

Article

Platinum-Cobalt Nanowires for Efficient Alcohol Oxidation Electrocatalysis

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Abstract: The compositions and surface facets of platinum (Pt)-based electrocatalysts are of great significance for the development of direct alcohol fuel cells (DAFCs). We reported an approach for preparing ultrathin Pt_nCo_{100−n} nanowire (NW) catalysts with high activity. The Pt_nCo_{100−n} NW alloy catalysts synthesized by single-phase surfactant-free synthesis have adjustable compositions and (111) plane and strain lattices. X-ray diffraction (XRD) results indicate that the alloy composition can adjust the lattice shrinkage or expansion of Pt_nCo_{100−n} NWs. X-ray photoelectron spectroscopy (XPS) results show that the electron structure of Pt is changed by the alloying effect caused by electron modulation in the d band, and the chemical adsorption strength of Pt is decreased, thus the catalytic activity of Pt is increased. The experimental results show that the activity of Pt_nCo_{100−n} for the oxidation of methanol and ethanol is related to the exposed crystal surface, strain lattice and composition of catalysts. The Pt_nCo_{100−n} NWs exhibit stronger electrocatalytic performance for both methanol oxidation reaction (MOR) and ethanol oxidation reaction (EOR). The dominant (111) plane Pt₅₃Co₄₇ exhibits the highest electrocatalytic activity in MOR, which is supported by the results of XPS. This discovery provides a new pathway to design high activity, stability nanocatalysts to enhance direct alcohol fuel cells.

Keywords: direct alcohol fuel cells; nanowire; nanocatalysts; alloy effect; lattice strain



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1. Introduction

Direct alcohol fuel cells are gaining increased attention owing to advantages such as good durability, high energy conversion efficiency, clean and low temperature operation, environmental protection and sustainable regeneration [1,2]. Pt is the best catalyst for fuel cells and determines the properties of a fuel cell to a large extent, and is also an effective catalyst for the EOR and MOR. Huang's team found that synthetic Pt catalyst showed very interesting dimension-dependent activity against ORR and MOR [3]. The conventional catalyst for MOR is Pt because of its excellent performance in selectivity and activity [4]; however, the use of Pt is restricted due to its high price and surface poisoning. Additionally, conserving expensive Pt catalysts is essential to advance their scalable usage in a sustainable energy economy. Pt-based catalysts are in a stage of rapid development, and the introduction of non-precious metals is notable for increasing the performance of Pt and reducing the amount of Pt required. Pt was alloyed with first-row transition metals [5–8], which is an acceptable method to solve the above-mentioned problems by virtue of the special electronic structure and geometric configuration [9,10]. One-dimensional (1D) nanostructured electrocatalysts, such as NWs, have been favored to solve the challenge with nanoparticles (NPs) [11–14]. Pt-based NWs possess significant activities [15]. CoPt nanowires have shown good performance in oxygen reduction [16,17] and methanol oxidation [18]. In addition, Pt NWs promote electron transfer during the reaction due to their larger accessible surface area [19]. Therefore, a clear pathway to

enhance the performance of electrocatalysts is to develop Pt-M NWs with ultralong and ultrathin structures. $\text{Pt}_n\text{Co}_{100-n}$ NWs were prepared through a seed-mediated process in which superior nanostructures met the requirements for high density and exponential surfaces, showing better performance than NPs [20]. Using $\text{Co}_{25}\text{Pt}_{75}$ nanoparticles as catalyst, the highest reported current density is 47.1 mA cm^{-2} (Xia et al. [21]). For $\text{Co}_{40}\text{Pt}_{60}$ alloys used as catalysts in the shape of nanowires, the highest reported current density is 14 mA/cm^2 (Bertin et al. [22]), while using $\text{Co}_{23}\text{Pt}_{77}$ NWs as catalyst, the highest MOR activity reported by Serrà et al. [23] is 7 mA cm^{-2} .

In this work, we used a straightforward one-step hydrothermal technique to prepare $\text{Pt}_n\text{Co}_{100-n}$ NWs with tunable compositions, highly active facets, and lattice strain. We firstly found that the electrocatalytic performance of ultrathin $\text{Pt}_n\text{Co}_{100-n}$ nanowires (NWs) ($\approx 2.1 \text{ nm}$) is improved by regulating the compositions, high active facets and lattice strain of catalysts for alcohol oxidation. In mass activity (MA) and specific activity (SA), $\text{Pt}_n\text{Co}_{100-n}$ NW catalyst demonstrates higher MOR and EOR activity and stability in comparison to Pt/C. The structure and chemical makeup of the catalysts were examined using high-resolution transmission electron microscopy (HR-TEM) and XRD. The findings demonstrated that the catalyst compositions alter the lattice strain and dominating (111) facets of $\text{Pt}_n\text{Co}_{100-n}$ NWs with adjustable compositions. The improved activity and stability for MOR and EOR of ultrathin $\text{Pt}_n\text{Co}_{100-n}$ NWs reveals the relation among morphology, facets, lattice strain and bimetallic compositions. Additionally, the inclusion of Co alters the crystal structure, redesigning the electrical structure and considerably reducing the consumption of Pt.

2. Experimental Section

2.1. Chemicals

Ethanol, platinum (II) acetylacetonate ($\text{Pt}(\text{acac})_2$), oleylamine, cobalt (II) acetylacetonate ($\text{Co}(\text{acac})_2$), *N, N*-dimethylformamide (DMF), and 1-heptanol were bought from Aladdin. Hexane and KOH were obtained from Stem Chemicals. Potassium chloride and ethylene glycol (EG) were purchased from Deen reagent. All gases were purchased from Airgas.

2.2. Preparation of $\text{Pt}_n\text{Co}_{100-n}$ NWs

The $\text{Pt}_n\text{Co}_{100-n}$ catalysts were prepared using a one-step hydrothermal method. In this synthesis [24], KOH (1.0 g) was entirely dispersed in a solution including 5.0 mL 1-heptanol, 10 mL DMF, 20 mL oleylamine and 36 mL EG by magnetic stirring. Then, different atomic ratios of $\text{Pt}(\text{acac})_2$ precursors and $\text{Co}(\text{acac})_2$ precursors were dissolved in the above solution, stirred overnight, and subsequently transferred to an autoclave and kept at 180°C for 8 h to obtain $\text{Pt}_{70}\text{Co}_{30}$ NWs. The ratio of $\text{Co}(\text{acac})_2$ and $\text{Pt}(\text{acac})_2$ precursors was adjusted to control compositions. The $\text{Pt}_n\text{Co}_{100-n}$ NWs were washed with ethanol and collected through centrifugation and dispersed in ethanol for further use.

The as-obtained $\text{Pt}_n\text{Co}_{100-n}$ NWs were loaded on Vulcan XC-72 (20 wt% metal loading) carbon to prepare $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ for catalytic measurement. Vulcan XC-72 carbon was added into ethanol and stirred to form a uniform dispersion, and then the as-obtained $\text{Pt}_n\text{Co}_{100-n}$ NWs catalyst was added to the above dispersion and stirred to ensure the catalysts loaded on carbon, which was collected and dried to obtain the final catalyst. Pt NWs was synthesized in the same way without adding the $\text{Co}(\text{acac})_2$ precursor.

2.3. Characterizations

Inductively coupled plasma-optical emission spectroscopy (ICP-OES) was used to analyze the elementary compositions and loading for catalysts [25]. Transmission electron microscope (200 kV) scanning performed on an FEI Titan G2 F20 microscope was used to investigate the morphology of the resulting catalysts [26]. XRD was conducted to analyze the structures of $\text{Pt}_n\text{Co}_{100-n}/\text{C}$. The XPS technique was used to analyze the chemical state of the catalysts [27].

2.4. Electrochemical Measurements

The ink was created by dispersing 2.0 mg of catalysts using ultrasonication for 50 min in a solution including isopropanol, water, and 5 wt% Nafion (19:1:0.015, *v/v/v*). To form a uniformly thin film that served as the working electrode, the catalyst ink (10 μ L) was pipetted onto a polished glassy carbon rotating disk electrode (0.196 cm²). On a CHI 760E electrochemical workstation, electrochemical tests were conducted in a three-electrode cell (CH Instrument, Inc., Bee Cave, TX, USA). By using cyclic voltammetry (CV) and rotating disk electrode (RDE) curves, the electrocatalytic activity of the Pt_nCo_{100-n}/C was investigated. Before CV and RDE tests, the electrolyte was bubbled with pure N₂ and O₂, respectively, for more than 30 min to build a saturated testing environment. The properties of Pt_nCo_{100-n}/C were assessed mainly by MA as well as SA.

3. Results and Discussion

3.1. NW Morphology

KOH, Pt(acac)₂ and Co(acac)₂ were added to synthesize Pt_nCo_{100-n} nanoalloy catalysts in a mixture containing oleylamine DMF, 1-heptanol and EG, which was stirred overnight, transferred to an autoclave, and then kept at 180 °C to obtain the target product. 1-heptanol acted as a reducing agent of Co²⁺ to Co during the synthesis process. In terms of morphology and structure, TEM analysis showed that the as-obtained Pt_nCo_{100-n} exhibited a bundle network structure, which was composed of multiple ultra-thin nanowires interwoven (Figure 1a–c). The diameter of each nanowire was about 0.21 nm, and the ultra-long nanowires were several micrometers in length with a high aspect ratio. The reason why Pt_nCo_{100-n} NWs possess this special structure has two aspects. Firstly, the hydrophobic interaction of DMFs helps to minimize the free energy of Pt_nCo_{100-n} synthesis, resulting in a strong cohesive interaction on the NWs. Secondly, there is a propensity to develop active defects along the NWs, which combine with the different NWs contacted to form the network structure [28]. Structural defects, such as grain boundaries and holes, can enhance catalytic activity. Figure 1 shows TEM images of Pt_nCo_{100-n} NWs (Figure 1a–c) and HR-TEM observations (Figure 1d–f). The HR-TEM image shown in Figure 1 reveals the single crystal properties of each alloy nanowire. The lattice fringes of Pt_nCo_{100-n} fell in between Pt and Co, which was gradually increased, indicating that Co atoms successfully integrated into the Pt nanostructure. As the proportion of platinum increased, the morphology of the NWs was gradually thinned and terminated by the (111) facet, which was reported in our previous work [29,30].

The structure of Pt_nCo_{100-n}/C was characterized by XRD and XPS techniques. Figure 2a displays the XRD patterns of Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C, and Pt₇₀Co₃₀/C. The diffraction peaks of Pt_nCo_{100-n}/C with different compositions were located between Pt (JCPDS No. 04-0802) and Co (JCPDS No. 15-0806), and were obviously exponential with face-centered cubic (fcc) PtCo alloy [31]. The peaks located at 40.37°, 39.87° and 39.82° corresponded to the (111) planes of Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C and Pt₇₀Co₃₀/C, respectively. The (111) diffraction peaks' small downshift indicated a slight increase in Pt content in the NWs. The broad (111) peak, (200) peak and (220) peak indicated the formation of a nanoscale crystal structure. XRD data showed that the compositions of the nanoalloy catalysts for Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C and Pt₇₀Co₃₀/C were uniform, and there was no phase segregation in the samples. The diffraction peak of Pt_nCo_{100-n}/C became sharp and showed a slight blue shift with the increase in Co%, proving that the smaller Co atoms entered into the NWs and might result in lattice strain. The lattice strain would influence the electrocatalytic properties through a possible geometric effect. As shown in Figure 2b, Pt_nCo_{100-n}/C exhibited lattice expansion when the amount of Pt was less than 50%, and lattice shrinkage when Pt% was greater than 50%, conforming to Vegard's law. XPS analysis was performed to understand the chemical states of Pt and Co as well as surface compositions of Pt and Co in Pt_nCo_{100-n}/C. Figure 2c, d shows the Pt 4f and Co 2p XPS spectra, indicating that Co and Pt were in the metallic state. XPS analysis also confirmed the existence of Pt and Co in the as-obtained Pt_nCo_{100-n}/C with different compositions.

Figure 2c illustrates the Pt 4f XPS spectra for Pt₂₇Co₇₃/C, Pt₅₇Co₄₃/C and Pt₇₀Co₃₀/C. The peaks of Pt 4f_{7/2} appeared at 71.9, 71.9, and 72.2 eV, and the peaks of Pt 4f_{5/2} appeared at 75.1, 75.1 and 75.3 eV, respectively. The binding energy of Pt 4f showed an obviously blue shift relative to the XPS data for pure Pt, which suggested charge transfer from Pt to Co. The positive shift of the Pt binding energy suggested that the center of the d-band had shifted down.

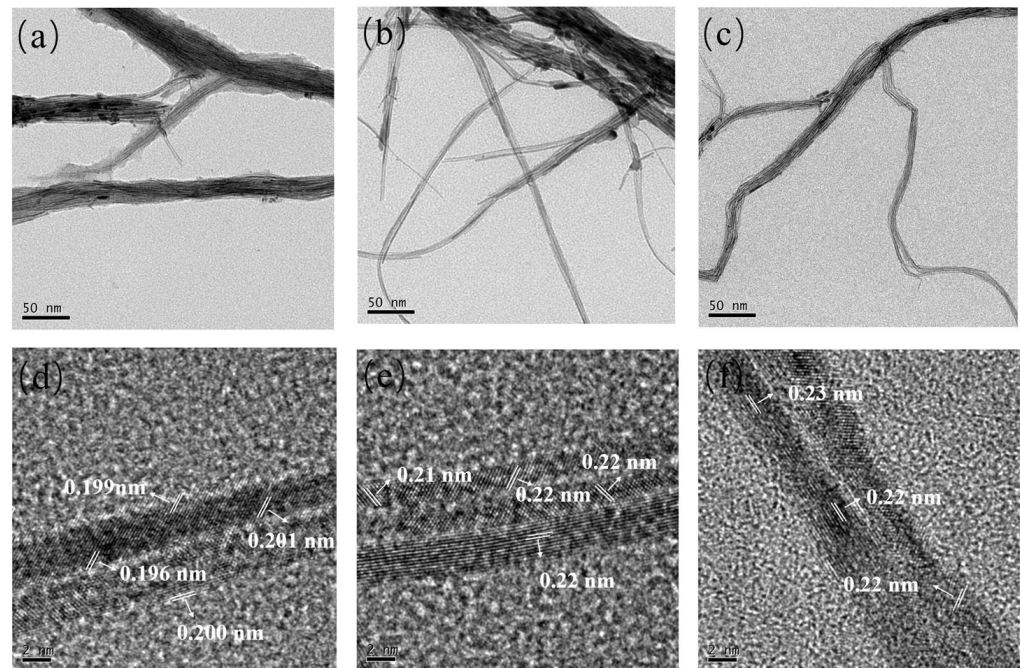


Figure 1. TEM images of Pt₂₇Co₇₃ (a), Pt₅₃Co₄₇ (b), Pt₇₀Co₃₀ (c). HR-TEM images of Pt₂₇Co₇₃ (d), Pt₅₃Co₄₇ (e) and Pt₇₀Co₃₀ (f) with lattice fringes.

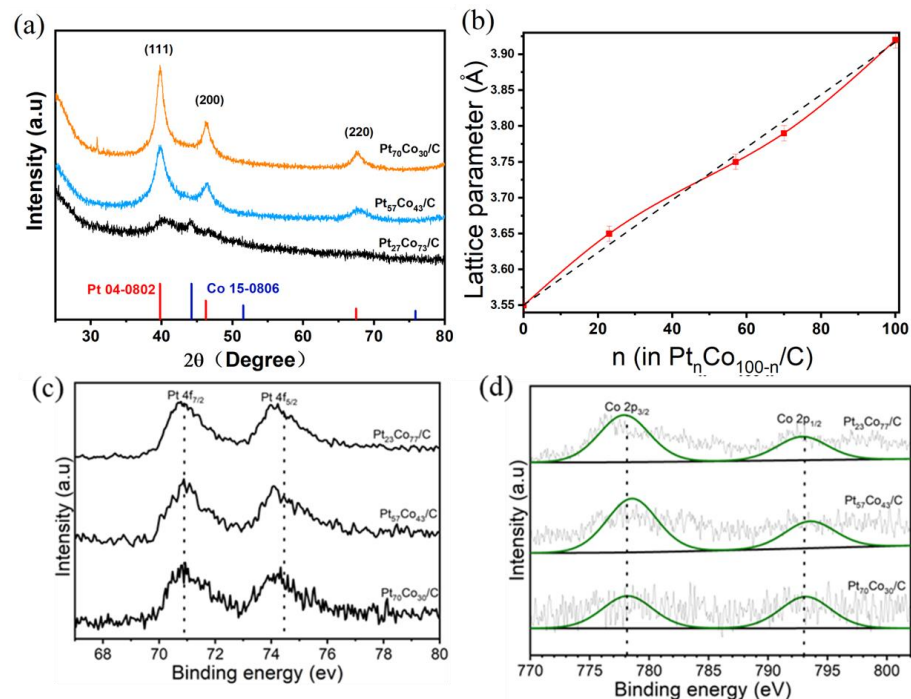


Figure 2. (a) XRD patterns of NWs. (b) Lattice parameters for NWs on the relative composition of Pt%. XPS spectra and deconvoluted peaks in the regions of (c) Pt 4f and (d) Co 2p. The international standard (C 1s) was used to calibrate the peak position.

3.2. Electrocatalytic Properties

The CV technique was used to study the MOR and EOR performance of the $\text{Pt}_n\text{Co}_{100-n}$ NWs/C as anodic catalysts. As shown in Figure 3, CVs were recorded in N_2 -saturated 0.1 M HClO_4 + 0.5 M $\text{CH}_3\text{OH}/\text{C}_2\text{H}_5\text{OH}$ solution with a 50 mV s^{-1} scan speed. In MOR and EOR, the MA and SA were the foremost parameters. The Pt loading of the $\text{Pt}_{27}\text{Co}_{73}/\text{C}$, $\text{Pt}_{53}\text{Co}_{47}/\text{C}$, $\text{Pt}_{70}\text{Co}_{30}/\text{C}$, Pt/C, and commercial Pt were determined to be 35%, 25%, 30%, 30% and 20% by ICP-OES, respectively. We first evaluated the electrocatalytic performance of Pt/C and $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ for MOR. The MOR performance of $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ catalysts is shown in Figure 3a. The peak currents and peak potentials showed that the $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ catalyst exhibited higher activity than the Pt/C catalyst. The $\text{Pt}_{53}\text{Co}_{47}/\text{C}$ exhibited an MA of $2.15 \text{ A mg Pt}^{-1}$, which was 1.38, 2.11, 3.58 and 4.78 times higher than those of $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ ($1.56 \text{ A mg Pt}^{-1}$), $\text{Pt}_{70}\text{Co}_{30}/\text{C}$ ($1.02 \text{ A mg Pt}^{-1}$), Pt/C ($0.60 \text{ A mg Pt}^{-1}$), and commercial Pt/C ($0.45 \text{ A mg Pt}^{-1}$), respectively. The $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ exhibited the highest SA of 3.31 mA/cm^2 , which was 1.655, 2.98, 5.02 and 4.60 times higher than those of $\text{Pt}_{53}\text{Co}_{47}/\text{C}$ (2.00 mA/cm^2), $\text{Pt}_{70}\text{Co}_{30}/\text{C}$ (1.11 mA/cm^2), Pt/C (0.66 mA/cm^2), and commercial Pt/C (0.72 mA/cm^2), respectively (Figure 3b). $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ and $\text{Pt}_{53}\text{Co}_{47}/\text{C}$ displayed higher activity for the MOR (Table 1) in comparison to the literature [32–35]. The EOR performance of commercial Pt/C and $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ catalysts is evaluated in Figure 3c,d. The $\text{Pt}_n\text{Co}_{100-n}/\text{C}$ catalyst exhibited higher activity than commercial Pt/C, as shown in Figure 3c. Figure 3d and Table 2 show that $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ exhibited the largest MA value (2.11 A mg^{-1}) and largest SA (1.44 mA/cm^2) of EOR, which was 3.91 times and 1.67 times that of Pt/C (0.54 A mg^{-1} and 0.86 mA/cm^2), respectively. $\text{Pt}_{53}\text{Co}_{47}/\text{C}$ and $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ had higher activity for the EOR (Table 4) compared with the literature [36–39]. From the catalytic performance of MOR and EOR, it can be seen that the introduction of Co played an important part in the improvement of their performance. The I_f/I_b ratio can be used to certify the CO tolerance of catalysts [40]. The I_f/I_b ratio of $\text{Pt}_{53}\text{Co}_{47}/\text{C}$ catalyst was 1.21, larger than that of Pt/C catalysts (1.05) (Figure 3a), due to facilitating the oxidation of methanol via relieving the CO poisoning with Co. The I_f/I_b ratio of the $\text{Pt}_{27}\text{Co}_{73}/\text{C}$ catalyst was 1.14, larger than that of the Pt/C catalysts (0.83) (Figure 3c), which could be ascribed to its anti-poisoning enhancement in the presence of Co.

Figure 4a shows CVs of $\text{Pt}_{27}\text{Co}_{73}/\text{C}$, $\text{Pt}_{57}\text{Co}_{43}/\text{C}$, $\text{Pt}_{70}\text{Co}_{30}/\text{C}$ and commercial Pt/C catalysts in 0.1 M HClO_4 solution. As shown in Figure 4b, the electrochemical active surface area (ECSA) on behalf of the effective number of active catalytic sites, was determined from the H_2 desorption peak areas of CV. The $\text{Pt}_{27}\text{Co}_{73}/\text{C}$, $\text{Pt}_{57}\text{Co}_{43}/\text{C}$, $\text{Pt}_{70}\text{Co}_{30}/\text{C}$ and commercial Pt/C catalysts exhibited ECSA values of 30.6, 43.2, 54.9 and $49.9 \text{ m}^2\text{g}^{-1}$ Pt, respectively. The introduction of Co induced less exposure of the Pt active sites. However, the alloying with a higher atomic ratio of Co reduced the NW lengths and also led to an increase in the ECSA.

Table 1. Comparison of MOR activities of various catalysts.

Catalyst	Electrolyte	MA (A/mg _{Pt})	SA (mA/cm ²)	Reference
$\text{Pt}_{69}\text{Ni}_{16}\text{Rh}_{15}\text{NWs}/\text{C}$	0.1 M HClO_4 + 0.5 M methanol	1.72	2.49	[32]
UV-Pt@TONR/GN Pd ₉ Ru@Pt/FGN	0.5 M H_2SO_4 + 1 M CH_3OH	1.94 0.881	3.16 -	[33] [34]
1D PtFe alloy	0.1 M HClO_4 + 0.5 M Methanol	1.65	-	[35]
$\text{Pt}_{27}\text{Co}_{73}/\text{C}$ $\text{Pt}_{53}\text{Co}_{47}/\text{C}$	0.1 M HClO_4 + 0.5 M ethanol	1.56 2.15	3.31 2.00	This work This work

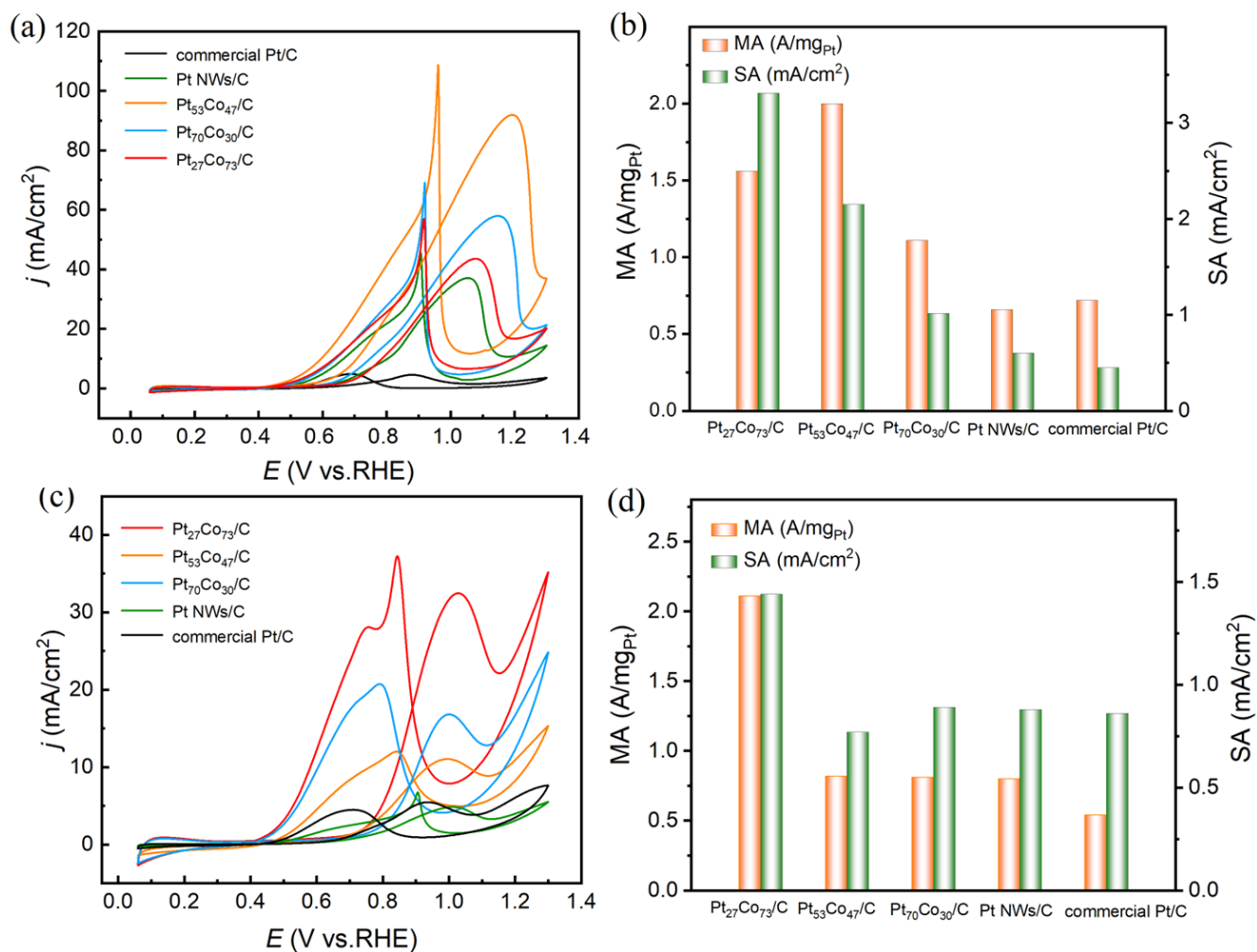


Figure 3. (a,c) CV curves of Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C, Pt₇₀Co₃₀/C, Pt/C and commercial Pt/C in 0.1 M HClO₄ + 0.5 M CH₃OH/C₂H₅OH solution purged with N₂-saturated solution; (b,d) mass activity and specific activity data of methanol oxidation and ethanol oxidation.

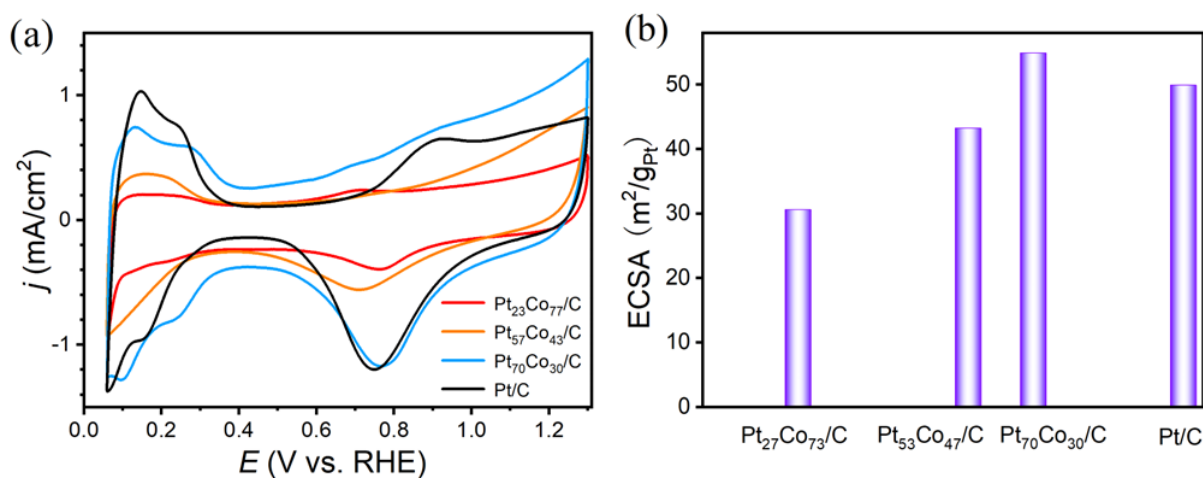


Figure 4. CV curves (a) with the corresponding ECSA (b) of Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C, Pt₇₀Co₃₀/C and commercial Pt/C in 0.1 M HClO₄ solution.

Table 2. Comparison of EOR activities of various catalysts.

Catalyst	Electrolyte	MA (A/mg _{Pt})	SA (mA/cm ²)	Reference
PtSn/XC-72	0.5 M H ₂ SO ₄ + 1 M ethanol	0.76	NA	[36]
Pt-Mo-Ni NWs	0.5 M H ₂ SO ₄ + 2 M ethanol	0.87	2.57	[37]
PtRhNi/C		0.34	NA	[38]
PtCu _{2.1} NWs		1.02	2.16	[39]
Pt ₂₇ Co ₇₃ /C	0.1 M HClO ₄ + 0.5 M ethanol	2.11	1.44	This work
Pt ₅₃ Co ₄₇ /C		0.82	0.77	This work

3.3. Durability

In practical applications, durability is also an important requirement for catalysts [41]. Hence, the initial stability of the Pt_nCo_{100-n}/C was assessed by chronoamperometric (CA) tests. Through a comparison of Pt_nCo_{100-n}/C catalyst with different components and commercial Pt/C, it was found that Pt₅₃Co₄₇/C and Pt₂₇Co₇₃/C have excellent catalytic activity. CA measurements of the catalysts were conducted at 0.65 V (vs. SCE) in N₂-saturated 0.1 M HClO₄ aqueous solution containing 0.5 M methanol/ethanol to evaluate their stability (Figure 4). CA curves indicated that the Pt₂₇Co₇₃/C catalyst and Pt₅₃Co₄₇/C electrocatalysts exhibited excellent stability. The Pt₂₇Co₇₃/C and Pt₅₃Co₄₇/C catalysts exhibited much higher initial current densities than Pt/C. The current density of Pt₂₇Co₇₃/C and Pt₅₃Co₄₇/C remaining were also much higher than that of Pt/C after 5000 s tests. There was about 22.1% and 21.8% activity retention of Pt₅₃Co₄₇ NWs/C and Pt₂₇Co₇₃ NWs/C, respectively, after 5000 s in MOR, much larger than that of commercial Pt/C (10% activity retention). There was about 26.1% and 24.8% activity retention of Pt₂₇Co₇₃ NWs/C and Pt₅₃Co₄₇ NWs/C, respectively, after 5000 s in EOR, much larger than that of commercial Pt/C (9% activity retention). The MOR current densities of Pt₅₃Co₄₇/C catalyst and Pt₂₇Co₇₃/C catalyst were 110.65 mA mg⁻¹ and 59.92 mA mg⁻¹ after 5000 s, which were 5.84 and 3.16 times higher than that of commercial Pt/C (18.95 mA mg⁻¹), respectively (Figure 5a). The EOR current density of catalyst Pt₂₇Co₇₃/C (118.88 mA mg⁻¹) and Pt₅₃Co₄₇/C (86.48 mA mg⁻¹) catalyst were 5.47 and 3.98 times that of commercial Pt/C (21.73 mA mg⁻¹) after 5000 s, respectively (Figure 5b). Moreover, compared with recently reported electrocatalysts [42–47], the Pt₅₃Co₄₇ NWs/C catalyst and Pt₂₇Co₇₃ NWs/C catalyst also exhibited excellent stability under the same conditions (Tables 3 and 4). CA tests of the Pt₅₃Co₄₇/C for MOR were extended to 7500 s at 0.65 V vs. RHE. The current densities remained 91.4 mA mg⁻¹ for MOR after 7500 s (Figure 5c). On the other hand, the electronic transfer rate was notably improved, due to the synergistic electronic interaction between Pt and Co atoms of Pt_nCo_{100-n}/C catalyst. As a consequence, these advantages enabled the catalyst to achieve superior catalytic performance and durability.

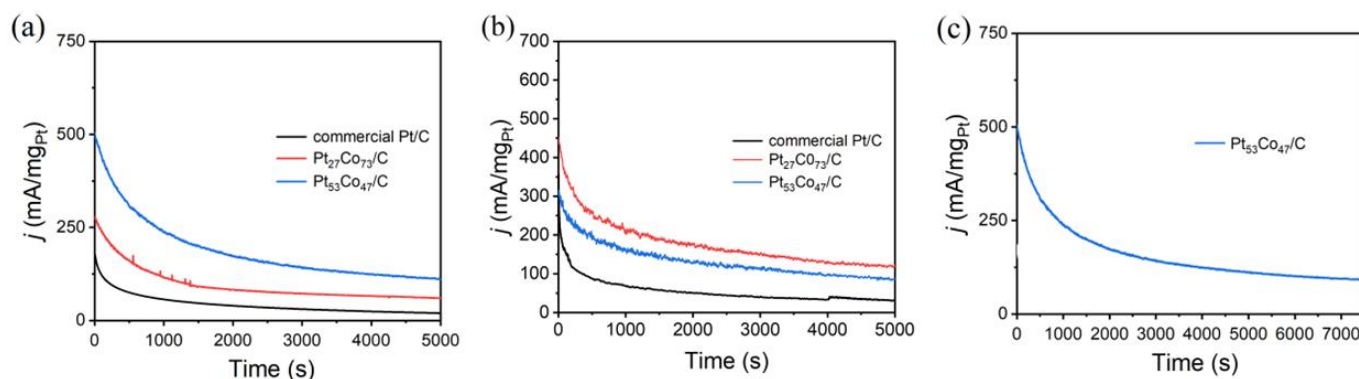


Figure 5. Electrocatalytic durability of the Pt₂₇Co₇₃/C, Pt₅₃Co₄₇/C and commercial Pt/C. Current-time curves of these catalysts recorded at 0.65 V. (a,c) chronoamperometric tests for MOR in 0.1 M HClO₄ + 0.5 M methanol. (b) chronoamperometric tests for EOR in 0.1 M HClO₄ + 0.5 M ethanol.

Table 3. Stability comparison of Pt₂₇Co₇₃ NWs/C and Pt-based electrocatalysts for EOR.

Catalyst	Electrolyte	CA Stability (Activity Retention)	Potential	Reference
Pt HCCLV Pt ₃ Sn/GO	0.5 M H ₂ SO ₄ + 1 M ethanol	~27.0% after 2000 s ~21.7% after 3000 s	0.60 V (vs. Ag/AgCl) peak potential	[45] [46]
Octahedral Pt _{2.3} Ni/C Pt ₂₇ Co ₇₃ NWs/C	0.1 M HClO ₄ + 0.5 M methanol	~14.7% after 1800 s ~25.7% after 5000 s	~0.63 V (vs. RHE) 0.65 V (vs. SCE)	[47] This work

Table 4. Stability comparison of Pt₅₃Co₄₇ NWs/C and Pt-based electrocatalysts for MOR.

Catalyst	Electrolyte	CA Stability (Activity Retention)	Potential	Reference
G@(PEI/Au) 3.5 @Pt Pt ₉₅ Co ₅ NWs	0.5 M H ₂ SO ₄ + 1 M methanol	~5.4% after 2000 s ~7.0% after 3600 s	0.60 V (vs. SCE) 0.60 V (vs. SCE)	[42] [43]
PdRuPt NWs Pt ₅₃ Co ₄₇ NWs/C	0.1 M HClO ₄ + 0.5 M methanol	~16.2% after 5000 s ~22.3% after 5000 s	0.60 V (vs. SCE) 0.65 V (vs. SCE)	[44] This work

The excellent MOR and EOR activity and stability of Pt_nCo_{100−n} NWs were mainly determined by the synergistic effect of Pt and Co bimetallic and the defect-rich nanowire structure. The ultrafine and ultrathin nanowire structure fully exposed the active sites. Ultrathin NWs with Boerdijk–Coxeter structure have different crystal plane orientations of the (111)-dominant facets that can control the electronic structure and generate lattice strain, which is beneficial to improve the performance of MOR and EOR.

4. Conclusions

In conclusion, we demonstrated a simple one-step hydrothermal method for the synthesis of ultrathin Pt_nCo_{100−n} NWs with adjustable compositions. With lattice strain and dominating (111) facets, ultrathin Pt_nCo_{100−n} NWs demonstrated significant MOR and EOR activity and stability while also enhancing alcohol oxidation catalysis. Pt_nCo_{100−n} NWs with controlled compositions have a low starting potential and enhanced MOR and EOR activity because of the robust electronic interaction between metals. The Pt₅₃Co₄₇/C exhibited the highest MA of 2.15 A mg Pt^{−1}, which was 1.38-fold and 4.78-fold higher than that of Pt₂₇Co₇₃/C and commercial Pt/C for MOR. The Pt₂₇Co₇₃/C exhibited the highest SA of 3.31 mA/cm², which was 1.655-fold and 4.60-fold higher than that of Pt₅₃Co₄₇/C and commercial Pt/C for MOR. The Pt₂₇Co₇₃/C exhibited the largest MA value (2.11 A mg^{−1}) and largest SA (1.44 mA/cm²) of EOR, which were about 3.91-fold and 1.67-fold that of Pt/C, respectively. The current densities of Pt₂₇Co₇₃/C and Pt₅₃Co₄₇ NWs/C were much higher than that of commercial Pt/C after 5000 s tests. This study provides an ideal strategy for adjusting the composition of Pt-based alloys. We believe that this study will offer good insights for the preparation of fuel cell electrocatalysts with excellent performance and remarkable durability and will promote the future development of fuel cell electrocatalysts for energy conversions.

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