

Review

# Metal Sulfide Photocatalysts for Hydrogen Generation: A Review of Recent Advances

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**Abstract:** Metal-sulfide nanostructures have piqued the interest of researchers for decades due to their intriguing optoelectronic properties. Indeed, significant advances and improvements have been made in various fundamental aspects for cutting-edge applications, such as water splitting and hydrogen production. Furthermore, rising demand for low-dimensional materials due to lower material consumption and improved performance due to quantum size effects has spurred research on semiconducting metal sulfides. Consequently, size-controllable nanostructures with diverse morphologies have been fabricated and studied for potential applications. However, the photocatalytic hydrogen evolution rate is still limited mainly by fast recombination rate, poor solar energy utilization and lack of surface-active sites for H<sub>2</sub> reduction. This review will highlight particularly recent findings in metal-sulfide-based photocatalysts for hydrogen evolution reactions, considering the swift development and excellent research in this field. Following a brief overview of fundamental properties, we will explore state-of-the-art strategies for enhancing H<sub>2</sub> generation efficiencies over the pristine, heterostructured and co-catalyzed metal-sulfide photocatalysts.



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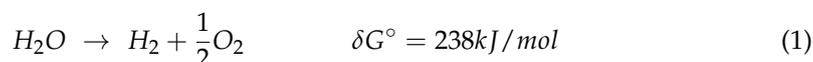


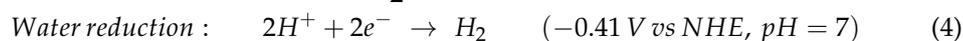
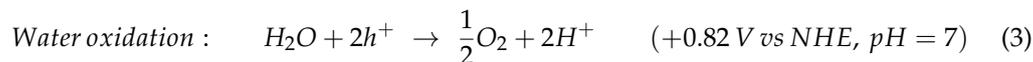
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**Keywords:** metal sulfide; photocatalyst; water splitting; hydrogen generation; heterogeneous photocatalyst; visible light response; co-catalyst; renewable energy

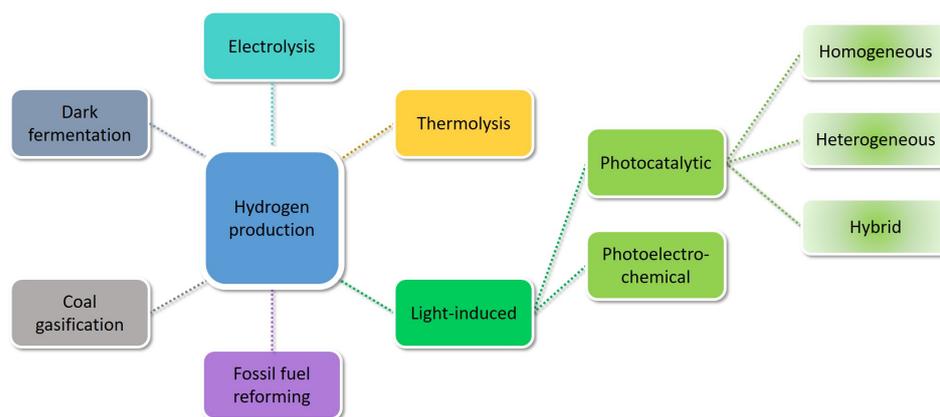
## 1. Introduction

Increasing demand for energy consumption triggers a search for new renewable and sustainable energy sources. On the other hand, the emerging climate crisis, such as increased CO<sub>2</sub>, NO<sub>2</sub>, and other air pollutants, as well as global warming, steadily urges the development of new materials to cope with these problems, alongside providing high alternative energy harvesting and storage capacities. According to the United Nations (UN), energy is the dominant contributor to climate change, providing almost 60 % of total global greenhouse gas emissions [1]. Therefore, investigation of new renewable energy sources has also been included (Goal 7) in the 17 Sustainable Development Goals implemented by the UN [1]. In this regard, as a sustainable energy source, H<sub>2</sub> production has been the focus of increasing fundamental and applied research. In general, hydrogen can be obtained from different sources such as coal or liquid fossil fuels by employing different chemical processes, as summarized in Figure 1. Among all other methods, photocatalytic water splitting has the simplicity of using semiconducting particles in solution with incident sunlight to produce hydrogen and oxygen, as follows.





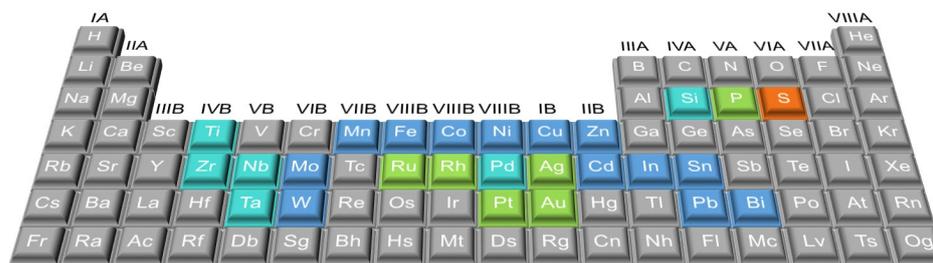
To initiate the oxidation reaction, the valence band (VB) of the photocatalyst must be more positive than the water oxidation potential (+0.82 V vs Normal Hydrogen Electrode (NHE)) (Equation (3)). Whereas, more negative conduction band (CB) in compare to hydrogen evolution potential (−0.41 V vs NHE) is required to drive the reduction reactions (Equation (4)). Ultimately, this process provides a clean and renewable source of hydrogen without producing greenhouse gases or any other harmful products. However, in practice, the simultaneous production of hydrogen and oxygen from water splitting by a single photocatalyst is a tough process since the photocatalyst must fulfill band structure and energy gap criteria. Therefore, combining co-catalysts is a commonly employed strategy for this purpose. In this regard, different catalysts, including homogeneous, heterogeneous, and hybrid, have been developed. Among all others, heterogeneous photocatalysts are known for their high chemical stability in a liquid medium, high recycling or robustness, easier recovery or separation from the reaction medium, nontoxicity, low material, and operational cost, enabling selective product formation, etc.



**Figure 1.** Most used possible routes for hydrogen generation. Currently, the solar-induced hydrogen evolution from water splitting is the most promising approach among others since it is environmentally benign, cost-effective, and offers the potential for high conversion efficiency.

Semiconductor metal sulfides have recently attracted great attention due to their superior physical and chemical properties. Up to nowadays, several binary metal sulfide (MS) compounds (such as PbS, CdS, ZnS, MoS<sub>2</sub>, SnS<sub>2</sub>, Bi<sub>2</sub>S<sub>3</sub>, In<sub>2</sub>S<sub>3</sub>, Cu<sub>2</sub>S, NiS/NiS<sub>2</sub>, and CoS<sub>2</sub>) as well as their derivatives and heterostructures have been intensively employed for versatile purposes [2–7]. In Figure 2, we have summarized the most employed metals as sulfide compounds for photocatalytic hydrogen generation. Most of these compounds have been reported with a remarkable visible light response, sufficient active sites, and appropriate reduction potential of H<sup>+</sup>/H<sub>2</sub> as an effective photocatalyst. Moreover, emerging quantum size effects enable further tunability towards fast charge transfer, enhanced excited state lifetime, etc. [8,9]. Thus, in the last decade, tremendous work has been devoted to improving solar hydrogen generation over the MS semiconductor photocatalysts [10–12]. Among other MS materials, ternary sulfides, such as CuInS<sub>2</sub>, CuGaS<sub>2</sub> have been shown to have high selectivity and a high photocatalytic production rate, particularly when coupled to titanium dioxide (TiO<sub>2</sub>) [13,14]. Especially, MS with reduced dimensionalities, such as 2D nanosheets, flakes, and anisotropic structures, are in the spotlight of current research. Moreover, the functionalization of low-dimensional MS structures is a growing research toward

high hydrogen production [15]. For example, using metals as dopants or other compounds (i.e., see Figure 2) as co-catalyst is a prominent way of enhancing photocatalytic activity.

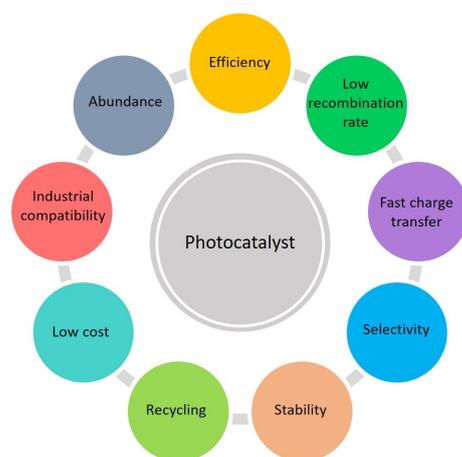


**Figure 2.** Chemical elements whose sulfide compounds have been widely used in photocatalytic hydrogen generation (blue boxes). Elements employed as dopants (green) and whose oxides or other compounds (cyan) were used to construct heterostructures with metal sulfides for hydrogen evolution are also marked. Please note that lanthanides and actinides are not shown.

Metal sulfide nanostructures have been increasingly employed in hydrogen evolution reactions, especially in the last couple of years [2,16,17]. In this review article, we will discuss the recent breakthrough, mainly in the last two years, regarding nanostructured metal-sulfide photocatalysts for hydrogen evolution as an alternative energy. Following this general introduction, we will briefly discuss the fundamental properties of MS-based photocatalytic systems. This will be followed by discussions on the specific MS systems that have been the focus of very recent research regarding photocatalytic hydrogen production. In each subsection, we will present the possible and most promising routes to photocatalytic enhancement.

#### *Metal Sulfide Photocatalysts: Design and Mechanisms*

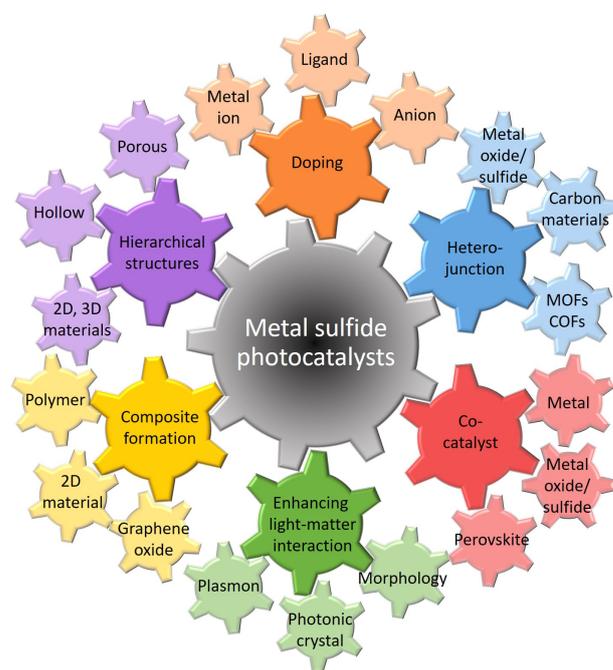
As listed in Figure 3, there are specific parameters that are important to improve photocatalytic reaction yield in addition to general environmental and industrial requirements. Certainly, the search for an excellent photocatalyst to meet all requirements has triggered tremendous research resulting in versatile new photocatalysts and photocatalytic reaction routes. Any catalytic process involves fundamental steps like adsorption, interaction, desorption, etc., which determine the overall reaction mechanism, products, and yield [18]. On the other hand, perfect parametrization of these factors enables the design of an effective photocatalyst.



**Figure 3.** General requirements for designing high-efficiency photocatalysts.

Although photocatalysis based on MS semiconductors is considered an economical, environmentally benign, renewable, and clean technology, the usefulness of these photocatalysts is limited by the low solar energy utilization and the fast recombination of

photoexcited electron-hole pairs. Several strategies have been developed to overcome these intrinsic limitations, that we will discuss in the following parts of this review. General strategies and the most employed combination for the related strategy for metal sulfide-based photocatalysts are listed in Figure 4. Designing new morphologies and decoration of particles are recent trends to enhance the light spectrum absorbed by the semiconducting photocatalysts. For example, hollow structures, porous morphology, nanorods (NRs), 1D nanowires (NWs), 0D quantum dots (QDs), 2D sheets, and layered structures are the focus of current research and development regarding the photocatalytic hydrogen production. In this regard, mesoporous structures have been shown to facilitate the transport of reactant and reaction products [19]. Besides that, the porous medium improves sunlight absorption due to multiple reflections of light waves in porous networks [20]. Moreover, porous structures possess a high Brunauer-Emmett-Teller (BET) surface area with more active sites, which facilitates the photocatalytic reaction. Besides enhancing solar energy utilization, photocatalytic enhancement also requires improved separation efficiency of photogenerated charge carriers, which can also be achieved by designing appropriate architectures with the help of low-dimensional nanostructures. According to Tang et al., the charge carrier recombination on the surface of semiconducting photocatalyst is  $\sim 2 \times 10^7$  fs, which is three orders of magnitude faster than chemical oxidation/reduction reactions ( $10^{10}$ – $10^{11}$  fs) [21]. Therefore, single component photocatalysts are frequently ineffective for  $H_2$  generation reaction. In order to alleviate this shortcoming combining metal sulfides with metal oxide co-catalysts has been demonstrated as a powerful approach. This type of composite photocatalyst reduces charge carrier recombination of MS while also broadening the light absorption range of wide band gap metal oxide.



**Figure 4.** The most frequently employed strategies and representative components for enhancing hydrogen production over the metal sulfide photocatalysts. The figure was designed in accordance with the recent literature discussed in this review.

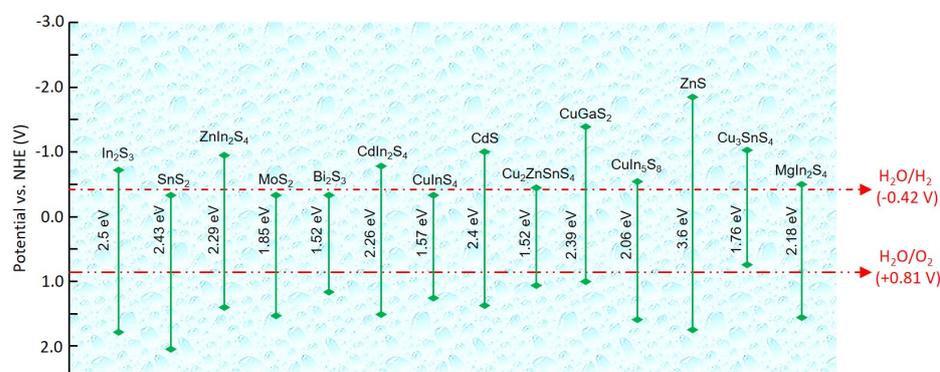
Notably, size reduction improves the touching surface areas and strengthens the inter-particle interactions, facilitating rapid photoexcited charge transport. Moreover, emerging quantum size effects bring about strong electronic interaction developing fast charge transfer. Also, the formed discrete states and trap states at quantum particles further improve the electron-hole separation, leading to a reduced charge recombination rate [6,22–24]. The robustness and recycling cycles are being improved with more rugged materials and pro-

protective surface coatings. Other than these, coupling the MS photocatalysts with plasmonic metallic structures has been intensively investigated over the last few years. When noble metal (e.g., Au, Ag, Pt) NPs are attached to the semiconductor nanostructures, the photo-generated electrons in the CB of semiconductor can transfer to the adjacent metals due to their lower Fermi energy level.

The Schottky barrier formed at the metal/semiconductor interface promote electron/hole separation in photocatalysis, while the metal act as electron sinks. The photoinduced the electrons in noble metals and holes in the VB of semiconductor participate in subsequent redox reactions. The whole process increases charge carrier lifetime and improves photocatalytic activity. The unique localized surface plasmon resonance (LSPR) of noble metal NPs can boost photocatalytic activity by increasing incoming photon absorption. Furthermore, the facile tunability of LSPR (i.e., by adjusting size and shape) allows for a broad spectrum of light absorption.

Another useful parameter to achieve higher hydrogen evolution rate (HER) is a high active crystallographic orientation or facet. In this context, zinc-blende CdS nanocubes with a well-controlled charge flow and selective transfer of photogenerated electrons and holes to the (111) and (100) facets, respectively, demonstrated better charge carrier separation and significant photocatalytic H<sub>2</sub> production by water splitting [25].

Usual high-performance photocatalytic materials with large band gaps, such as TiO<sub>2</sub>, ZnO, and others, absorb light exclusively in the near-UV region ( $\lambda = 400$  nm). Most low-bandgap photocatalysts, such as CdS, on the other hand, retain photoactivity for short time due to photocorrosion, which occurs when the sulfur ions are attacked by photogenerated holes and oxidized to elemental sulfur. Also, early researches on metal chalcogenide photocatalysts revealed that hole transport plays a limiting role in these nanostructures. To tackle this shortcoming, the isolation of photoinduced holes from the reaction site has been suggested [26]. For example, in Pt/CdS-CdSe heterostructures, the holes are confined to the CdSe, whereas the delocalized electrons are transferred to the Pt tip. This was due to larger VB edge of CdSe than CdS, and consequently resulted in significant photocatalytic activity [11]. Compared to photocatalysts made up of only one type of semiconductor, heterojunction structures often possess altered band alignment at the interface, enhancing carrier separation and limiting carrier recombination for enhanced photocatalytic activity. Sulfide atoms near metal ions in typical metal sulfides potentially act as a catalytic site for the attachment and detachment of H atoms. Therefore, the increased number of sulfur sites is another parameter for enhancing photocatalytic activity. Noteworthy, compared to crystalline metal sulfides, amorphous metal sulfides often exhibit more unsaturated sulfur atoms with excessive chemical reactivity, functioning as active sites for H<sub>2</sub> production. In Figure 5, the band gap energy with respect to the water oxidation and reduction potential is summarized for representative semiconductor metal sulfides.

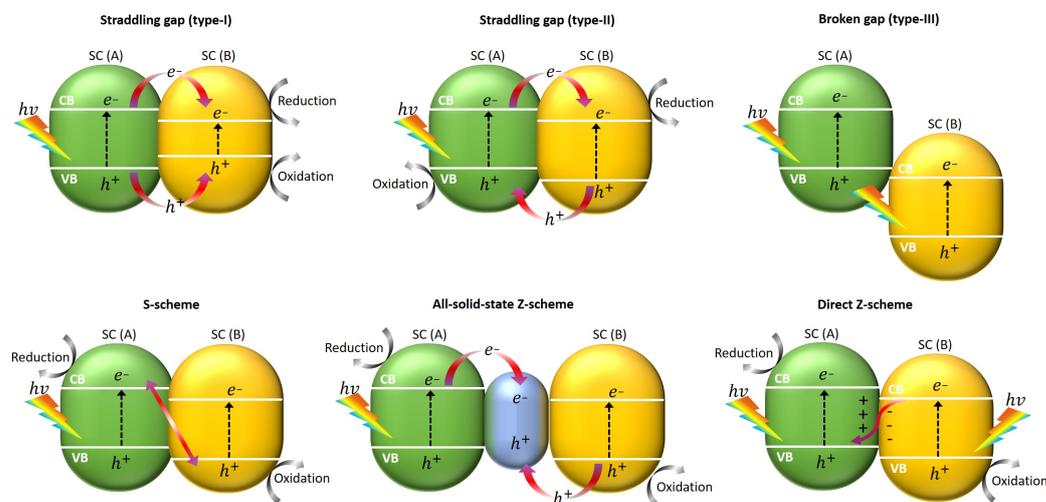


**Figure 5.** The band gap and band positions of the selected MS semiconductors, along with the needed oxidation and redox potential for water splitting. Data obtained from, In<sub>2</sub>S<sub>3</sub> [27], SnS<sub>2</sub> [28], ZnIn<sub>2</sub>S<sub>4</sub> [29], MoS<sub>2</sub> [30], Bi<sub>2</sub>S<sub>3</sub> [31], CdIn<sub>2</sub>S<sub>4</sub> [32], CuInS<sub>2</sub> [33], CdS [34], Cu<sub>2</sub>ZnSnS<sub>4</sub> [35], CuGaS<sub>2</sub> [36], CuIn<sub>5</sub>S<sub>8</sub> [37], ZnS [38], Cu<sub>3</sub>SnS<sub>4</sub> [39], MgIn<sub>2</sub>S<sub>4</sub> [40].

Morphological modifications and the construction of heterojunction photocatalysts are the two most employed techniques to increase the performance of photocatalytic hydrogen evolution significantly. The close distance between the two nanosheet subunits induces a strong interaction, facilitating the separation and transfer of charge carriers. In this regard, a reverse cation exchange-mediated and a  $\text{Cu}_{2-x}\text{S}$  nanocube template-assisted growth strategy for fabricating hollow multinary metal sulfide was reported recently [41]. Thus, it was found that in contrast to the typical cation exchange mechanism mediated by the metal sulfide constant, the presence of tri-*n*-butylphosphine (TBP) can reverse cation exchange to produce a series of hollow metal sulfides. Various hollow metal sulfide cubic NSs were demonstrated using this method, including binary compounds ( $\text{CdS}$ ,  $\text{ZnS}$ ,  $\text{Ag}_2\text{S}$ ,  $\text{PbS}$ ,  $\text{SnS}$ ), ternary compounds ( $\text{CuInS}_2$ ,  $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ ), and quaternary compounds (single-atom platinum attached  $\text{Zn}_x\text{Cd}_{1-x}\text{S}$ ;  $\text{Zn}_x\text{Cd}_{1-x}\text{S-Pt}_1$ ). Also, a light-induced deposition approach to precisely load nickel sulfide nanodots onto the surface of  $\text{g-C}_2\text{N}_2$  resulted in both enhanced photocatalytic hydrogen evolution and oxygen reduction performances over composite photocatalysts [42]. In general, as demonstrated in Equations (1)–(4), photocatalytic water splitting involves three significant steps: the generation of electron-hole pairs by photon absorption, the separation of electron-hole pairs and migration to the photocatalyst surface, and the splitting of water molecules via charge carriers on the photocatalyst surface.

One should emphasize that depending on the selected co-catalyst or heterojunction, the reaction pathway follows different routes, which are mainly dictated by the chemical potential of the components. In this regard, different routes have been reported in the literature, which are also valid for metal sulfide photocatalysts. The most pronounced photocatalytic reaction types are illustrated in Figure 6. In type-I heterojunction, a built-in electric field is established by a photogenerated charge carrier inside the large band gap semiconductor before it is transmitted into the narrower band gap semiconductors; thus, electron-hole pairs cannot be spatially separated. On the contrary, in the type-II heterojunctions between two semiconductors, rapid generation of a built-in electric field induces a prolonged lifetime of photogenerated charge carriers. Therefore, in engineering a photocatalytic system for enhanced charge carrier separation in heterojunction, facilitating a suitable type-II interfacial contact with effective carrier migration pathways is critical (i.e., see Figure 6). For instance, Liu et al. synthesized a type-II hexagonal cadmium sulfide ( $\text{CdS}$ )/molybdenum disulfide ( $\text{MoS}_2$ ) heterostructure for photocatalytic hydrogen production [43]. The synergetic enhancement in the charge dynamics of the excited carriers on the heterostructure interface of the  $\text{CdS}$  on the  $\text{MoS}_2$  film was investigated by transient absorption spectroscopy. Type-III heterostructures are called Van der Waals broken band gap heterojunctions, which are less investigated in applying two-dimensional (2D) photocatalytic materials for hydrogen production. That is because the photogenerated charge carriers are not synergistically coupled; thus, electron-hole pairs exhibit a fast decay. In the S-scheme heterojunction, strong photogenerated holes and electrons in the valence and conduction bands, respectively, participate in the redox reactions. Meanwhile, weak photogenerated carriers are recombined by the internal electric field and Coulomb gravity. As a result, more efficient electrons for photocatalytic hydrogen reduction can be used [44]. In all-solid-state Z-scheme photocatalysts, NPs such as noble and transition metals ( $\text{Ag}$ ,  $\text{Au}$ ,  $\text{Rh}$ ,  $\text{Ru}$ ,  $\text{Cu}$ ,  $\text{Pt}$ , etc.) are used as electron conductors/mediators between two SCs due to their strong reduction ability. This type of heterojunction has been studied extensively over the last decade with different combinations of the SCs after the pioneering work of the anisotropic  $\text{CdS-Au-TiO}_2$  nano junction SCs [45]. The direct Z-scheme photocatalytic process, which involves two steps of photoexcitation, mimics the natural photosynthesis process, which also requires the spatial isolation of photogenerated electrons and holes. By pooling photogenerated electrons, an electron-rich region forms in the CB of SC I, and by pooling photogenerated holes, a hole-rich region generates in the VB of SC II, which reduces bulk electron-hole recombination in this type of heterojunction. Meanwhile,

the photogenerated electrons in the CB of PC II and holes in the VB of PC I with less redox power recombine.



**Figure 6.** Schematic illustration of the most frequently reported charge transfer mechanisms for any photocatalytic systems. SC(A) and SC(B) stand for the two different semiconductor components of the corresponding photocatalyst.

## 2. Hydrogen Generation as an Energy Material

### 2.1. Binary Metal Sulfide Compounds

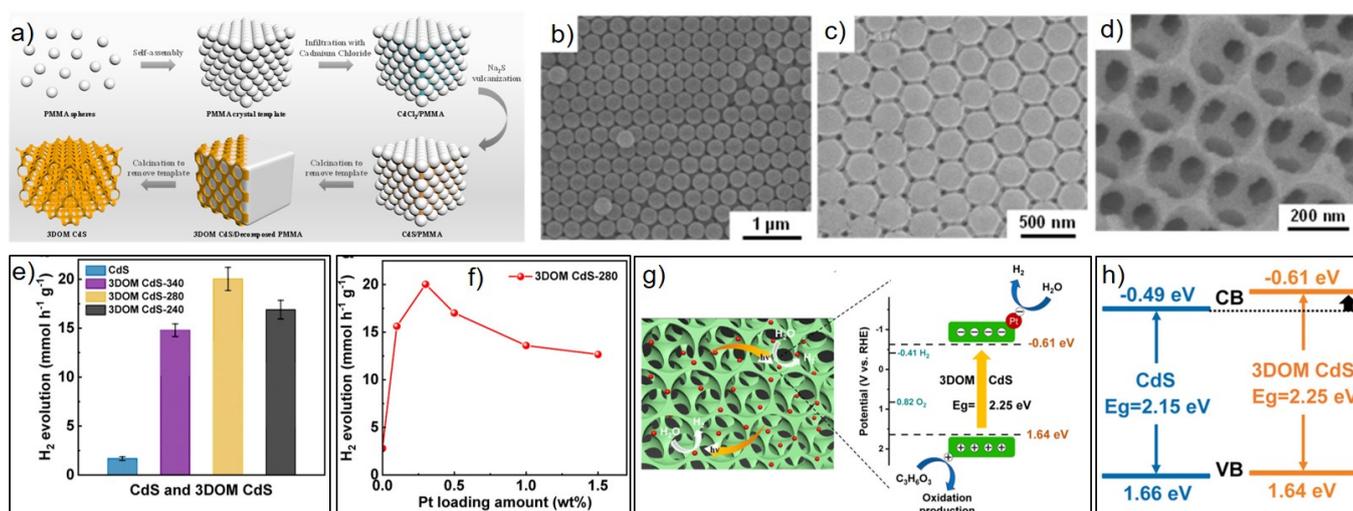
Semiconductors, with a large band gap energy, can only be activated under UV irradiation, which limits their efficiency in terms of using a narrow light spectrum. CdS, with its energy level of CB being more negative than the potential of water reduction, is one of the most suitable semiconductors for water splitting by utilizing visible-light irradiation [46–51]. However, a general drawback, photocorrosion, is also related to CdS. Thus, sulfur anions ( $S^{2-}$ ) tend to be oxidized easier than water at the surface of the anode. Ultimately, hole scavengers, which can easily provide electrons reacting with holes such as alcohol,  $S^{2-}$ ,  $SO_3^{2-}$ , etc., are required to react with  $h^+$  to neutralize the reaction for continuous  $H_2$  evolution. Thus, introducing CdS as the main photocatalyst or as a component in different chemical compounds is one of the effective ways to improve its usefulness for water splitting. For example, it is known that nickel boride ( $Ni_xB$ ), which is a well-known low-cost photocatalyst, suffers from the weakening of atomic H adsorption on Ni active site by the electron transfer from B to Ni atoms, which causes the inhibition of  $H_2$  evolution. In order to enhance the photocatalytic  $H_2$  generation over CdS, Zhang et al. have prepared a  $Ni_xB$ -rGO/CdS hetero-photocatalyst, where the  $Ni_xB$  was doped by metalloid P to enhance its H adsorption ability [52]. The 8–10 nm  $Ni_xB$  are loaded on reduced graphene oxide (rGO) and coupled with CdS nanoparticles. The as-synthesized photocatalysts exhibited  $H_2$  evolution rate of  $5.79 \text{ mmol h}^{-1} \text{ g}^{-1}$ , being 1.8 and 9.9 times higher than the  $Ni_xB$ -rGO/CdS and rGO/CdS photocatalysts, respectively. Similarly, CdS@ $Ni_2P$  was prepared by a simple hydrothermal method followed by in situ photodeposition [53]. The highest photocatalytic hydrogen production rate achieved with this composite was  $287 \mu\text{mol g}^{-1} \text{ h}^{-1}$ , that is  $\sim 98.3$  times that of pristine CdS. The higher activity was related to the effective interface between the  $Ni_2P$  co-catalyst and CdS nanorods, which leads to better separation and transport of photogenerated electrons.

For solar-driven water splitting,  $NiWO_4$ /CdS nanosheets-on-nanorods direct Z-Scheme heterostructure was developed by using a facile in-situ approach (see Figure 6 for Z-scheme photocatalysts) [54]. The fabricated  $NiWO_4$ /CdS heterostructure photocatalysts showed a  $H_2$  evolution rate of  $26.43 \text{ mmol g}^{-1} \text{ h}^{-1}$ , which is 75 times higher than that of bare CdS. It was shown that the nanostructures with multiple active sites, intimate

contact interface, and improved charge separation rate are responsible for the enhanced photocatalytic performance.

Another strategy to improve the photocatalytic activity of semiconductor materials is employing metalloid compounds, which form a Schottky junction with the semiconductor at the heterogeneous interface. In this regard, tungsten nitride (WN) nanosheets, which have excellent electron conductivity, were employed as a new noble-metal-free cocatalyst to improve the H<sub>2</sub> evolution performance of CdS NPs via the one-pot solvothermal method [55]. The H<sub>2</sub> production rate of the WN/CdS composite photocatalyst under visible light irradiation reached 24.13 mmol g<sup>-1</sup> h<sup>-1</sup>, which is 9.28 times that of pure CdS NPs. Similarly, P-rich platinum diphosphide (PtP<sub>2</sub>) nanodots were used to improve the photoactivity of CdS by a facile phytic acid-assisted pyrolysis [56]. The obtained PtP<sub>2</sub>@C/CdS(10 wt%) photocatalyst presents an optimum H<sub>2</sub> evolution rate of 9.76 mmol g<sup>-1</sup> h<sup>-1</sup> with apparent quantum efficiency (AQE) of 41.67% (420 nm), which is 2.2 and 34.8 times higher than that of Pt@C/CdS and clean CdS, respectively. It was found that the electron-enriched P-rich PtP<sub>2</sub> can serve as the effective H-proton adsorption active site to enhance the photocatalytic activity greatly.

As discussed previously, designing porous photocatalysts is one of the major trends for HER enhancement. In this perspective, three-dimensional ordered macroporous (3DOM) structures have received increased attention because their open interconnected hierarchical macroporous architecture facilitates the transport and diffusion processes of species, leading to enhanced photocatalytic enhancement. Furthermore, the regularly ordered structure of 3DOM results in photon band gaps and the slow photon effect of photonic crystals, which can significantly improve the photoinduced carriers and boost photocatalytic activity. Additionally, changes to the pore walls typically do not modify the 3DOM structure, making it easy to build efficient photocatalysts with diverse materials. In this regard, recently, a CdS photocatalyst with a 3DOM structure using polymethylmethacrylate as a template was reported [57]. In line with the expectations, photocatalytic H<sub>2</sub> evolution rate of 20.1 mmol g<sup>-1</sup> h<sup>-1</sup> was achieved, which is remarkably improved in comparison with the CdS NPs. Moreover, as shown in Figure 7f, HER was further enhanced by loading Pt metal. The schematic view of the preparation procedure, corresponding structural characterization, HER results, and proposed H<sub>2</sub> evolution mechanism over CdS 3DOM photonic crystals are summarized in Figure 7.

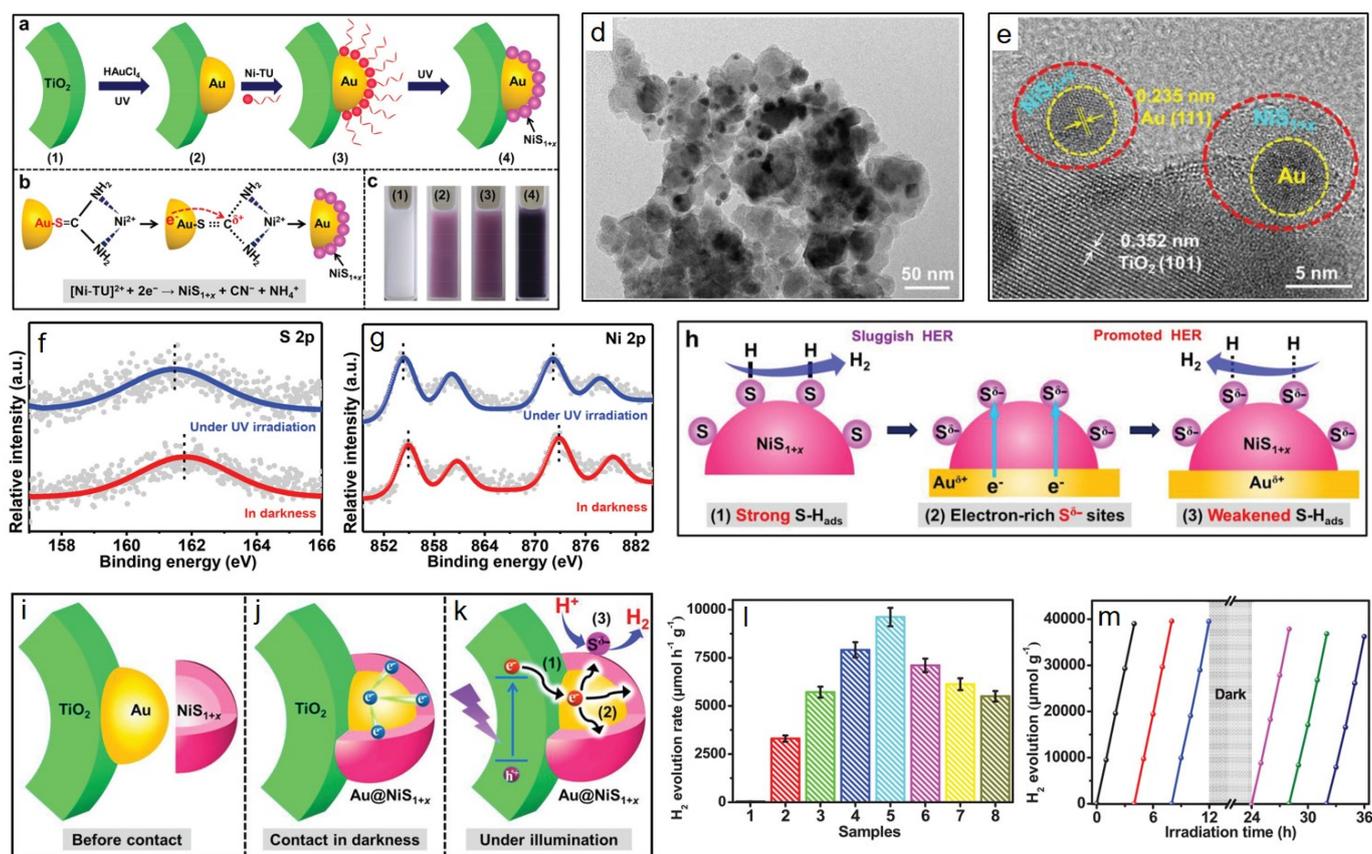


**Figure 7.** (a) Schematic illustration of the overall synthetic strategy of 3D ordered macroporous (3DOM) CdS. (b) PMMA crystal template. (c) PMMA/CdS. (d) 3DOM CdS. (e) HER efficiency over the CdS NPs and 3DOM CdS with 0.3 wt% Pt loading. (f) Efficiency of HER over the 3DOM CdS-280 with different Pt loadings. (g) Schematic illustration of 3DOM CdS for photocatalytic H<sub>2</sub> production. (h) Schematic illustration of the band gap states of CdS NPs and 3DOM CdS. [57] Copyright 2022, American Chemical Society

Also, a hybrid heterojunction consisting of CdS NPs and structurally defined graphene nanoribbons (GNRs) was reported with an enhanced HER [58]. Through photocurrent, electrochemical impedance, steady-state photoluminescence, and time-resolved fluorescence decay spectroscopy, it was revealed that the CdS/GNR heterojunction could efficiently reduce the photoexcited electron-hole pair recombination. Notably, the CdS/GNR hybrid displayed H<sub>2</sub> evolution rate up to 22.4 mmol g<sup>-1</sup> h<sup>-1</sup> under Vis-light irradiation ( $\lambda = 420$  nm), which is much greater than that of individual CdS and GNRs. Lei et al. employed the facile solid-liquid mixing method to load CdS onto WC [59]. 3 wt% WC-loaded CdS photocatalyst (CdS/WC) exhibiting the highest HER of 9180  $\mu\text{mol g}^{-1} \text{h}^{-1}$  with an AQE of 14.3% at 420 nm. Thus, incorporating WC improves the light absorption capability of CdS NPs. Moreover, a more conductive WC provides a fast electron transfer channel and water reduction site, which enhances the spatial photogenerated charge separation while facilitating the H<sub>2</sub> production.

Alongside metal sulfide-based heterostructures, doping the metal sulfides with metals is also a promising trend toward photocatalytic H<sub>2</sub> generation. For example, Ni<sub>x</sub>Cd<sub>y</sub>S composite photocatalysts possessing porous structure, large specific surface, and highly dispersed metal active sites were synthesized using Ni-doped CdS [60]. The fabricated Ni<sub>x</sub>Cd<sub>y</sub>S photocatalysts showed three times higher H<sub>2</sub> yield (i.e., 8.45 mmol g<sup>-1</sup> h<sup>-1</sup>) than the CdS. NiS<sub>1+x</sub> compounds have recently gained interest as low-cost transition-metal chalcogenides for effective photocatalytic H<sub>2</sub> evolution [61,62]. Co-catalyst-free NiS ( $\alpha$  and  $\beta$  phases) nanostructures for photocatalytic H<sub>2</sub> generation was reported [63]. Unfortunately, due to the limited number of exposed sulfur sites and their strong interaction with adsorbed hydrogen atoms, these materials are ineffective for H<sub>2</sub> generation. To overcome these limitations, Gao et al. developed a nanostructured core-shell Au@NiS<sub>1+x</sub> co-catalyst and adjoined with TiO<sub>2</sub> by employing a two-step route [61]. The synthetic route for producing the core shell structures, as well as their characterisation, is shown in Figure 8a–g. The optimized TiO<sub>2</sub>/Au@NiS<sub>1+x</sub> (1.7:1.3) exhibited high HER of 9616  $\mu\text{mol g}^{-1} \text{h}^{-1}$  with an AQE of 46.0% at 365 nm, which is 2.9 and 1.7 times higher than the HER obtained for the TiO<sub>2</sub>/NiS<sub>1+x</sub> and TiO<sub>2</sub>/Au photocatalysts, respectively. By XPS studies complemented by DFT calculations, it was shown that the conductive-electrons of Au can transfer to sulfur-enriched NiS<sub>1+x</sub> resulting in electron-enriched active sulfur centers. This weakens the strong bonds between sulfur and adsorbed hydrogen atoms and facilitates desorption of adsorbed hydrogen for rapid hydrogen formation. Figure 8 depicts the photocatalytic performance as well as the associated charge transfer process.

Besides the aforementioned compounds, Fe<sub>3</sub>S<sub>4</sub> (greigite), which is a ferromagnetic material, has also recently been extensively investigated for energy applications, including high-energy batteries, electrochemical hydrogen storage because of its low-cost, and environmental benignity and natural abundance. In addition to the sulfur ions acting as H-adsorption/desorption sites, the mixed Fe<sup>2+</sup>/Fe<sup>3+</sup> valence states in the inverse spinel structure also endow this material with relatively higher electric conductivity than the FeS and FeS<sub>2</sub>, which makes it more suitable photocatalysts for H<sub>2</sub> generation. However, Fe<sub>3</sub>S<sub>4</sub> crystal possesses inappropriate surface adsorption energy for H atoms, which can be altered by doping with additional metal atoms. In this regard, Fe<sub>3</sub>S<sub>4</sub> nanorods were derived from a metal-organic framework by a facile method and its HER capability was further enhanced by Ni doping [64]. Thus, the organic dye (eosin-Y) sensitized, 1%Ni-doped Fe<sub>3</sub>S<sub>4</sub> nanorods showed remarkably improved H<sub>2</sub> generation rate of 3240  $\mu\text{mol g}^{-1} \text{h}^{-1}$  with an AQE of 12% under  $\lambda = 500$  nm. Due to its unique localized surface plasmon resonance in the near-IR region, the Cu<sub>2-x</sub>S has showed substantial promise in the field of broad-spectrum driven photocatalyst engineering. Nonetheless, inefficient carrier separation and severe photocorrosion limit its practical uses.



**Figure 8.** Schematic illustration of the synthesis procedure of the core-shell Au@NiS<sub>1+x</sub> combined TiO<sub>2</sub> photocatalyst (a) and the formation mechanism of NiS<sub>1+x</sub> on Au surface (b). (c) shows the corresponding photographs for (1) TiO<sub>2</sub>, (2) TiO<sub>2</sub>/Au, (3) TiO<sub>2</sub>/Au@Ni-TU (TU stands for thiourea), and (4) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>. (d) TEM and (e) HRTEM images of the TiO<sub>2</sub>/Au@NiS<sub>1+x</sub> (1.7:1.3) photocatalyst. (h) Schematic illustration for activating H-adsorbed S on Au-mediated NiS<sub>1+x</sub> by forming electron-enriched active S sites. In situ irradiated XPS spectra of S 2p (f) and Ni 2p (g) for TiO<sub>2</sub>/Au@NiS<sub>1+x</sub> in dark and under UV-light illumination. (h) Schematic view of charge transfer within TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>: (i) before contact, (j) contact in the dark and (k) during UV irradiation: the (1) photoelectron trapping by the Au mediator, (2) photoelectron transfer from Au to NiS<sub>1+x</sub>, and (3) fast H<sub>2</sub> evolution on the electron-enriched S active sites of Au@NiS<sub>1+x</sub> co-catalyst. (l) Photocatalytic HER over (1) TiO<sub>2</sub>, (2) TiO<sub>2</sub>/NiS<sub>1+x</sub>, (3) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(2.5:0.5), (4) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(2:1), (5) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(1.7:1.3), (6) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(1.5:1.5), (7) TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(1:2), and (8) TiO<sub>2</sub>/Au. (m) Recycling test for hydrogen evolution over the TiO<sub>2</sub>/Au@NiS<sub>1+x</sub>(1.7:1.3) photocatalyst. [61] Copyright 2022, Wiley-VCH.

Recently, multinary compounds based on Cu<sub>x</sub>S synthesized by a facile colloidal method were also reported as effective photocatalysts for the H<sub>x</sub> evolution [65]. There, the authors incorporated Ni, Co, Mn, Zn metals to enhance activity of Cu<sub>x</sub>S and reaction rates of H<sub>x</sub> generation was found in the order of Ni:Cu<sub>x</sub>S > Co:Cu<sub>x</sub>S > Mn:Cu<sub>x</sub>S > Zn:Cu<sub>x</sub>S as 4.0, 1.2, 0.9 and 0.7 mmol g<sup>-1</sup> h<sup>-1</sup>, respectively. The higher efficiency of the Ni-doped Cu<sub>x</sub>S nanorods can be attributed to the effective charge carrier separation and fast transport similar to the Ni-CdS, as discussed above. Molybdenum disulfide (MoS<sub>2</sub>) is another potential low-cost photocatalyst for hydrogen production reactions. According to the previous studies, the active sites for the catalytic reaction are mainly associated with the edge sites of 2D-layered MoS<sub>2</sub>, however, the basal plane shows poor activity toward catalytic reactions. Thus, activating the basal planes is another route for enhancing its HER. In this regard, Ni-doping and exfoliation were reported to convert the in-plane sites to active edge sites [66]. Furthermore, the activated MoS<sub>2</sub> was utilized for enhancing

the photocatalytic activity of CdS toward H<sub>2</sub> production. The resulting Ni-MoS<sub>2</sub>/CdS nanocomposite photocatalysts showed a H<sub>2</sub> evolution rate of 249 mmol h<sup>-1</sup> g<sup>-1</sup>, which is 70 times higher than pristine CdS NPs.

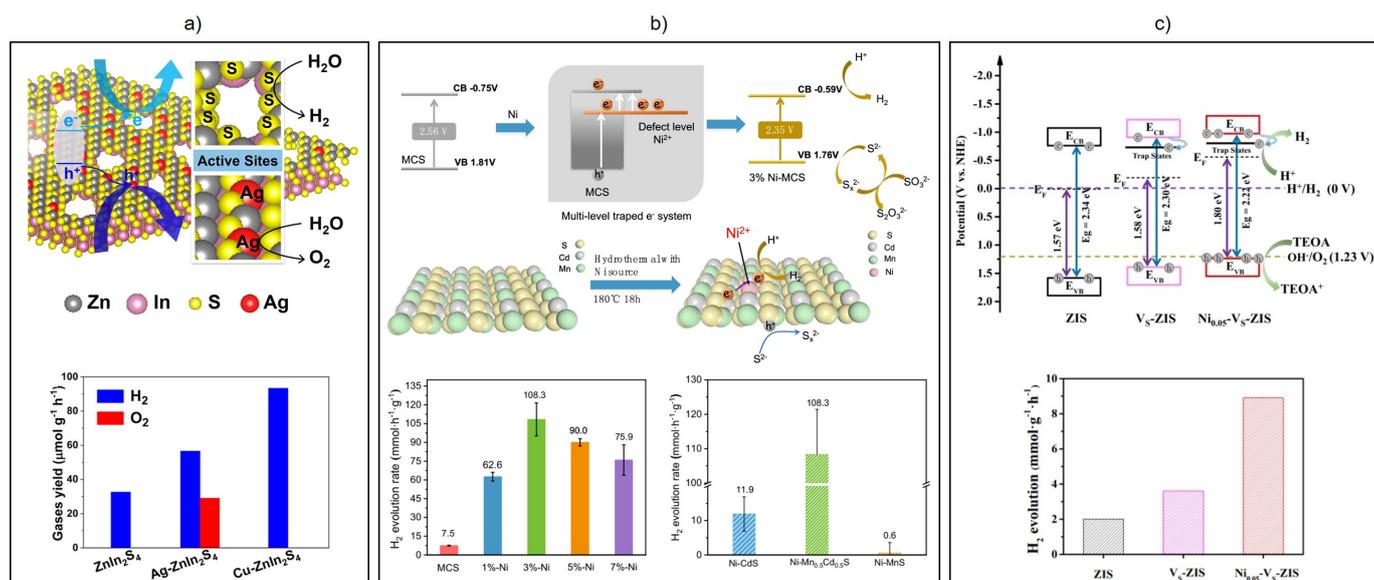
## 2.2. Ternary Metal Sulfide Compounds

Bimetals sulfides display more outstanding HER performance during continuous catalysis due to their unique bimetal synergistic effect and adjustable potential. Zinc indium sulfide (ZnIn<sub>2</sub>S<sub>4</sub>), one of the emblematic AB<sub>2</sub>S<sub>4</sub> ternary metal chalcogenides, is a layered structure n-type semiconductor with a direct band gap. ZnIn<sub>2</sub>S<sub>4</sub> can exist in three different crystal polymorphs (cubic, trigonal, and hexagonal), with the hexagonal phase being the most thermodynamically stable structure.

Tao et al. employed the wet-chemical method for the preparation of ultrathin MIn<sub>2</sub>S<sub>4</sub> (M = Zn, Cd, and Ni) nanoplates with a thickness of 2 nm, using as-grown β-In<sub>2</sub>S<sub>3</sub> nanoplates as a template [67]. Further optimization showed that loading of Ni NPs as co-catalysts, the 3.0 wt% Ni/ZnIn<sub>2</sub>S<sub>4</sub> photocatalysts exhibited the highest H<sub>2</sub> evolution rate of 19.9 mmol h<sup>-1</sup> g<sup>-1</sup>, which is six times larger than the ZnIn<sub>2</sub>S<sub>4</sub> NPs. Recently, ZnIn<sub>2</sub>S<sub>4</sub> monolayers with dual defects (Ag dopants and nanoholes) for improved HER were also reported [68]. While the undoped ZnIn<sub>2</sub>S<sub>4</sub> could produce HER of 32.6 μmol h<sup>-1</sup> g<sup>-1</sup>, the Ag-ZnIn<sub>2</sub>S<sub>4</sub> showed 56.6 μmol h<sup>-1</sup> g<sup>-1</sup>. Most importantly, the Ag-doped ZnIn<sub>2</sub>S<sub>4</sub> could achieve 29.1 μmol h<sup>-1</sup> g<sup>-1</sup> O<sub>2</sub> evolution, which was not seen for the undoped ZnIn<sub>2</sub>S<sub>4</sub>. The proposed mechanism and the obtained reaction yield are given in Figure 9a.

In order to overcome the intrinsic limitations of the MoS<sub>2</sub>, Lei et al. prepared a cobalt-modified amorphous molybdenum sulfide (Co-MoS<sub>x</sub>) co-catalyst [69]. The optimized ZnCdS/Co-MoS<sub>x</sub> (Co/Mo = 1/4, 1 mol%) photocatalyst showed the highest HER rate of 551.48 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 21.7% at λ = 420 nm. The achieved rate was superior to that of ZnCdS/MoS<sub>x</sub> and pristine ZnCdS by 1.4 times and 23 times, respectively.

As stated above, the photocatalytic activity of NiS towards hydrogen generation is practically limited by the strong sulfur-hydrogen bonds. The weakening of this bond strength was achieved by introducing Cu atoms into the amorphous NiS<sub>x</sub> (a-NiS<sub>x</sub>) co-catalyst, which led to the formation of the electron-rich active S sites [70]. The as-synthesized co-catalyst was anchored onto TiO<sub>2</sub> to form a-NiCuS<sub>x</sub>/TiO<sub>2</sub> composite photocatalysts. The optimized CuS<sub>x</sub>/TiO<sub>2</sub> (3:1) photocatalyst exhibited an improved H<sub>2</sub> evolution rate of 427.9 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 34.67% using methanol as a sacrificial reagent, which is superior compared to the a-NiS<sub>x</sub>/TiO<sub>2</sub> with 226.1 μmol g<sup>-1</sup> h<sup>-1</sup>. Xu et al. reported a CuGaS<sub>2</sub> (CGS) photocatalyst synthesized by a one-step solid-phase sintering method [71]. It was shown that when the Cu to Ga atom ratio is 1: 1.4 in the raw material, the prepared CuGaS<sub>2</sub> exhibited the highest visible-light (λ > 420 nm) H<sub>2</sub> production rate reaching to 1.12 mmol g<sup>-1</sup> h<sup>-1</sup>. Moreover, loading with Ruthenium as a co-catalyst, the H<sub>2</sub> production rate of CuGaS<sub>2</sub>@Ru was further promoted to 3.38 mmol g<sup>-1</sup> h<sup>-1</sup>. Wang et al. prepared CuInS<sub>2</sub> microflowers via the solvothermal method and decorated them with CdSe QDs on the surface. The prepared CdSe/CuInS<sub>2</sub> microflowers showed an improved photocatalytic H<sub>2</sub> production rate of 10,610.37 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 48.97% at 420 nm [72]. The enhanced photocatalytic H<sub>2</sub> evolution was related to the formation of a p-n heterostructure with improved light absorption ability, enhanced charge transfer efficiency, and a reduced recombination rate of the photogenerated electrons and holes. Also, the one-step alkaline hydrothermal strategy was used to prepare Ni-doped Mn<sub>0.5</sub>Cd<sub>0.5</sub>S with different Ni doping concentrations [73]. It was shown that 3% Ni-doped Mn<sub>0.5</sub>Cd<sub>0.5</sub>S could obtain the highest rate of 108.3 mmol g<sup>-1</sup> h<sup>-1</sup> at λ > 420 nm, which is nearly 14.4 times more than undoped counterpart (7.5 mmol g<sup>-1</sup> h<sup>-1</sup>). The proposed charge transfer mechanism, together with the obtained reaction yield, is shown in Figure 9b.



**Figure 9.** (a) Schematic illustration of the dual-defect (Ag dopants and nanoholes) configuration established on ZnIn<sub>2</sub>S<sub>4</sub> monolayers and their role in promoting O<sub>2</sub> and H<sub>2</sub> evolution reactions during photocatalytic water splitting [68]. Copyright 2021, American Chemical Society. (b) Schematic illustration for the H<sub>2</sub> evolution mechanism for 3% Ni doped Mn<sub>0.5</sub>Cd<sub>0.5</sub>S photocatalyst [73]. Copyright 2022, Wiley-VCH. (c) Schematic illustration of the band structure of ZIS, VS-ZIS and Ni<sub>0.05</sub>-VS-ZIS [74]. Copyright 2021, Wiley-VCH. The corresponding HER yields for each optimized photocatalyst are given in the bottom panel.

Engineering sulfur vacancies supplemented by metal doping have been shown as a promising route to improve photocatalytic H<sub>2</sub> evolution. In this regard, ultrathin ZnIn<sub>2</sub>S<sub>4</sub> nanosheets functionalized by sulfur vacancies and nickel dopants were reported [74]. It was shown that the S vacancies lower the CB minimum, enhance the photogenerated charge reduction capacity, and improve the lifetime of photoexcited states. Moreover, introducing Ni dopants shifts the Fermi level upward, leading to enhanced charge carrier separation and a reduced recombination rate. Thus, the synergistic effects lead to an 8.91 mmol g<sup>-1</sup> h<sup>-1</sup> H<sub>2</sub> evolution rate under Vis-light irradiation. The proposed charge transfer mechanism, together with the obtained reaction yield, is shown in Figure 9c. In order to improve photocatalytic hydrogen yield, the biomineralization method has also been recently employed. Thus, Zn-doped CdS (Cd<sub>1-x</sub>Zn<sub>x</sub>S) with various Zn<sup>2+</sup>/Cd<sup>2+</sup> ratios was synthesized by engineered *Escherichia coli* with a surface-displayed biomineralization system [75]. More specifically, bacterial-synthesized Cd<sub>1-x</sub>Zn<sub>x</sub>S with an average particle size of <50 nm was deposited on the bacterial cell surface, which led to more material loading. The optimized Cd<sub>1-x</sub>Zn<sub>x</sub>S with a 0.044 molar ratio of Zn/Cd showed an improved HER of 1.146 μmol h<sup>-1</sup> g<sup>-1</sup>.

As already given examples above, one of the main trends to improve photocatalytic H<sub>2</sub> production is a search for new architectures. For example, hollow Cu<sub>2</sub>MoS<sub>4</sub>/ZnIn<sub>2</sub>S<sub>4</sub> heterostructural nanocubes with intimate-contact interface were reported as a self-template method [76]. The optimized 15 wt% Cu<sub>2</sub>MoS<sub>4</sub>/ZnIn<sub>2</sub>S<sub>4</sub> sample exhibits the highest hydrogen production rate of 8103 μmol h<sup>-1</sup> g<sup>-1</sup>, which is four-fold improvement compared to the clean ZnIn<sub>2</sub>S<sub>4</sub>. The improved activity was ascribed to more active sites, increased light absorption by hollow nanocubes, and a built-in electric field, which facilitates photogenerated holes from ZnIn<sub>2</sub>S<sub>4</sub> to Cu<sub>2</sub>MoS<sub>4</sub> with enhanced charge separation. A sphere-like, structured ZnIn<sub>2</sub>S<sub>4</sub>-rGO-CuInS<sub>2</sub> ternary heterojunction catalyst for visible-active photocatalytic hydrogen evolution from water splitting was reported [77]. The obtained photocatalyst exhibited hydrogen generation of 2531 μmol/g after 5 h. Based on photocurrent density and photoluminescence measurements, it was discussed that rGO acts as an electron transfer medium by reducing excited charge recombination and facilitating electron transfer in the ternary

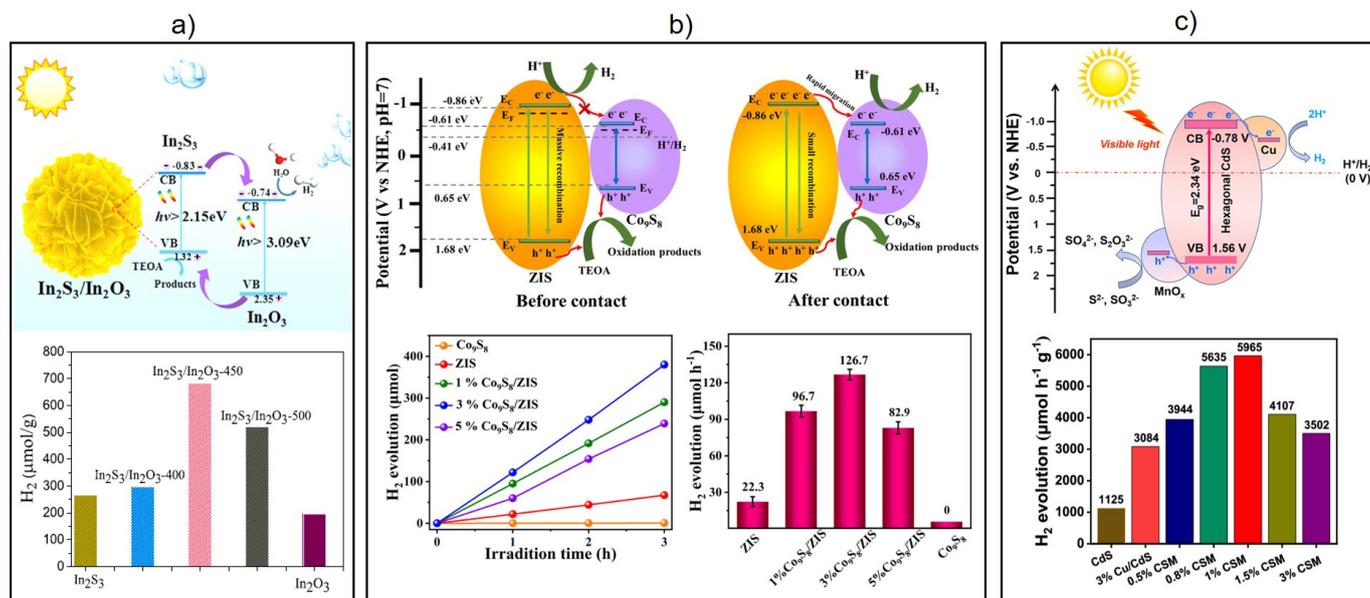
composite. Chong et al. reported on atomic-level charge steering for photocatalytic water splitting in P-doped ZnIn<sub>2</sub>S<sub>4</sub> (ZIS) bilayer nanosheets via Density Functional Theory [78]. It was revealed that the replacement of the S3 atom by substitutional phosphorus gave rise to the most stable P-ZIS structure. Furthermore, P-ZIS was shown to have a lower band gap energy, an upshift of the valence band maximum (VBM), an increase in electron density at the VBM, and a lower H<sup>\*</sup> adsorption-desorption barrier, which are essential parameters for H<sub>2</sub> evolution kinetics. Thus, the p-type doping strategy was shown as a promising route for improved H<sub>2</sub> evolution from water splitting.

### 2.3. Metal Sulfide and Metal Oxide Heterostructures

In order to design an efficient interface, Lin et al. produced a 2D/2D atomic double-layer WS<sub>2</sub>/Nb<sub>2</sub>O<sub>5</sub> shell/core photocatalyst (DLWS/Nb<sub>2</sub>O<sub>5</sub>) [79]. It was revealed that due to the unusual interface, the average lifetime of charge carriers for DLWS/Nb<sub>2</sub>O<sub>5</sub> (180.97 ps) is remarkably reduced as compared to that of Nb<sub>2</sub>O<sub>5</sub> (230.50 ps). Thus, the ultrafast charge transfer from Nb<sub>2</sub>O<sub>5</sub> to atomic double-layer WS<sub>2</sub> was achieved, which gave rise to a high photocatalytic H<sub>2</sub> evolution rate of 237.6 μmol g<sup>-1</sup> h<sup>-1</sup>. Also, CoS/Nb<sub>2</sub>O<sub>5</sub> heterojunctions were constructed by in-situ loading CoS co-catalyst on the surface of Nb<sub>2</sub>O<sub>5</sub> nanosheets [80]. The femtosecond-resolved transient absorption spectroscopy revealed that the average lifetime of charge carriers for 10 wt% CoS/Nb<sub>2</sub>O<sub>5</sub> (159.6 ps) is drastically shortened by contrast with that of Nb<sub>2</sub>O<sub>5</sub> (5531.9 ps), thus similar to the DLWS/Nb<sub>2</sub>O<sub>5</sub> rapid charge transfer from Nb<sub>2</sub>O<sub>5</sub> to CoS could be achieved. The designed photocatalyst showed H<sub>2</sub> evolution rate of 355 μmol g<sup>-1</sup> h<sup>-1</sup>, which is up to 17.5 times higher than the pristine Nb<sub>2</sub>O<sub>5</sub>. By using the hydrothermal-physical mixture method, a Mo<sub>2</sub>S<sub>3</sub>/Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> composite with a unique n-n heterojunction and 2D structure was successfully prepared [81]. The optimized co-catalyst with the mass ratio of Mo/Bi-3% showed the highest H<sub>2</sub> evolution rate of ~160 μmol.

Bismuth vanadate (BiVO<sub>4</sub>) is another promising catalyst, which has been employed for solar-driven water splitting and H<sub>2</sub> production [82]. Recently, a step-scheme photocatalyst was developed by spatial deposition of MnO<sub>x</sub> and CdS NWs on a 3D decahedron BiVO<sub>4</sub> surface [83]. The synthesized CdS/MnO<sub>x</sub>-BiVO<sub>4</sub> photocatalyst containing 5% of CdS demonstrated the highest H<sub>2</sub> and O<sub>2</sub> production rates of 1010 and 510 μmol g<sup>-1</sup> h<sup>-1</sup>, respectively, with an AQE of 11.3% without any sacrificial reagent. The existence of oxygen vacancies, effective charge separation and transport, and a significant interaction of co-catalysts (MnO<sub>x</sub> and Pt) with photocatalysts (BiVO<sub>4</sub> and CdS) in the CdS/MnO<sub>x</sub>-BiVO<sub>4</sub> heterojunction were shown to be the major factors for boosted photocatalytic activity. It should be noted that especially for simultaneous O<sub>2</sub> generation, the single MS photocatalysts are not favorable because of the severe self-photooxidation or photocorrosion, which, however, can be altered by combining with other co-catalysts. In this regard, the development of MS-based Z-scheme photocatalyst, where the photogenerated holes in MS can be successfully carried away, is an effective approach (see Figure 6 for the Z-scheme photocatalysts) [84]. Heterostructure composite photocatalyst was synthesized by combining the Nb<sub>2</sub>O<sub>5</sub>-MoS<sub>2</sub> (NM) with CdS/graphene and CuS/graphene (NM-CdS/G, NM-CuS/G) via hydrothermal method [85]. While the NM photocatalyst showed a relatively low photocatalytic activity with a H<sub>2</sub> evolution rate of 1.6 mmol g<sup>-1</sup> h<sup>-1</sup>, the NM-CuS/graphene and NM-CdS/graphene composites showed remarkably enhanced H<sub>2</sub> evolution rates of 7.1 and 8.7 mmol g<sup>-1</sup> h<sup>-1</sup>, respectively. Thus, the combination of wide and narrow band gap semiconductors resulted in a fast charge separation process and a possibly improved lifetime of the charge carriers. Also, hierarchically porous hydrangea-like In<sub>2</sub>S<sub>3</sub>/In<sub>2</sub>O<sub>3</sub> heterostructures were reported through a simple in situ oxidization process [86]. The new heterostructures exhibited superior photocatalytic activity (683.8 μmol/g) to the In<sub>2</sub>S<sub>3</sub> and In<sub>2</sub>O<sub>3</sub>, that was assigned to enhanced interfacial charge transfer because of heterostructures and hierarchically porous heterostructures, which increases the number of active sites and improves the permeability. See Figure 10b for the proposed photocatalytic charge transfer mechanism and HER plot.

A Vis-light active hierarchical  $\text{Co}_9\text{S}_8/\text{ZnIn}_2\text{S}_4$  photocatalyst was designed by decorating  $\text{Co}_9\text{S}_8$  NPs onto the  $\text{ZnIn}_2\text{S}_4$  nanoflower through a two-step solvothermal process and applied for photocatalytic  $\text{H}_2$  evolution [87]. Due to sulfide-sulfide integrations in a hierarchical flower-like  $\text{Co}_9\text{S}_8/\text{ZnIn}_2\text{S}_4$  photocatalytic heterojunction, large surface areas with rich active centers, effective separation and transfer of photoinduced electrons and holes were achieved. Correspondingly, the optimized  $\text{Co}_9\text{S}_8/\text{ZnIn}_2\text{S}_4$  photocatalyst showed remarkably improved HER of  $12.67 \text{ mmol h}^{-1} \text{ g}^{-1}$  with high stability, which is 6 times higher than that of single  $\text{ZnIn}_2\text{S}_4$  photocatalyst under visible-light irradiation. The proposed photocatalytic charge transfer mechanism, together with the obtained yield, is shown in Figure 10b.



**Figure 10.** Schematic illustration of photocatalytic hydrogen evolution mechanism and corresponding HER plots for different metal sulfide and metal oxide heterostructures. (a)  $\text{In}_2\text{S}_3/\text{In}_2\text{O}_3$  [86] Copyright 2022, Elsevier. (b)  $\text{Co}_9\text{S}_8/\text{ZnIn}_2\text{S}_4$  [87] Copyright 2022, American Chemical Society. (c)  $\text{Cu/CdS/MnO}_x$  [88] Copyright 2021, American Chemical Society.

$\text{Ta}_2\text{O}_5$  is another promising photocatalyst, which has been studied in the last years [89], also for the photocatalytic  $\text{H}_2$  generation [90]. Recently, a  $\text{Cu}_2\text{S}/\text{Ta}_2\text{O}_5/\text{CdS}$  ternary nanocomposite photocatalyst was fabricated by a solid-state approach and applied for photocatalytic hydrogen evolution [91]. It was shown that in the resulting structure, the  $\text{Cu}_2\text{S}$ ,  $\text{Ta}_2\text{O}_5$ , and  $\text{CdS}$  form six-petals, nanorods, and random-spherical morphology, respectively. The optimized  $\text{Cu}_2\text{S}/\text{Ta}_2\text{O}_5/\text{CdS}$  nanocomposite demonstrated a high rate of  $\text{H}_2$  evolution of  $131 \text{ mmol h}^{-1} \text{ g}^{-1}$  being remarkably higher than that of its constituents.

It is known that preparing photocatalysts deposited on semiconductors can effectively restrain charge recombination. In this regard, a  $\text{Cu/CdS/MnO}_x$  (CSM) heterostructured photocatalyst with a spatial separation of the photoinduced charge carriers was reported [88]. It was shown that while a reduction co-catalyst  $\text{Cu}$  NPs tend to capture electrons, the oxidation co-catalyst,  $\text{MnO}_x$  NPs, collects the photogenerated holes. Time-resolved fluorescence spectra (TRPL) and photoelectrochemical characterization showed that the lifetime of photogenerated carriers improved, and surface-reactive HER kinetics were accelerated. As a result, the optimized 1% CSM photocatalyst showed a maximum HER of  $5965.03 \text{ h}^{-1} \text{ g}^{-1}$  being  $\sim 5.3$  higher than the single  $\text{CdS}$  photocatalyst. The proposed photocatalytic charge transfer mechanism, together with the obtained HER, is given in Figure 10c.

Step-scheme heterojunction composite photocatalyst based on  $\text{W}_{18}\text{O}_{49}/\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$  was produced via the physical stirring method and employed for photocatalytic hydrogen production under Vis-light illumination [44]. It was revealed that the  $\text{H}_2$  evolution rate was

highest ( $5373 \mu\text{mol h}^{-1} \text{g}^{-1}$ ) for the load ratio of  $\text{W}_{18}\text{O}_{49}$  was 5%. Consequently, the formation of S-scheme heterojunction between  $\text{W}_{18}\text{O}_{49}$  and  $\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$  contact interface was shown as the reason for the improved  $\text{H}_2$  production. Thus, the S-scheme heterojunction promotes the consumption of ineffective electrons and holes, improves the effective charge separation and transfer, and lowers the activation energy of the catalytic reaction system, leading to enhanced photocatalytic  $\text{H}_2$  evolution. Also,  $\text{MoS}_2@\text{MoO}_3$  step S-scheme heterojunction photocatalyst was prepared through in-situ partial sulfidation [92]. The heterostructure interface was engineered through the in-situ vulcanization strategy, which gradually corrodes from the outside to the inside of the nanostructures. Thus, introducing S-atoms can replace oxygen atoms to build a S-rich surface and generate molybdenum sulfide. Afterward, the amount of thioacetamide (S source) was adjusted to control vulcanization and optimize the experimental conditions. The synthesized heterostructure photocatalyst showed  $\text{H}_2$  production rate of  $12,416.8 \mu\text{mol h}^{-1} \text{g}^{-1}$ .

### Metal Sulfide and $\text{TiO}_2$ -Based Photocatalysts

$\text{TiO}_2$  is one of the most studied and applied photocatalysts for versatile purposes ranging from wastewater treatment and indoor air purification [93] to organic synthesis [94,95]. Moreover, metal sulfides/ $\text{TiO}_2$  heterostructures with a porous architecture possess the capability to enhance the photocatalytic activity of  $\text{TiO}_2$ . First of all, the porous structure provides an abundance of carrier charge transfer routes and surface reaction sites. Moreover, the appropriate number of MS deposited on  $\text{TiO}_2$  significantly improves the electrical conductivity and light absorption capabilities.

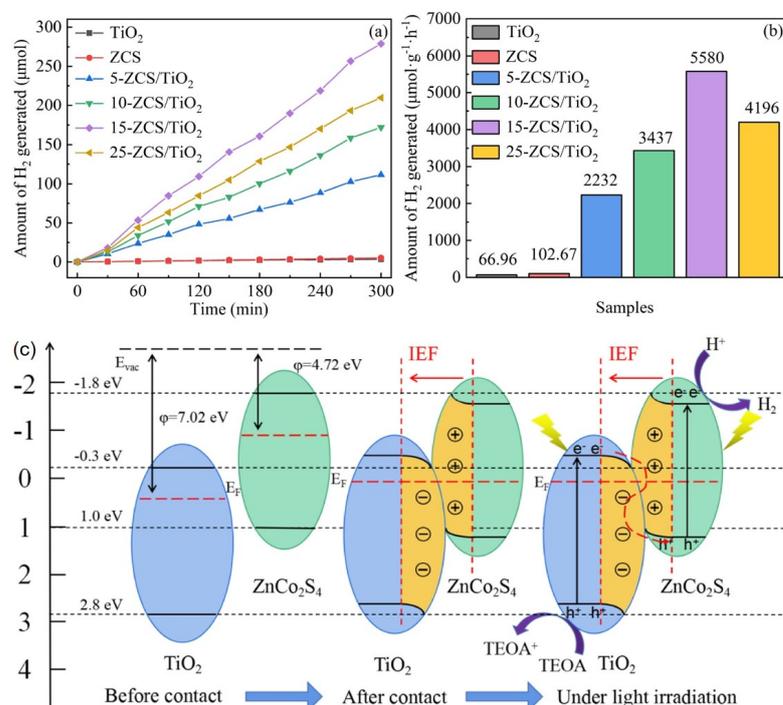
Lin et al. reported a sea-urchin-like  $\text{ReS}_2$  nanosheet/ $\text{TiO}_2$  NP heterojunctions with an unusual charge edge-collection effect leading to a significant acceleration of charge separation and transfer [96]. The designed photocatalysts yielded  $\text{H}_2$  evolution rate of  $3.71 \text{ mmol h}^{-1} \text{g}^{-1}$  with an AQE of 16.09%, which is  $\sim 232$  times higher than the commercial P25  $\text{TiO}_2$ . Anatase phase  $\text{TiO}_2$  QDs-sensitized  $\text{Cu}_2\text{S}$  ( $\text{Cu}_2\text{S}/\text{TiO}_2$ ) nanocomposites with varying concentrations of  $\text{TiO}_2$  QDs were synthesized via a facile two-stage hydrothermal-wet impregnation method [97]. The as-synthesized co-catalyst showed  $\text{H}_2$  evolution rate of  $\sim 45.6 \text{ mmol h}^{-1} \text{g}^{-1}$  under simulated solar irradiation, which is  $\sim 5$  and 2.4-fold higher than that of the pristine  $\text{TiO}_2$  and  $\text{Cu}_2\text{S}$ , respectively.

Chen et al. synthesized the  $\text{CoS}_x\text{-rGO}/\text{TiO}_2$  composite photocatalyst by employing two steps: first, the rGO nanosheets hydrothermally loaded onto  $\text{TiO}_2$  NPs, and then  $\text{CoS}_x$  nanodots (0.5–2 nm) photodeposited on the rGO nanosheets [98]. The produced  $\text{CoS}_x\text{-rGO}/\text{TiO}_2$  (10%) composite photocatalyst exhibited  $\text{H}_2$  production efficiency of  $256.97 \mu\text{mol g}^{-1} \text{h}^{-1}$ , being  $\sim 12$  times higher than that of  $\text{rGO}/\text{TiO}_2$  photocatalyst and  $\sim 2$  times higher than that of  $\text{CoS}_x/\text{TiO}_2$ . It was proposed that rGO nanosheets facilitate the rapid transfer of photo-induced carriers from  $\text{TiO}_2$  to  $\text{CoS}_x$  nanodots and that  $\text{CoS}_x$  nanodots thereafter function as active sites for  $\text{H}_2$  production. In another study, the  $\text{TiO}_2@\text{CuS}$  double-shell nanoboxes were constructed via a multistep control strategy involving template participation [99]. The optimized  $\text{TiO}_2@\text{CuS}$  double-shell nanoboxes exhibited  $\text{H}_2$  production rate of 2467 and  $173 \mu\text{mol g}^{-1} \text{h}^{-1}$  under full-spectrum and near-IR irradiation, respectively.

Bismuth sulfide ( $\text{Bi}_2\text{S}_3$ ) has also recently been considered as an effective co-catalyst (i.e., with  $\text{TiO}_2$ ) because of its small band gap (1.28 eV), strong visible light absorption capacity (450–700 nm), and large negative conduction band (CB) potential. The amorphous  $\text{BiS}_x$  nanodots with the size of 0.5–2 nm were loaded on  $\text{TiO}_2$  surface to facilitate the photocatalytic  $\text{H}_2$  generation [100]. The  $\text{BiS}_x/\text{TiO}_2$  photocatalyst was synthesized by an Ethylenediaminetetraacetic acid (EDTA)-assisted two-step process, which included the adsorption of the Bi(III)-EDTA ions on  $\text{TiO}_2$  and the in-situ formation of amorphous  $\text{BiS}_x$  nanodots. The optimized  $\text{BiS}_x/\text{TiO}_2$  (1.0 wt%) photocatalyst showed the maximum photocatalytic  $\text{H}_2$  production rate of  $803.2 \mu\text{mol h}^{-1} \text{g}^{-1}$  with an AQE of 3.86%, which is 83.6 and 1.6 folds of the clean  $\text{TiO}_2$  and crystalline  $\text{Bi}_2\text{S}_3$ -modified  $\text{TiO}_2$  ( $c\text{-Bi}_2\text{S}_3/\text{TiO}_2$ ), respectively. The enhanced photocatalytic  $\text{H}_2$  production was assigned to the amorphous  $\text{BiS}_x$  nanodots

providing more unsaturated active S atoms, which act as the efficient H<sub>2</sub> evolution active sites. Effective hydrogen evolution from water splitting under solar light irradiation was investigated using ZnS-PbS QDs loaded onto a Au/TiO<sub>2</sub> photocatalyst [101]. In the presence of 20 wt% methanol as a sacrificial reagent to photocatalyze for 5 h, 5011  $\mu\text{mol g}^{-1} \text{h}^{-1}$  HER was observed. Nguyen et al. utilized polyvinyl alcohol (PVA) as a binder compound for combining MoS<sub>2</sub> and TiO<sub>2</sub> nanotube arrays (TNAs) to improve photoelectrochemical (PEC) water splitting ability [102]. They showed that by a thermal treatment process, the formation of the  $\pi$ -conjunction in the PVA structure enhanced the PEC performance of MoS<sub>2</sub>/TNAs, exhibiting linear sweeps in the anodic direction with the current density over 65  $\mu\text{A}/\text{cm}^2$  at zero bias voltage vs. Ag/AgCl.

In general, ternary-metal-sulfide/TiO<sub>2</sub> photocatalysts are expected to have lower toxicity, a tunable band structure, and favorable chemical stability, enabling a higher HER over the binary-metal-sulfide/TiO<sub>2</sub>. In this regard, NiWS<sub>x</sub> nanodot-modified TiO<sub>2</sub> (NiWS<sub>x</sub>-ND/TiO<sub>2</sub>) photocatalysts were prepared by evenly anchoring ultras-small amorphous NiWS<sub>x</sub> nanodots (~2 nm) onto TiO<sub>2</sub> surface by means of a facile photodeposition strategy [103]. The optimized NiWS<sub>x</sub>-ND/TiO<sub>2</sub> (3 wt%) photocatalyst manifested the maximum H<sub>2</sub> generation efficiency of 4580  $\mu\text{mol h}^{-1} \text{g}^{-1}$  with the corresponding AQE of 13%. Dai et al. reported a S-scheme heterojunction photocatalyst by coupling the ZnCo<sub>2</sub>S<sub>4</sub> NPs on the surface of TiO<sub>2</sub> through simple solvothermal method [104]. The prepared nanocomposite photocatalyst exhibited H<sub>2</sub> evolution efficiency of 5580  $\mu\text{mol h}^{-1} \text{g}^{-1}$  with an AQE of 11.5% at 420 nm, which is 88.3 times and 54.3 times that of TiO<sub>2</sub> and ZnCo<sub>2</sub>S<sub>4</sub> NPs, respectively. It was discussed that the S-scheme electron transfer path facilitates better electron-hole separation alongside improved charge transfer efficiency. While the ZnCo<sub>2</sub>S<sub>4</sub> NPs significantly enhance the visible light absorption, leading to a high H<sub>2</sub> yield. The resulting efficiency and proposed photocatalytic mechanism are summarized in Figure 11.

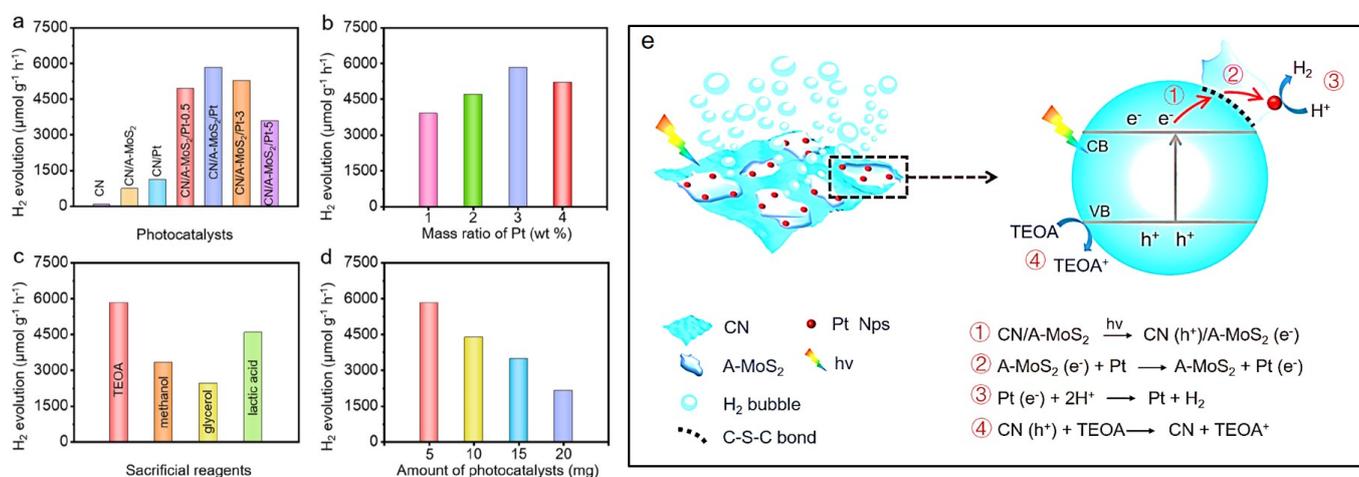


**Figure 11.** Time evolution (a) and average rate (b) of photocatalytic hydrogen production activity over the TiO<sub>2</sub>, ZnCo<sub>2</sub>S<sub>4</sub> (ZCS), 5-ZCS/TiO<sub>2</sub>, 10-ZCS/TiO<sub>2</sub>, 15-ZCS/TiO<sub>2</sub> and 25-ZCS/TiO<sub>2</sub> photocatalysts. 5, 10, 15, and 25 are the mass proportion of ZCS in the composite. (c) Schematic illustration of the photocatalytic mechanism over the 15-ZCS/TiO<sub>2</sub> photocatalyst: Thus, under Vis-light irradiation, the internal electric field (IEF) induces charge transfer (a), separation, and the formation of S-scheme heterojunction (b) to realize photocatalytic hydrogen evolution (c) [104]. Copyright 2022, Elsevier.

Also, a TiO<sub>2</sub> nanoplate/amorphous MoS<sub>2</sub> QD (a-MoS<sub>2</sub>) composite was prepared via in situ sulfuration of MoO<sub>2</sub>/TiO<sub>2</sub> [105]. It was shown that in this composite, the intimate chemical bonding instead of electrostatic adsorption between TiO<sub>2</sub> and MoO<sub>2</sub> enables effective electron transfer from TiO<sub>2</sub> to a-MoS<sub>2</sub>. The optimized a-MoS<sub>2</sub>/TiO<sub>2</sub> nanocomposite exhibited a maximum H<sub>2</sub> yield of 880.3  $\mu\text{mol h}^{-1} \text{g}^{-1}$ . In another study, ultrathin TiO<sub>2</sub> nanosheets were integrated into the growth of ZnIn<sub>2</sub>S<sub>4</sub> to produce TiO<sub>2</sub>-ZnIn<sub>2</sub>S<sub>4</sub> heterostructure nanoflowers in order to design a direct Z-scheme heterojunction photocatalyst with an improved PHE [106]. It was revealed that the formed direct Z-scheme heterojunction can consume residual photogenerated holes of ZnIn<sub>2</sub>S<sub>4</sub> and enable water oxidation to occur solely at the TiO<sub>2</sub> site, improving the overall stability of the system. The optimized DZH nanoflowers showed enhanced photocatalytic water splitting activity of 214.9  $\mu\text{mol h}^{-1} \text{g}^{-1}$  for H<sub>2</sub> evolution and 81.7  $\mu\text{mol h}^{-1} \text{g}^{-1}$  for O<sub>2</sub> evolution.

#### 2.4. Metal Sulfide/g-C<sub>3</sub>N<sub>4</sub> Heterostructures

Combining suitable co-catalysts and simultaneous interface engineering is another promising strategy for photocatalytic HER enhancement. In this regard, Wang et al. reported a CN/A-MoS<sub>2</sub>/Pt photocatalyst using amorphous MoS<sub>2</sub> (a-MoS<sub>2</sub>) as the conformal multifunctional intermediate layer between g-C<sub>3</sub>N<sub>4</sub> and Pt [107]. They showed that a conformal chemically bonded interface and low overpotential for the H<sub>2</sub> evolution reaction lead to enhanced interface charge transfer and photochemical reaction kinetics. Thus, under Vis-light illumination, 5830  $\mu\text{mol h}^{-1} \text{g}^{-1}$  HER was achieved, which is more than an 83-fold enhancement compared with the g-C<sub>3</sub>N<sub>4</sub>. The detailed parameterization and corresponding HERs, together with the proposed photocatalytic reaction mechanism, are given in Figure 12.



**Figure 12.** (a) Photocatalytic hydrogen production rate of various photocatalysts. (b) photocatalytic rate of CN/A-MoS<sub>2</sub>/Pt with different mass ratios of Pt. (c) photocatalytic rate of CN/A-MoS<sub>2</sub>/Pt measured in different sacrificial reagents. (d) photocatalytic rate of CN/A-MoS<sub>2</sub>/Pt tested with different masses of the photocatalyst. (e) Proposed charge transfer mechanism for the CN/A-MoS<sub>2</sub>/Pt photocatalysts [107]. Copyright 2021, American Chemical Society.

It should be noted that the effective surface area of the co-catalysts has a significant impact on photocatalytic performance. In this regard, it has been reported that g-C<sub>3</sub>N<sub>4</sub> structures with a large surface area can be produced using a rapid high-temperature hydrothermal post treatment strategy with fast heat transfer and short processing time. Thus, the extended surface area resulted in about 5.1 higher HER compared to pristine g-C<sub>3</sub>N<sub>4</sub> under Vis-light ( $\lambda > 400 \text{ nm}$ ) irradiation [108]. A ternary composite catalyst of MoS<sub>2</sub>/Fe<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> via hydrothermal method was reported [109]. The fabricated composite photocatalyst demonstrated H<sub>2</sub> generation rate of 7.82  $\text{mmol g}^{-1} \text{h}^{-1}$ , which is almost 5 times higher than the single-component g-C<sub>3</sub>N<sub>4</sub> photocatalyst (1.56  $\text{mmol g}^{-1} \text{h}^{-1}$ ). The improvement

of its photocatalytic activity can be mainly attributed to its enhanced visible light absorption and the increased specific surface area. According to the earlier DFT investigations, in the MoS<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composite, the CB minimum and VB maximum of the g-C<sub>3</sub>N<sub>4</sub> monolayer are roughly 0.83 eV, and 0.15 eV higher than those of the MoS<sub>2</sub> sheet [110]. As a result of the formed type-II band alignment, photoexcited electrons can easily migrate from the g-C<sub>3</sub>N<sub>4</sub> monolayer to the MoS<sub>2</sub> sheet, resulting in a high hydrogen evolution reaction activity. Additionally, the charge transfer between MoS<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> generates a polarized field at the boundary, enabling effective separation of photogenerated carriers. A transition metal sulfide (Co<sub>3</sub>S<sub>4</sub>) produced from MOF was used to build an ohmic contacting Co<sub>3</sub>S<sub>4</sub>-CN heterojunction interface for enhanced photocatalytic H<sub>2</sub> generation [111]. It was revealed that electron injection from the lower work function Co<sub>3</sub>S<sub>4</sub> to g-CN causes a significant interfacial electric field resulting in downward band bending in a g-CN semiconductor. Furthermore, the Co<sub>3</sub>S<sub>4</sub>-CN heterojunction facilitates exciton dissociation while removing a possible barrier for electron extraction from g-CN, facilitating charge carrier separation synergistically. The obtained co-catalyst showed H<sub>2</sub> evolution of ~217.0 μmol g<sup>-1</sup> h<sup>-1</sup> upon visible light illumination. Chen et al. employed a light-induced deposition approach to precisely load NiS nanodots onto the surface of g-C<sub>3</sub>N<sub>4</sub> to produce a bifunctional photocatalyst [42]. The obtained photocatalyst showed an enhanced photocatalytic H<sub>2</sub> evolution reaction and photocatalytic oxygen reduction. Particularly, the NiS@g-C<sub>3</sub>N<sub>4</sub>-30 displayed a H<sub>2</sub> evolution rate of 3297 μmol g<sup>-1</sup> h<sup>-1</sup> under visible-light illumination. It was discussed that the close interaction of the loaded NiS nanodots with the electron transfer sites of g-C<sub>3</sub>N<sub>4</sub> nanosheets provides effective separation of photogenerated charge carriers and further accelerates the chemical reaction on the surface of the photocatalyst.

Considering that loading more NiS co-catalyst to function as active sites of g-C<sub>3</sub>N<sub>4</sub> for improved photocatalytic H<sub>2</sub> generation, a facile sulfur-mediated photodeposition approach was developed [112]. It was shown that photogenerated electrons excited by Vis light reduce the S molecules absorbed on g-C<sub>3</sub>N<sub>4</sub> surface to S<sup>2-</sup>, and subsequently, a NiS co-catalyst is formed in situ on the g-C<sub>3</sub>N<sub>4</sub> surface by a combination of Ni<sup>2+</sup> and S<sup>2-</sup>. The resulting NiS/g-C<sub>3</sub>N<sub>4</sub> photocatalyst exhibited H<sub>2</sub> generation rate of 244 μmol g<sup>-1</sup> h<sup>-1</sup>.

The PtS/g-C<sub>3</sub>N<sub>4</sub> co-catalyst was produced by loading 2D PtS nanorectangles with a length of ~7 nm and width of ~5 nm on the surface of g-C<sub>3</sub>N<sub>4</sub> nanosheets utilizing in situ vapor-phase growth [113]. The constructed compound showed an unusual metal sulfide-support interaction, which promoted stabilization of the geometrical structure and energy-band structure, leading to acceleration of charge transfer, and reduction of hydrogen adsorption free energy of PtS/CN. The characteristics mentioned above gave rise to a high H<sub>2</sub> generation rate of 1072.6 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 45.7% at 420 nm. This is ~13.3 and ~1532.3 times enhancement in comparison with the Pt NPs/g-C<sub>3</sub>N<sub>4</sub> nanosheets and pristine g-C<sub>3</sub>N<sub>4</sub> nanosheets, respectively. In another study, ultrasmall fine Pt NPs and MoS<sub>2</sub> nanosheets were deposited on the surface of g-C<sub>3</sub>N<sub>4</sub> as co-catalyst [114]. While MoS<sub>2</sub> could efficiently induce structural reconstruction of g-C<sub>3</sub>N<sub>4</sub>, resulting in surface charge redistribution, the boosted electrons localized in the S and N atoms were conducive to the reduction of Pt<sup>4+</sup> via strong metal-support contact, hence enhancing charge separation and transfer. Consequently, the synthesized Pt/MoS<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> composite exhibited a high photocatalytic hydrogen production rate up to 1595.3 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 30.9% at 435 nm, which is 87.6 times higher than that of the clean g-C<sub>3</sub>N<sub>4</sub>.

Wei et al. synthesized ternary ZnS-NiS<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> and NiS<sub>2</sub>-ZnS/g-C<sub>3</sub>N<sub>4</sub> composites by a thermal polymerization and post-co-deposition-sulfidation methods to promote visible light absorption as well as charge separation and transfer [115]. The as-synthesized NiS<sub>2</sub>-ZnS/CN(0.4) and ZnS-NiS<sub>2</sub>/CN(0.4) photocatalysts showed PEH rate of 283.3 and 302.7 μmol h<sup>-1</sup> g<sup>-1</sup>, respectively. The 2D SnS/g-C<sub>3</sub>N<sub>4</sub> nanosheets prepared by a facile ultrasonic and microwave heating method were also reported for enhanced H<sub>2</sub> evolution [116]. Furthermore, after being loaded with MoO<sub>3</sub> particles, the photocatalyst stability improved significantly due to the produced Z-scheme heterojunction. As a result, the enhanced light-harvesting capability of the composite, reduced charge-transfer resistance, improved

electrical conductivity, and the co-catalyst effect of SnS facilitated a higher H<sub>2</sub> generation rate of 818.93  $\mu\text{mol h}^{-1} \text{g}^{-1}$  under AM1.5G irradiation. Also, g-CN/MoS<sub>2</sub> composite was synthesized via an in-situ gas-solid reaction where rodlike MoO<sub>3</sub> was sulfurized to form MoS<sub>2</sub> by the byproduct generated during the thermal condensation of thiourea [117]. The H<sub>2</sub> evolution rate improved from 0.99 to 13.31  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which is 13.44 times higher than that of clean g-CN. Similarly, 1T/2H hybrid WS<sub>2</sub> was prepared as a non-noble metal co-catalyst for g-C<sub>3</sub>N<sub>4</sub> using a straightforward synthesis technique based on an in situ gas-solid phase reaction [118]. Because of the hybrid crystal phase and the presence of S defects in WS<sub>2</sub>, the photocatalytic hydrogen evolution efficiency of the co-catalyst improved noticeably, increasing from 0.32 to 34.76  $\mu\text{mol g}^{-1} \text{h}^{-1}$ . In another study, red-phosphor-doped Co<sub>9</sub>S<sub>8</sub> (P-Co<sub>9</sub>S<sub>8</sub>) was synthesized by a hydrothermal process [119]. It was revealed that the introduction of P-Co<sub>9</sub>S<sub>8</sub> improved the electrical conductivity and surface area of the composite along with reducing the over-potential of H<sub>2</sub> evolution. Consequently, the optimized 25% P-Co<sub>9</sub>S<sub>8</sub>/g-C<sub>3</sub>N<sub>4</sub> exhibited H<sub>2</sub> evolution rate of 4362  $\mu\text{mol h}^{-1} \text{g}^{-1}$  under UV and visible light, which is roughly 121.2 times improved in comparison with the pristine g-C<sub>3</sub>N<sub>4</sub>. According to the photoelectrochemical analysis, the designed photocatalyst was assigned as a Type-I system, where more electrons on the CB of P-Co<sub>9</sub>S<sub>8</sub> to participate in the H<sub>2</sub> evolution processes.

In parallel to the binary metal sulfides, ternary metal sulfides were also employed to produce co-catalyst together with g-C<sub>3</sub>N<sub>4</sub> for photocatalytic H<sub>2</sub> generation. In this regard, in order to improve the photocatalytic water splitting reactivity of g-C<sub>3</sub>N<sub>4</sub>, ternary ZnCo<sub>2</sub>S<sub>4</sub> was coated on its surface [120]. The optimized 25 wt%-ZnCo<sub>2</sub>S<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> showed H<sub>2</sub> evolution rate of 6619  $\mu\text{mol h}^{-1} \text{g}^{-1}$ , which is 55.2 times higher than that of g-C<sub>3</sub>N<sub>4</sub> alone. Thus, the enhanced H<sub>2</sub> production activity was assigned to the presence of both zinc and cobalt ions in ZnCo<sub>2</sub>S<sub>4</sub>, which reduced the H<sub>2</sub> evolution overpotential and charge recombination rate. The obtained photocatalytic process was ascribed as an S-scheme charge transfer route based on trapping experiments for active species. Also, copper-nickel sulfide was anchored on the g-C<sub>3</sub>N<sub>4</sub> for increasing the photocatalytic H<sub>2</sub> evolution [121]. It was revealed that copper-nickel sulfides can effectively suppress deep trapping states of active charge in g-C<sub>3</sub>N<sub>4</sub> improving the efficiency of the shallow trapped electron transfer through a single C-S bond for enhancing Vis-light-driven photocatalytic H<sub>2</sub> production. The synthesized co-catalyst showed H<sub>2</sub> evolution rate up to 752.8  $\mu\text{mol h}^{-1} \text{g}^{-1}$  being 470 times higher than that of clean g-C<sub>3</sub>N<sub>4</sub> [121]. Employing femtosecond time-resolved spectroscopy, it was also shown that the strong interaction between copper-nickel sulfide and g-C<sub>3</sub>N<sub>4</sub> favored the hole transfer from copper-nickel sulfide to the trap state of g-C<sub>3</sub>N<sub>4</sub> leading to a remarkable near-IR-driven photocatalytic H<sub>2</sub> production (0.32  $\mu\text{mol h}^{-1} \text{g}^{-1}$ ,  $\lambda > 800 \text{ nm}$ ).

### 2.5. Metal Sulfide/MOFs (COFs) Heterojunction Photocatalysts

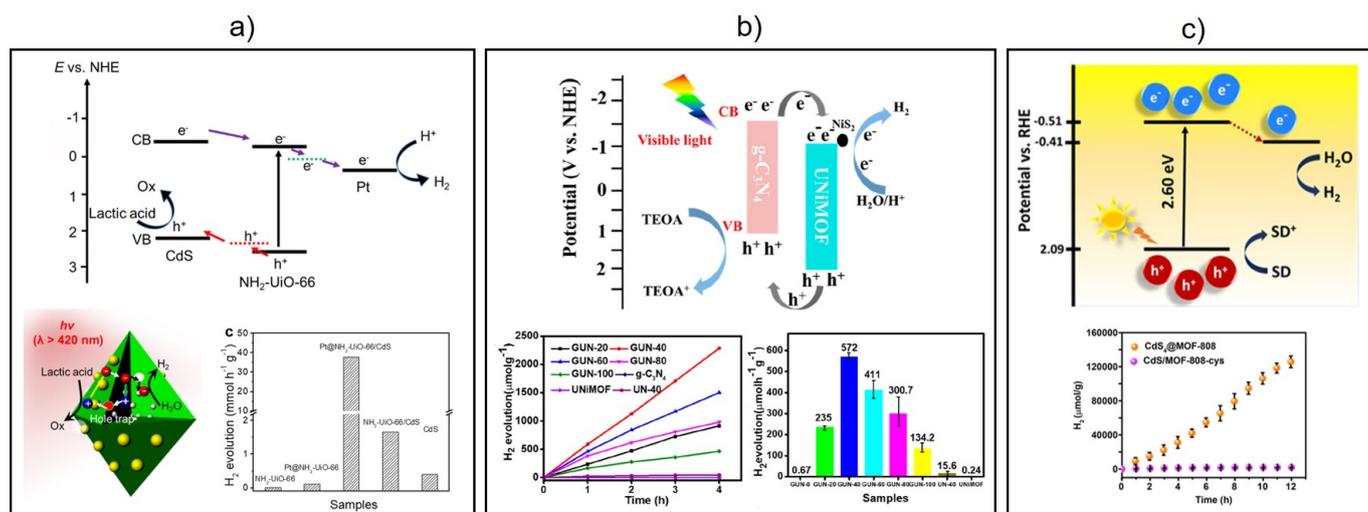
One of the viable techniques for solar energy-driven water splitting is the development of metal-organic frameworks and covalent-organic framework (COF)-based heterostructure. Wang et al. fabricated a direct Z-scheme heterostructure COF-metal sulfide hybrid (T-COF@CdS) with shell-core architecture by self-polymerization of 1,3,5-benzenetricarboxaldehyde and 2,4,6-tris(4-aminophenyl)-1,3,5-triazine in situ on CdS NPs [122]. In these shell-core structures, the C-S chemical bonding generated between T-COF and CdS results in a tight and stable interface. Moreover, the T-COF shell acts as an oxidation site to avoid the photocorrosion of CdS and protects the catalytic centers of CdS against deactivation, leading to long-term stability. Thus, the fabricated Z-scheme T-COF@CdS-3 heterostructure exhibited a high AQE of 37.8% at  $\lambda = 365 \text{ nm}$ .

MOFs have been widely applied in photocatalytic hydrogen production because of their high surface area, semiconductor-like behavior, and great active sites. For example, Mn<sub>0.2</sub>Cd<sub>0.8</sub>S NRs assembled on the surface of Ni-MOF-74 with flake morphology, where the Ni-S bonds between Ni-MOF-74 and Mn<sub>0.2</sub>Cd<sub>0.8</sub>S form a unique transfer channel for photo-induced charge carriers [123]. Thus, the potential energy difference between the two

compounds enables fast injection of the photogenerated electrons in the CB of  $\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$  into the CB of Ni-MOF-74, which remarkably inhibits the electron-hole recombination. Thus the improved carrier lifetime, fast charge transfer rate, and lower overpotential lead to a high  $\text{H}_2$  production rate reaching  $7.104 \text{ mmol g}^{-1} \text{ h}^{-1}$ , which is 6.96 times that of  $\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$ . A  $\text{CdS@NiS}$  photocatalyst using bimetallic MOFs as the precursor was also reported [124]. It was demonstrated that bimetallic Cd/Ni-MOFs can result in well-contacted heterogeneous interfaces between CdS and NiS, effectively facilitating photogenerated carrier separation and enhancing electron interface transport. The designed  $\text{CdS@NiS}$  photocatalysts showed a high photocatalytic HER of  $42.7 \text{ mmol g}^{-1} \text{ h}^{-1}$ , and the corresponding AQE reaches 13.23% at 450 nm.

In another study, the microwave irradiation method was utilized to create the  $\text{UiO-66(Ce)/ZnCdS}$  composite using Ce-based MOF components [UiO-66(Ce)] [125]. It was revealed that ZnCdS NPs were decorated on the surface of UiO-66(Ce) and that UiO-66(Ce) considerably increased the  $\text{H}_2$  generation ability of ZnCdS NPs by effectively enhancing the charge carrier separation. Thus, the  $\text{UiO-66(Ce)/ZnCdS}$  nanocomposite photocatalysts showed  $\text{H}_2$  production rate of  $3.958 \text{ mmol g}^{-1} \text{ h}^{-1}$ , which is roughly 1.95 times higher than that of ZnCdS ( $2.031 \text{ mmol g}^{-1} \text{ h}^{-1}$ ).

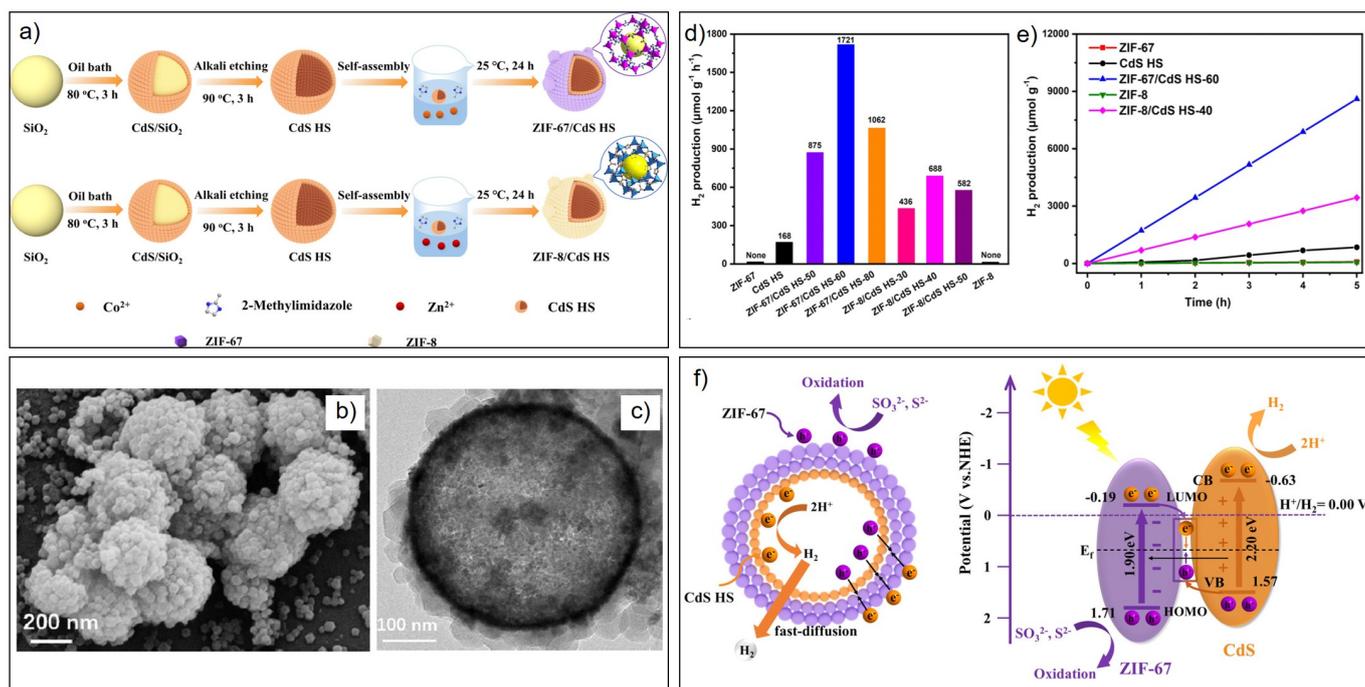
Also, as an efficient  $\text{ZnIn}_2\text{S}_4$ -MOFL photocatalysts for robust photocatalytic hydrogen evolution, hollow tubular  $\text{ZnIn}_2\text{S}_4$  modified by MOF-layers with In-MOF as precursor was demonstrated [126]. The DFT studies proved the decreased density of states (DOS) of  $\text{ZnIn}_2\text{S}_4$  after coupling with the organic ligands. The established direct Z-scheme photocatalytic system synergistically promoted the separation of electron and hole pairs leading to high photocatalytic  $\text{H}_2$  evolution. The  $\text{H}_2$  evolution using this photocatalyst reached  $28.2 \text{ mmol g}^{-1} \text{ h}^{-1}$ , which is roughly 14.8 times higher than that of pristine  $\text{ZnIn}_2\text{S}_4$ . Moreover, the photocatalyst showed AQEs of 22.67% at monochromatic light (350 nm) and the stable  $\text{H}_2$  generation capability ( $5.7 \text{ mmol g}^{-1} \text{ h}^{-1}$ ) under natural sunlight irradiation. Aiming for efficient electron-hole separation and carrier utilization in photocatalytic  $\text{H}_2$  generation, Lian et al. employed MOF ( $\text{NH}_2\text{-UiO-66}$ ) modified with inner Pt NPs and outer CdS NPs [127]. The designed ternary composite  $\text{Pt@NH}_2\text{-UiO-66/CdS}$  possessing a spatially separated, the hierarchical structure was employed for visible-light-driven  $\text{H}_2$  evolution, which exhibited higher  $\text{H}_2$  yields with an AQE of 40.3% at  $\lambda = 400 \text{ nm}$  irradiation. The proposed photocatalytic mechanism and  $\text{H}_2$  evolution rate are given in Figure 13a.



**Figure 13.** Proposed reaction mechanisms for the photocatalytic  $\text{H}_2$  evolution and obtained  $\text{H}_2$  production rates over different MS-based heterojunction photocatalysts. (a)  $\text{ZnIn}_2\text{S}_4\text{@CoS}_2$  [127] Copyright © 2022 Springer Nature. (b)  $\text{g-C}_3\text{N}_4/\text{UNiMOF}/\text{NiS}_2$  (GUN) [128] Copyright 2022, American Chemical Society. (c)  $\text{CdS@MOF-808}$  [129] Copyright © 2022, American Chemical Society.

As we already discussed through this overview, stabilizing photoexcited states by suppressing charge recombination is one of the most important parameters for being good photocatalysts. In this regard, using two-dimensional (2D) UNiMOF as a precursor and by applying the partial hydrothermal sulfidation method, a  $g\text{-C}_3\text{N}_4/\text{UNiMOF}/\text{NiS}_2$  (GUN) ternary heterojunction (TH) photocatalyst was synthesized [128]. As expected, the optimized GUN-40 heterojunction containing 40 mg of UNiMOF (GUN-40) exhibited enhanced HER of  $572 \mu\text{mol g}^{-1} \text{h}^{-1}$  with a longer exciton recombination lifetime of  $\tau = 12.12 \text{ ns}$ . This is remarkably higher than that of the pristine  $g\text{-C}_3\text{N}_4$  ( $0.67 \mu\text{mol g}^{-1} \text{h}^{-1}$ ,  $\tau = 7.87 \text{ ns}$ ) under Vis-light illumination. Also, the optimized GUN-40 photocatalyst showed an improved AQE up to 1.54 % at  $\lambda = 450 \text{ nm}$ . Thus, the GUN TH structure significantly accelerates electron transport to the  $\text{NiS}_2$  active site, improving the excited state lifetime, and minimizes the charge recombination rate, thus enhancing photocatalytic performance. The proposed photocatalytic mechanism and  $\text{H}_2$  evolution rate are given in Figure 13b. Similarly, a  $\text{CdS}@\text{MOF-808}$  composite photocatalyst where CdS NPs are confined inside the nanosized pores of  $\text{Zr}^{4+}$ -based MOF-808 was recently proposed [129]. It was shown that the confinement of the CdS NPs inside the MOF pores, close to the  $\text{Zr}^{4+}$  cluster, forms a shorter electron transfer route from CdS to the catalytic  $\text{Zr}^{4+}$  cluster. Thus, with an optimized load of 3.56 wt% CdS HER of  $10.41 \text{ mmol g}^{-1} \text{h}^{-1}$  from the water splitting was achieved. The proposed photocatalytic mechanism and HER are shown in Figure 13c. Moreover, a similar composite containing CdS NPs on the external surface, instead of cavities of MOF-808, showed a lower photocatalytic HER of  $0.15 \text{ mmol g}^{-1} \text{h}^{-1}$ .

Among other MOFs, zeolitic imidazolate frameworks (ZIFs), which are topologically isomorphic with zeolite, possess exposed metal sites, accessible carbon/nitrogen ligands, and good chemical stability, are attractive for solar-driven hydrogen generation. Recently, ZIF-67/CdS hollow sphere (HS) and ZIF-8/CdS HS heterostructures were synthesized through an in-situ self-assembly process, where the ZIF-67 and ZIF-8 are closely coated on CdS HSs to form “double-shell”-like structures with porous outer layers [130]. The optimal proportion of ZIF-67/CdS HS showed a HER of  $1721 \mu\text{mol g}^{-1} \text{h}^{-1}$  under Vis-light illumination, which was  $\sim 3$  times higher than that of ZIF-8/CdS HS ( $555 \mu\text{mol g}^{-1} \text{h}^{-1}$ ). The photocatalytic mechanism for the ZIF-8/CdS HS and ZIF-67/CdS HS was assigned as type-II and Z-scheme mechanisms, respectively. The synthetic route to composite formation alongside the morphological characterization, corresponding model structure and photocatalytic mechanism for the optimized ZIF-67/CdS HS are summarized in Figure 14. In this line, lollipop-shaped  $\text{Co}_9\text{S}_8/\text{CdS}$  nanocomposite via direct annealing of zeolitic imidazolate framework-67 (ZIF-67) as a template and CdS as a sulfur source was reported [131]. It was mentioned that the close proximity of these two materials facilitates electron transport and increases the number of active sites toward the desirable redox reaction. The as-prepared  $\text{Co}_9\text{S}_8/\text{CdS}$  resulted in a remarkably improved HER of  $1852 \mu\text{mol g}^{-1} \text{h}^{-1}$  under Vis-light illumination. In another study, the ZIF-67 derived  $\text{Co}_3\text{O}_4$  was used as precursor to prepare  $\text{ZnIn}_2\text{S}_4@\text{CoS}_2$  photocatalyst [132]. The optimized structure possessing 5%  $\text{CoS}_2$  showed the highest HER performance of  $2768 \mu\text{mol g}^{-1} \text{h}^{-1}$  under the irradiation of simulated sunlight, which is three times higher than the  $\text{ZnIn}_2\text{S}_4$  ( $879 \mu\text{mol g}^{-1} \text{h}^{-1}$ ).



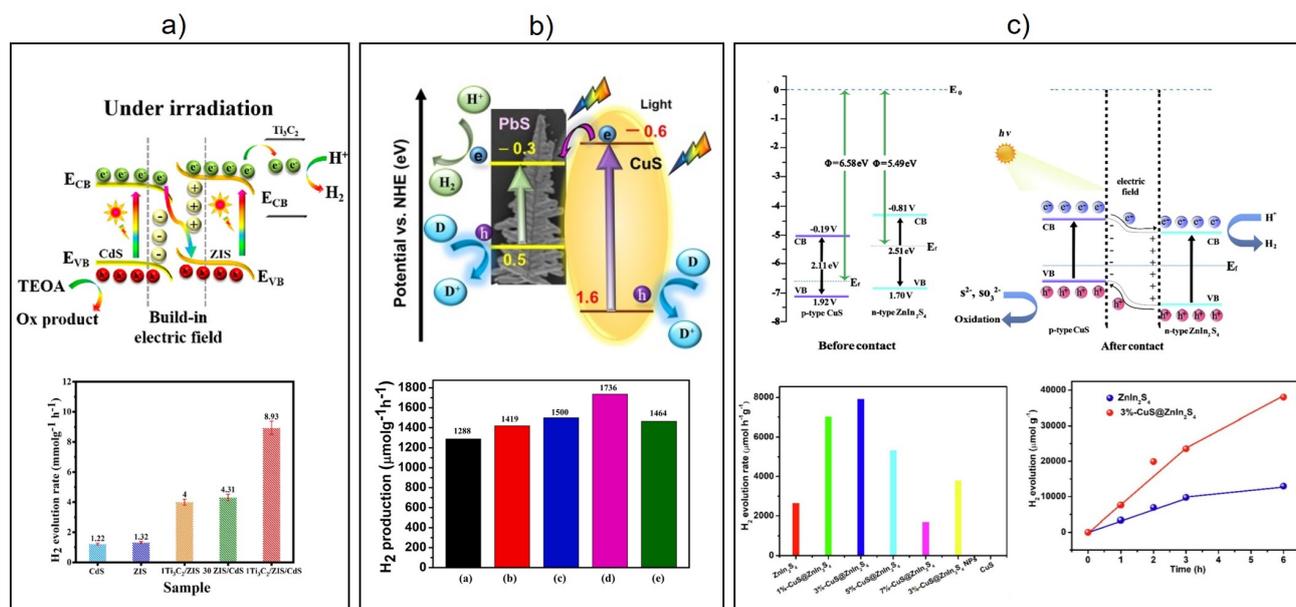
**Figure 14.** (a) The synthesis procedure of the ZIF-67/CdS HS and ZIF-8/CdS HS. (b) TEM and (c) SEM images of the ZIF-8/CdS HS heterostructures. CdS HSs are rapped with the ZIF-8 to form “double-shell” like structure, indicating close contact between the two components. (d) Hydrogen evolution rates and (e) time-dependent photocatalytic performances of various catalysts. (f) Schematic view of the charge transfer mechanism over the ZIF-67/CdS HS photocatalyst under Vis-light illumination [130]. Copyright 2022, American Chemical Society.

## 2.6. Hierarchical Metal Sulfide Photocatalysts

Hierarchical nano- and mesostructures are potential systems for catalysis, electrocatalysis, solar cells, membranes, and photocatalysis, among other applications. Regarding photocatalytic applications the layered (e.g., 2D) and multidimensional (e.g., 1D/2D, 1D/3D, etc.) hybrid heterostructures are particularly appealing. In addition to dimensional advantages (tunable band gap, large surface and interaction area, etc.), these structures enable formation of high energy facets and easy morphological modification, such as the design of membranes [133] and photonic crystals [57], which may enable further photocatalytic enhancement. Recently promising results have been achieved in this field. For example, an ultrathin 2D/2D MoS<sub>2</sub>/ZnIn<sub>2</sub>S<sub>4</sub> co-catalyst was designed and prepared employing two steps of the in situ hydro/solvothermal method [134]. It was suggested that the reduced dimensionality of the components enables the fast charge transfer and suppression of the photogenerated charge carriers recombination. 4 wt% ut-MoS<sub>2</sub>/ZnIn<sub>2</sub>S<sub>4</sub> exhibited the highest H<sub>2</sub> evolution rate of 221.71 μmol g<sup>-1</sup> h<sup>-1</sup> under Vis-light irradiation (λ > 420 nm) with a corresponding AQE of 1.8%.

Another trend of the strategies toward photocatalytic enhancement is the improvement of sulfur vacancies (SV) in metal sulfide compounds. In this regard, CuS NPs, which possess suitable band gap energy [135,136], were employed to prepare CdS@CuS core-shell heterostructure photocatalyst by hydrothermal treatment and subsequent cation-exchange reactions [137]. The CdS-sulfur vacancies enriched by CuS via photoinduced interfacial charge transfer. The developed CdS-SV@CuS(5%) catalyst showed a significantly higher H<sub>2</sub> production rate of 1654.53 μmol g<sup>-1</sup> h<sup>-1</sup>. Similarly, 0D/2D NiS/CdS nanocomposite heterojunction photocatalyst with enhanced photocatalytic HER was reported [138]. There, the composite containing 25% molar ratio of NiS showed the highest HER at about 18.1 mmol g<sup>-1</sup> h<sup>-1</sup>. A Co<sub>9</sub>S<sub>8</sub>@CdIn<sub>2</sub>S<sub>4</sub> heterostructure with hierarchical hollow construction and closely connected heterogeneous shells was also reported for photocatalytic H<sub>2</sub>

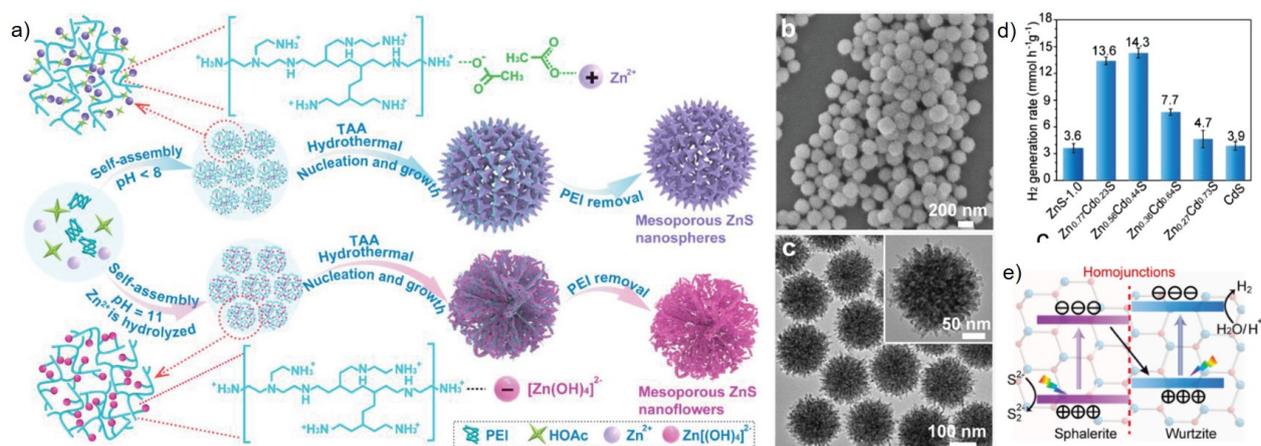
evolution under visible light [47]. It was demonstrated that such a unique architecture may efficiently separate and transmit photoinduced charges while also providing a large surface area and exposing rich sites for photocatalytic redox processes. The designed photocatalyst showed HER of  $4604 \mu\text{mol h}^{-1} \text{g}^{-1}$  under 1.5 AM simulated light irradiation. In other study,  $\text{CoS}_x$ /twinned- $\text{Cd}_{0.5}\text{Zn}_{0.5}\text{S}$  ( $\text{CoS}_x$ /T-CZS) photocatalyst was produced through an in-situ deposition method [139]. Particularly, the formation of the zinc-blende/wurtzite (ZB/WZ) homojunction in T-CZS, as well as the heterojunction between  $\text{CoS}_x$  and T-CZS promoted the charge separation in the bulk and surface of T-CZS. Consequently, T-CZS exhibited much higher activity compared to the zinc blende  $\text{Cd}_{0.5}\text{Zn}_{0.5}\text{S}$  (ZB-CZS) and wurtzite  $\text{Cd}_{0.5}\text{Zn}_{0.5}\text{S}$  (WZ-CZS) with single-phase. Introduction of amorphous  $\text{CoS}_x$  improved  $\text{H}_2$  evolution rate of 4 wt%  $\text{CoS}_x$ /T-CZS to  $76.2 \text{ mmol g}^{-1} \text{h}^{-1}$ . Recently, various hollow  $\text{CoS}$ -supported  $\text{CdS}$  catalyst ( $\text{H-CoS/CdS}$ ) prepared by loading  $\text{CdS}$  onto the surface of  $\text{H-CoS}$  was also reported for  $\text{H}_2$  production in pure water and simulated seawater [140]. The  $\text{H}_2$  production rate of the as-synthesized co-catalyst from simulated seawater reached to  $572.4 \mu\text{mol g}^{-1} \text{h}^{-1}$ . Thus, incorporating the  $\text{h-CoS}$  co-catalyst extended the light absorption range of  $\text{CdS}$ , improved the chemical stability, and significantly enhanced the charge separation leading to high photocatalytic efficiency. In another study, cobalt sulfide was employed to produce 1D-2D  $\text{CdIn}_2\text{S}_4$ /carbon nanofibers (CNFs)/ $\text{Co}_4\text{S}_3$  tandem Schottky heterojunction via in-situ electrospinning combined with a hydrothermal method [141]. The optimised composite showed the  $\text{H}_2$  production activity rate of  $25.87 \text{ mmol g}^{-1} \text{h}^{-1}$  with an AQE of 19.56% at  $\lambda = 365 \text{ nm}$ . Similarly, in order to mitigate the recombination of electrons and holes,  $\text{MoS}_2$  nanosheets with rich active sites were modified on the surface of  $\text{CdS}$  NRs by a room-temperature sonication treatment [142]. The obtained 6%  $\text{MoS}_2$ / $\text{CdS}$  nanocomposites co-catalyst showed improved  $\text{H}_2$  evolution rates of 63.71 and  $71.24 \text{ mmol g}^{-1} \text{h}^{-1}$  in visible light and simulated solar light irradiation, respectively. The AQE of the 6%  $\text{MoS}_2$ / $\text{CdS}$  nanocomposite at 420 nm reached 33.62%. The enhanced photocatalytic activity was assigned to the extra photogenerated charge carriers, which enhanced the charge separation and transfer ability of the  $\text{MoS}_2$ / $\text{CdS}$  co-catalyst. By in-situ epitaxially growing 2D  $\text{ZnIn}_2\text{S}_4$  nanosheets onto the surface of 1D faulty  $\text{CdS}$  NRs, atomic  $\text{CdS/ZnIn}_2\text{S}_4$  heterojunctions were produced [143]. The optimized ZIS/ $\text{CdS}$ -0.3 heterojunctions showed the highest activity ( $5.80 \text{ mmol/g}$ ), which is improved in comparison with the pure  $\text{CdS}$  ( $1.28 \text{ mmol/g}$ ) and ZIS ( $5.14 \text{ mmol/g}$ ). As already discussed, the efficiency of charge carrier separation and the acceleration of the surface reductive process in MS semiconductor photocatalysts are essential parameters in promoting photocatalytic  $\text{H}_2$  evolution from water. The  $\text{Ti}_3\text{C}_2$  (Mxene) modified all-sulfide 2D/2D S-scheme heterojunction  $\text{Ti}_3\text{C}_2/\text{ZnIn}_2\text{S}_4$  (ZIS)/ $\text{CdS}$  composite material was produced in two steps to regulate both parameters [144]. It was demonstrated that the produced S-scheme heterojunction enhances electron-hole separation efficiency (see Figure 6 for S-scheme photocatalysts). In contrast, the intimate 2D/2D van der Waals structure provides a high interaction force and a wide contact area to promote charge transfer. Furthermore, the presence of 2D  $\text{Ti}_3\text{C}_2$  provides the accumulation layer, minimizing electron and hole recombination. Consequently, the obtained  $\text{Ti}_3\text{C}_2$ /ZIS/ $\text{CdS}$  composite photocatalyst could reach to  $\text{H}_2$  evolution rate of  $8.93 \text{ mmol g}^{-1} \text{h}^{-1}$ . The proposed charge transfer mechanism and obtained  $\text{H}_2$  evolution rate are given in Figure 15a.



**Figure 15.** Photogenerated charge carrier transfer mechanisms for different MS-based hierarchical photocatalysts during the hydrogen evolution reaction. (a)  $\text{Zn}_{1-x}\text{Cd}_x\text{S}/\text{CdS}$  [144] Copyright 2022, Elsevier. (b)  $\text{CuS}/\text{PbS}$  [145] Copyright 2021, Elsevier. (c)  $\text{CuS}@/\text{ZnIn}_2\text{S}_4$  [146] Copyright 2022, Elsevier. Corresponding HER plots are shown in the bottom panel. In fig (b) bottom axis, (a) corresponds to clean PbS, while (b–e) correspond to the Cu/Pb ratio of 0.414, 0.446, 0.578, 0.452, respectively.

Recently two-component co-catalyst, dendritic  $\text{PbS}@/\text{CuS}$  core-shell photocatalysts were synthesized via a two-step method. Thus, a dendritic PbS core was formed through an ionic liquid/surfactant-assisted hydrothermal method synthesis, and a CuS shell was grown with an ion-exchange method [145]. In situ Cu L<sub>3</sub>-edge near-edge X-ray-absorption fine-structure (NEXAFS) spectroscopy before and after exposure to solar light showed the photoelectron transfer from CuS to PbS. The CuS shell proportions in these dendritic  $\text{PbS}@/\text{CuS}$  photocatalysts were adjusted to attain a maximal activity of 1736 mol g<sup>-1</sup> h<sup>-1</sup> hydrogen generation. Also, see Figure 15b for the reported HER and proposed charge transfer mechanism.  $\text{Co}_9\text{S}_8$  is another rising member of MS photocatalysts because of its narrow band gap. Using  $\text{Co}_9\text{S}_8$ , two-component (0D) $\text{Co}_9\text{S}_8$  / (3D)  $\text{Bi}_2\text{S}_3$  S-scheme heterojunction for efficient photocatalytic hydrogen evolution was reported [147]. The unique 0D/3D architecture and formed S-scheme heterojunction were shown as an efficient and convenient path for transferring photogenerated charges, which further facilitate the effective separation and diversion of the electrons. A core-shell  $\text{Cu}_{1.94}\text{S}-\text{MnS}$  nanosized heterostructure photocatalyst prepared via cation exchange reaction on djurleite  $\text{Cu}_{1.94}\text{S}$  nanospheres template also reported exhibiting high H<sub>2</sub> evolution rate of 878.1 μmol g<sup>-1</sup> h<sup>-1</sup> [148]. In another study, CuS nanosheets-assembled hollow cubic cages were used to construct 3D hierarchical nanocages photocatalysts. Thus, 3D hierarchical nanocages  $\text{CuS}@/\text{ZnIn}_2\text{S}_4$  photocatalyst with abundant and compact nanosheets 2D/2D hetero-interface was designed [146]. The as prepared structure exhibited an improved photocatalytic H<sub>2</sub> evolution rate up to 7910 μmol g<sup>-1</sup> h<sup>-1</sup> in the absence of any co-catalyst. According to further experimental and theoretical studies, the strong interaction between CuS and  $\text{ZnIn}_2\text{S}_4$  2D/2D hetero-interfaces can remarkably facilitate the separation of photogenerated charge carriers and accelerate the electron transfer. The photocatalytic mechanism and optimization of HER are given in Figure 15c. The CoN with high HER performance was introduced into  $\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$  through simple electrostatic self-assembly [48]. The inclusion of CoN boosts the absorption of Vis-light by the photocatalyst. As a result of its ideal CB location and strong conductivity, CoN receives photogenerated electrons from  $\text{Mn}_{0.2}\text{Cd}_{0.8}\text{S}$  as an excellent electron acceptor in the form of active sites. Remarkably, CoN significantly improved the HER of photocatalysts up to 14.612 mmol g<sup>-1</sup> h<sup>-1</sup>.

As stated above, one of the promising routes for photocatalytic HER enhancement is forming porous structures, which possess large surface area and enhanced excited state lifetime. However, due to the fast precipitation between the metal ions and  $S^{2-}$  ions and the large volume contraction during the sulfide formation, synthesizing such porous structures is not a straightforward process. To cope with this problem, Zhout et al. employed a general polymer-oriented acid-mediated self-assembly method to synthesize highly crystalline Mesoporous metal sulfides (MMSs) (e.g., ZnS, CdS,  $Ni_3S_4$ , CuS, and  $Zn_xCd_{1-x}S$ ) by using polyethylenimine (PEI) as pore-forming agent [149]. The synthesis procedure, together with the structural analysis, is shown in Figure 16a–c. In order to adjust the pH and control the interactions between the inorganic precursors and the PEI, acetic acid was employed. It was found that the absorption edge of  $Zn_{0.56}Cd_{0.44}S$  exhibits a slight red shift as the Cd content increases, indicating band gap narrowing. The optimized MMSs  $Zn_{0.56}Cd_{0.44}S$  photocatalyst having abundant mesopores and homojunction enabled high HER of  $14.3 \text{ mmol g}^{-1} \text{ h}^{-1}$ . The optimized HER plot and proposed charge transfer mechanism are shown in Figure 16d,e, respectively.

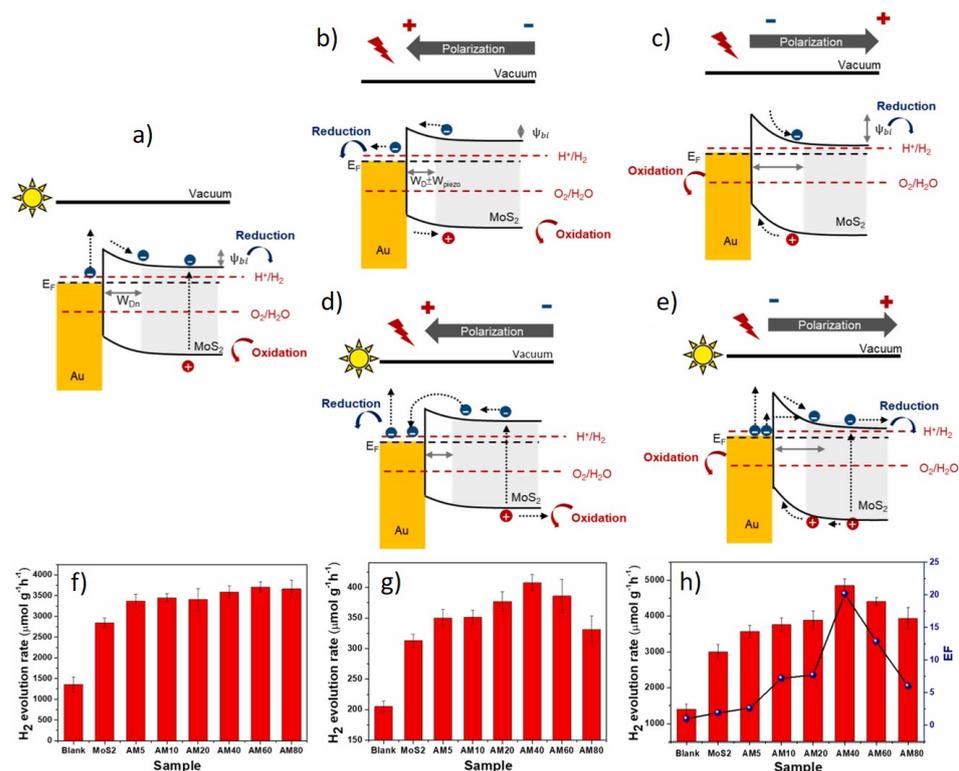


**Figure 16.** (a) Schematic illustration of the synthesis procedure of mesoporous ZnS nanostructures. (b,c) SEM and TEM images of mesoporous ZnS-1.0, respectively. (d) Photocatalytic HER of mesoporous  $Zn_xCd_{1-x}S$  with different Cd content. (e) Schematic view of a photocatalytic charge transfer mechanism for mesoporous  $Zn_{0.56}Cd_{0.44}$  in  $H_2$  evolution reaction. [149] Copyright 2021, Wiley-VCH.

### 2.7. Plasmonic Co-Catalyst Modified Metal Sulfide Photocatalysts

Plasmons, the collective oscillation of the excited conductive charge carriers, have been in the focus of fundamental and applied sciences [150–155]. The optical response of plasmonic nanostructures can be tuned from the ultraviolet through the visible to the near-infrared wavelength, by adjusting the size, shape, composition and geometrical arrangement of the NPs. In this regard, developing new photocatalysts by employing metallic nanostructures with localized surface plasmon resonance (LSPR) is also considered state-of-the-art research. For this purpose, Au and Ag NPs are the most frequently used materials since their light absorption can effectively be shifted to Vis-light region by varying the particle size and shape [156]. In a simple picture, the enhanced electric fields around the plasmonic particle near the surface of the main MS photocatalyst increase the rate of electron-hole pair formation. As a result, the readily separated charge carriers can effectively migrate to the MS surface and drive photocatalytic process. Due to this specific pathway, combining the MS semiconductor photocatalyst with plasmonic-metal NPs can drastically reduce recombination rate. Thus, by combining enhanced light absorption capabilities through light-scattering, near-field amplification, hot-charge injection, or plasmon-induced resonance energy transfer mechanisms, such hybrid structures lead to improved photocatalytic activity and  $H_2$  evolution rate. For example, Au-PbS-CdS nanodumbbells with double Au caps on the ends were produced using a straightforward three-step wet-chemical technique [157]. It was demonstrated that by modifying the size

and spacing of the Au caps, the LSPR-induced lateral electric field distribution surrounding the hybrid structure can be beneficially produced. Thus, the dual-plasmon-induced broad spectral coupling between the metal and semiconductor could be obtained. Moreover, multi-interface structures (e.g., PbS-CdS, Au-PbS, and Au-CdS) provide fast and multiple transfer channels for the photogenerated charges. As a result, the designed Au-PbS-CdS nanodumbbells provided photocatalytic HER of  $513.63 \mu\text{mol g}^{-1} \text{h}^{-1}$ , which is superior to mixed Au, PbS, CdS NPs ( $97.37 \mu\text{mol g}^{-1} \text{h}^{-1}$ ). Also, the CdS/TiN core-shell structure with various thicknesses of the TiN outer layer was studied [158]. The optimized heterostructure involving 15 nm TiN showed a  $\text{H}_2$  production rate of  $86.1 \text{ mmol g}^{-1} \text{h}^{-1}$ , which is a 362% improvement compared to the CdS NWs. This remarkable improvement due to the TiN outer layer was assigned to a synergetic coupling of three main factors; the improvement of light absorption and electron concentration caused by LSPR, the reduction of the photo-induced carrier recombination rate, and the improved chemical stability during photocatalytic water splitting. Tu et al. have also shown that the  $\text{H}_2$  evolution rate of Au NPs encapsulated by  $\text{MoS}_2$  nanoflowers ( $\text{Au}@\text{MoS}_2$  NFs) was remarkably improved due to the coupling of LSPR with the piezophototronic effect [159]. Consequently, applying simultaneous light illumination and mechanical vibration to the obtained structure enhanced the  $\text{H}_2$  evolution rate from 2981 to  $4808 \mu\text{mol g}^{-1} \text{h}^{-1}$  (see Figure 17). Thus, the piezoelectric generation of hot electrons through LSPR was a primary reason for the observed photocatalytic enhancement. The reported mechanism of the photocatalytic process involving LSPR and piezophototronic effects over the  $\text{Au}@\text{MoS}_2$  NFs are described in Figure 17.



**Figure 17.** (a) The schematic representation of  $\text{Au}@\text{MoS}_2$  photocatalyst during light illumination, considering the contact between Au and  $\text{MoS}_2$  with the Fermi level at equilibrium. (b,c)  $\text{Au}@\text{MoS}_2$  photocatalyst under applied mechanical stress, with piezoelectric polarization point to the left side (i.e., under forwarding bias) and piezoelectric polarization point to the right side (i.e., under reverse bias), respectively. The simultaneous light illumination and ultrasonic vibration of  $\text{Au}@\text{MoS}_2$  NFs with piezoelectric polarization point to the left side (d) and piezoelectric polarization point to the right side (e). The bottom panel shows the hydrogen evolution rate obtained for  $\text{Au}@\text{MoS}_2$  NFs photocatalyst with different size Au NPs (i.e., 5, 10, 20, 40, 60 and 80 nm) under light irradiation (f), under ultrasonic vibration (g) and simultaneous light irradiation and ultrasonic vibration (h) [159]. Copyright 2021, Elsevier.

By assembling Au@CdS core-shell NPs on Ag@SiO<sub>2</sub> shell-isolated MPs dual-plasmonic-antenna nanocomposites to achieve the simultaneous efficient generation of energetic hot electrons and strong electromagnetic field enhancement for efficient photocatalytic H<sub>2</sub> evolution was reported [160]. Where, the Au@CdS antenna generates hot electrons to facilitate H<sub>2</sub> evolution reaction, while the Ag@SiO<sub>2</sub> antenna enables strong electromagnetic field enhancement to promote the generation and separation of hot carriers. The existence of both hot-electron transfer and plasmon-induced resonance energy transfer in the designed system was confirmed by transient absorption spectroscopic experiments and electromagnetic field simulations. The designed photocatalyst showed 191.2 mmol g<sup>-1</sup> h<sup>-1</sup> HER under Vis-light irradiation ( $\lambda > 400$  nm). Cheng et al. studied modification of the Schottky barriers over Ni/S vacancy-rich Mn<sub>0.3</sub>Cd<sub>0.7</sub>S (Ni/MCS-s) composites prepared by the self-assembly and photochemical method [161]. The optimized Ni/MCS-s heterostructures exhibited enhanced HER up to 164.1 mmol g<sup>-1</sup> h<sup>-1</sup> in simulated seawater (3.5 wt% NaCl) with the maximum QYE of 60.4% at 420 nm. They showed that coupling plasmonic Ni and S vacancies can effectively lower the Schottky barrier and enhance hot electrons across the interface for the photocatalytic process. In addition, introducing the Ni layer remarkably reduced the corrosion of the catalyst in seawater. Similarly, plasmonic quaternary CuS-AgVO<sub>3</sub>/Ag-TNR hetero nanostructures (HNSs) were employed to improve photocatalytic H<sub>2</sub> generation [162]. The optimized 1 wt% CuS-AgVO<sub>3</sub>/Ag-TNR HNS showed the highest HER of 756  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>. The achieved HER was related to higher charge separation, improved conductivity, LSPR, and electron-storing capacity. Au/g-C<sub>3</sub>N<sub>4</sub>/ZnIn<sub>2</sub>S<sub>4</sub> plasma photocatalyst heterojunction composite with 3D hierarchical microarchitecture was also reported [163]. They showed that integrating Au NPs, the resulting Au/g-C<sub>3</sub>N<sub>4</sub>/ZnIn<sub>2</sub>S<sub>4</sub> composite photocatalyst showed highest HER of 0.973 mmol g<sup>-1</sup> h<sup>-1</sup> under Vis-light irradiation and reached up to the maximum for the Au/g-C<sub>3</sub>N<sub>4</sub> is 40 wt% of ZnIn<sub>2</sub>S<sub>4</sub>. In addition to the implementation of catalysts, the Zn<sub>1-x</sub>Cr<sub>x</sub>In<sub>2</sub>S<sub>4</sub> solid-solution photocatalysts synthesized via a facial hydrothermal method was also reported with an improved HER [164]. The elevated HER was ascribed to the LSPR modes originating from amorphous domains, which facilitated interfacial charge transfer between amorphous and crystalline domains. Moreover, with the increase of Cr percentage in the stoichiometry, the highest photocatalytic activity  $\sim 7$  mmol g<sup>-1</sup> h<sup>-1</sup> for the Z<sub>0.75</sub>C<sub>0.25</sub>IS after 6 hrs reaction time was achieved. This is almost 2.2 times higher than the well-known ZnIn<sub>2</sub>S<sub>2</sub> photocatalyst. As already discussed, incorporating plasmonic materials with metal sulfide photocatalysts results in enhanced hydrogen production activity. Due to the fact that established control of the plasmonic signal over the spectral range and designing effective architectures remains a promising route for further enhancement in hydrogen evolution rate. In this regard, another parameter to optimize is the amount of co-catalyst particles. Feldmann et al., for example, found that shortening the path by increasing the number of co-catalyst particles boosts the transfer rate but decreases photocatalytic performance [165]. This is due to the fact that subsequent electron transport to the acceptor is significantly slower; as a result, the recombination rate with neighboring holes increases, compensating any gain from the enhanced electron transfer rate.

### 3. Summary and Outlook

In this state-of-the-art review paper, we discussed very recent, namely the last two years, developments regarding metal-sulfide-based photocatalysts for H<sub>2</sub> generation. This survey showed that within a short time, a tremendous amount of effort has been devoted to this topic, which has resulted in an improvement of our fundamental understanding of the MS photocatalyzed H<sub>2</sub> production as well as a remarkable enhancement over the H<sub>2</sub> evolution rate, as summarized in Table 1. This is due to the fact that MS compounds, with their immense tunable stoichiometry, crystal structure, and electronic band structure, serve as superior research grounds over other semiconductors. Moreover, we saw that the interaction mechanism and strength of the MS with other co-catalyst and incorporated materials can easily be tuned by modifying the particle size and morphology. On the other

hand, such size reduction and corresponding controllable electronic modification make the MS materials the perfect playground for studying light-matter interaction, which is highly important for designing and developing a new generation of effective photocatalytic systems. This survey also showed that in addition to the new emerging photocatalysts, the conventional ones also still require more studies and stay promising for vast-scale application. Especially promising is combining such well-known photocatalysts with porous materials to overcome the short excited state lifetime and enhance the light absorption capability. Evidently, hierarchical nanostructured are also promising photocatalysts for H<sub>2</sub> evolution, which however investigated only in a limited manner and materials. Hence, more research is required in order to find suitable co-catalysts, to optimize nanostructural morphology and finding suitable doping agent to enhance light absorption range. In this regard, developing photocatalysts with earth-abundant metal based co-catalysts (such as NiS<sub>2</sub>, MoS<sub>2</sub>, CuS, etc.) and noble metal-free co-catalysts should be more prioritized. Moreover, building hierarchical structures based on low-dimensional and metal free materials such as graphene, g-CN, graphene oxide, etc. is more promising also to reduce metal contamination and production cost. Another interesting and very promising photocatalytic systems would be membrane systems having advantage of photocatalysis and membrane separation. Eventually, a better understanding of charge generation, separation, and transport through the photocatalytic process, reactions at the surface, interaction of heterogeneous photocatalyst with the environment is necessary to design stable photocatalysts with an enhanced performance towards scalable applications. One of the most important researches in this direction should be developing facile, controllable and cost effective synthesis methods for photocatalytic systems.

**Table 1.** Summary of recently reported metal sulfide based photocatalysts with improved efficiency for the H<sub>2</sub> evolution.

Photocatalyst	Type/Junction	Rate ( $\mu\text{mol g}^{-1} \text{h}^{-1}$ )	QYE (%)	Incident Light ( $\lambda$ , nm)	Ref.
WN/CdS	Schottky (p-n)	24,130	19.8	>420	[55]
CdS@Ni <sub>2</sub> P	n-n	287	-	480	[53]
NiWO <sub>4</sub> /CdS nanosheets	Z-scheme	26,430	22	420	[54]
PtP <sub>2</sub> @C/CdS (10 wt%)	-	9760	41.67	420	[56]
3 wt% WC/CdS	Schottky	9180	14.3	420	[59]
red-P/Co <sub>9</sub> S <sub>8</sub>	Schottky	4362	-	UV-Vis	[119]
25 wt%-ZnCo <sub>2</sub> S <sub>4</sub> /g-C <sub>3</sub> N <sub>4</sub>	S-scheme	6619	-	Lab-solar	[120]
CuNiS/g-C <sub>3</sub> N <sub>4</sub>	p-n	752.8	-	800	[121]
T-COF/CdS	Z-scheme	500	37.8	365	[122]
Mn <sub>0.2</sub> Cd <sub>0.8</sub> S/Ni-MOF-74	type-II	7104	-	>420	[123]
CdS@NiS/MOFs	p-n	42,700	13.23	450	[124]
ZnCdS/UIO-66(Ce)	type-II	3958	-	UV-Vis	[125]
ZnIn <sub>2</sub> S <sub>4</sub> -MOF layer	-	28,200	22.67	350	[126]
Pt@NH <sub>2</sub> -UiO-66/CdS	type-II	38,000	40.3	400	[127]
CN/amorphous-MoS <sub>2</sub> /Pt	Schottky	5830	8.51	400	[107]
CoS <sub>x</sub> -rGO/TiO <sub>2</sub> (10%)	-	256.97	14.62	365	[98]

Table 1. Cont.

Photocatalyst	Type/Junction	Rate ( $\mu\text{mol g}^{-1} \text{h}^{-1}$ )	QYE (%)	Incident Light ( $\lambda$ , nm)	Ref.
TiO <sub>2</sub> @CuS	LSPR-Schottky	2467	13.4/3.7	UV-vis/near-IR	[99]
BiS <sub>x</sub> /TiO <sub>2</sub> (1.0 wt%)	-	803.2	3.86	365	[100]
NiWS <sub>x</sub> -ND/TiO <sub>2</sub> (3 wt%)	Schottky	4580	13	365	[103]
ZnCo <sub>2</sub> S <sub>4</sub> /TiO <sub>2</sub>	S-scheme	5580	11.5	420	[104]
amorphous MoO <sub>2</sub> /TiO <sub>2</sub>	co-catalyst	880.3	-	UV-Vis	[105]
TiO <sub>2</sub> -ZnIn <sub>2</sub> S <sub>4</sub>	direct Z-scheme	214.9	36.7/11.6	UV/Vis	[106]
MoS <sub>2</sub> /Fe <sub>2</sub> O <sub>3</sub> /g-C <sub>3</sub> N <sub>4</sub>	Z-scheme	7820	-	Vis	[109]
Co <sub>3</sub> S <sub>4</sub> /g-CN	Ohmic	536.0	7.55	400	[111]
NiS@g-C <sub>3</sub> N <sub>4</sub> -30	co-catalyst	3297	-	Vis	[42]
NiS/g-C <sub>3</sub> N <sub>4</sub>	co-catalyst	244	-	420	[112]
PtS/g-C <sub>3</sub> N <sub>4</sub> nanosheets	co-catalyst	1072.6	45.7	420	[113]
Pt/MoS <sub>2</sub> /g-C <sub>3</sub> N <sub>4</sub>	co-catalyst	1595.3	30.9	435	[114]
ZnS-NiS <sub>2</sub> /g-C <sub>3</sub> N <sub>4</sub>	dual co-catalyst	302.7	1.8	>420	[115]
2D SnS/g-C <sub>3</sub> N <sub>4</sub> nanosheets	Z-scheme	818.93	0.55	Solar-simulator	[116]
Ni <sub>x</sub> PB-rGO/CdS	p-n	5790	9	420	[52]
MoS <sub>2</sub> /ZnIn <sub>2</sub> S <sub>4</sub>	co-catalyst	221.71	11.8	420	[134]
CdS-SV@CuS(5%)	-	1654.53	6.51	450	[137]
W <sub>18</sub> O <sub>49</sub> /Cd <sub>0.5</sub> Zn <sub>0.5</sub> S	S-scheme	147,700	45.3	<460	[166]
MoS <sub>2</sub> @MoO <sub>3</sub>	S-scheme	12,416.8	8.43	500	[92]
CdSe/CuInS <sub>2</sub> microflowers	p-n	10,610.37	48.97	420	[72]
3% Ni-doped Mn <sub>0.5</sub> Cd <sub>0.5</sub> S	-	108,300	-	420	[73]
a-NiCuS <sub>x</sub> /TiO <sub>2</sub> (3:1)	-	427.9	34.67	UV	[167]
Co <sub>9</sub> S <sub>8</sub> @CdIn <sub>2</sub> S <sub>4</sub>	type-I	4604	-	Vis	[47]

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