

Article

High-Pressure Vibrational and Structural Studies of the Chemically Engineered Ferroelectric Phase of Sodium Niobate

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Abstract: Pure NaNbO₃ has an antiferroelectric phase at ambient pressure. The structural behaviour of the chemically engineered ferroelectric phase of sodium niobate, NNBT05: [(0.95) NaNbO₃–(0.05) BaTiO₃], under high-pressure has been studied using Raman scattering and angle-dispersive synchrotron X-ray diffraction techniques. At pressure > 1 GPa, noticeable changes in the Raman spectra can be seen in the low wavenumber modes (150–300 cm^{−1}). Large changes in the positions and intensities of the Raman bands as a function of pressure provide evidence for structural phase transition. The results indicate significant changes in the bond-lengths and the orientation of the NbO₆ octahedra at ~1 GPa, and a transition to the paraelectric phase at ~5 GPa, which are at lower pressures than previously found in pure NaNbO₃. The powder X-ray diffraction pattern shows an appreciable change in the peak profile in terms of position and width on increasing pressure. The pressure dependences of the structural parameters show that the response of the lattice parameters to pressure is strongly anisotropic. By fitting the pressure–volume data using the Birch–Murnaghan equation of state, the isothermal bulk modulus was estimated. The experimental results suggest that on doping BaTiO₃ in NaNbO₃, the bulk modulus increases. The bulk modulus of NNBT05 has been estimated to be 164.5 GPa, which is fairly close to 157.5 GPa, as previously observed in NaNbO₃.

Keywords: ferroelectric; antiferroelectric; niobate; Raman scattering; high pressure diffraction

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1. Introduction

Materials with perovskite structures exhibit diverse crystal structures and physical properties such as (anti)ferroelectricity, relaxor, magnetic, multiferroic properties, etc., and remain pertinent in next generation applications [1–14]. The perovskite structure consists of two types of polyhedra, the octahedron surrounding the B-cation, and the A-cation, coordinated by 12 oxygen atoms forming a cuboctahedron. In the perovskite family, alkaline niobates and their derivatives have attracted a great deal of attention, and are promising candidates for eco-friendly lead-free piezoceramics [10,11,15–17].

The application of hydrostatic pressure strongly modifies the short-range interatomic and long-range Coulomb interactions, which are responsible for the structural stability of ferroelectric and antiferrodistortive phases. These phases are associated with the freezing of zone-centre and zone-boundary phonon instabilities. Sodium niobate (NaNbO₃) has both these competing instabilities simultaneously. At ambient conditions, sodium niobate crystallizes in the antiferroelectric orthorhombic phase. Sodium niobate perovskite is a textbook example for understanding a complicated series of structural phase changes that occur with temperature and pressure [17–28]. Using high pressure Raman scattering techniques, Shiratori et al. [28] have reported transitions at around 2, 6, and 9 GPa, respectively. The pressure-dependent neutron diffraction studied showed that the antiferroelectric (*Pbcm*) phase transformed to a paraelectric (*Pbnm*) phase at 8 GPa [19]. Neutron diffraction also

revealed changes in the baric behaviour of Nb-O-Nb bond angles as a result of complex reorientations of NbO₆ octahedra at high pressures of ~2 GPa [19]. Recent, high pressure X-ray diffraction data suggest that the paraelectric (*Pbnm*) phase is stable up to 30 GPa [29].

In addition to temperature and pressure, the structure and physical properties of materials can also be tuned by chemical doping. Specific doping elements are used to produce subtle distortions in crystal structures and to regulate changes in their physical properties. For example, when doping the BaTiO₃ in the NaNbO₃ matrix, there is a significant enhancement in the dielectric and piezoelectric response of the material [30–34]. The structural, dielectric, ferroelectric and piezoelectric properties of NNBT_x [(1 – x) NaNbO₃ – xBaTiO₃] ceramics have been investigated by various researchers [10,16,30,31,33–36]. This solid solution is similar to K_x Na_{1–x}NbO₃ (KNN), except that the KNbO₃ has been replaced by BaTiO₃. The main problem with KNN is the poor densification of the ceramic due to the high volatility of alkaline oxides, which significantly affects the functional performance of this ceramic [37]. As suggested by Zeng et al. [37], the problem of the poor densification of KNN can be resolved by doping BaTiO₃ instead of KNbO₃ in the NaNbO₃ matrix, because both show similar structural phase transitions. It can also be seen that BaTiO₃ based piezo-ceramics show very good electromechanical properties. Raveskii et al. [32] and other researchers have studied the various compositions in the full composition range. They have reported extensive electro-physical properties of this solid solution, with expected structures. It is reported that on increasing the BaTiO₃ composition, the dielectric maximum shifts to the lower temperatures and reaches near room temperature for x = 0.25. They also found P-E hysteresis loops with good saturation for x = 0.10. For the same sample, they observed the coercive field E_C (11 kV/cm), the spontaneous polarization P_S (15 μC/cm²) and residual polarization P_R (13 μC/cm²) at 50 Hz. However, the electromechanical properties of the NNBT_x in the NaNbO₃ are exceptionally remarkable. The literature contains very little information about the structural phase transition with composition and temperature. Most of the reports contain only knowledge of expected room temperature structures probed using X-ray diffraction studies. Raveskii et al. [32] have proposed the monoclinic structure of the compositions x~0.05–0.07 at room temperature.

Earlier, we reported the crystal structure and phase stability of the NNBT_x system for the small doping of BaTiO₃ (x = 0.0 to 0.15). We reported an orthorhombic antiferroelectric phase (space group: *Pbcm*) for x < 0.02, an orthorhombic ferroelectric phase (space group: *Pmc2*₁) for 0.02 < x < 0.10, and an orthorhombic ferroelectric phase (space group: *Amm2*) for x ≥ 0.10. We also reported the phase transitions as a function of temperature for NNBT03 and 05 using powder X-ray and neutron diffraction studies in conjunction with dielectric and Raman scattering measurements [30,31,35,36]. Apart from this, we provided evidence of the existence of a functional monoclinic phase at low temperatures. This monoclinic (*Cc*) phase is expected to provide easy polarization rotation at low temperatures for the compositions x = 0.03 (NNBT03) and x = 0.05 (NNBT05).

The application of pressure either distorts or symmetrises the structure and causes the alteration of the crystal symmetry with huge consequences on the physical properties. The present study gives insight into the unique pressure-dependent phase stability of chemically engineered ferroelectric NaNbO₃, i.e., NNBT05, and aims to deepen the understanding of structural distortion in this material. It also helps in the understanding of the role of doping on pressure responses and the phase stability of the ferroelectric phase. The pressure dependence of the powder X-diffraction and Raman scattering experiments will be used to understand the microscopic origin of the structural and vibrational properties of the sample. At pressure > 1 GPa, noticeable changes in the Raman spectra are seen in the low wavenumber modes (150–300 cm⁻¹). These observations are most likely associated with octahedral reorientations in the parent ferroelectric phase. A detailed analysis of the high-pressure X-ray diffraction data suggests that the response of the lattice parameters to pressure is strongly anisotropic. By fitting the pressure–volume data using the Birch–Murnaghan equation of state, the isothermal bulk modulus can be estimated. We found that on doping BaTiO₃ in NaNbO₃, the bulk modulus increases.

2. Materials and Methods

The solid solution NNBT05 was prepared with the solid-state reaction method. Dried powders of high purity Na_2CO_3 , BaCO_3 , Nb_2O_5 and TiO_2 were taken in the stoichiometric ratios and were mixed in a planetary ball mill for 12 h in acetone medium. The mixtures were calcined in the air at 1173 K. The phase purity of the solid solutions was confirmed with powder diffraction data. The calcined powders were then sintered at 1273 K for 6 h. The sintered pellets were crushed into fine powders and were used for the high-pressure X-ray diffraction and Raman scattering measurements.

Angle-dispersive X-ray powder diffraction patterns at high pressures up to 30 GPa and at ambient temperature were measured at the BL-11 beamline at INDUS-2 India. The diffraction images were collected with the wavelength $\lambda = 0.79997 \text{ \AA}$ on the MAR345 detector located at a distance of 190 mm from the sample. The two-dimensional X-ray diffraction (XRD) images were converted to one-dimensional diffraction patterns using the FIT2D program. The data at each pressure level were analysed with the Rietveld refinement method using the FULLPROF program [38]. A Thompson–Cox–Hastings pseudo-Voigt with Axial divergence asymmetry function was used to model the peak profiles. The background was fitted using a Chebychev polynomial. Except for the occupancy parameters of the atoms, which were fixed corresponding to the nominal composition, all other parameters, i.e., scale factor, zero displacement, and isotropic profile parameters, lattice parameters, isotropic thermal parameters, and positional coordinates, were refined.

High pressure Raman experiments were performed using an in-house-made Mao-Bell type of diamond anvil cell. Raman spectra were recorded at increasing and decreasing pressures using a 514.5 nm excitation wavelength from a mixed gas laser (model: Stabilite 2018 from Spectra Physics) and a triple grating Raman spectrometer (T64000 from Horiba Jobin Yvon) equipped with a liquid nitrogen cooled charge coupled device. We used the holographic grating with 1800 g/mm and a 20-micron exit slit with a spectrometer focal length of 1 m, which resulted in 0.7 cm^{-1} optical resolution of the spectra.

High pressure X-ray diffraction and Raman experiments were carried out using a Mao Bell kind of diamond anvil cell, which was equipped with diamonds with a culet size of ~ 400 microns. The sample was loaded into a hole of $100 \mu\text{m}$ and drilled in a pre-indented $50 \mu\text{m}$ thick tungsten gasket. Methanol/ethanol in the ratio of 4:1, which is known to be quasi hydrostatic upto 10 GPa [39], was used as the pressure transmitting medium, and fluorescence peaks from ruby sphere were used to estimate the pressure inside the sample chamber of the diamond anvil cell. Care was taken to load a very tiny sample so that the environment around the sample remained quasi hydrostatic [40]. The limitations of the present Raman spectroscopic setup made it difficult to investigate Raman spectra below 150 cm^{-1} .

3. Results and Discussion

3.1. High Pressure Raman Scattering Study

The high symmetry phase of the ABO_3 perovskite compound had a cubic structure. In these compounds, the condensation of different phonon instabilities led to phase transitions. The A atoms (Na/Ba) were located at the cubic structure's eight corners, while the B atoms (Nb/Ti) were situated in the body's centre. The oxygen (six atoms) was arranged at the face-centred site and forms an octahedron. The Raman modes, also known as translational (Tr), librational (L), bending (B), and stretching (S) modes, were produced as a result of these atoms' vibrations. The roto-vibrational motion of the oxygen octahedra, the translational motion of the A ions against the BO_6 octahedra, and the bending of the BO_6 octahedra linkages gave rise to these modes. For systems of the ABO_3 type, these are collectively referred to as external modes. The bending and stretching motion of the oxygen octahedra's O-B-O bonds caused the internal modes of the octahedra, known as the bending (B) and stretching (S) modes.

As said earlier, the NNBT05 ambient phase structure had orthorhombic symmetry (space group: $Pmc2_1$; tilt system: $a^- a^- c^+$) with unit cell dimensions $2a_p \times \sqrt{2}a_p \times \sqrt{2}a_p$

and four numbers of formula units, i.e., a total of 20 atoms (see Figure 1a), and it was ferroelectric in nature. Here, a_p is the lattice constant of the parent cubic phase. The group theoretical analysis gave 57 Raman active modes, represented by $\Gamma_{\text{Raman}} = 16A_1 + 13A_2 + 12B_1 + 16B_2$. Due to the non-centrosymmetric structure, except for the A_2 mode, these Raman active modes were simultaneously IR active. The number of observed modes was lower compared to the expected number of modes due to the peaks in the experimental Raman spectrum not being properly resolved, and due to accidental degeneracy. Generally, at elevated temperatures and pressures, the cation displacements from the centre of the octahedra (which results in nonzero polarization) may become vanishingly small and the structure may transform to the centrosymmetric orthorhombic paraelectric (*Pnma*) phase. This phase also had similar cell dimensions to ferroelectric (*Pmc2₁*) and in total 20 atoms in the primitive cell (Figure 1b). The number of Raman active modes in this phase (in another setting of *Pbnm*) was $24, 7A_g + 5B_{2g} + 7B_{2g} + 5B_{3g}$.

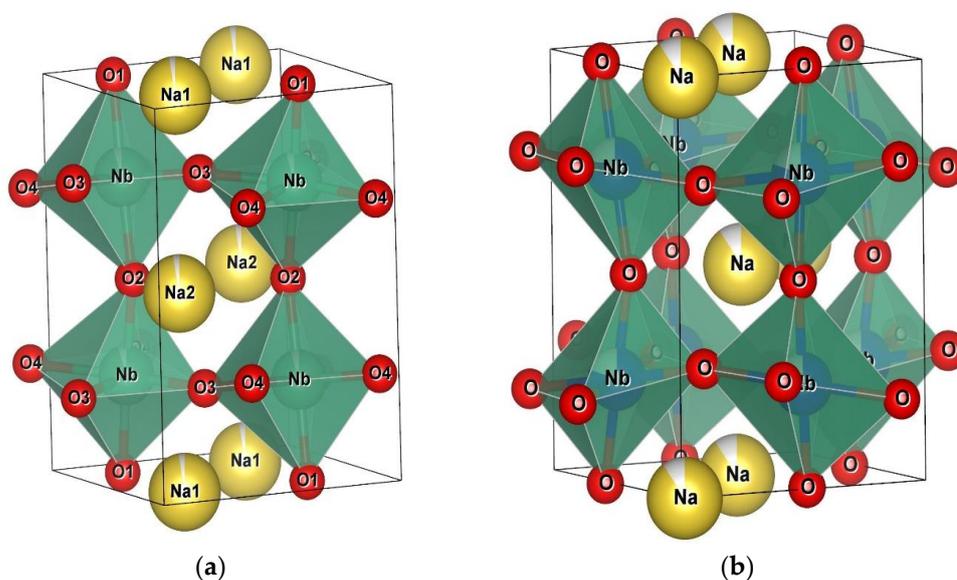


Figure 1. Crystal structure of (a) ferroelectric phase (space group: *Pmc2₁*) and (b) paraelectric phase (space group: *Pbnm*).

Figure 2 depicts the evolution of the high-pressure Raman spectra of NNBT05 at selected pressures and room temperature. For pressure > 1 GPa, distinct changes in the low wavenumber modes ($150\text{--}300\text{ cm}^{-1}$) region of the Raman spectra were observed. The peaks containing the most prominent changes were denoted ν_1 , ν_2 , ν_3 , and ν_4 (bending mode: $150\text{--}300\text{ cm}^{-1}$) and showed drastic changes in their relative intensity in the pressure range 1 to 5.4 GPa. These observations at ~ 1 GPa are most likely associated with octahedral reorientation in the parent ferroelectric phase, as has also been found earlier in pure NaNbO_3 . The change in Raman spectra at ~ 4.6 GPa may be a signature of structural phase transition from the ferroelectric phase (*Pmc2₁*) to the paraelectric phase (*Pbnm*). We noted that the paraelectric transition was observed in pure NaNbO_3 at ~ 8 GPa. On further increasing the pressure above 5.4 GPa, the intensities of these peaks reduced and the stretching modes ($500\text{--}750\text{ cm}^{-1}$) centred around 584 cm^{-1} became merged. Figure 2b shows the fitted positions of the Raman bands. In literature [41], it was shown that even the 4:1 methanol/ethanol transmitting medium had Raman active modes assigned to the C-C stretch at ~ 450 and 880 cm^{-1} . However, we would like to mention here that these modes were very weak and were not discernible in the presence of relatively stronger Raman scatterers. In the Raman spectra depicted in Figure 2a, we can clearly see that there was no interference from these modes.

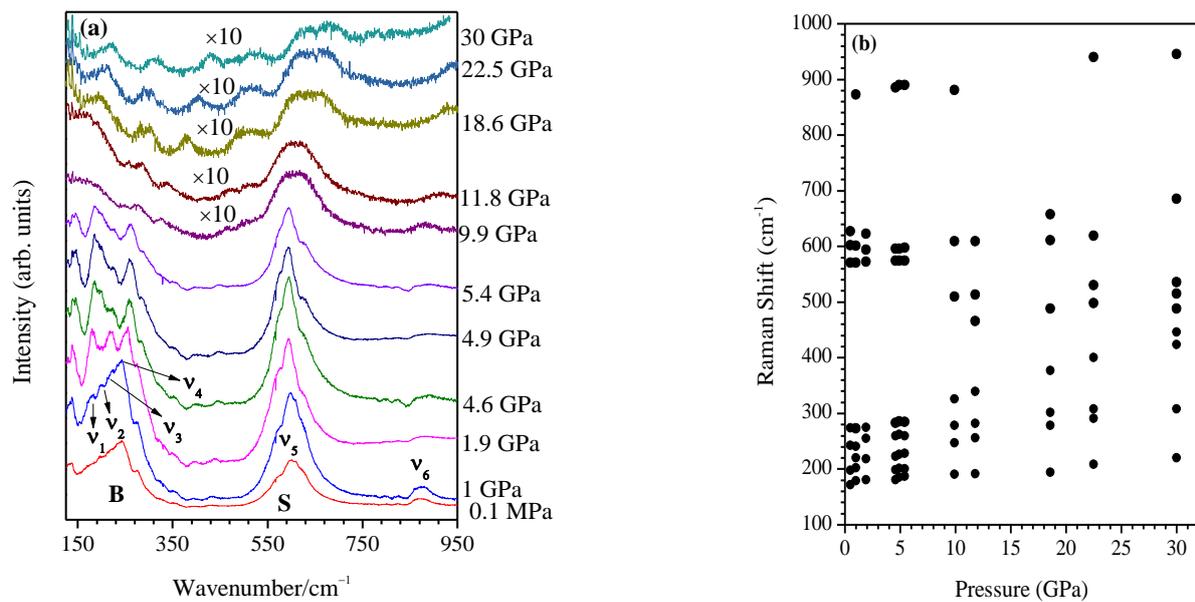


Figure 2. (a) Evolution of the Raman spectra collected at selected pressures for NNBT05. The pressure dependencies of the vibration modes (Raman shift) are shown in (b). The error bar on the pressure is the size of symbol. The intensities of Raman spectra at higher pressure >5.4 GPa are expanded 10 times.

The frequency evolution showed a blue-shift trend with increasing pressure. It is evident from Figure 2 that the frequencies of Nb–O stretching modes at $\sim 500\text{--}700\text{ cm}^{-1}$ increased with the pressure, implying a shortening of the Nb–O bond length. For pressure above 5.4 GPa, the Raman modes became very broad and weak, which could be due to an onset of disorder or due to a phase with a weak Raman response. In nanoporous silicon, it has been observed that though the Raman spectra showed very weak broad Raman modes and were termed a glassy phase, it actually was a crystalline phase which had very weak Raman activity, also because it was metallic [42]. At this point, we would also like to mention that even though the sample environment was quasi-hydrostatic, even up to ~ 10 GPa, we cannot rule out the presence of some small pressure inhomogeneities, which could lead to the broadening of the Raman modes. However, on the release of pressure, the Raman modes of the parent phase were observed and so the broadening and reduction in the intensity of the Raman modes could be attributed to a reversible phase transition and not to the pressure inhomogeneities. To ascertain if indeed the high-pressure phase was disordered or it was a new crystalline phase, we carried out high pressure X-ray diffraction studies.

3.2. High Pressure X-ray Diffraction Study

High pressure X-ray diffraction patterns of NNBT05 were collected at 11 pressures up to 40 GPa in a pressure-increasing cycle. Figure 3 depicts the quality of the powder X-ray diffraction patterns at selected pressure levels. It is evident from this figure that on increasing the pressure, the diffraction peaks systematically shift towards higher angles, indicating the compression of the lattice parameters. To investigate the structural parameters with pressure, we carried out a detailed Rietveld analysis of the high-pressure X-ray diffraction data.

The detailed Rietveld refinement of the powder-diffraction data showed that the synchrotron powder diffraction pattern at 300 K could be indexed using the orthorhombic structure (space group $Pmc2_1$) in a low pressure range. Figure 3b shows the quality of Rietveld refinement of high-pressure powder diffraction data at 4.1 GPa. For a pressure range $P > 4.1$, the ferroelectric ($Pmc2_1$) structural model in the refinements gave highly correlated atomic positions.

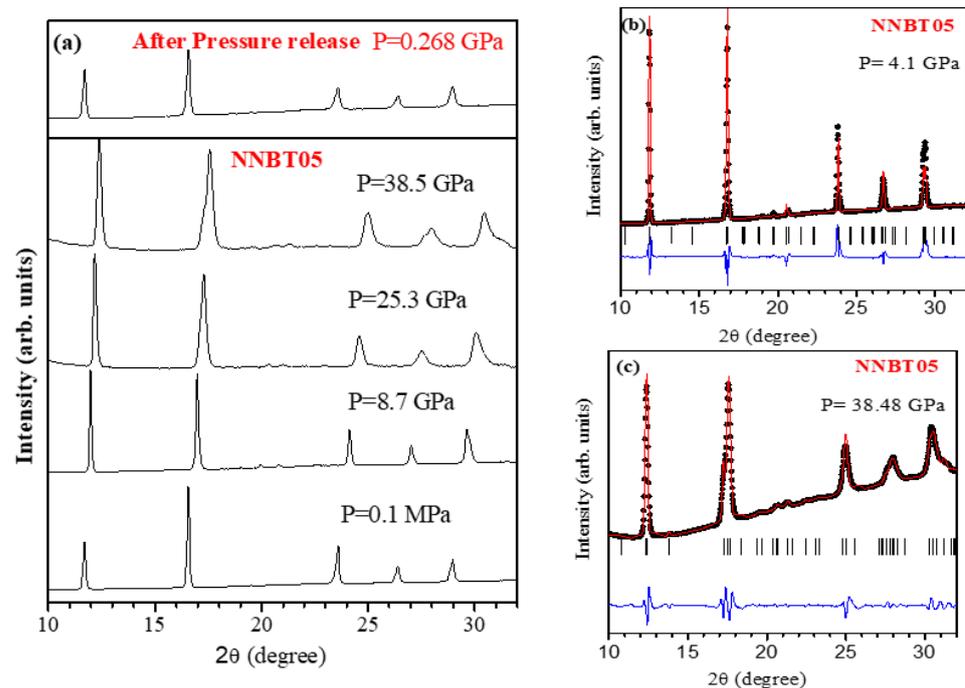


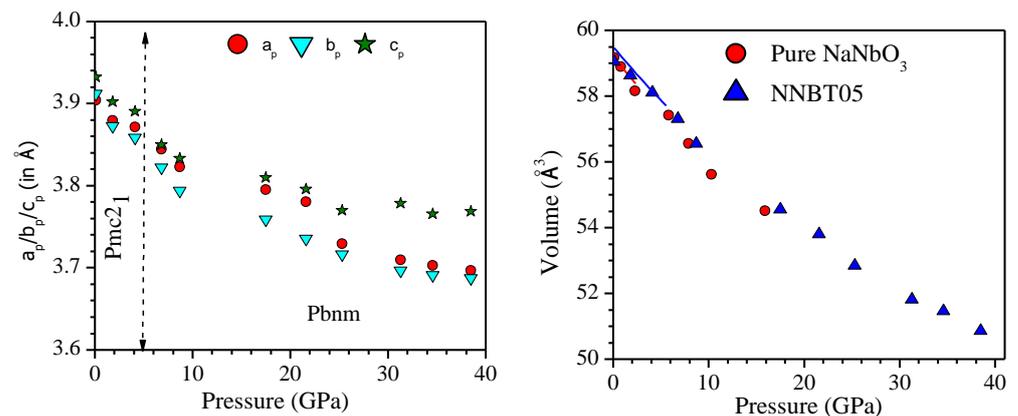
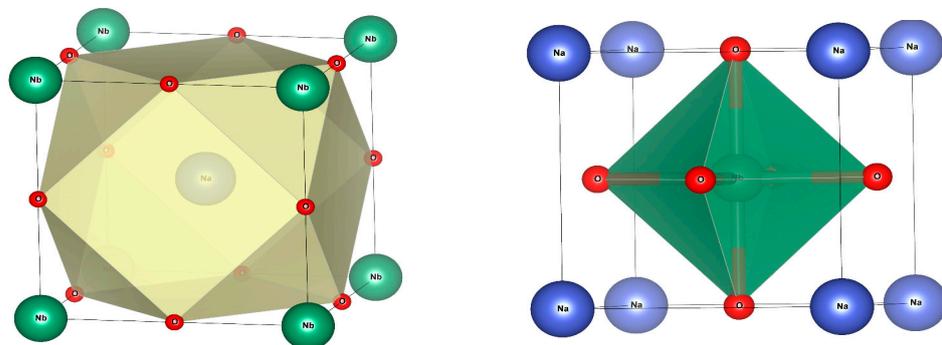
Figure 3. (Colour online) (a) Evolution of the powder—X-ray diffraction patterns of NNBT05 at selected pressures. calculated (continuous red line) and difference (bottom blue line) profiles obtained from Rietveld refinement using an orthorhombic ferroelectric phase (space group: $Pmc2_1$) at 4.1 GPa (b), and paraelectric (space group: $Pbnm$) at 38.5 GPa (c).

The high pressure neutron diffraction study on pure sodium niobate clearly revealed a structural phase transition from the orthorhombic antiferroelectric to paraelectric phase ($Pbnm$) above 8 GPa. In view of this, we refined the high-pressure synchrotron X-ray diffraction data using the paraelectric orthorhombic phase $Pbnm$, which accounted for all the reflections and gave nearly similar goodness-of-fit parameters compared with the space group $Pmc2_1$. However, it should be noted that the total number of variable structural parameters for $Pmc2_1$ was doubled (20) compared to the $Pbnm$ space group (10 parameters). Even doubling the variable parameters did not help to improve the fitting. Hence, the latter phase has been preferred over the ferroelectric ($Pmc2_1$) phase. The Rietveld refinements using the orthorhombic ($Pbnm$) phase proceeded smoothly, revealing a monotonic decrease in lattice constants and cell volume with the pressure increasing up to 38.5 GPa. Structural parameters obtained after the Rietveld refinement for the compound are given in Table 1. The X-ray powder diffraction results indicate that the broadening and weakening of Raman modes beyond 5.4 GPa cannot be attributed to a disordered phase. It is possible that this high-pressure phase is weakly Raman active.

The pressure dependences of the lattice parameters and unit cell volume obtained after the Rietveld refinement of high-pressure synchrotron diffraction data are shown in Figure 4. For easy comparison with Figure 2, we used pseudocubic lattice parameters for the orthorhombic phase using the relations $a_p = A_o/\sqrt{2}$, $b_p = B_o/\sqrt{2}$, and $c_p = C_o/2$, where a_p , b_p , c_p , A_o , B_o , and C_o are lattice parameters corresponding to the equivalent pseudocubic and orthorhombic phases, respectively. It is clear from Figure 5 that on increasing the pressure, the lattice parameters monotonically decreased in the entire range of our measurements for NNBT05. However, unlike the 'b' and 'c' lattice constants, 'a' showed a sudden decrease around 23 GPa. This could not be attributed to the non-hydrostatic stresses caused due to the freezing of the pressure transmitting medium. If we look at the structure of sodium niobate along the b and c axes, we can see that there was only one octahedra, whereas along the 'a' axes there were two corner-shared octahedra. With pressure, there is a possibility of bending of the octahedral hinge, thus drastically reducing the lattice constant along the 'a' axes.

Table 1. Structural parameters obtained through Rietveld refinement of high-pressure X-ray diffraction of NNBT05 using orthorhombic ferroelectric (space group: $Pmc2_1$) and paraelectric (space group: $Pbnm$).

Atom	Pressure = 4.1 GPa Space Group $Pmc2_1$				Pressure = 38.48 GPa Space Group $Pbnm$			
	Positional Coordinates				Positional Coordinates			
	x	y	z	B (\AA^2)	x	y	z	B (\AA^2)
Na1/Ba1	0.0000	0.2525(4)	0.7637(7)	1.98(5)	−0.059(5)	0.505 (5)	0.2500	2.81(5)
Na2/Ba2	0.5000	0.2514(9)	0.7691(5)	0.35(7)				
Nb/Ti	0.7507(9)	0.7480(8)	0.7925(3)	0.72(2)	0.0000	0.0000	0.0000	1.30 (2)
O1	0.0000	0.2185(7)	0.3217(8)	1.48(1)	0.010 (3)	0.019 (6)	0.2500	1.23 (5)
O2	0.5000	0.3109(9)	0.2986(9)	1.08(8)	0.281(7)	0.307 (4)	0.039 (3)	1.557(2)
O3	0.2265(8)	0.4482(6)	0.2140(6)	2.18(8)				
O4	0.2667(6)	−0.086(5)	0.6128(2)	1.55(3)				
$A_0 = 7.7425(4) \text{ \AA}$; $B_0 = 5.4571(5) \text{ \AA}$; $C_0 = 5.5018(5) \text{ \AA}$ Unit cell volume (V) = $232.46(9) \text{ \AA}^3$ $R_p = 3.68$ $R_{wp} = 6.61$; $R_{exp} = 5.45$; $\chi^2 = 1.22$					$A_0 = 5.2147(7) \text{ \AA}$; $B_0 = 5.3295(6) \text{ \AA}$; $C_0 = 7.3933(5) \text{ \AA}$ Unit cell volume (V) = $205.41(9) \text{ \AA}^3$ $R_p = 5.68$ $R_{wp} = 8.03$; $R_{exp} = 7.65$; $\chi^2 = 1.10$			

**Figure 4.** Evolution of lattice parameters and unit cell volume of NNBT05 as functions of pressure. For comparison, the pressure dependence of the volume of pure sodium niobate is also plotted. The solid lines are fitted to the experimental data using the Birch–Murnaghan equation of state. The error bar on the pressure is the size of the symbol.**Figure 5.** Schematic representation of A-containing cubooctahedra (left side) and BO_6 octahedra (right side).

From Figure 4, it can be seen that the lattice compression of the orthorhombic phase of NNBT05 was anisotropic. The compressibility of the ‘c’ parameter was smaller in comparison with those of the ‘a’ and ‘b’ parameters. Compressibility (at ambient conditions ($1/l$ (dl/dP))) along a, b, and c was 0.0028 , 0.0025 and 0.0020 GPa^{-1} , respectively. This may be compared with the values for pure NaNbO_3 , which showed the compressibility

along *a*, *b*, and *c* as 0.0027, 0.0022, and 0.0018 GPa^{−1}, respectively. The compressibility in the high-pressure paraelectric phase in NaNbO₃ was very different from that in the low-pressure phase. However, in NNBT05, we did not find any significant discontinuous change in the compressibility as a function of pressure. Fitting the pressure versus volume data with a third-order Birch–Murnaghan equation of state gave a bulk modulus *B* (≈164.5 ± 1.0) GPa (with *B'* = 4 fixed) for the orthorhombic phase at room temperature. This experimental bulk modulus in NNBT05 was fairly close to that in NaNbO₃ (*B* ≈ 157.5 GPa) at room temperature.

It is evident from the foregoing discussion that the doping of BaTiO₃ in the NaNbO₃ matrix results in reduced compressibility, because on doping the free space in the A-containing cuboctahedron reduces. It was found that in perovskites in which the A cation has a lower formal charge than the B cation, the AO₁₂ sites are more compressible than the BO₆ octahedra (see Figure 5). To reduce the unit cell volume, the tilts of the BO₆ octahedra increased with increasing pressure [43]. Thus, the application of pressure drives this type of perovskite structure away from the phase transition boundary to higher-symmetry structures.

On the basis of the Raman experimental results, we can conclude that NNBT05 showed changes in the Raman spectra from 1 to 1.9 GPa, 1.9 to 4.6 GPa, and above 5.4 GPa. Below 5.4 GPa, the spectra evolved continuous changes. It can be seen that the intensity of phonons located at 150–350 cm^{−1} gradually changed with increasing pressure, up to 5.4 GPa. Then, these Raman peaks merged and became broad peaks as the pressure reached 9.9 GPa. Meanwhile, BO₆ octahedra showed a growing distortion under higher pressure, as a result of the volume shrinkage. Hence, additional changes to the Raman spectra occurred when the pressure held at more than 10 GPa. As shown for pure NaNbO₃, the observed anomalies in Raman spectra at a low pressure of ~2 GPa were mainly attributed to significant changes in Nb–O–Nb bond angles as a result of complex reorientations of NbO₆ octahedra [19,29]. On the other hand, the lattice beyond 8 GPa belonged to a paraelectric orthorhombic phase with space group *Pbnm*. Increasing the pressure resulted in a shrinking of the polyhedral volume, and the cation was displaced towards the symmetric position to avoid over bonding. In addition to this, the ab-initio calculation of the enthalpy in the various phases of NaNbO₃ [19] also supported the stabilization of the paraelectric (*Pbnm*) phase at high pressure. In view of this, we may conclude that on application of pressure, the ferroelectric phase of NNBT05 may transform to the paraelectric phase at a high pressure of ~5 GPa, which is lower than the transition pressure of ~8 GPa in pure NaNbO₃. Since structural changes in the structure of the dodecahedra are very small due to changes in the average ionic radii in A and B positions, at 5% substitution, the lowering of pressure could be attributed to the chemical pressure introduced due to the substitution of BaTiO₃.

4. Conclusions

In summary, we have investigated the effect of pressure on the crystal structure and structural phase transition behaviour in the engineered ferroelectric phase of sodium niobate using Raman scattering and powder X-ray diffraction techniques. At pressure > 1 GPa, noticeable changes in the Raman spectra were seen in the low wavenumber modes (150–300 cm^{−1}). These observations are most likely associated with significant changes in Nb–O–Nb bond angles as result of complex reorientations of NbO₆ octahedra in the parent ferroelectric phase, as in pure NaNbO₃. Further changes were observed at ~4.6 GPa, which may be due to a transition to the paraelectric phase, which has a lower pressure than that of ~8 GPa in pure NaNbO₃. For pressure levels greater than 5.4 GPa, the Raman spectrum became diffusive and may be associated with an increase in inhomogeneities in the crystal. We have determined the bulk modulus of NNBT05 to be 164.5 GPa, which is fairly close to that of 157.5 GPa observed in NaNbO₃.

Author Contributions: S.K.M., R.M., N.G. and S.L.C. formulated the problem. N.G. carried out the high-pressure Raman and X-ray diffraction experiments. S.G. performed high-pressure Raman experiments. S.K.M. and N.G. analyzed experimental data. All the authors discussed the results and wrote the manuscript. All authors have read and agreed to the published version of the manuscript.

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Conflicts of Interest: The authors declare that they have no known competing financial interest or personal relationship that could have appeared to influence the work reported in this paper.

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