



# Article DFT Analysis of the Electronic and Structural Properties of Lanthanide Nitride Cluster Fullerenes Ln<sub>3</sub>N@C<sub>80</sub>

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Abstract: We have undertaken a DFT study of the nitride cluster fullerenes (NCFs)  $Ln_3N@C_{80}$  for the complete series of fourteen lanthanides plus lanthanum by using the PBE functional with the Grimme's dispersion correction (PBE-D2). We tested the DN and DND basis sets, which are equivalent to 6-31G and 6-31G(d) Pople-type basis sets, respectively. Due to the known convergence problems when treating lanthanide-containing systems, only with the DN basis set was it possible to complete the calculations (geometry optimization and analysis of selected electronic parameters) for all the fifteen NCFs. We found that the bending of the Ln<sub>3</sub>N cluster increases as the ionic radius increases, in general agreement with the available X-ray diffraction data. The Ln<sub>3</sub>N cluster becomes more planar as the Ln–N bond length is contracted, and the  $C_{80}$  cavity slightly deforms. The HOMO-LUMO energies and distribution, as well as the charge and spin of the encapsulated metal ions, are analyzed.

Keywords: nitride cluster fullerenes; lanthanides; density functional theory; geometry; electronic parameters



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## 1. Introduction

At the end of the last century, it was discovered experimentally that it was possible trap atoms or molecules (A) inside fullerene cages ( $C_n$ ): the resulting complexes are referred to as endohedral fullerenes (EFs or endohedral metallofullerenes, EMFs, when A species contains metals) of the general formula A@C<sub>n</sub>. The new structures were related to promises of application in many areas of medicine as magnetic resonance imaging contrast agents, single-molecule magnets, electron-spin quantum computing, etc. [1–3]. This was the logical next step in the chemistry of carbon nanoclusters after the experimental discovery of the  $C_{60}$  molecule via mass spectrometry in 1985 [4,5], when the efforts to obtain new related systems with unique geometrical and electronic properties dramatically increased [6,7]. The A@C<sub>n</sub> structures described until present are based on the combination of different atoms/molecules and fullerenes of variable size. In 1999, Dorn and collaborators [8] reported on the synthesis of  $Sc_3N@I_h(7)-C_{80}$ , a member of a new family of EFs named nitride clusterfullerenes (NCFs). The NCFs encapsulate an M<sub>3</sub>N cluster where M usually is a rare-earth metal (Sc, Y or lanthanide), and the fullerene  $C_{2n}$  cage comprises different sizes (60 < 2n < 88) [9]. The M<sub>3</sub>N cluster is made up of M<sup>3+</sup> ions located at the vertices of a triangle, with the nitride  $N^{3-}$  in its center. Neither the  $M_3N$  cluster nor the  $C_{2n}-I_h$  cage can be synthesized separately due to their lability, whereas their combination leads to the stabilization of NCF via the electron transfer of six electrons from the cluster to the carbon cage as it is suggested by experimental and theoretical data [10-12].

One of the most interesting NCF types is the one where the lanthanide (Ln) species are present in the internal cluster  $(Ln_3N)$  due to the presence of 4*f*-electrons in their electronic configuration. The Ln-based combinations exhibit long spin relaxation lifetimes [13–15] and intriguing geometrical characteristics [16-18]. In some cases, the encapsulated Ln<sub>3</sub>N cluster suffers deformation from planar topology, depending on the Ln metal and cavity size. In addition, changes can be observed in the Ln-N bond length, the Ln-N-Ln angle between Ln and N species, as well as the  $C_{2n}$  cage geometry. Such changes have been reported for a broad variety of cavity sizes and species [2,3,9].

A particularly interesting and important family is Ln<sub>3</sub>N@C<sub>80</sub> NCFs. Here, one should emphasize that the previous experimental and theoretical research reports were limited to consideration of a few selected members of this family. The topology of the inner cluster has been analyzed both experimentally and theoretically. For example, this is the case for X-ray studies of Ln<sub>3</sub>N bending in Gd<sub>3</sub>N@C<sub>80</sub> [19], Tb<sub>3</sub>N@C<sub>80</sub> [20], Ho<sub>3</sub>N@C<sub>80</sub> [21] and  $Er_3N@C_{80}$  [21,22]. For Lu<sub>3</sub>N@C<sub>80</sub> [23], the encapsulated species has a planar structure, explained by the fact that Lu has the smallest ionic radius of all lanthanides. Interestingly, the X-ray structural analysis turns out to be a very challenging task, since until now it has been impossible to crystallize pure  $Ln_3N@C_{80}$  NCFs; the 'trick' which is used to obtain experimental geometries is to co-crystallize NCF with nickel (II) octaethylporphyrin [19-23]. The interaction between the two interacting units is very strong, so one cannot be totally sure whether it does not alter the  $Ln_3N@C_{80}$  geometry. In this regard, theoretical calculations within the framework of density functional theory (DFT) on isolated NCFs have an evident advantage in terms of excluding the external influence of the above sort. Within this area, one should mention several efforts to characterize the geometrical and electronic properties of NCFs. For example, Aparicio-Anglès et al. [16] showed for  $M_3N@C_{80}$  (where M = Sc, Y, La-Nd, Gd, Tb, Er, Tm and Lu), as well as for some other NCFs, that the bending of the M<sub>3</sub>N cluster increases as the atomic radius increases. Additionally, Schlesier et al. [24] used DFT calculations to analyze the geometrical features of some  $M_3N@C_{80}$  members (namely, mixed dysprosium–lanthanide  $DyM_2N@C_{80}$  and  $Dy_2MN@C_{80}$ , where M = Gd, Er, Tm and Lu), with a similar conclusion that the N atom is pushed out of the plane of Ln ions. On the other hand, to the best of our knowledge, the information on the frontier molecular orbital and spin density distribution in NCFs is rather scarce [7,25,26], with no published information on how they change during the step-by-step adding of 4*f*-electrons from La to Lu.

The changes in Ln–N bond lengths were analyzed previously both experimentally [21,27,28] and theoretically [16,24,29,30] for several NCFs. In particular, Olmstead et al. [21], Yang et al. [27] and Zuo et al. [28] reported the relevant experimental data for the Ho, Er, Dy and Tm NCFs. Aparicio-Anglès et al. [16] presented the DFT-calculated values for the Ln–N bond length for selected lanthanides including La-Nd, Gd, Tb and Er-Lu, and found it shortened as the ionic radius decreased. Both the above values and the ones reported by Schlesier et al. [24] and Gan et al. [29] for Gd, Dy, Er and Lu NCFs are in good agreement with the experimental parameters. On the other hand, again, there are no published DFT-calculated values comprising the complete lanthanide series.

In the present work we made an attempt to trace how the main geometrical and electronic properties of  $Ln_3N@C_{80}$  NCFs change for the complete series from lanthanum to lutetium, that is, during the step-by-step addition of 4*f*-electrons. The DFT functional of choice was PBE, same as in the most systematic related study by Aparicio-Anglès et al. [16] (though, periodic plane wave calculations instead of cluster calculations were employed in their work). Nevertheless, our approach has an important difference from the latter, where for Tb, Er, Tm and Lu ions, having very deep 4*f* orbitals, the *f* electrons were considered for simplicity to be in the core (that is, included in the pseudopotential); in our case, to fulfill our goal, all the 4*f* electrons had to be included explicitly.

#### 2. Results and Discussion

Selected geometrical parameters, orbital energies and other electronic properties for  $Ln_3N@C_{80}$  NCFs were calculated by using the DN and DND basis sets (Tables 1–4 and Figures 1–7). We considered the complete series of lanthanides with the purpose of tracing the effect of a step-by-step adding of 4*f*-electrons on the selected geometrical and electronic parameters of NCFs. It was also important to verify to what degree the results can vary when using the basis sets of different sizes.

**Table 1.** Three Ln–N bond lengths (in Å), Ln–N–Ln angles (in °), the pyramidalization ( $\theta$ ) angle of the Ln<sub>3</sub>N cluster (in °), Ln···Ln distances (in Å), and the N···C<sub>C80</sub> distance between N and the nearest C<sub>80</sub> aromatic ring (in Å) were calculated by using the DN basis set.

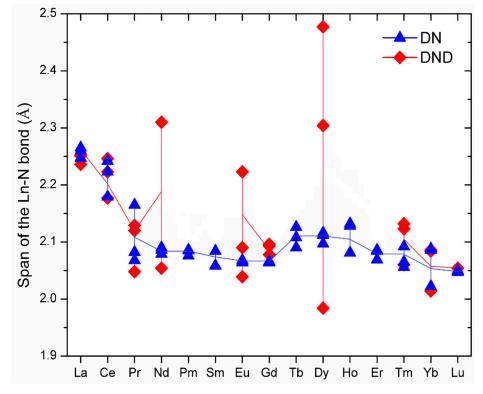
NFC	Ln–N (Å)	Ln–N–Ln (°)	θ (°)	Ln···Ln (Å)	N· · · C <sub>C80</sub> (Å)
La <sub>3</sub> N@C <sub>80</sub>	2.247, 2.260, 2.265	94.6, 95.8, 96.0	31.3	3.325, 3.345, 3.352	2.949
Ce <sub>3</sub> N@C <sub>80</sub>	2.179, 2.223, 2.242	90.6, 93.5, 95.2	33.0	3.174, 3.206, 3.266	2.962
Pr <sub>3</sub> N@C <sub>80</sub>	2.068, 2.082, 2.165	111.1, 113.5, 117.1	14.5	3.503, 3.540, 3.540	3.597
Nd <sub>3</sub> N@C <sub>80</sub>	2.079, 2.086, 2.090	115.5, 116.7, 117.9	10.6	3.533, 3.550, 3.568	3.758
Pm3N@C80	2.076, 2.085, 2.086	115.7, 117.0, 118.0	10.2	3.532, 3.550, 3.568	3.770
Sm3N@C80	2.058, 2.059, 2.084	117.7, 118.8, 119.9	6.4	3.543, 3.564, 3.566	3.925
Eu <sub>3</sub> N@C <sub>80</sub>	2.064, 2.065, 2.068	118.3, 119.2, 119.7	5.6	3.548, 3.564, 3.569	3.948
Gd <sub>3</sub> N@C <sub>80</sub>	2.064, 2.065, 2.066	118.2, 119.1, 119.5	5.9	3.546, 3.56, 3.567	3.957
Tb <sub>3</sub> N@C <sub>80</sub>	2.090, 2.108, 2.126	107.8, 109.4, 112.0	19.2	3.422, 3.441, 3.481	3.443
Dy3N@C80	2.097, 2.112, 2.116	109.2, 109.2, 111.2	18.4	3.431, 3.446, 3.477	3.461
Ho3N@C80	2.081, 2.128, 2.132	105.4, 109.3, 112.2	20.0	3.433, 3.496, 3.889	3.423
Er <sub>3</sub> N@C <sub>80</sub>	2.069, 2.084, 2.086	112.0, 112.7, 115.3	15.3	3.444, 3.472, 3.508	3.617
Tm3N@C80	2.056, 2.065, 2.092	110.8, 110.9, 118.6	15.1	3.415, 3.423, 3.543	3.593
Yb3N@C80	2.022, 2.086, 2.088	109.0, 114.9, 116.7	15.0	3.398, 3.464, 3.497	3.543
Lu3N@C80	2.047, 2.049, 2.050	118.8, 119.8, 120.0	3.9	3.528, 3.545, 3.548	3.985

**Table 2.** Three Ln–N bond lengths (in Å), Ln–N–Ln angles (in °), the pyramidalization ( $\theta$ ) angle of the Ln<sub>3</sub>N cluster (in °), Ln···Ln distances (in Å), and the N···C<sub>C80</sub> distance between N and the nearest C<sub>80</sub> aromatic ring (in Å) were calculated by using the DND basis set.

NFC	Ln–N (Å)	Ln–N–Ln (°)	θ (°)	Ln∙ · · Ln (Å)	$N \cdots C_{C80}$ (Å)
La <sub>3</sub> N@C <sub>80</sub>	2.236, 2.251, 2.252	95.6, 97.0, 97.2	30.4	3.360, 3.366, 3.337	2.983
Ce <sub>3</sub> N@C <sub>80</sub>	2.177, 2.223, 2.246	88.9, 92.7, 94.0	33.9	3.126, 3.184, 3.235	2.933
Pr <sub>3</sub> N@C <sub>80</sub>	2.048, 2.120, 2.129	111.1, 114.2, 115.7	14.8	3.503, 3.508, 3.530	3.563
Nd <sub>3</sub> N@C <sub>80</sub>	2.054, 2.310, 2.310	78.3, 99.8, 99.8	32.5	2.917, 3.343, 3.343	2.998
Eu <sub>3</sub> N@C <sub>80</sub>	2.039, 2.090, 2.223	105.6, 109.3, 118.4	17.6	3.446, 3.487, 3.548	3.462
Gd <sub>3</sub> N@C <sub>80</sub>	2.078, 2.093, 2.096	115.1, 115.3, 115.3	12.8	3.520, 3.526, 3.539	3.692
Dy3N@C80	1.984, 2.304, 2.477	63.3, 93.8, 99.4	35.9	3.275, 3.277, 2.513	2.935
Tm <sub>3</sub> N@C <sub>80</sub>	2.060, 2.123, 2.132	104.8, 107.8, 110.2	21.3	3.370, 3.388, 3.429	3.366
Yb3N@C80	2.014, 2.084, 2.086	117.0, 117.1, 108.0	14.4	3.373, 3.494, 3.497	3.551
Lu <sub>3</sub> N@C <sub>80</sub>	2.054, 2.054, 2.054	119.1, 119.6, 119.8	4.2	3.541, 3.550, 3.554	3.994

From La to Lu, the Ln<sub>3</sub>N cluster shows geometrical changes in Ln–N bond lengths, Ln–N–Ln angles, Ln··· Ln distances, N··· C<sub>C80</sub> distances (between N and the nearest C<sub>80</sub> ring), and the bending degree. These changes are due to the electrostatic repulsion between the fullerene cage and the Ln<sub>3</sub>N unit, which depends on the size of the metal ion (Tables 1 and 2; Figures 1 and 2). In the case of Ln–N bond lengths, the DN results are close to the ones obtained previously experimentally [19–22,27,28,31] and theoretically [22–24,29,32,33], with the present DFT values tending to be slightly higher (except for Nd [16], Gd and Lu [16,24,33] NCFs). The average calculated values for La and Ho NCFs reach a maximum discrepancy of 4.4% and 3.7% in comparison with the computational results of Gan et al. [29] and experimental values of Olmstead et al. [21], respectively. In general, the trend in Ln–N bond length changes can be described as follows: a notable contraction is observed from

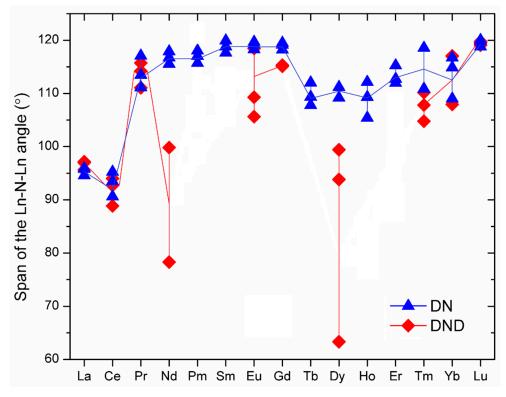
La<sub>3</sub>N@C<sub>80</sub> (2.247–2.265 Å; Table 1 and Figure 1) to Gd<sub>3</sub>N@C<sub>80</sub> (2.064–2.066 Å), then an increase for Tb<sub>3</sub>N@C<sub>80</sub> (2.090–2.126 Å), followed by a gradual decrease until the end of the series, that is, Lu<sub>3</sub>N@C<sub>80</sub> (2.047–2.05 Å). The largest span of Ln–N bond lengths was found for Pr (0.097 Å). The experimental values by Olmstead et al. [21], Yang et al. [27] and Zuo et al. [28] showed a similar reduction in Ln–N bond length in the series of Dy-Tm NCFs, as with the DFT-computed data reported previously [22–24,29,32,33]. In the present study, with the DN basis set, within the entire series two general situations can be observed: in most cases, only two Ln–N bonds have the same (or almost the same) length (La, Pr-Sm and Dy-Yb), whereas for the Eu, Gd and Lu NCFs, all three bond lengths are almost exactly equal.



**Figure 1.** Comparison of the three Ln–N bond lengths in Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DN and DND basis sets.

Regarding the geometries obtained with the DND basis sets (Table 2 and Figure 1), the general trend of Ln–N bond contraction can be seen in two (incomplete) series: La-Gd and especially Dy-Lu. However, this conclusion is not as straightforward as the one made for the DN basis set. The reason is that the use of the DND basis set results in too-rampant differences between the three Ln–N bond lengths, which are supposed to be more uniform, according to the experimental geometries ([21] and references therein). An especially 'conflictive' case is  $Dy_3N@C_{80}$ , for which the difference span reaches 0.493 Å, with the Dy–N bond lengths being 1.984, 2.304 and 2.477 Å; whereas the experimental ones are 2.017, 2.087 and 2.059 Å [21]; and the ones obtained with the DN basis set are 2.097, 2.112 and 2.116 Å (Table 1). The comparison of the two sets of results leads to a clear conclusion that, despite the prevailing idea that the use of a larger basis guarantees more realistic geometries (among other computed parameters), in the present case the smaller DN basis set is preferred to DND. In other words, for Ln<sub>3</sub>N@C<sub>80</sub> NCFs we observe the same tendency as in the case of lanthanide 'double-decker' bisphthalocyanines that have been reported recently [34]. Taken together with the results of another study on  $Ln@C_{60}$ endohedral fullerenes [35], where no substantial effect of the basis set (DN vs. DND) on the calculated parameters was found, one can strongly suggest the use of DN for further theoretical investigations of lanthanide-containing NCFs, polyazamacrocyclic and probably

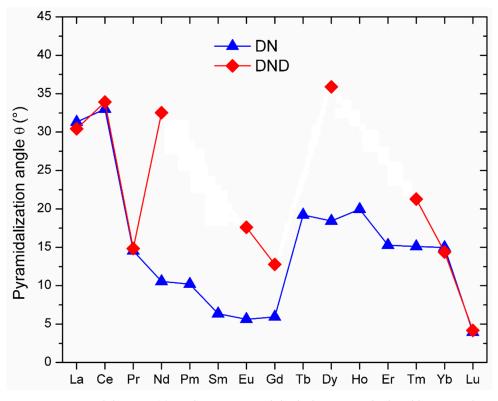
some other classes of Ln compounds (at least when using the DMol<sup>3</sup> module). Hereafter in this section we will place emphasis on the parameters obtained with the DN basis set for the same reasons. Coming back to the Ln–N bond lengths (Figure 1), one final observation to mention is that a reasonably good match between the DN and DND-computed values was obtained only for five lanthanides: La, Ce, Gd, Yb and Lu.



**Figure 2.** Comparison of the three Ln–N–Ln angles in Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DN and DND basis sets.

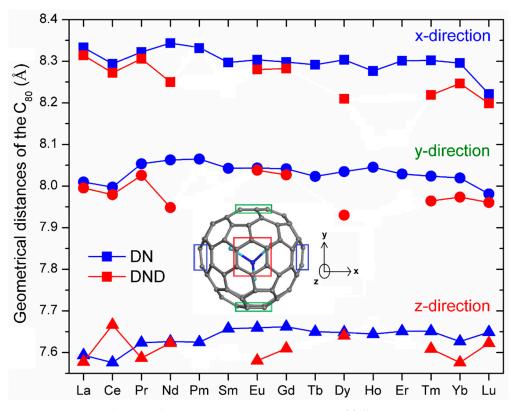
Another important and closely related geometric parameter is the Ln–N–Ln angle (Tables 1 and 2; Figure 2), which directly reflects the effect of Ln<sub>3</sub>N confinement within the C<sub>80</sub> cage. In the case of the DN basis set, the NCFs studied can be classified into two groups (though the difference between them is rather subtle): the three Ln–N–Ln values are notably different for Ce-Sm, Tb, Ho, Er and Yb; whereas for La, Eu, Gd, Dy, Tm and Lu, one value is different from the other two. The phenomenon of lanthanide contraction manifests itself in the general trend of increasing Ln–N–Ln angles in two separate series: Pr-Gd and Tb-Lu; in other words, the effect is not continuous from Pr to Lu. In addition, with the values for La and Ce NCFs, which have the largest Ln<sup>3+</sup> ionic radii, the angles are much smaller than for all other Ln<sub>3</sub>N@C<sub>80</sub> species, due to the strongest confinement. As a whole, this behavior correlates with the trends in the changes of Ln–N bond length; the larger Ln–N–Ln angles match the smaller Ln–N values. In the case of the DND basis set (Table 2; Figure 2), the changes in Ln–N–Ln angles from La to Lu are much more rampant, with very big differences between the three values for most NCFs, which is no surprise considering the results of the Ln–N bond length analysis presented above.

When comparing our results (obtained with the more reliable DN basis set) with that reported by other authors, for  $\text{Er}_3\text{N}@\text{C}_{80}$  we found an average Er-N-Er angle of 113.3°, which differs from the experimental value [22] by 6%. In the case of Tb<sub>3</sub>N@C<sub>80</sub> and Tm<sub>3</sub>N@C<sub>80</sub>, we obtained the average angles of 109.7° and 113.4°, respectively, which are up to 7% smaller than the values calculated by Aparicio-Anglès et al. [16], who used a very different theoretical approach (periodic calculations without inclusion of a dispersion correction and explicit consideration of the 4*f*-electrons).



**Figure 3.** Pyramidalization ( $\theta$ ) angle in Ln<sub>3</sub>N endohedral species calculated by using the DN and DND basis sets.

The bending of the  $Ln_3N$  cluster caused by the confinement within the  $C_{80}$  cage can also be characterized by the pyramidalization angle  $\theta$ , which is referred to as one of the most important geometric parameters for NCFs [36]. Several experimental [19–21,23,37] and theoretical [6,16,24,33] reports showed that the Ln<sub>3</sub>N cluster can acquire a planar or bent shape depending on the Ln metal and the size of the  $C_{2n}$  cavity. In most cases, large fullerene cages favor planar geometry [24]. The  $\theta$  values obtained are summarized in Tables 1 and 2 and Figure 3. The ones obtained with the DN basis set exhibit more uniform behavior compared to the rampant ones calculated with DND, in particular, a more uniform decrease, which is split into two series of Ce-Gd and Tb-Lu. Only for La, Ce, Yb and Lu did the DN and DND values approximately coincide, with the lowest  $\theta$  values of 3.9 and 4.2°, respectively, found for the Lu3N cluster. As a whole, the  $\theta$  angles presented here are larger compared to the experimental [19–21,28] and DFT-calculated ones [16–20,28] reported by other authors, and again, the DN basis set yields more realistic results. On the other hand, it is difficult to explain why the experimental  $\theta$  values for the central lanthanides Gd, Tb and Dy uniformly decrease (14.6, 12.6 and  $7.4^{\circ}$ , respectively; [21] and references therein), but the DN-computed angles jump from  $5.9^{\circ}$  for Gd to  $19.2^{\circ}$  and to  $18.4^{\circ}$  for Tb and Dy, respectively. One should remember, however, that the experimental values were obtained for NCF–Ni(II) octaethylporphyrin co-crystals (where solvent molecules are present as well), and it remains unclear how the solid-state environment can influence the geometry of encapsulated Ln<sub>3</sub>N species.



**Figure 4.** Average distances between opposite aromatic rings of fullerene cage in  $Ln_3N@C_{80}$  NCFs calculated by using the DN and DND basis sets. The *x*- (horizontal), *y*- (vertical) and *z*-directions (into the plain of view) are considered, as explained by the inset.

Other geometric parameters of interest are the Ln···Ln and N···C<sub>C80</sub> distances (between the N atom and the nearest hexagon/pentagon ring of C<sub>80</sub> cage) [16,29], which directly depend on the bending of the Ln<sub>3</sub>N cluster (Tables 1 and 2; Figures S1 and S2 of the Electronic Supplementary Information). For example, the N···C<sub>C80</sub> distances exhibit behavior opposite to the behavior of the  $\theta$  values, since the increase in pyramidalization results in a closer approach of the N atom to the fullerene cage.

The presence of encapsulated bulky  $Ln_3N$  clusters leads to slight geometrical changes in the geometry of the  $C_{80}$  cage. It can be expressed in different ways. As Figure 4 explains, the way we did it is to measure average distances between two opposite C–C bonds in the *x*-direction, between two opposite pentagonal rings in the *y*-direction, and between two opposite hexagons in the *z*-direction. As calculated with the DN basis set, the  $C_{80}$  cage exhibits relatively stable geometry with variations of 0.12 (*x*-direction), 0.08 (*y*-direction) and 0.09 Å (*z*-direction) between the maximum (La and Nd for *x*-; Pm for *y*-; Sm-Gd for *z*-direction) and minimum (Lu for *x*- and *y*-directions; Ce for *z*-direction) distances. As a general trend, the cage is compressed in the *x*- and *y*-directions, and at the same time slightly elongates in the *z*-direction, as the atomic number of lanthanides increases. Like for all the above parameters analyzed, the use of the DND basis sets yields very rampant values without clear trends.

NEC	E <sub>HOMO</sub> (eV)	E <sub>LUMO</sub> (eV)	E <sub>gap</sub> (eV)	Charge (e)		Spin (e) <sup>a</sup>	
NFC				Ln	Ν	Ln	Ν
La <sub>3</sub> N@C <sub>80</sub>	-6.012	-4.640	1.371	0.447, 0.467, 0.486	-0.925	0, 0, 0 (0)	0
Ce <sub>3</sub> N@C <sub>80</sub>	-5.818	-4.956	0.862	0.397, 0.415, 0.439	-0.809	0.948, 0.992, 1.044 (1)	0.042
Pr <sub>3</sub> N@C <sub>80</sub>	-5.500	-4.798	0.702	0.303, 0.352, 0.353	-0.453	2.063, 2.072, 2.103 (2)	0.051
Nd <sub>3</sub> N@C <sub>80</sub>	-5.923	-4.983	0.941	0.160, 0.178, 0.180	-0.634	3.147, 3.162, 3.179 (3)	0.179
Pm <sub>3</sub> N@C <sub>80</sub>	-5.976	-5.248	0.728	0.054, 0.070, 0.080	-0.696	4.223, 4.256, 4.268 (4)	0.066
Sm <sub>3</sub> N@C <sub>80</sub>	-5.956	-5.660	0.296	0.424, 0.434, 0.463	-0.837	5.441, 5.446, 5.474 (5)	0.446
Eu <sub>3</sub> N@C <sub>80</sub>	-5.940	-5.868	0.072	0.426, 0.441, 0.469	-0.857	6.646, 6.649, 6.661 (6)	0.627
Gd <sub>3</sub> N@C <sub>80</sub>	-6.046	-4.589	1.457	0.496, 0.512, 0.527	-0.992	6.899, 6.934, 6.935 (7)	0.028
Tb <sub>3</sub> N@C <sub>80</sub>	-5.979	-5.791	0.189	0.409, 0.430, 0.454	-0.827	5.519, 5.520, 5.530 (6)	0.198
Dy <sub>3</sub> N@C <sub>80</sub>	-5.974	-5.900	0.074	0.412, 0.412, 0.435	-0.824	4.338, 4.387, 4.409 (5)	0.109
Ho <sub>3</sub> N@C <sub>80</sub>	-5.952	-5.882	0.070	0.457, 0.499, 0.505	-0.862	3.350, 3.360, 3.384 (4)	0.058
Er <sub>3</sub> N@C <sub>80</sub>	-5.961	-5.872	0.089	0.467, 0.487, 0.542	-0.878	2.377, 2.393, 2.421 (3)	0.083
Tm <sub>3</sub> N@C <sub>80</sub>	-5.960	-5.835	0.125	0.368, 0.375, 0.420	-0.822	1.263, 1.268, 1.273 (2)	0.005
Yb <sub>3</sub> N@C <sub>80</sub>	-5.942	-5.859	0.083	0.417, 0.420, 0.445	-0.855	0.012, 0.099, 0.110, (1)	0.053
Lu3N@C80	-6.034	-4.506	1.528	0.557, 0.584, 0.595	-1.074	0, 0, 0 (0)	0

**Table 3.** HOMO, LUMO and HOMO-LUMO gap energies (in eV), as well as the charge and spin values (in *e*; formal spin of Ln atom in parenthesis) for  $Ln_3N@C_{80}$  NCFs calculated by using the DN basis set.

<sup>a</sup> Absolute values.

Tables 3 and 4 summarize the calculated HOMO, LUMO and HOMO-LUMO gap ( $E_{gap}$ ) energies, as well as the charge and spin (from Mulliken population analysis) of Ln and N atoms as calculated with the DN and DND basis sets. For the frontier orbital energies, obtained with either the DN or DND basis set, no clear trend is observed (Tables 3 and 4), with La, Gd and Lu NCFs having similar values for LUMO energies with both basis sets. The DN-calculated  $E_{gap}$  energies (Table 3) tend to decrease in the entire NCF series, with the exception of the 'jumping-back' values for Nd, Gd and Lu derivatives. These species with totally empty (La), half-filled (Gd) and totally filled (Lu) 4*f* orbitals exhibit the highest gap values of 1.371, 1.457 and 1.528 eV, respectively; these results are in good agreement with the DFT-calculated values by Zhu et al. [38]. As a whole,  $E_{gap}$  spans from 0.070 (Ho<sub>3</sub>N@C<sub>80</sub>) to 1.528 eV (Lu<sub>3</sub>N@C<sub>80</sub>) with DN, and from 0.084 (Dy<sub>3</sub>N@C<sub>80</sub>) to 1.485 eV (La and Lu<sub>3</sub>N@C<sub>80</sub>), with no clear trend, when the DND basis set is used (Table 4). Here, it is appropriate to mention that the GGA functionals (including PBE) tend to underestimate HOMO-LUMO gap energies, and we present the values obtained simply to illustrate their relative behavior from La to Lu.

**Table 4.** HOMO, LUMO and HOMO-LUMO gap energies (in eV), as well as the charge and spin values (in *e*; formal spin of Ln atom in parenthesis) for Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DND basis set.

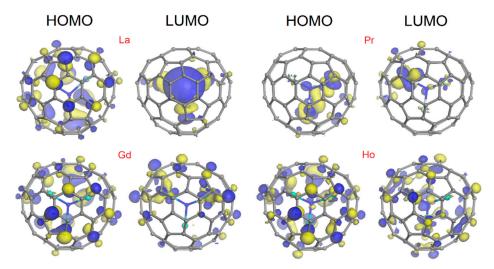
NFC	E <sub>HOMO</sub> (eV)	E <sub>LUMO</sub> (eV)	E <sub>gap</sub> (eV)	Charge (e)		Spin (e) <sup>a</sup>	
				Ln	Ν	Ln	Ν
La3N@C80	-5.769	-4.284	1.485	0.962, 0.962, 0.962	-1.021	0, 0, 0 (0)	0
Ce <sub>3</sub> N@C <sub>80</sub>	-5.459	-4.695	0.764	0.829, 0.837, 0.838	-0.889	1.052, 1.079, 1.101 (1)	0.116
Pr <sub>3</sub> N@C <sub>80</sub>	-5.214	-4.537	0.677	0.050, 0.082, 0.086	-0.758	2.047, 2.047, 2.065 (2)	0.051
Nd <sub>3</sub> N@C <sub>80</sub>	-5.579	-5.140	0.438	0.060, 0.164, 0.174	-0.545	3.185, 3.220, 3.297 (3)	0.127
Eu <sub>3</sub> N@C <sub>80</sub>	-5.649	-5.558	0.090	0.888, 0.907, 0.907	-0.933	6.567, 6.608, 6.703 (6)	0.243

E <sub>HOMO</sub> (eV)	E <sub>LUMO</sub> (eV)	E <sub>gap</sub> (eV)	Charge (e)		Spin (e) <sup>a</sup>	
			Ln	Ν	Ln	Ν
-5.739	-4.365	1.373	0.854, 0.857, 0.857	-1.136	6.866, 6.910, 6.913 (7)	0.031
-5.634	-5.550	0.084	0.696, 0.867, 0.901	-0.742	4.175, 4.159, 4.446 (5)	0.261
-5.673	-5.573	0.100	0.819, 0.823, 0.832	-0.964	1.266, 1.282, 1.288 (2)	0.114
-5.650	-5.563	0.088	0.845, 0.862, 0.862	-1.012	0.014, 0.114, 0.122 (1)	0.085
-5.758	-4.272	1.485	0.882, 0.884, 0.884	-1.215	0, 0, 0 (0)	0
	-5.739 -5.634 -5.673 -5.650	$\begin{array}{ccc} -5.739 & -4.365 \\ -5.634 & -5.550 \\ -5.673 & -5.573 \\ -5.650 & -5.563 \end{array}$	-5.739         -4.365         1.373           -5.634         -5.550         0.084           -5.673         -5.573         0.100           -5.650         -5.563         0.088	$E_{\rm HOMO}$ (eV) $E_{\rm LUMO}$ (eV) $E_{\rm gap}$ (eV) $Ln$ -5.739         -4.365         1.373         0.854, 0.857, 0.857           -5.634         -5.550         0.084         0.696, 0.867, 0.901           -5.673         -5.573         0.100         0.819, 0.823, 0.832           -5.650         -5.563         0.088         0.845, 0.862, 0.862	$E_{\rm HOMO}$ (eV) $E_{\rm LUMO}$ (eV) $E_{\rm gap}$ (eV) $Ln$ N           -5.739         -4.365         1.373         0.854, 0.857, 0.857         -1.136           -5.634         -5.550         0.084         0.696, 0.867, 0.901         -0.742           -5.673         -5.573         0.100         0.819, 0.823, 0.832         -0.964           -5.650         -5.563         0.088         0.845, 0.862, 0.862         -1.012	$E_{\rm HOMO}$ (eV) $E_{\rm LUMO}$ (eV) $E_{\rm gap}$ (eV) $I$ $I$ $-5.739$ $-4.365$ $1.373$ $0.854, 0.857, 0.857$ $-1.136$ $6.866, 6.910, 6.913$ (7) $-5.634$ $-5.550$ $0.084$ $0.696, 0.867, 0.901$ $-0.742$ $4.175, 4.159, 4.446$ (5) $-5.673$ $-5.573$ $0.100$ $0.819, 0.823, 0.832$ $-0.964$ $1.266, 1.282, 1.288$ (2) $-5.650$ $-5.563$ $0.088$ $0.845, 0.862, 0.862$ $-1.012$ $0.014, 0.114, 0.122$ (1)

Table 4. Cont.

<sup>*a*</sup> Absolute values.

As we mentioned in Introduction, the  $I_h$ -C<sub>80</sub> fullerene is a very reactive molecule per se [1,3], but its combination with some other species  $(M_3N, M_2, etc.)$  gives stable complexes [1,2,9,10,39,40]; this stabilizing effect is due to electron transfer from the encapsulated species to the carbon cage. Formally, in the case of rare-earth derivatives  $M_3N@C_{80}$ , the  $M_3N$  cluster is supposed to donate six electrons to  $C_{80}$ . Nevertheless, it turns out that the "actual" charges are very different from "formal" charges, which are integers and imply that the cluster-cage bonding is purely ionic. This aspect was addressed in several theoretical studies (on representative  $Sc_3N@C_{80}$ ), which found that the calculated net  $Sc_3N$  charges are also much smaller than the values expected for the purely ionic bonding [41,42]; for example, around +1, either from the Mulliken or Hirshfeld population analysis [42]. This effect is referred to as "back-donation" [43,44]. Additionally, in our calculations, the actual positive charge on Ln atoms (Tables 3 and 4; from the Mulliken population analysis) was much lower than the "formal" charge of +3; in particular, it spans from 0.054-0.080 e (Pm<sub>3</sub>N) to 0.557–0.595 e (Lu<sub>3</sub>N) with the DN basis set, and from 0.050–0.086 e (Pr<sub>3</sub>N) to 0.962 e (La<sub>3</sub>N) with DND. For the N atom, the values vary from -0.453 e (Pr<sub>3</sub>N) to -1.074 e (Lu<sub>3</sub>N) with DN, and from -0.545 e (Nd<sub>3</sub>N) to -1.215 e (Lu<sub>3</sub>N) with DND. The DN-calculated absolute values are generally lower than the DND ones. It is interesting and important to mention the observation that the use of a smaller basis set produces more reliable results for the Mulliken population analysis [45–47].

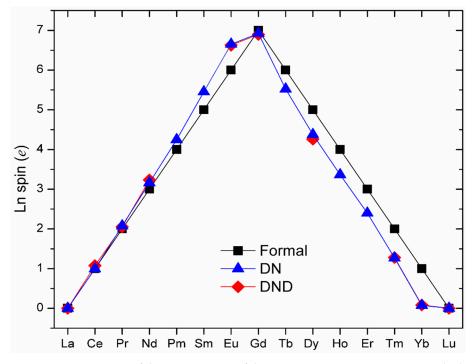


**Figure 5.** Selected patterns of HOMO-LUMO distributions (isosurfaces at 0.03 a.u.) in  $Ln_3N@C_{80}$  NCFs calculated by using the DN basis set.

Figure 5 shows the selected patterns of the HOMO and LUMO distribution obtained with the DN basis set. After analyzing these plots for the complete NCF series, it was not possible to classify them in a simple way due to a large variability in the results (with both basis sets). Here, we show the four most distinctive cases (La, Pr, Gd and Ho NCFs) obtained with the DN basis set (the reader can find all the plots for the fifteen NCFs in

Electronic Supplementary Information, in Figures S3–S6; with the aim of completeness of the graphical presentation, both the front and the side views are included.) As can be seen in Figure 5, the HOMO lobes in the case of La and Gd NFCs are very similar: they are localized on the  $C_{80}$  cage, with no apparent contribution from the  $Ln_3N$  cluster. In the case of Ho NCF, the HOMO distribution on fullerene cage is rather similar to the one found for La and Gd complexes, but a minor contribution from the Ho<sub>3</sub>N cluster can be observed as well. For  $Pr_3N@C_{80}$ , the HOMO lobes are localized mostly on two Pr atoms, whereas the contribution from the  $C_{80}$  cage is minimal. For the LUMO density, the lobes on Ln and N atoms are of highly variable size; of the ones presented in Figure 5, the largest ones can be seen for the  $La_3N$  cluster. The LUMO lobes on the fullerene cage almost vanish for La, Ce, Nd, Pm, Sm and Tb NCFs. The results obtained with DND (Figures S5–S6) only partially match the ones calculated by using the DN basis set.

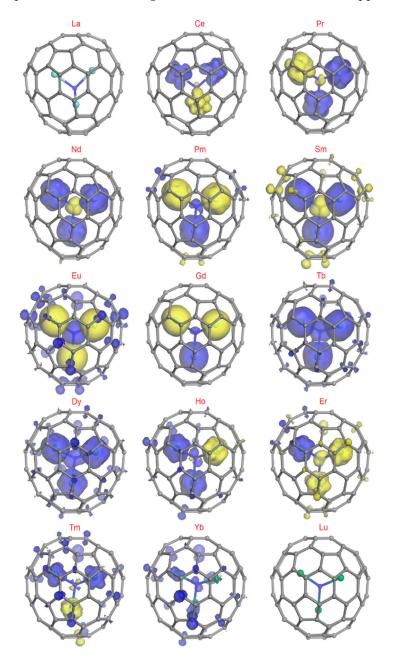
As compared to the formal spin for isolated  $Ln^{3+}$  ions, the results obtained with the DN and DND basis sets (Tables 3 and 4; Figure 6) show the possibility of spin transfer both to and from lanthanide species. In the DN data set (Table 3), spin depletion from Ln can be observed for Ce (but from two cerium atoms only) and the late lanthanides Gd-Yb (by up to 0.890–0.988 *e* for Yb, compared to to the formal spin of 1 *e* for isolated Yb<sup>3+</sup>); this can be also seen from Figure 6 (here we plotted the average values for simplicity). The early lanthanides Pr-Eu tend to acquire additional spin, with the strongest increase by 0.646–0.661 *e* observed for Eu (compared to the formal spin of 6 *e* for isolated Eu<sup>3+</sup>). For La and Lu ions with totally empty and totally filled 4*f* orbitals, respectively, the spin remains to be zero. The limited available values calculated by using the DND basis set (Table 4 and Figure 6) follow a similar trend.



**Figure 6.** Comparison of the average spin of three Ln ions in  $Ln_3N@C_{80}$  NCFs calculated by using the DN and DND basis sets with the formal spin of isolated  $Ln^{3+}$  ions.

The spin density plots for  $Ln_3N@C_{80}$  NCFs presented in Figure 7 (the DN data set) cannot serve as a similar quantitative measure of spin transfer, but instead are more illustrative, especially in order to show variability in spin alignment. In the most simplistic terms of spin distribution, they can be classified into three general cases. The first one includes La and Lu NCFs (with the closed-shell electronic configuration of Ln species), where no unpaired electrons can be found on either the  $Ln_3N$  cluster or  $C_{80}$  cage (the corresponding side views are shown in Figure S7). In the case of Ce-Nd and Gd NCFs, the main contribu-

tion comes from the  $Ln_3N$  cluster, where the largest lobes are localized on metal species. For the remaining NCFs (Pm-Eu, Tb-Yb), spin density can be found on both the cluster and fullerene cage. In the calculations with the DND basis set, qualitatively similar spin density plots were obtained (Figures S8 and S9 of the Electronic Supplementary Information).



**Figure 7.** Spin density plots (isosurfaces at 0.02 a.u.) for Ln<sub>3</sub>N@C<sub>80</sub> complexes as calculated by using the DN basis set. Blue and yellow lobes correspond to spin-up and spin-down electrons, respectively.

As regards the spin-up and spin-down directions, they turned out to be broadly variable as well. When considering  $Ln_3N$  cluster only, all the NCFs can be classified into the following groups: (1) the spin directions of all four atoms coincide (spin-up) for Tb, Dy and Yb; (2) the spin directions of three Ln atoms coincide, but for the N atom it is the opposite, which is observed for Nd, Sm and Eu; (3) the spin directions of Ln atoms differ (2 + 1) for all the remaining NCFs (Ce, Pr, Pm, Gd, Ho, Er and Tm). One should also note that in three cases of Tb, Dy and Yb derivatives, the spin directions of  $Ln_3N$  atoms (spin-up) are the same as those of the small lobes found on the carbon atoms of the  $C_{80}$  cage.

### 3. Computational Methods

The theoretical analysis of the geometries and electronic properties of  $Ln_3N@C_{80}$  NCFs was performed by using the numerical-based DFT module DMol<sup>3</sup>, which is a part of the Materials Studio 8.0 software package [48–51]. The Perdew–Burke–Ernzerhof (PBE) general gradient approximation functional [52] was employed in combination with the empirical dispersion correction developed by Grimme [53], usually referred to as PBE-D (or PBE-D2). The previous computational method was successfully used in our previous studies involving C<sub>80</sub>-based dyads [25,54], as well as in numerous studies by other research groups on systems that include metal species and carbon nanoclusters.

The DN (with no polarization functions added) and DND (with a polarization *d*-function added on all non-hydrogen atoms) double-numerical basis sets were employed, which are equivalent to the 6-31G and 6-31G(d) Pople-type basis sets. The full geometry optimization and the calculation of the electronic parameters were performed with the DFT semi-core pseudopotentials (DSPPs), including relativistic and spin-orbit coupling effects, whose account is important due to the presence of the lanthanide species. In the calculations with the DN basis set, the real space cutoff (defined by the presence of lanthanide species) was 4.3 Å and a DND of 5.0 Å. The convergence criteria were as follows: For the DN basis set, an energy gradient of  $10^{-4}$  Ha/Å, maximum force of 0.02 Ha/Å, maximum displacement of 0.05 Å, and SCF tolerance of  $10^{-4}$  Ha/Å, maximum displacement of 0.05 Å, and SCF tolerance of 0.004 Ha/Å, maximum displacement of 0.05 Å, and SCF tolerance of 0.004 Ha/Å, maximum displacement of 0.05 Å.

The existence of 4*f* orbitals results in a high number of degenerate states near the Fermi level, which has a serious negative impact on the self-consistent field (SCF) convergence. Thermal smearing was employed as an indispensable tool to solve the (almost always inevitable) SCF convergence problems according to the general protocol explained in the previous studies [55,56]. It was found that the use of a very low value of 0.0001 Ha (equivalent temperature of 31.6 K) yields stable and consistent results. At the same time, even with the use of thermal smearing tools, only for the DN basis set was a successful convergence afforded for all  $Ln_3N@C_{80}$  NCFs. With DND, we were unable to finish the calculations for Pm-, Sm-, Tb-, Ho- and Er-containing systems.

### 4. Conclusions

The most important results can be summarized as follows.

(1) The geometry optimization and electronic parameter calculation for the series of fifteen  $Ln_3N@C_{80}$  NCFs (Ln = La-Lu) was possible to complete only with the DN basis set. The use of DND basis sets leads to SCF convergence problems, due to which it was impossible to obtain the results for Pm-, Sm-, Tb-, Ho- and Er-containing systems. Furthermore, within the NCFs for which the computations seemed to be successful, too-rampant behavior was observed for most important geometric parameters (including the Ln–N bond lengths, Ln–N–Ln angles, and pyramidalization angle  $\theta$ ), which is supposed to be more uniform and logical.

(2) For the frontier orbital energies, obtained with either the DN or DND basis set, no clear trend is observed. The DN-calculated  $E_{gap}$  energies tend to decrease in the entire NCF series, with the exception of the 'jumping-back' values for Nd, Gd and Lu derivatives. These species with totally empty (La), half-filled (Gd) and totally filled (Lu) 4*f* orbitals exhibit the highest gap values of 1.371, 1.457 and 1.528 eV, respectively, which are in good agreement with the DFT results reported elsewhere. The corresponding HOMO and LUMO distribution patterns are highly variable.

(3) The calculated "actual" positive charge on Ln atoms, as estimated from the Mulliken population analysis, is much lower than the "formal" charge of +3; it spans from 0.054–0.080 *e* (Pm<sub>3</sub>N) to 0.557–0.595 *e* (Lu<sub>3</sub>N) with the DN basis set, and from 0.050–0.086 *e* (Pr<sub>3</sub>N) to 0.962 *e* (La<sub>3</sub>N) with DND. Similarly, for the N atom, the values vary from -0.453 e (Pr<sub>3</sub>N) to -1.074 e (Lu<sub>3</sub>N) with DN, and from -0.545 e (Nd<sub>3</sub>N) to -1.215 e (Lu<sub>3</sub>N) with DND. These results are in good agreement with the known effect of "back-donation".

(4) As compared to the formal spin for isolated  $Ln^{3+}$  ions, the results obtained show the possibility of spin transfer both to and from lanthanide species. In the DN data set, spin depletion from Ln can be observed for Ce (but from two cerium atoms only) and the late lanthanides Gd-Yb, with it being up to 0.890–0.988 *e* for Yb. The early lanthanides Pr-Eu tend to acquire additional spin, with the strongest increase by 0.646–0.661 *e* observed for Eu. For La and Lu ions with totally empty and totally filled 4*f* orbitals, respectively, the spin remains to be zero. The spin density plots reveal interesting features in terms of localization of unpaired electrons, as well as spin alignment.

Supplementary Materials: Supplementary data associated with this article can be found in the online version at https://www.mdpi.com/article/10.3390/inorganics11050223/s1. Figure S1: Comparison of the Ln··· Ln distance (Å) in Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DN and DND basis sets; Figure S2: Comparison of the  $N \cdots C_{C80}$  distance (Å) of the Ln<sub>3</sub>N@C<sub>80</sub> complexes calculated by using the DN and DND basis sets; Figure S3: Complete patterns of HOMO-LUMO distribution (front views with isosurfaces at 0.03 a.u.) in  $Ln_3N@C_{80}$  NCFs calculated by using the DN basis set; Figure S4: Complete patterns of HOMO-LUMO distribution (isosurfaces at 0.03 a.u.) in Ln<sub>3</sub>N@C<sub>80</sub> NCFs but rotated at an angle of 90° to the left (side view) with respect to Figure S3; Figure S5: Patterns of HOMO-LUMO distribution (front views with isosurfaces at 0.03 a.u.) in Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DND basis set; Figure S6: Patterns of HOMO-LUMO distribution (isosurfaces at 0.03 a.u.) in  $Ln_3N@C_{80}$  NCFs but rotated at an angle of 90° to the left (side views) with respect to Figure S5; Figure S7: Spin density plots (isosurfaces at 0.02 a.u.) for Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DN basis set but rotated an angle of  $90^{\circ}$  to the left (side views) with respect to Figure 7. Blue and yellow lobes correspond to spin-up and spin-down electrons, respectively; Figure S8: Spin density plots (isosurfaces at 0.02 a.u.) for  $Ln_3N@C_{80}$  NCFs were calculated by using the DND basis set. Blue and yellow lobes correspond to spin-up and spin-down electrons, respectively; Figure S9: Spin density plots (isosurfaces at 0.02 a.u.) for Ln<sub>3</sub>N@C<sub>80</sub> NCFs calculated by using the DND basis set but rotated an angle of  $90^{\circ}$  to the left (side views) with respect to Figure S8.

**Author Contributions:** Conceptualization, V.A.B.; methodology, V.A.B.; validation, C.M.-F.; formal analysis, C.M.-F.; investigation, C.M.-F.; data curation, C.M.-F.; writing—original draft preparation, C.M.-F.; writing—review and editing, V.A.B.; visualization, C.M.-F.; supervision, V.A.B.; funding acquisition, V.A.B. All authors have read and agreed to the published version of the manuscript.

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