely collapsed and the only phase detected by PXRD was ZnO (Figure 3).

Table 6 includes the volume-weighted mean particle diameter,  $D[4,3]$ , values for each of the samples, both before and after ultrasound treatment [46,47]. In general, the lowest particle size values were obtained for the samples synthesized in the presence of TEA, obtaining values even lower than those of the samples obtained without amines in the reaction medium (reported in our previous work [27]). For all series of samples, similar  $D[4,3]$  values were found, regardless of the MW treatment time, whereas, when HT hydrothermal treatment was applied, as the treatment time increased, the  $D[4,3]$  value slightly decreased. The smallest particle size, 114  $\mu\text{m}$ , was found for sample ZA2EAHT7, where, as previously commented, the structure had collapsed to zinc oxide.

**Table 6.** Volume-weighted mean particle diameter,  $D[4,3]$ , values of prepared samples.

Sample	$D[4,3]$ Before Sonication	$D[4,3]$ After Sonication
ZA2EASTH	428	399
ZA2EAMW60	280	258
ZA2EAMW300	298	248
ZA2EAHT1	327	230
ZA2EAHT7	114	16
ZA2DEASTH	347	324
ZA2DEAMW60	268	263
ZA2DEAMW300	275	210
ZA2DEAHT1	272	94
ZA2DEAHT7	232	22
ZA2TEASTH	274	212
ZA2TEAMW60	221	219
ZA2TEAMW300	201	112
ZA2TEAHT1	253	79
ZA2TEAHT7	231	36

The application of a sonication treatment had a great impact on the  $D[4,3]$  value of the samples subjected to HT hydrothermal treatment, as can be deduced from the data collected in Table 6. Thus, the lowest value was obtained for sample ZA2DEAHT7, with a mean particle diameter of 36  $\mu\text{m}$ .

It can be concluded that the samples subjected to MW treatment were formed by tightly bound particle agglomerates, deduced by the low disintegration after the sonication treatment. Only for samples with long periods of MW treatment was a slight disintegration of the agglomerates observed, obtaining the lowest  $D[4,3]$  value for the sample ZA2TEAMW300, with 112  $\mu\text{m}$ .

#### 4. Conclusions

The effect on the properties of Zn-Al-NO<sub>3</sub> LDHs, prepared using amines with two carbon atoms in the organic chain, together with the application of hydrothermal treatments using different energy sources, conventional or microwave heating, was studied. Well crystallized compounds were obtained, the crystallinity of which improved a lot after prolonged conventional hydrothermal treatment when diethylamine or triethylamine were used as precipitant agents. When ethylamine was used in the synthesis media, two LDH phases were obtained, in which the nitrate anion had two different orientations in the interlayer space. Furthermore, the LDH structure of solids prepared using ethylamine

collapsed easily with a conventional hydrothermal treatment to form zinc oxide. When the samples were subjected to conventional hydrothermal treatment, the formation of ZnO was also observed when DEA or TEA were used, although to a smaller amount. So, the ZnO content is lower as the degree of substitution of the amino group of the compound used as precipitating agent increases. The results also showed the formation of aggregates which could be disaggregated by sonication. So, it was possible to obtain solids with high crystal sizes and low particle size distributions when a conventional treatment was used.

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