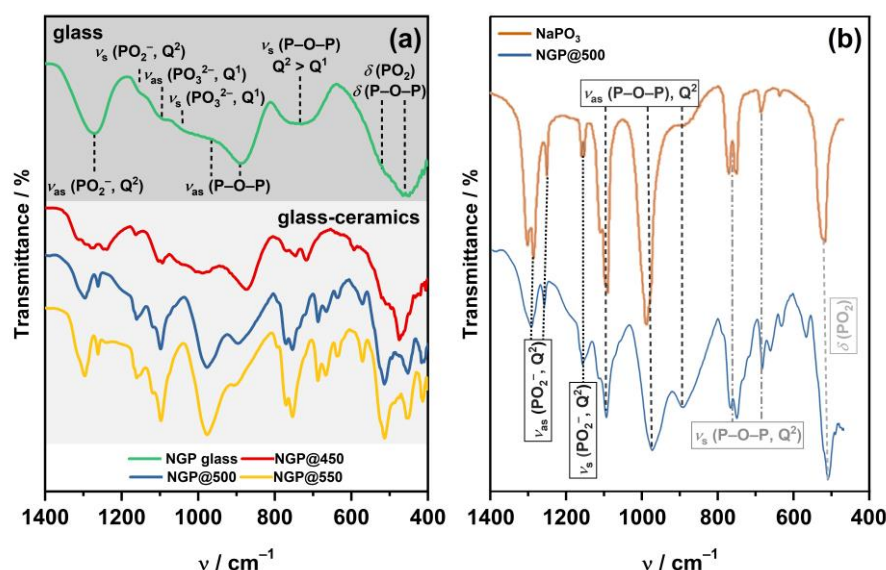


# The Crystallization Behavior of a Na<sub>2</sub>O-GeO<sub>2</sub>-P<sub>2</sub>O<sub>5</sub> Glass System: A (Micro)Structural, Electrical, and Dielectric Study

Sara Marijan <sup>1</sup>, Marta Razum <sup>1</sup>, Kristina Sklepić Kerhač <sup>1</sup>, Petr Mošner <sup>2</sup>, Ladislav Koudelka <sup>2</sup>, Jana Pisk <sup>3</sup>, Andrea Moguš-Milanković <sup>1</sup>, Željko Skoko <sup>4,\*</sup> and Luka Pavić <sup>1,\*</sup>

- <sup>1</sup> Division of Materials Chemistry, Ruđer Bošković Institute, Bijenička 54, 10000 Zagreb, Croatia; smarijan@irb.hr (S.M.); mrazum@irb.hr (M.R.); ksklepickerhac@gmail.com (K.S.K.); mogus@irb.hr (A.M.-M.)
  - <sup>2</sup> Department of General and Inorganic Chemistry, Faculty of Chemical Technology, University of Pardubice, 53210 Pardubice, Czech Republic; petr.mosner@upce.cz (P.M.); ladislav.koudelka@upce.cz (L.K.)
  - <sup>3</sup> Department of Chemistry, Faculty of Science, University of Zagreb, Horvátovac 102a, 10000 Zagreb, Croatia; jana.pisk@chem.pmf.hr
  - <sup>4</sup> Department of Physics, Faculty of Science, University of Zagreb, Bijenička 32, 10000 Zagreb, Croatia
- \* Correspondence: zskoko@phy.hr (Ž.S.); lpavic@irb.hr (L.P.)

## 1. IR-ATR spectroscopy



**Figure S1.** (a) IR-ATR spectra of all samples from this study and (b) comparison of IR-ATR spectra of NGP@500 GC and NaPO<sub>3</sub>.

The IR-ATR spectrum of NGP glass, see Figure S1(a), exhibits distinct bands characteristic of metatraphosphate (Q<sup>2</sup>) glasses, and it also reveals indications of pyrophosphate units (Q<sup>1</sup>), which is consistent with the O/P ratio of 3.1 [1]. The intense band observed at 1275 cm<sup>-1</sup>, followed by shoulder at 1110 cm<sup>-1</sup> is indicative of the asymmetric and symmetric stretching of non-bridging oxygen atoms in Q<sup>2</sup> units, respectively [2,3], while the bands at 970 cm<sup>-1</sup> and 890 cm<sup>-1</sup> can be attributed to the asymmetric stretching vibrations of P–O–P groups in metaphosphate units, manifesting as chains, rings, and terminal groups [2,3]. Additionally, low intensity shoulders at 1150 cm<sup>-1</sup> and 1035 cm<sup>-1</sup> can be assigned to asymmetric and symmetric stretching of non-

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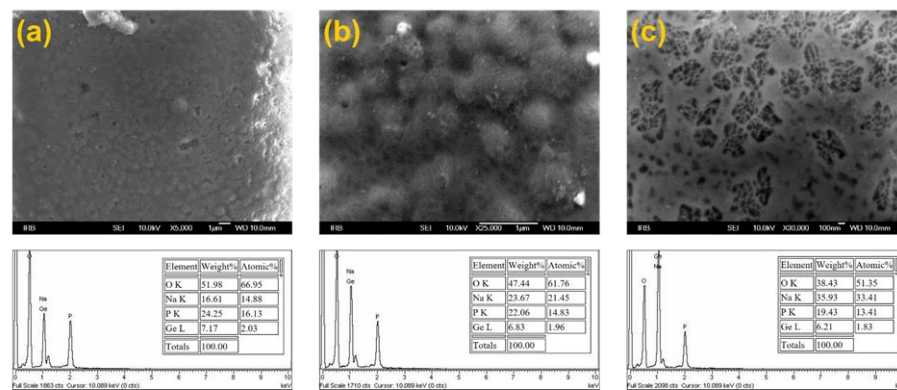


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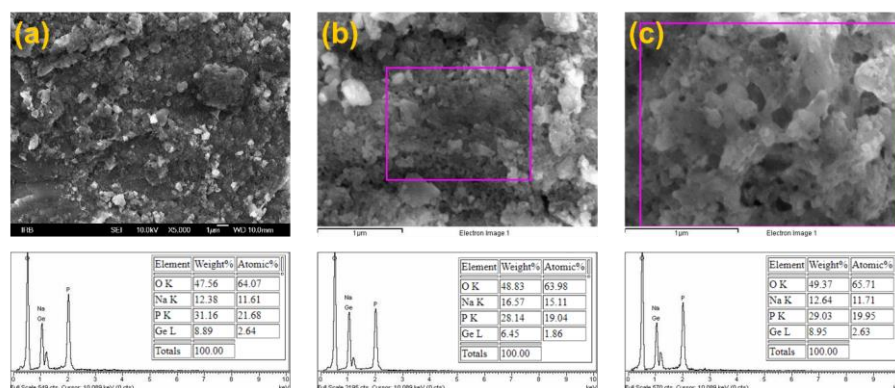
bridging oxygen atoms in  $Q^1$  units, respectively, which presence in small quantities is in accordance with O/P ratio in studied glass. Furthermore, the symmetric stretching vibrations of P–O–P within  $Q^2$  and  $Q^1$  units are found in the range between 640–810  $\text{cm}^{-1}$  [2,3]. Moreover, the bands around  $\sim 460 \text{ cm}^{-1}$  and  $\sim 520 \text{ cm}^{-1}$  can be ascribed to the bending vibrations of P–O–P and  $\text{PO}_2$  units within the metaphosphate framework [2,3]. Notably, the IR-ATR spectroscopy results for the initial glass correlate well with prior Raman spectroscopy findings [1].

In contrast to the broad and diffuse bands observed in the spectrum of the initial glass, the spectra of the prepared glass-ceramics exhibit well-defined, sharp signals characteristic of crystalline materials. As the heat-treatment temperature is elevated from  $450^\circ\text{C}$  to  $550^\circ\text{C}$ , a notable transformation in the appearance of the spectra becomes apparent. The spectrum of NGP@450 glass-ceramic exhibits initial signs of crystallization and the presence of a glass matrix. In contrast, the spectra of NGP@500 and NGP@550 glass-ceramics display remarkable similarity, reflecting a notably advanced stage of crystallization. Here, it's noteworthy to highlight that the acquired IR-ATR spectra of GC samples, characterized by a substantial amount of  $\text{NaPO}_3$  crystal phase, exhibit a striking similarity to the results of the structural analysis of  $\text{NaPO}_3$  documented in [4]. Additionally, the strong agreement observed between the IR-ATR spectra of the prepared GCs and the IR spectrum of the  $\text{NaPO}_3$  crystal phase (SpectraBase®), see Figure S1(b), provides additional confirmation that  $\text{NaPO}_3$  is the predominant phase within the samples.

## 2. SEM-EDS analysis



**Figure S2.** SEM images and EDS spectra of the NGP@450 glass-ceramic surface: (a) area, (b) surface – separation and (c) surface - separation 2.



**Figure S3.** SEM images and EDS spectra of the NGP@500 glass-ceramic surface.

### 3. Electrical analysis

#### 3.1. Complex impedance plane– electrical equivalent circuit modelling

**Table S1.** Fitting parameters obtained from EEC modeling of complex impedance spectra measured at 150 °C for the initial glass and glass-ceramics from this study

Parameters	NGP	NGP@450	NGP@500	NGP@550
$R_1^a$ ( $\Omega$ )	$3.65 \times 10^5$	$3.06 \times 10^6$	$5.03 \times 10^7$	$1.96 \times 10^8$
$A_1^b$ ( $s^\alpha \Omega^{-1}$ )	$2.27 \times 10^{-11}$	$9.41 \times 10^{-12}$	$4.29 \times 10^{-11}$	$1.42 \times 10^{-11}$
$\alpha_1^c$	0.90	0.94	0.86	0.84
$C_1^d$ (F)	$6.22 \times 10^{-12}$	$4.92 \times 10^{-12}$	$1.50 \times 10^{-11}$	$4.63 \times 10^{-12}$
$R_2^a$ ( $\Omega$ )	-	$6.51 \times 10^6$	$3.59 \times 10^8$	$2.58 \times 10^8$
$A_2^b$ ( $s^\alpha \Omega^{-1}$ )	-	$9.23 \times 10^{-10}$	$6.56 \times 10^{-11}$	$2.09 \times 10^{-10}$
$\alpha_2^c$	-	0.60	0.71	0.66
$C_2^d$ (F)	-	$2.95 \times 10^{-11}$	$1.38 \times 10^{-11}$	$4.80 \times 10^{-11}$
$R_3^a$ ( $\Omega$ )	-	-	-	$4.12 \times 10^8$
$A_3^b$ ( $s^\alpha \Omega^{-1}$ )	-	-	-	$2.77 \times 10^{-9}$
$\alpha_3^c$	-	-	-	0.49
$C_3^d$ (F)	-	-	-	$3.18 \times 10^{-9}$
$A_4^b$ ( $s^\alpha \Omega^{-1}$ )	$8.85 \times 10^{-6}$	$2.81 \times 10^{-7}$	$2.37 \times 10^{-8}$	$7.16 \times 10^{-8}$
$\alpha_4^c$	0.91	0.67	0.40	0.81

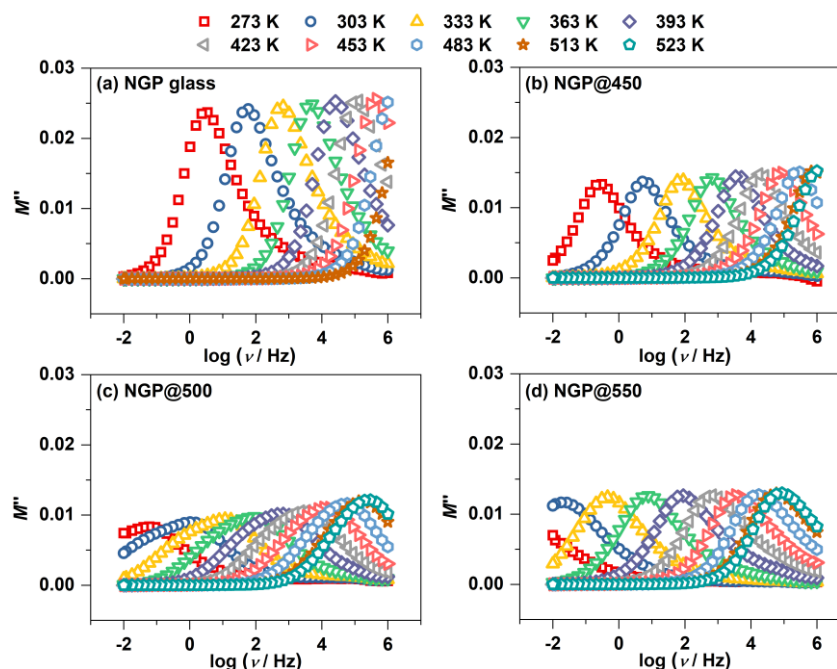
<sup>a</sup> Individual resistance ( $R$ ) values for each R-CPE circuit element in the proposed model.

<sup>b</sup>  $A$  – a constant (CPE capacitance) derived from the empirical impedance function,  $Z_{CPE}^* = 1/A(i\omega)^\alpha$ , measured in  $s^\alpha \Omega^{-1}$  unit. <sup>c</sup>  $\alpha$  – a constant ( $0 \leq \alpha \leq 1$ ) determined from the empirical function,  $Z_{CPE}^* = 1/A(i\omega)^\alpha$ . When  $\alpha = 1$  (true capacitor), the unit becomes  $s \Omega^{-1} = F$ . <sup>d</sup>

Individual capacitance ( $C$ ) values calculated for each R-CPE circuit element in the proposed model using the formula  $C = A(\omega_{max})^{\alpha-1}$ .

## 4. Dielectric analysis

### 4.1. Complex Modulus Formalism



**Figure S4.** Spectra of imaginary moduli,  $M''(\omega)$ , for all samples.

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