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# Influence of Precession Electron Diffraction Parameters and Energy Filtering on Reduced Density Function Analysis of Thin Amorphous Silica Films—Implications for Structural Studies

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**Abstract:** We investigated the influence of precession angle, energy filtering and sample thickness on the structural parameters of amorphous SiO<sub>2</sub> thin films from the electron reduced density functions obtained by applying precession electron diffraction. The results demonstrate that the peak positions in the electron reduced density functions are generally insensitive to the studied experimental conditions, while both precession angle and energy filtering influence peak heights considerably. It is also shown that introducing precession with small angles of up to 2 degrees and energy filtering results in higher coordination numbers that are closer to the expected theoretical values of 4 and 2 for Si and O, respectively, for data obtained from a thicker sample.

**Keywords:** transmission electron microscopy; precession electron diffraction; amorphous silica; reduced density function; pair distribution function



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## 1. Introduction

Amorphous structures are an important class of materials for a number of applications, e.g., optical fibres [1] and biological scaffolds [2]. Understanding these materials' properties and building meaningful structure–property correlations depends on the accurate determination of atomistic structures. Due to the structural complexity of amorphous materials, describing an averaged building block or structure is one of the more generally used approaches to assess these structures [3]. Diffraction techniques are among the experimental techniques that are suitable for obtaining an averaged structure [4–6]. In this approach, the diffraction pattern can be converted into a pair distribution function (PDF) that describes an averaged structure, and this can be subsequently used to build an atomistic model.

Diffraction techniques have been applied extensively to determine atomic structures using various types of radiation including X-rays, neutrons and electrons. The major advantage of electron diffraction is that electrons have larger scattering cross sections (than X-rays), with the ability to form fine probes of controlled geometry. This enables the observation of the local atomic structure of materials from very small volumes when compared to those required for X-ray and neutron diffraction [7]. In addition, the imaging mode in a transmission electron microscope (TEM) allows for the straightforward selection of the sample area of interest from where the diffraction data are collected.

Electron reduced density function (eRDF) analysis, which is equivalent to pair distribution function (PDF) analysis, is a direct way to characterise the atomistic structure of amorphous thin films by recording and analysing diffraction patterns in a TEM.

The kinematic scattering approximation is often used when comparing atomistic models to experimental electron diffraction data, a practice which assumes that only one elastic scattering event occurs for each electron passing through the specimen [8]. However, in practice, the electrons can be scattered multiple times and scattering events can be inelastic, with characteristic energy losses.

The contribution of inelastic scattering can be removed by energy filtering [9]. Additionally, precession electron diffraction (PED) has been shown to reduce the effect of multiple scattering when interpreting diffraction data from crystalline samples [10–12].

The contributions of both multiple and inelastic scattering are expected to increase with sample thickness and therefore this is an important factor that will influence the results of an RDF analysis based on the kinematic scattering approximation.

Whilst eRDF analysis has been extensively applied in various studies [13–17], the effect of various experimental parameters on the eRDF analysis and especially on the refined atomistic models has not been systematically investigated for amorphous materials. The purpose of this study is to quantify the influence of precession angle in PED, together with the effects of energy filtering and sample thickness on the eRDF of amorphous SiO<sub>2</sub> thin films and, subsequently, on the determined structural parameters.

#### Electron Reduced Density Function (eRDF)

The total electron scattering intensity,  $I(q)$ , for a particular arrangement of different atomic species for all orientations of the sample can be calculated using Equation (1) [18], where  $f_i(q)$  is the atomic scattering factor,  $q$  is the scattering vector and  $r_{ij} = |r_i - r_j|$  is the distance between two atoms.

$$I(q) = \sum_i f_i^2(q) + \sum_{i \neq j} \sum_j f_i(q) f_j(q) \sin(qr_{ij}) / (qr_{ij}) \quad (1)$$

Following Cockayne [7], this function can be rewritten in terms of a pair distribution function,  $g(r) = 4\pi r^2 \rho dr$ , by integrating over the sample volume which is directly related to the local average density, (Equation (2)).

$$I(q) = \sum_i f_i^2(q) + 4\pi \left( \sum_i f_i(q) \right)^2 \int_0^\infty [g(r) - \rho_0] \frac{r}{q} \sin(qr) dr \quad (2)$$

In Equation (2), the atomic scattering intensity term,  $\sum_i f_i^2(q)$ , represents the atoms scattering independently and the second term describes the deviation from the mean intensity when the local density at  $r$  deviates from the mean density  $\rho_0$ . Used to extract the RDF, the reduced scattering intensity function  $\varphi(q)$  is defined as in Equation (3).

$$\varphi(q) = \frac{[I(q) - \sum_i f_i^2(q)] q}{(\sum_i f_i(q))^2} \quad (3)$$

Finally, the function  $\varphi(q)$  is Fourier-transformed to give the eRDF, (Equation (4)).

$$G(r) = 4\pi r [g(r) - \rho_0] = 4 \int_0^\infty \varphi(q) \sin(qr) dq \quad (4)$$

In practice, the eRDF (Equation (5)) is calculated by the integration of the reduced scattering function over an experimentally limited range of  $q$  between  $q_{min}$  and  $q_{max}$ . In this case, to reduce the influence of data truncation on the Fourier transform, the data are

multiplied by an exponential function with a small artificial temperature factor  $b$ , which is usually set at  $b = 0.02$ :

$$G(r) = 4 \int_{q_{min}}^{q_{max}} \varphi(q) \sin(qr) \exp(-bq^2) dq \quad (5)$$

## 2. Materials and Methods

### 2.1. Experimental

Electron diffraction data were collected from amorphous SiO<sub>2</sub> films samples with two different thickness of 234 Å and 471 Å (purchased from Agar Scientific Ltd., Essex, UK). The thicknesses were determined using an ARM-200CF TEM (JEOL, Akishima, Japan) from the field of electron energy loss spectroscopy (EELS) at 200 kV using the log-ratio technique [19] by averaging over three windows in each film sample. A Libra 120 TEM (Zeiss, Oberkochen, Germany), operating at 120 kV with an in-column energy filter, was used to collect diffraction data and electron diffraction patterns were recorded on a TRS full-frame 16-bit 2 k × 2 K-pixel CCD camera, with a camera length of 144 mm and a 5 μ condenser aperture giving a parallel nanobeam of 150 nm in diameter. The diffraction patterns were energy-filtered at the zero-loss peak, with a slit width of 20 eV. A NanoMEGAS Digistar P1000 was used for precession electron diffraction. When collecting diffraction patterns, a beam stop was used to block the central beam in order to collect scattering intensity data at larger scattering angles while avoiding the risk of damaging the CCD during long exposures.

### 2.2. eRDF Analysis

The calculation of eRDF curves from the experimental data was performed using the free open source eRDF Analyser software [20], which is one of several similar software suites available, including ProcessDiffraction [21], SUEPDF [22], ePDF tools [23], and RDF Tools [24].

The eRDF Analyser provides an interactive graphical user interface and allows users to mask beam stops or any distorted areas, which are then excluded from the calculation of the final results. It also allows for the optimisation of the position of the centre of the diffraction pattern in the experimental diffraction data. Identifying the position of the centre is important for computing an accurate azimuthally averaged intensity distribution function,  $I(q)$ .

A polycrystalline aluminum specimen (SPI Supplies, West Chester, PA, USA) was used as a standard to calibrate the diffraction patterns. A customised automated fitting routine provided in the eRDF Analyser software was used to extract the atomic scattering background for calculations of the reduced scattering intensities according to Equation (3). This was performed based on least-squares fitting of the atomic scattering curve, with either the full range or the last third of the experimental data selected as the region of fitting. The reduced scattering intensity function  $\varphi(q)$  was obtained from Equation (3), and the eRDF curve was obtained by a Fourier transform of  $\varphi(q)$  following Equation (5). These functions were subsequently used in a reverse Monte Carlo refinement [25] to obtain an atomistic model, from which average coordination numbers and average bond lengths were calculated. The reduced scattering functions were obtained in the same range from 1.541 to 18.355 Å<sup>-1</sup>, with a 0.014 Å<sup>-1</sup> step to facilitate comparison between experiments.

### 2.3. Structure Refinement

Model structures were obtained from matched reduced scattering functions  $\varphi(q)$  calculated from the model using Equation (6). Then, they were filtered to remove the contribution of frequencies, resulting in unphysically short distances in the RDF below 1 Å

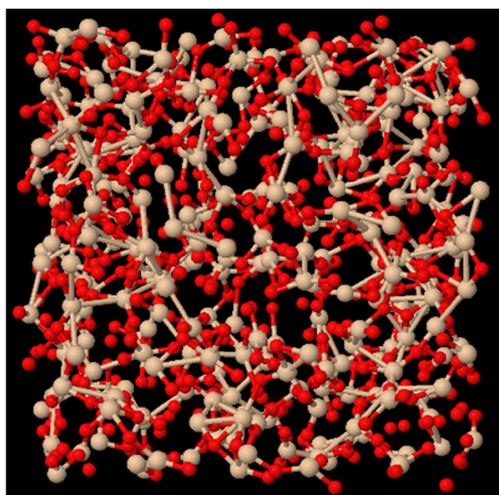
to the corresponding experimental scattering intensities, as described by McGreevy [25] and performed previously [26].

$$\varphi(q) = \frac{q \sum_{i \neq j} \sum_j f_i(q) f_j(q) \sin(qr_{ij}) / (qr_{ij})}{(\sum_i f_i(q))^2} \quad (6)$$

Initially, a small model of SiO<sub>2</sub> with 102 atoms was prepared by random dense packing of atoms approximated as soft spheres, in a cubic cell with a lattice parameter corresponding to a density of 2.3 g cm<sup>-3</sup>.

DFT calculations using the CASTEP code [27] were used to perform liquid quench molecular dynamics (MD) simulations using on-the-fly-generated pseudopotentials within the generalized gradient approximation using the Perdew–Burke–Ernzerhof exchange–correlation functional [28]. The generated pseudopotentials are a function of atomic species and, once generated in the beginning of the calculations, do not change during simulations. MD simulations were performed using a canonical (constant NVT) ensemble. The starting model was first melted at 3000 K for 2 ps. It was then simulated at 2000 K for 10 ps, and then further quenched to 400 K, reducing the temperature by 200 K in 5 ps equilibration steps. Finally, the structure was equilibrated at 300 K for 10 ps. The total simulation time was therefore 62 ps, with time steps of 2 fs throughout. The temperature was controlled using an implementation of the Nosé–Hoover thermostat [29]. The electron density was only sampled at the gamma point, with the pseudopotential energy cut-off set at 300 eV.

A larger model of 816 atoms for SiO<sub>2</sub> was constructed from the smaller model, obtained in the DFT MD simulations at 300 K, by assembling 2 × 2 × 2 small models with random orientations. These larger models were then used in RMC refinements. Small displacements of atoms in each of the RMC refinement steps were used to retain the structural features from the MD simulation, such as the preferred bond types and bond angle distributions. An example of a refined model is shown in Figure 1



**Figure 1.** An example of SiO<sub>2</sub> model of 816 atoms after RMC refinement.

Unfiltered and filtered reduced scattering intensity functions are shown in Figure 2a. The filter is a low-order polynomial contribution to the background that cannot be fit to the refinements as it would mean the presence of unphysically close interatomic distances [25]. A comparison of the RDF obtained from the refined model to the experimental one computed from filtered scattering intensities is shown in Figure 2b.

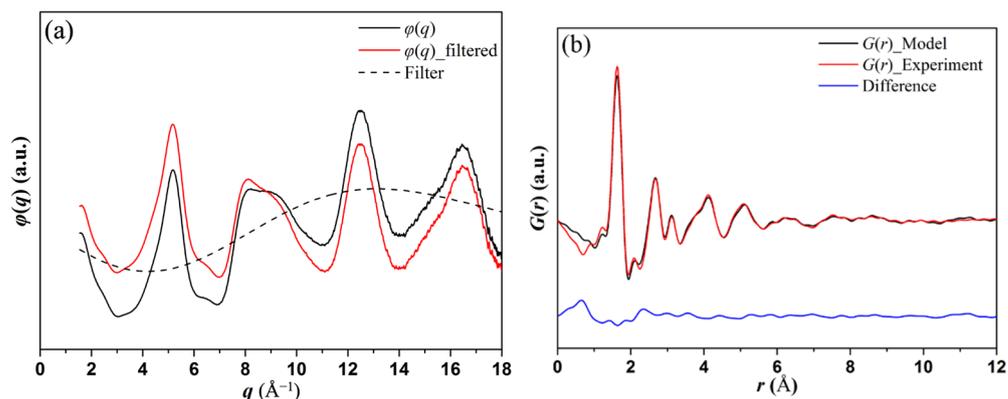


Figure 2. (a)  $\varphi(q)$  filtering and (b) comparison between experimental and model  $G(r)$  curves.

### 3. Results

Figure 3 shows a typical selected area diffraction pattern acquired from polycrystalline aluminium and amorphous  $\text{SiO}_2$  films without energy filtering. The patterns were obtained with the central beam shifted to the corner of the detector to maximize the recorded scattering angles. The thinnest part of the beam stopper was used to block the unscattered beam in order to collect the scattered intensity at low angles. A pixel calibration factor of  $0.00223 \text{ \AA}^{-1}$  was determined using the eRDF Analyser by first identifying and refining the centre of the diffraction pattern obtained without energy filtering and then calculating  $I(q)$  and relating the peak positions to the known reflections of polycrystalline Al, as summarised in Table 1.

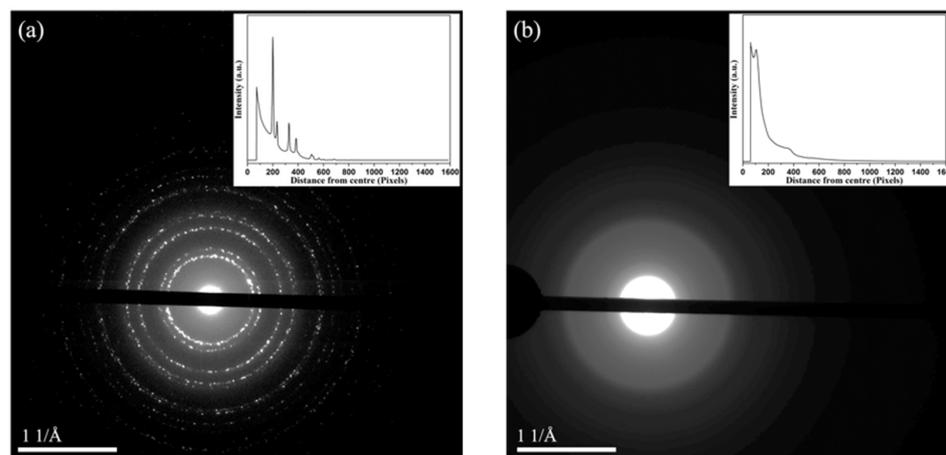


Figure 3. Typical electron diffraction patterns of (a) polycrystalline Al and (b) amorphous  $\text{SiO}_2$  specimen with azimuthally averaged total scattering intensities shown inset.

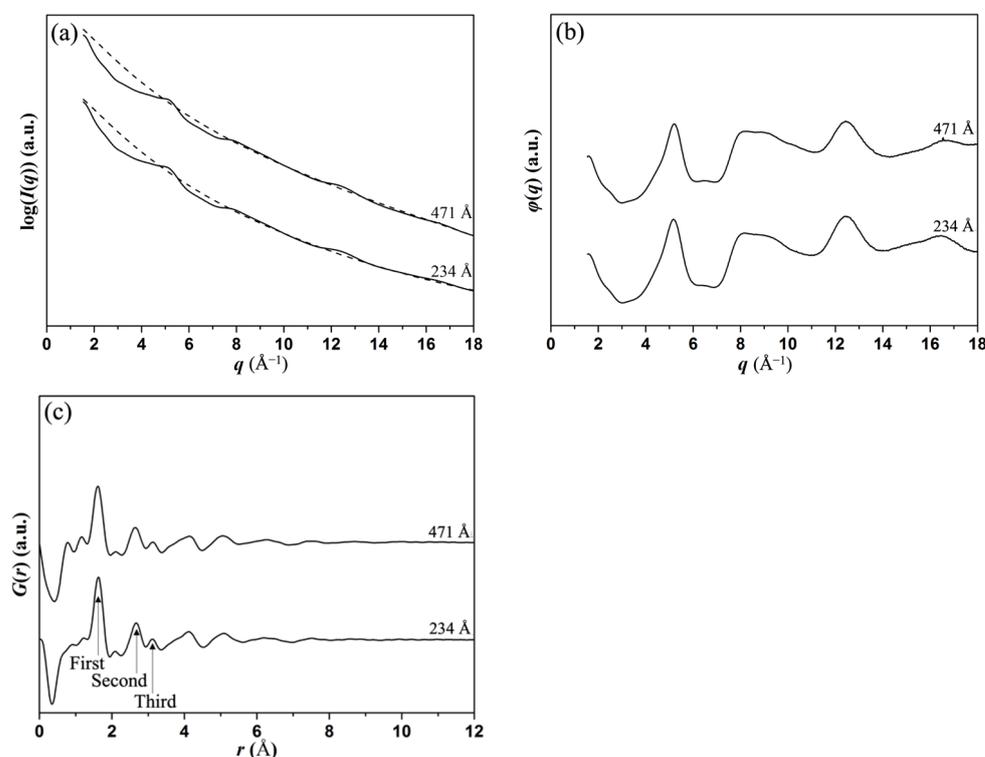
Table 1. Computed pixel calibration factors for polycrystalline Al.

{hkl}	Pixel Number n (Pixel)	Lattice Spacing d (Å)	Calibration Factor (1/(n·d))
{111}	382	2.34	0.00224
{200}	445	2.02	0.00222
{220}	628	1.43	0.00223
{311}	734	1.22	0.00223
			Average = 0.00223

#### 3.1. Influence of Specimen Thickness

Figure 4 compares the results of atomic scattering background fitting to the total experimental scattered intensity curves  $I(q)$ ,  $\varphi(q)$  and  $G(r)$ , recorded from amorphous

SiO<sub>2</sub> specimens with thicknesses of 234 Å and 471 Å, in experiments with 0° precession (no precession) and without energy filtering. Noticeable differences are observed in scattering intensities (Figure 4) at low scattering angles ( $q$  from 2 to 6 Å<sup>-1</sup>) when comparing the data from the 471 Å film to the 234 Å data. This is due to the atomic scattering background for thinner specimens fitting better through the oscillations of the total scattered intensity curve in this region due to there being a smaller contribution from multiple and inelastic scattering, as seen in Figure 4. In the  $G(r)$  curves, the first peak corresponds to the Si-O distances, while the second and third peaks are due to a mixture of Si...O, Si...Si and O...O distances. The determined Si-O bond length of 1.61–1.63 Å also agrees with the value of 1.612(7) Å drawn from an X-ray diffraction study [30]. Measurements of peak position and height are given in Table 2. The maximum difference in peak positions for the two sample thicknesses is observed for the second peak and is about 0.03 Å, or less than 2%. The peak heights, however, show larger deviations for the two films, being of 12.6%, 6.4% and 21.9% for the first, second and third peaks, respectively. The peak heights were defined as the peak amplitude in the calculated eRDF curves. The significance of large change in the peak heights is that it affects the coordination numbers that can be calculated from the results.



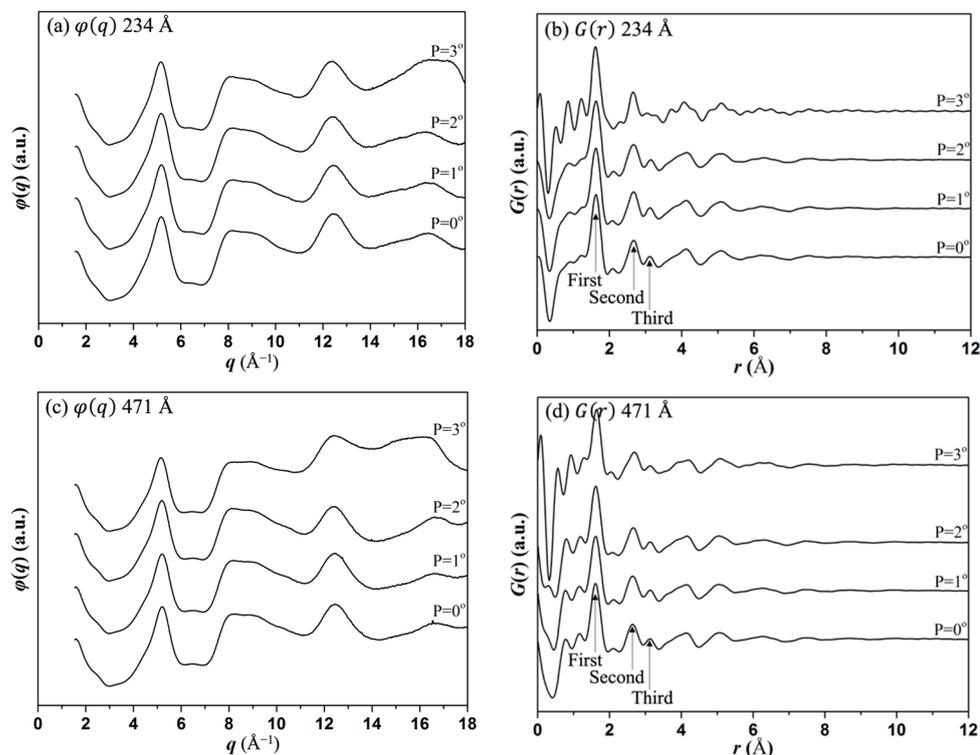
**Figure 4.** Influence of specimen thickness on atomic scattering background fitting to (a)  $I(q)$ , (b)  $\phi(q)$  and (c)  $G(r)$  for samples with two different thicknesses, 234 Å and 471 Å.

**Table 2.** Influence of specimen thickness on (a) positions and (b) heights of the eRDF peaks.

(a) Peak Position/(Å)	First	Second	Third
234 Å	1.63	2.67	3.12
471 Å	1.61	2.64	3.13
(b) Peak Amplitude/(a.u.)	First	Second	Third
234 Å	2.70	1.10	0.32
471 Å	2.36	1.03	0.25

### 3.2. Influence of Precession Angle

PED was measured with precession angles from  $0^\circ$  to  $3^\circ$  for samples of both thicknesses and the resulting  $\varphi(q)$  and  $G(r)$  for both 234 Å and 471 Å samples are shown in Figure 5. The main parameters are given in Table 3. These results show that only small variations are observed when comparing the data without precession ( $0^\circ$ ) to those with precession ( $1^\circ$  to  $2^\circ$ ). However, when the precession angle reaches  $3^\circ$ ,  $\varphi(q)$  data at high  $q$  (above  $16 \text{ \AA}^{-1}$ ) appears larger, which leads to larger oscillations in the resulting  $G(r)$ . Furthermore, it can be seen from Table 3 that the peak positions differ very little when the precession angles vary from  $0^\circ$  (no precession) to  $2^\circ$ , with differences being less than  $0.02 \text{ \AA}$ . However, for the  $3^\circ$  precession, the third peak position decreases by  $0.08 \text{ \AA}$ . The same trend is observed for the peak heights. While experiments with precession angles from  $0^\circ$  to  $2^\circ$  show small differences of around 6.3% for all peaks, the  $3^\circ$  precession shows larger differences of 13.3%, 13.6%, and 65.6% for the first, second and third peaks, respectively. Similar trends are observed for the peak heights for the thicker sample of 471 Å with differences of 10.9%, 18.4%, and 47.1% for the first, second and third peaks, respectively, of the sample with  $3^\circ$  precession. This can be understood as the principle that applying large precession angle effectively increases the thickness of the sample. As suggested previously [31], the effect of sample thickness on the scattered intensity can be regarded as a convolution of the scattered intensity from a thin slice with itself as the electron wave propagates through a thicker sample. This results in a reduction of the amplitudes of all peaks in the total scattering intensity and, therefore, reduction in peak heights in  $G(r)$ . Importantly, the large change in the peak heights has consequences for the determination of the coordination numbers from the results.



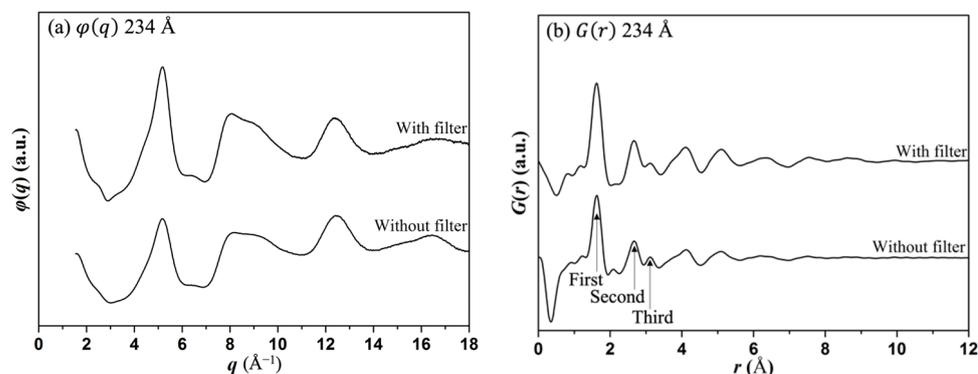
**Figure 5.** Influence of precession angle on  $\varphi(q)$  and  $G(r)$  for (a,b) 234 Å and (c,d) 471 Å samples.

**Table 3.** Influence of precession angle on (a) positions and (b) heights of the eRDF peaks for the 234 Å and 471 Å samples.

(a) Peak Position/Å		First	Second	Third
234 Å	P = 0°	1.63	2.67	3.12
	P = 1°	1.62	2.67	3.12
	P = 2°	1.63	2.67	3.14
	P = 3°	1.61	2.66	3.04
471 Å	P = 0°	1.61	2.64	3.13
	P = 1°	1.61	2.65	3.14
	P = 2°	1.62	2.66	3.13
	P = 3°	1.64	2.69	3.13
(b) Peak Amplitude/(a.u.)		First	Second	Third
234 Å	P = 0°	2.70	1.10	0.32
	P = 1°	2.66	1.11	0.34
	P = 2°	2.59	1.06	0.30
	P = 3°	3.06	1.25	0.11
471 Å	P = 0°	2.36	1.03	0.25
	P = 1°	2.31	1.03	0.27
	P = 2°	2.39	1.02	0.28
	P = 3°	2.65	0.87	0.17

### 3.3. Influence of Energy Filtering on the RDF

Figure 6 shows  $\varphi(q)$  and  $G(r)$  measured with and without energy filtering for both 234 Å and 471 Å samples. Although the overall shapes of  $\varphi(q)$  are similar with and without energy filtering for both thicknesses, noticeable differences can be observed in curve oscillation amplitudes. The use of higher-amplitude oscillations in the  $\varphi(q)$  in the experiments with energy filtering led to higher peaks in the corresponding  $G(r)$  curves. Measurements of the eRDF curves parameters are shown in Table 4. It can be seen that peak positions in both  $G(r)$  curves are identical, except for a small 0.01 Å difference for the first (Si-O) peak. However, the peak heights are 30.0%, 27.3%, and 56.3% smaller for the first, second and third peaks, respectively, when recorded without energy filtering. The trends for the 471 Å thick sample are similar. The difference in peak positions is small, being less than 0.02 Å, while the peak heights are 25.8%, 29.5%, and 13.8% smaller for the first, second and third peaks, respectively, without energy filtering. As the energy of the electrons defines their wavelength, the scattering of electrons of different energies will result in slightly shifted (relative to the incident beam) scattering profiles. Averaging over such scattering profiles at the detector is expected to have the effect of flattening the peaks, similar to the effects of the multiple scattering.

**Figure 6.** Cont.

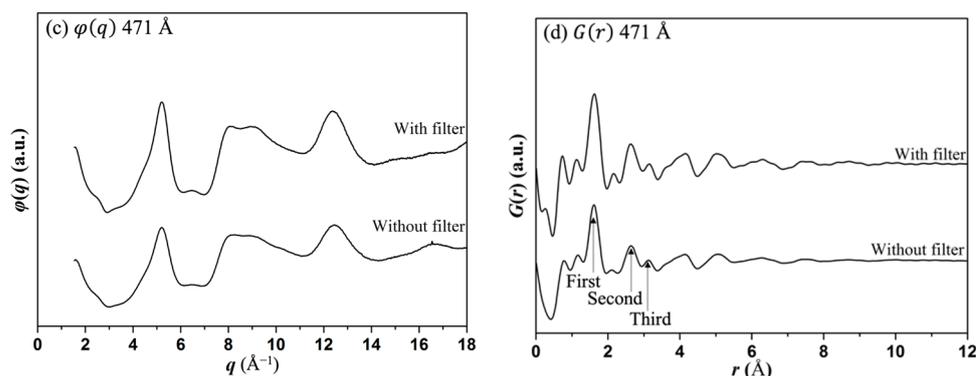


Figure 6. Influence of energy filtering on  $\varphi(q)$  and  $G(r)$  for (a,b) 234 Å and (c,d) 471 Å samples.

Table 4. Influence of energy filtering on (a) positions and (b) height of the eRDF peaks for 234 Å and 471 Å samples.

(a) Peak Position/(Å)		First	Second	Third
234 Å	Without energy filter	1.63	2.67	3.12
	With energy filter	1.62	2.67	3.12
471 Å	Without energy filter	1.61	2.64	3.13
	With energy filter	1.62	2.63	3.15
(b) Peak Amplitude/(a.u.)		First	Second	Third
234 Å	Without energy filter	2.70	1.10	0.32
	With energy filter	3.51	1.40	0.14
471 Å	Without energy filter	2.36	1.03	0.25
	With energy filter	3.18	1.46	0.29

### 3.4. Influence of Experimental Parameters on the Refined Structure

Table 5 summarises the coordination numbers,  $Z$ , and average Si-O bond length computed from the refined models of the amorphous  $\text{SiO}_2$  films obtained under different experimental conditions, while Figure 7 shows the change in coordination numbers for various experimental conditions.

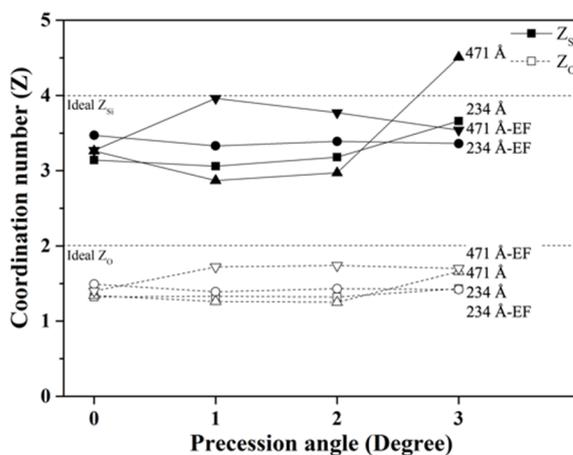


Figure 7. Variations in coordination numbers of  $Z_{\text{Si}}$  and  $Z_{\text{O}}$  with sample thickness, energy filtering and precession angle obtained from structure refinements of amorphous  $\text{SiO}_2$  models.

**Table 5.** Influence of experimental parameters on refined coordination numbers and structural parameters of amorphous SiO<sub>2</sub> models.

Thickness/Å	Energy Filtering	Precession Angle/Deg	Z <sub>Si</sub> (Ideal 4)	Z <sub>O</sub> (Ideal 2)	Si-O/Å	R-Factor/%
234	No	0	3.14	1.32	1.67(19)	12.5
		1	3.06	1.33	1.68(20)	11.9
		2	3.18	1.32	1.67(19)	12.4
		3	3.66	1.43	1.63(19)	15.3
	Yes	0	3.47	1.49	1.64(18)	14.7
		1	3.33	1.39	1.64(15)	15.0
		2	3.39	1.43	1.64(14)	19.1
		3	3.36	1.42	1.64(12)	14.0
471	No	0	3.26	1.34	1.67(20)	12.8
		1	2.87	1.26	1.69(22)	12.5
		2	2.97	1.25	1.67(21)	15.7
		3	4.51	1.66	1.53(25)	19.7
	Yes	0	3.27	1.4	1.67(19)	16.8
		1	3.96	1.72	1.61(19)	16.9
		2	3.77	1.74	1.61(18)	17.6
		3	3.54	1.7	1.63(13)	16.1

## 4. Discussion

### 4.1. Influence of Sample Thickness

We have observed that the heights of all peaks are reduced with increasing thickness independently of precession or energy filtering, while the peak positions remain the same.

Increasing the sample thickness decreases the quality of fit, as seen in the R-factors in Table 5. This is to be expected as the contribution of inelastic and multiple scattering increases with sample thickness. This observation also agrees with earlier research showing that multiple scattering does not affect peak positions [31].

Although, the peak heights generally reduce with thickness, (Table 2) there is a less pronounced trend in the change in the coordination number obtained from the refined models (Figure 7). It can be seen, however, that the coordination number increases with thickness when energy filtering and precession are applied. Without energy filtering, the coordination number tends to be lower for the thicker sample when using smaller precession angles (Figure 7).

### 4.2. Influence of Precession

Smaller precession angles (1°, 2°) slightly decrease the height of the first peak as compared to data recorded without precession (Table 3), whilst a higher precession angle (3°) increases the height of the first peak considerably while notably reducing the height of the third peak. Its influence on fit quality is not straightforward. Introduction of small (1°) precession generally improves the fit quality. The improvement of fitting between model and experiment with the introduction of precession has also been seen in previous experiments with metallic nanoparticles [32]. The fit quality, however, reduces with larger precession angle. It can be suggested that larger precession angles effectively increase the sample thickness, which leads to larger contribution of inelastic scattering and, therefore, worse fit. This decrease in fit quality is notably larger for the thicker 471 Å sample (Table 5). Other possible contributing factors cannot be excluded, such as an increased probability of descanning errors.

Using higher precession angles increased the coordination numbers for the thinner sample in the present investigation. For the thicker sample, however, a high degree of precession resulted in unreasonably high coordination number (4.5 for Si), which can be attributed to the worse fit due to effective increase in sample thickness and increased contribution of inelastic scattering at higher precession.

#### 4.3. Influence of Energy Filtering

Energy filtering, as for precession, does not affect all peaks uniformly. For example, the heights of the first and second peaks increase considerably after energy filtering, (Table 4) as compared to the third peak, which reduces for the 234 Å thin sample or increases only slightly for the 471 Å thick one.

When energy filtering is introduced, coordination numbers increase for both sample thicknesses as compared to unfiltered data, but the goodness of fit reduces slightly (Table 5). The improvement in coordination number can be attributed to the removal of the inelastic scattering contribution by the energy filter and is a consequence of increased peak heights. The slight decrease in the fitting quality may be due to some experimental error introduced by the energy filter. The magnitude of change in coordination number with introduction of energy filtering is more notable for the thicker sample, as would be expected by the removal of the larger contribution of inelastic scattering.

## 5. Conclusions

The study demonstrates that the peak positions in the obtained eRDF curves are generally insensitive to sample thickness, precession angle or energy filtering.

However, the studied experimental conditions influence the peak heights considerably, which results in different coordination numbers being obtained from the atomistic models.

It is shown that introducing precession and energy filtering results in higher coordination numbers, bringing them closer to the ideal theoretical values for the material when the data are obtained from a thicker sample. It is also observed that energy filtering and precession, may influence the goodness-of-fit of atomistic models to the experimental data.

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