



## Supplementary Material

# Synthesis of Porous Hierarchical In<sub>2</sub>O<sub>3</sub> Nanostructures with High Methane Sensing Property at Low Working Temperature

Huiju Zhang <sup>1</sup>, Jiangnan Chang <sup>1</sup>, Yan Wang <sup>2</sup> and Jianliang Cao <sup>1,\*</sup><sup>1</sup> College of Chemistry and Chemical Engineering, Henan Polytechnic University, Jiaozuo 454000, China<sup>2</sup> State Collaborative Innovation Center of Coal Work Safety and Clean-Efficiency Utilization, Henan Polytechnic University, Jiaozuo 454000, China

\* Correspondence: caojianliang@hpu.edu.cn

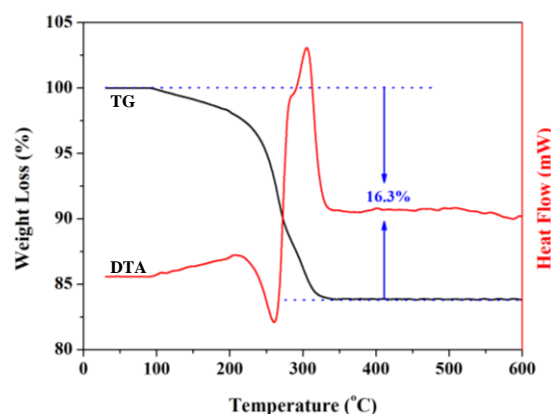


Figure S1. TG-DTA curves of the prepared In(OH)<sub>3</sub>-precursor.

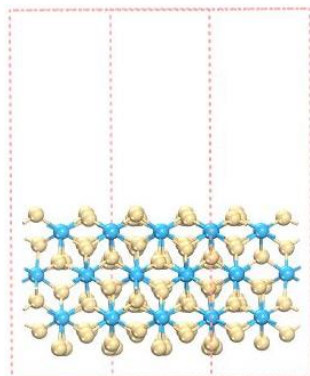
In order to determine the calcination temperature for the synthesis of In<sub>2</sub>O<sub>3</sub>, the precursor In(OH)<sub>3</sub> synthesized by hydrothermal method was tested by thermal analysis. As shown in Figure S1, with the temperature increasing from 30 to 600 °C, a big weight loss (16.3 %) was observed in the TG curve, being consistence with the theoretical weight loss (16.28%) for the decomposition of In(OH)<sub>3</sub>. Such a weight loss corresponds two continuous phase transformation processes, a endothermic process at around 265 °C and a exothermic process at around 305 °C. When the temperature is higher than 340 °C, the TG and DTA curves become of flat, indicating the formation of thermally stable In<sub>2</sub>O<sub>3</sub>. Based on above results, the calcination temperature of the precursor In(OH)<sub>3</sub> was set at 400 °C to ensure the formation of In<sub>2</sub>O<sub>3</sub> with good crystallinity.

The DFT computations were carried out using Cambridge Sequential Total Energy Package (CASTEP) program package [1]. At the level of the Generalized Gradient Approximation (GGA), the exchange-collection functional of Perdew–Burke–Ernzerhof (PBE) was employed [2,3]. We used a 400 eV plane-wave cutoff and a 3 × 3 × 1 k-point Monkhorst-Pack mesh in the geometry optimization and energy calculations. The vacuum between slabs was set up to 15 Å to preclude the interaction of the slabs and a slab model of ten-layer-thick In<sub>2</sub>O<sub>3</sub> (110) was used. The super cell is expanded to 2 × 2, then the bottom six layers are fixed, and the top four layers are fully relaxed. The calculated model of In<sub>2</sub>O<sub>3</sub> (110) is obtained as shown in Figure S2. Oxygen defects are bound to form on the surface of In<sub>2</sub>O<sub>3</sub> after high temperature heat treatment during sample preparation [4]. Therefore, calculate the adsorption capacity of oxygen molecules and methane molecules on the surface of In<sub>2</sub>O<sub>3</sub> (110) containing oxygen defects. It is calculated that the oxygen-containing defect In<sub>2</sub>O<sub>3</sub> (110) surface has an adsorption energy of −2.492 for oxygen molecules, indicating that the In<sub>2</sub>O<sub>3</sub> (110) surface can adsorb oxygen molecules spontaneously and has a strong adsorption capacity. The adsorption energy ( $\Delta E_{\text{ads}}$ ) of CH<sub>4</sub> molecule adsorbed on In<sub>2</sub>O<sub>3</sub> (110) defect surface is calculated as follows:

$$\Delta E_{\text{ads}} = E_{\text{ads}} - E_{\text{gas}} - E_{\text{def}}$$

where  $E_{\text{ads}}$  represents the total energy of CH<sub>4</sub> molecule adsorbed on In<sub>2</sub>O<sub>3</sub> (110) defect surface,  $E_{\text{gas}}$  and  $E_{\text{def}}$  represent the total energy of gas molecule (CH<sub>4</sub>) and In<sub>2</sub>O<sub>3</sub> (110) defect surface before adsorption, respectively. In general, the

negative value of  $\Delta E_{\text{ads}}$  indicates that the gas molecule adsorption reaction is exothermic and can occur spontaneously under certain conditions, so the adsorption system is energy stable [5,6].



**Figure S2.** The calculated model of  $\text{In}_2\text{O}_3$  (110), (Blue ball represents Sn atom, yellow ball represents O atom).

## References

1. Payne, M.C.; Teter, M.P.; Allan, D.C.; Arias, T.; Joannopoulos, J. Iterative minimization techniques for ab initio total-energy calculations: molecular dynamics and conjugate gradients. *Rev. Mod. Phys.* **1992**, *64*, 1045–1097.
2. Perdew, J.P.; Chevary, J.; Vosko, S.; Jackson, K.A.; Pederson, M.R.; Singh, D.; Fiolhais, C. Atoms, molecules, solids, and surfaces: applications of the generalized gradient approximation for exchange and correlation. *Phys. Rev. B* **1992**, *46*, 6671–6687.
3. White, J.; Bird, D. Implementation of gradient-corrected exchange-correlation potentials in Car-Parrinello total-energy calculations. *Phys. Rev. B* **1994**, *50*, 4954–4957.
4. Park, S.; Ahn, H.-S.; Lee, C.-K.; Kim, H.; Jin, H.; Lee, H.-S.; Seo, S.; Yu, J.; Han, S. Interaction and ordering of vacancy defects in NiO. *Phys. Rev. B* **2008**, *77*, 134103.
5. Wu, G.X.; Zhang, J.; Wu, Y.; Li, Q.; Chou, K.; Bao, X. Adsorption and dissociation of hydrogen on MgO surface: a first-principles study. *J. Alloys Compd.* **2009**, *480*, 788–793.
6. Zeng, W.; Liu, T.M.; Liu, D.J.; Han, E.J. Hydrogen sensing and mechanism of M-doped  $\text{SnO}_2$  ( $\text{M} = \text{Cr}^{3+}$ ,  $\text{Cu}^{2+}$  and  $\text{Pd}^{2+}$ ) nanocomposite. *Sens. Actuators B* **2011**, *160*, 455–462.